

Static and Dynamic of the Azide Anion Confined in an Armchair Carbon Nanotube

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1 Computational Details

1.1 Ab-initio

In this work, the (5,5) armchair carbon nanotubes have been treated as finite-size systems, with the addition of hydrogen atoms at both ends in order to complete the valence shell. The nanotubes geometries have been optimized at DFT level of theory, employing the B97D3 exchange-correlation functional[1] in conjunction with the correlation consistent cc-pvtz gaussian basis set[2]. All the optimized structures retained their high symmetry, formally belonging to the D_{5d} molecular point group. Since nanotubes of different lengths have been considered throughout the work, we introduce the notation Λ -CNT(5,5) to label a nanotube composed by Λ [10]cyclophenacene units (cf. [cite]). The C–C bonds parallel ($C-C_{\parallel}$) and perpendicular ($C-C_{\perp}$) to the principal axis show the alternation pattern typical of these systems when treated as finite clusters[3, 4].

The geometry of the N_3^- azide anion has been optimized at DFT level of theory too, with the same functional, but in conjunction with the augmented version of the Dunning basis set, the aug-cc-pvtz in order to better describe the more diffuse electron density. The N–N bond length measures 1.1874 Å and is in excellent agreement with the experimental value obtained in gas phase of 1.1884 Å[5]. The geometry optimizations of the fragments alone have been performed with the Gaussian 09 software package, revision d01[6], using the default convergence thresholds and the ultrafine grid for the functional integral calculation.

1.1.1 Azide anion inside CNT

It has been recently shown[?] that the interaction between the two fragments does not change significantly their structure and that the completely relaxed complex structure is found to be with the azide anion sitting in the center of the nanotube, aligned to the principal axis. Therefore, the optimal adsorption distance between the azide anion and the inner wall of the carbon nanotube has been obtained by relaxing all internal coordinates connecting the two fragments, while keeping frozen those strictly belonging to the monomers only. To ensure that the relaxation process did not remain trapped in a local minimum and without knowing a priori the optimal adsorption distance of the N_3^- molecule, the optimization procedure has been started from four different geometries. In all cases, the anion has been placed parallel to the principal axis of the nanotube. Three starting geometries were chosen according to possible “adsorption” sites in which the central nitrogen atom of the azide was placed either over C–C bonds (*bond* (b) and *zigzag* (z) geometries) or in the center of an hexagonal ring of the CNT wall (*hollow* (h) geometry). The three starting sites are shown in Figure 1, along with their labels that will be used in

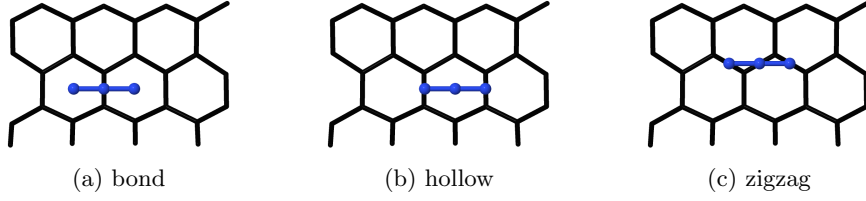


Figure 1: Starting sites of the bond, hollow and zigzag geometries, respectively.

the following to refer to them. For all three cases, the azide anion has been placed at a distance of about 2.25\AA from the nanotube wall, well below the optima adsorption distance. The fourth starting geometry is shown in Figure 2 and it simply consists in the N_3^- anion placed exactly in

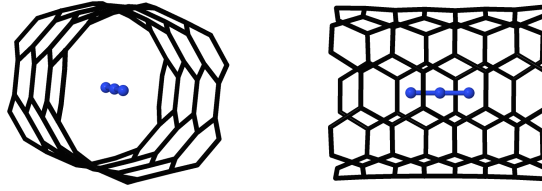


Figure 2: Central starting geometry (c). Note that the nanotube model has dangling hydrogen atoms at the two ends which are not shown here.

the center of the nanotube. We will refer to this starting geometry as *central* (c) geometry. In order to study the edge effects of the finite nanotube model, CNTs of four different lengths have been considered. The relaxation has been performed at density functional theory level using the B97D3 functional with the cc-pvdz basis set on the carbon nanotube and the aug-cc-pvtz on the N_3^- . The resolution-of-the-identity (RI) approximation has been used to compute the Coulomb part, adopting the universal fitting basis set def2/j[?]. The calculations have been performed using the ORCA 4 program package[?].

