

Modelling of Iodine induced Stress Corrosion Cracking (I-SCC) of zirconium alloys with MFRONT

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Outline

1. Context

Pellet Cladding Interaction (PCI)

Iodine induced Stress Corrosion Cracking (I-SCC)

2. Laboratory experiments

Inner pressure tests in iodine vapor

3. Implementation in MFRONT

One element in tension

Cyclic behavior

4. Applications

Inner Pressure tests

Polycrystalline simulations

5. Conclusions and Perspectives





1 ■ Context

Pellet Cladding Interaction (PCI)

Iodine Induced Stress Corrosion Cracking (I-SCC)

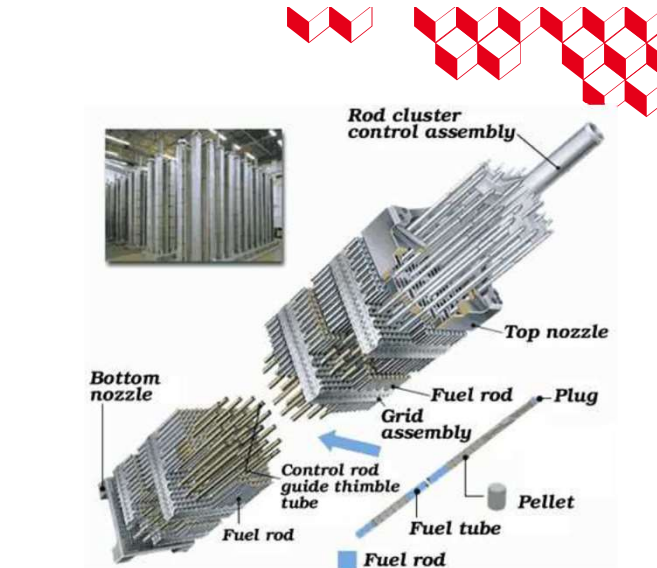
Context

Fuel cladding and Pellet-Cladding Interaction (PCI)

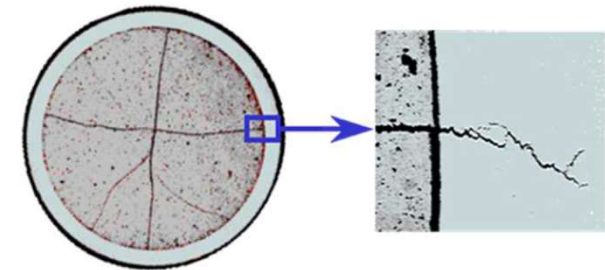
- **Zirconium alloys** are widely used in the nuclear industry as **fuel cladding material** due to their good properties. They are the primary containment barrier for fuel pellets and fission products in pressurised water reactors (PWR).
- During incidental power transients in the reactor, the fuel pellets increase in temperature and expand. Each pellet exerts a high stress on the fuel cladding, and this phenomenon is known as **Pellet-Cladding Interaction (PCI)**.

Iodine-Induced Stress Corrosion Cracking (ISCC)

- During large power variations or incidental situations, the combination of **the mechanical load imposed** on the cladding and **the corrosive environment** could lead to a risk of **Iodine-Induced Stress Corrosion Cracking (ISCC)**



Scheme of Pressurized Water Reactor fuel assembly

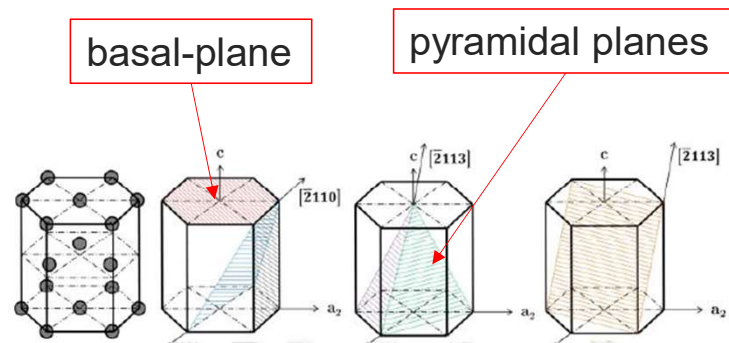


Fuel partitioning in the fuel pellet (radial cracks) and ISCC crack [Davies, 1977].

Context

Iodine-Induced Stress Corrosion Cracking (ISCC)

- The initiation of an ISCC crack is mainly intergranular
- The propagation is mainly transgranular
 - Quasi-clivage on basal-planes
 - Fluting on pyramidal planes



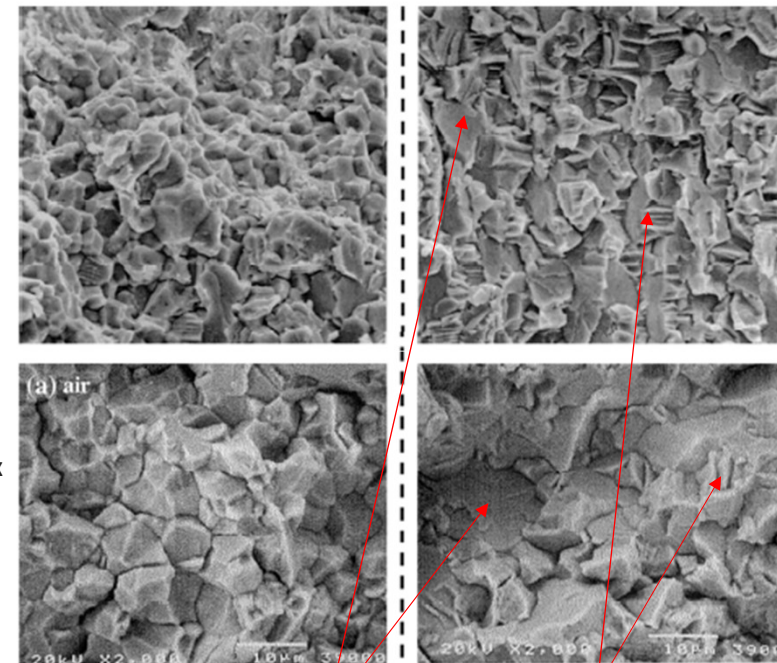
Tenckhoff 2005; Linga Murty et Charit 2006

Essai en rampe au GETR
d'une gaine de Zircaloy-2
(Davies 1977)

Essai de laboratoire sur Zy-4 RX
(Frégonèse 1999)