The relaxed structures starting from the (c) geometry correspond to a (local) minimum, as after one optimization step the convergence is reached, with the N_3^- anion insignificantly ($\approx 0.002\text{\AA}$) shifted out of the principal axis. For this reason we will consider from now on the (c) geometry as the perfectly aligned anion to the principal axis and sitting in the center of the nanotube. For the three other starting conformations, namely the the (b), (h) and (z) geometries, the relaxation leads to different (local) minima. In order to analyze the differences in the final structures, three geometrical parameters and an energetic one have been considered. The geometrical parameters are depicted in Figure 3 and consist in the axial distance d_{cm} between the center of mass of the

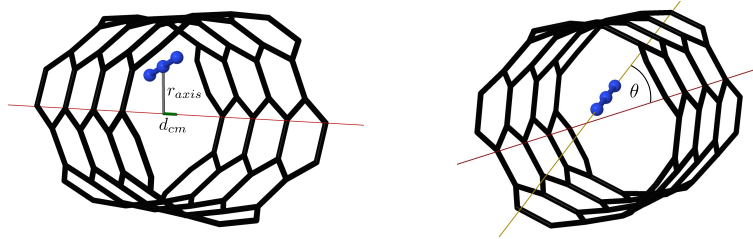


Figure 3: Geometrical parameters.

azide anion and the center of mass of the carbon nanotube, the radial distance r_{axis} of the azide anion center of mass from the principal axis of the nanotube and the angle θ formed between the principal axis and an imaginary line passing through the external nitrogen atoms of the azide anion. The energetic parameter is the interaction energy E_{int} between the two fragments, computed as

$$E_{int} = E_{complex} - (E_{cnt} + E_{azide}) \quad (1)$$

and corrected for the basis set superposition error according to the Boys and Bernardi scheme[7]. Note that although geometry optimizations have been carried out on the complex systems, the geometry of the two fragments was kept frozen at the optimal fragment conformation, thus the value obtained from Equation (1) is only the interaction energy and not the binding energy of the system (i.e. it does not account for deformation effects).

The four parameters computed for all systems are listed in Table 1. Since the (c) geometry

Λ	site	d_{cm} (Å)	r_{axis} (Å)	θ (°)	ΔE_{int} (kcal/mol)
3	<i>bond</i>	0.018	0.002	0.06	-0.001
	<i>hollow</i>	0.002	0.003	0.04	0.000
	<i>zigzag</i>	0.000	0.003	0.27	-0.002
5	<i>bond</i>	0.604	0.009	0.32	0.112
	<i>hollow</i>	-0.144	0.002	0.32	0.017
	<i>zigzag</i>	0.000	0.003	0.43	0.000
7	<i>bond</i>	0.470	0.029	1.10	0.588
	<i>hollow</i>	-0.233	0.027	0.54	0.417
	<i>zigzag</i>	0.001	0.038	0.18	0.363
9	<i>bond</i>	0.622	0.015	0.74	0.400
	<i>hollow</i>	-1.304	0.003	0.17	0.030
	<i>zigzag</i>	-0.007	0.004	0.53	0.141

Table 1: Geometrical parameters and interaction energy difference with respect to the (c) geometry. Note that the d_{cm} , r_{axis} and θ are all zero for the (c) geometry.

consists in the azide ion sitting in the exact center of the nanotube, the three conformational parameters are all zero in this case, thus for geometries (b), (h) and (z), the parameters can be considered as differences from the (c) geometry. Accordingly, instead of tabulating the interaction energy for every geometry, the values listed in the last column of Table 1 correspond to the interaction energy difference between the (c) geometry and the other three, i.e. $\Delta E_{int} = E_{int}(c) - E_{int}(site)$. A negative value for ΔE_{int} means that the interaction energy is more favorable (i.e. stronger) for the (c) geometry, while a positive value the opposite.