Intergranular

Transgranular



Quasi-clivage

Fluting



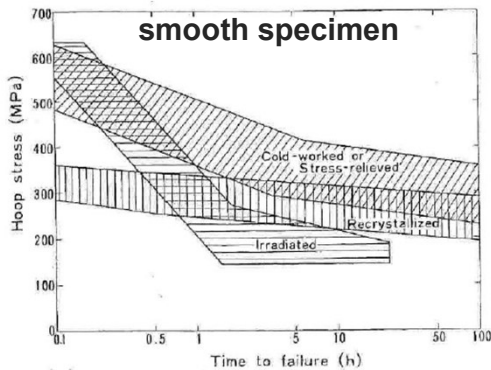
2. Laboratory tests

Inner Pressure tests in iodine vapor

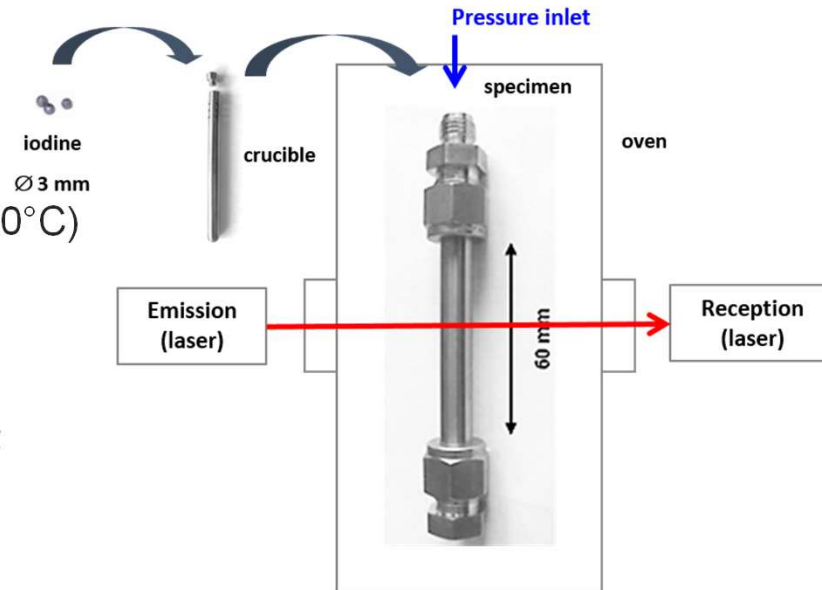
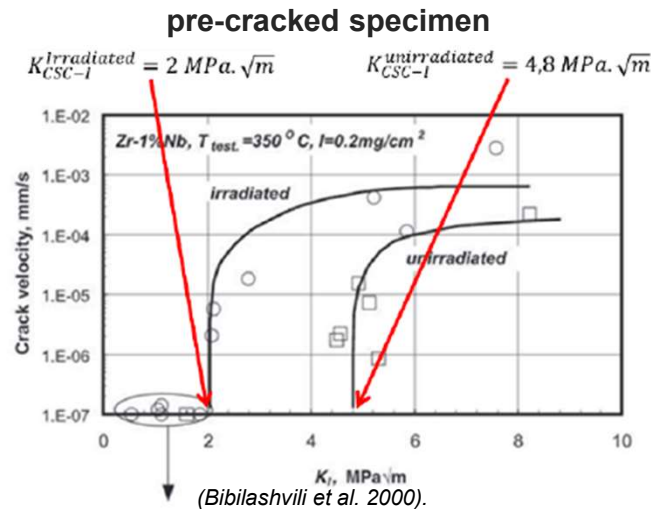
Laboratory tests (example)

Inner Pressure tests with iodine vapor

- Inner Pressure P experimentally controlled
- At PWR's temperature of the inner wall of the cladding (320°C - 420°C)
- On smooth specimen \rightarrow initiation of I-SCC cracks
- On pre-cracked specimen \rightarrow Propagation of I-SCC cracks



Influence de la contrainte tangentielle sur le temps de rupture pour différentes nuance de Zr (Une 1983)



$$\sigma_{\theta\theta} = P \cdot \frac{(R_{out} + R_{in})}{2 \cdot e}$$

$$pI_2 \approx 300\,000 \text{ Pa}$$



3. I-SCC modelling

Kachanov's damage law

KACHANOV's damage law (1/2)



- Coupled to viscoplastic models presented after
- KACHANOV's law suits well for brittle time-dependent failure modes.
 - The parameters depend on the local Iodine quantity.
 - It defines a local isotropic damage variable D to simulate the initiation and the propagation of an I-SCC crack

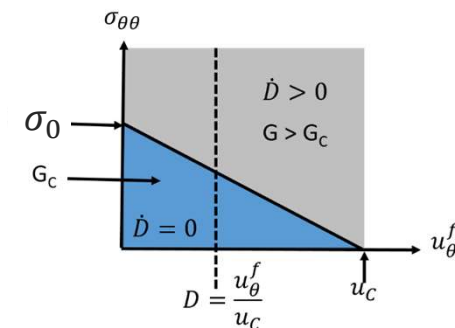
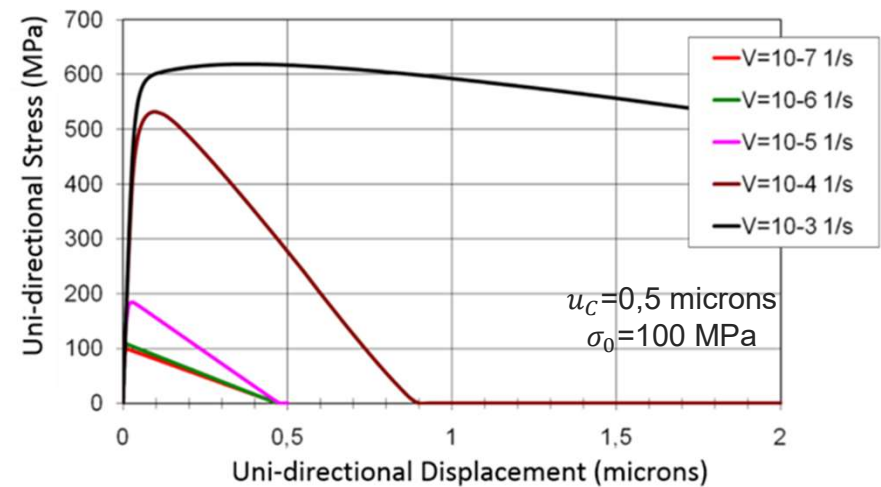
$$\dot{D} = A(I) \cdot e^{-\frac{Q}{T}} \cdot \left\langle \frac{\sigma_D}{1-D} - \sigma_0(I) \right\rangle^B$$