For all final geometries, the azide anion is found to be almost perfectly aligned to the principal axis of the nanotube since the angle θ is always very small (below 1° except in one case), confirming the results presented in [?]. Similarly, r_{axis} is also always particularly small (most of the times below 0.01Å), meaning that the anion has the preference to stay approximately at the same distance from the nanotube wall in all directions. In particular, independently from the length Λ , the azide ion is adsorbed at a distance of approximately 3.34Å (computed as the distance between the N_3^- center of mass and the closest side of the wall). This value is significantly larger than the adsorption distance of 3.17Å found for the confinement in a larger 3-CNT(6,6), obtained following the same computational procedure. There are two possible explanations for the difference in the adsorption distance. Either the different curvature of the two nanotubes is

such that the forces attracting the ion are stronger in the case of the (6,6) CNT, or in the case of the (5,5) CNT, there is a competition between the attractive forces exerted by the surrounding wall on the azide ion, which eventually stabilizes it at an (almost) equal radial distance in any direction. The analysis of the curvature effects on the adsorption distance inside the CNT are beyond the scope of this work, and thus not further investigated here [maybe there are refs for other systems, probably H₂ and H₂O]. The most fluctuating parameter is d_{cm} . Clearly, the periodic atomic pattern of the carbon nanotube makes the potential energy surface (PES) of the complex full of local minima, all of which approximately share the same well depth. Apart from the (z) conformation, for which d_{cm} remains virtually unchanged irrespective of the length, it is hard to rationalize the results for the (b) and (h) cases. The only exception is found for the 3-CNT(5,5), where the two open ends of the nanotube are so close that the energy barriers to move away from the center are particularly steep. Nonetheless, although the evident difference between the final geometries, the effect observed on the interaction energy, which is our first and most important parameter that we are interested in, is minimal and, importantly, below chemical accuracy (≈ 1 kcal/mol). As it appears a very hard task to find the *global minimum* for this type of systems and considering the small influence observed on the interaction energy as well as remembering that the accuracy of classical parametrized potentials is well below the differences ΔE_{int} listed in Table 1 [maybe ref?], the central geometry (c) will be used for all interaction energies computed in this work at a higher level of theory or for nanotubes longer than $\Lambda = 9$.

1.2 Molecular Dynamics

The Improved Lennard-Jones (ILJ) potential is given by

$$V_{ILJ}(r) = \epsilon \left[\frac{m}{n(r) - m} \cdot \left(\frac{r_0}{r} \right)^{n(r)} - \frac{n(r)}{n(r) - m} \cdot \left(\frac{r_0}{r} \right)^m \right], \quad (2)$$

where

$$n(r) = \beta + 4.0 \cdot \left(\frac{r}{r_0} \right)^2 \quad (3)$$

and $r = R_{i,j}$ is the distance between carbon atom i and nitrogen atom j . The parameters r_0 , ϵ and β are set according to the atoms interacting, while m is set according to the partial charges of the interacting species. The induction potential term for a N_3^- anion interacting with carbon atoms is given by (in meV)

$$V_{ind}(R_{i,1}, R_{i,2}, R_{i,3}) = -7200 \cdot \alpha_c \cdot \frac{n_2(R_{i,2})}{n_2(R_{i,2}) - 4} \left[\frac{q_1}{R_{i,1}^2} + \frac{q_2}{R_{i,2}^2} + \frac{q_3}{R_{i,3}^2} \right]^2, \quad (4)$$

where $R_{i,1}$, $R_{i,2}$ and $R_{i,3}$ are the distances between the carbon atom i and the three nitrogen atoms, q_1 , q_2 and q_3 are the partial charges on the nitrogen atoms, α_c is the polarizability of the carbon atoms and $n_2(R_{i,2})$ is equal to $n(r)$ using r_0 according to the interaction between $C - N_2$. The interaction energy between a trinitrogen anion confined inside a carbon nanotube is given by

$$E_{int} = \sum_{i=1}^{N_C} V_{ILJ}^{C-N_{1,3}}(R_{i,1}) + V_{ILJ}^{C-N_2}(R_{i,2}) + V_{ILJ}^{C-N_{1,3}}(R_{i,3}) + V_{ind}(R_{i,1}, R_{i,2}, R_{i,3}), \quad (5)$$

where N_C corresponds to the total number of carbon atoms, the ILJ potentials $V_{ILJ}^{C-N_{1,3}}$ and $V_{ILJ}^{C-N_2}$ are defined according to the parameters given in ?? and the polarizability α_c appearing

in V_{ind} is set to 1.2, the partial charges as $q_1 = q_3 = -0.56$ and $q_2 = 0.12$ and finally the term $\frac{n_2}{n_2-4}$ is simplified to 1.0 for the time being. In case of large polarization (large partial charges), the electrostatic interaction between interaction centers has to be taken into account. The fact that we have a charged species (N_3^-), enhances the polarization and it really needs to be taken into account.

Since the aim is to fit the potential to the max interaction energy only (i.e. to fit the well depth ϵ), a very accurate value for this interaction energy is required. In order to do so, an elaborate scheme is required, which involves several steps.

1. Geometry relaxation of the complex system and the monomers. The nanotube length is only 3 units, otherwise there are too many basis functions
 - (a) Using one pure functional among PBE, B97 and TPSS
 - (b) Using Grimme dispersion with Becke and Johnson damping (D3BJ)
 - (c) Using the cc-pvtz basis set on the CNT and the aug-cc-pvtz on the nitrogen (diffuse functions also on the tube lead to linear dependence)
 - (d) Now testing density fitting in Gaussian, if much faster, we use also that for the pure functional with the universal def2/j basis set
 - (e) Using one hybrid functional among B3LYP, APFD, what else?
2. All the geometry calculations are performed in Gaussian
3. To obtain accurate interaction energies, there is a total of 7 single point calculations to be performed
 - (a) Complex at the complex geometry in the full basis
 - (b) First monomer at the complex geometry in the full basis
 - (c) First monomer at the complex geometry in the monomer basis
 - (d) First monomer at the monomer geometry in the monomer basis
 - (e) Second monomer at the complex geometry in the full basis
 - (f) Second monomer at the complex geometry in the monomer basis
 - (g) Second monomer at the monomer geometry in the monomer basis
4. The series of 7 calculations just listed has to be done for two basis sets, to be able to extrapolate the results at CBS limit
5. The extrapolation schemes are different for SCF and correlated calculations, therefore they can be done separately
6. For the SCF extrapolation, we can compute in Gaussian using triple and quadruple zeta basis sets
7. For the correlated extrapolation, we can compute in ORCA using double and triple zeta basis sets (which will give us also dz scf points)
8. The SCF in this case does not use the RI approximation for both Gaussian and ORCA
9. The correlated method is the DLPNO-CCSD(T)

10. Partial charges have to be obtained and there are many ways to do it. First of all, we need to decide which density to use
 - (a) The density (relaxed, unrelaxed, linearized?) of the DLPNO-CCSD(T) method (I do not know how much expensive it is)
 - (b) Find a DFT functional in which we can trust (how?) and use its density
 - (c) Use the density of DLPNO-SCS-MP2 if it reproduces well the CCSD(T) one. But the same problem as above (relaxed, unrelaxed, linearized? Expensive?)
11. Once we know which density we want to use, calculate the charges
 - (a) NPA charges from NBO analysis
 - (b) MESP derived charges
 - (c) Many other possibilities...
12. Calculate multipole moments with the charges obtained and compare them with the moments obtained from the calculation

The above computational scheme will provide a very accurate interaction energy and can be ideally applied to different systems. The major pitfall of the above procedure, is the very first step, i.e. the geometry optimization.

1. The functional used for the geometry optimization might give wrong conformations
2. One should optimize the geometry using a wf method, but if it is too expensive we cannot
3. Moreover there are a plethora of minima in the PES, several starting geometries have to be relaxed and compared with each other

With the interaction energy and the partial charges, we can tune the ILJ potential such that it reproduces the interaction energy obtained with the above scheme.

There is a large amount of arbitrariness in this procedure to obtain the ILJ potential. The partial charges calculated are in principle arbitrary and considering them explicitly enhances the complexity of the problem. In particular, the charges will certainly vary depending on the length of the nanotube. This means that if we would like to use longer CNTs in the MD simulation, then the tuned ILJ potential which includes explicitly the charges might not be valid anymore. I believe that therefore we cannot pursue this approach.

A possible solution would be to fit the ILJ potential to a series of calculations of the complex system at different conformations. In this way, the electrostatic effects would be implicitly taken care of in the fitting procedure. Nonetheless, for such a fit, the above procedure is certainly too expensive and one should come up with a reasonable approximation.

Another possibility is to neglect completely the electrostatic term in the potential and tune it just according to dispersion and induction forces. This is basically the same as the above solution, but without fitting the entire potential and only relying on the single interaction energy value.

A last idea is to assume that in the limit of a very long nanotube, the polarization far away from the nanotube ends is minimal and has to be taken care only locally around the azide anion. This would mean that the partial charges are assigned only locally around the azide anion and they would need to be approximated in some way. One such way is to compute them for a nanotube of increasing length and observe how they change. If the change is minimal, they can be safely assumed to remain so for any length, if this is not the case, then probably this idea is not feasible.

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