$$\sigma_D = \sqrt{\langle \sigma_{\theta\theta} \rangle^2 + \beta(\sigma_{R\theta}^2 + \sigma_{Z\theta}^2)}$$

- A cracking strain is added ε_f to apply a cracking energy $G_C(I)$ in the calculations

- The cracking strain is added to the other strains of the material (viscoplasticity, thermal dilatancy, ...).
- Is defined with the opening u of the crack and a parameter with is the local size of the element of the mesh $L0$
- It allows to define a critical opening $u_C(I)$ of the I-SCC crack

$$\varepsilon_f = \frac{u}{L0}; D = \frac{u}{u_C(I)} \quad G_C(I) = \frac{1}{2} \cdot \sigma_0(I) \cdot u_C(I)$$



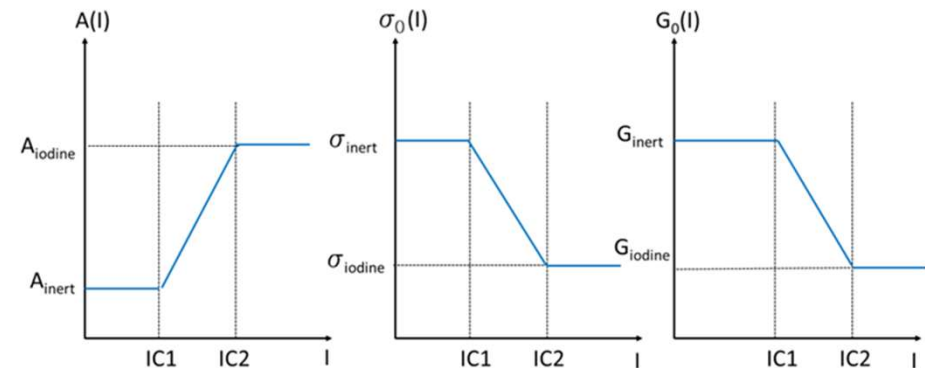


KACHANOV's damage law (2/3)

- Kachanov's model with embrittlement functions according to the iodine content
- Embrittlement functions according iodine content "I":
 - When "I" is lower than a threshold of iodine content "IC1", there is no embrittlement of the material. The Kachanov's model is supposed to model the damage of the material under an inert environment.
 - When "I" is higher than a saturation iodine content "IC2", the I-SCC susceptibility of the material does not change.
 - The local iodine content "I" and the parameters IC1 and IC2 can be physical quantities or dimensionless quantities.

$$\dot{D} = A(I) \cdot e^{-\frac{Q}{T}} \cdot \left\langle \frac{\sigma_{\theta}}{1 - D} - \sigma_0(I) \right\rangle^B$$

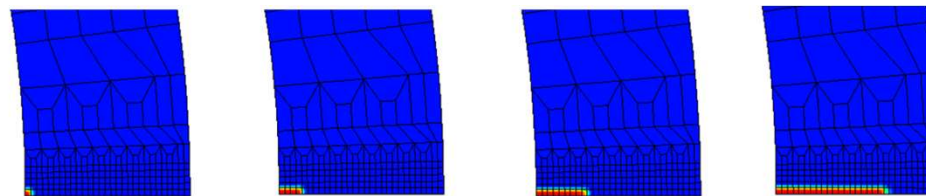
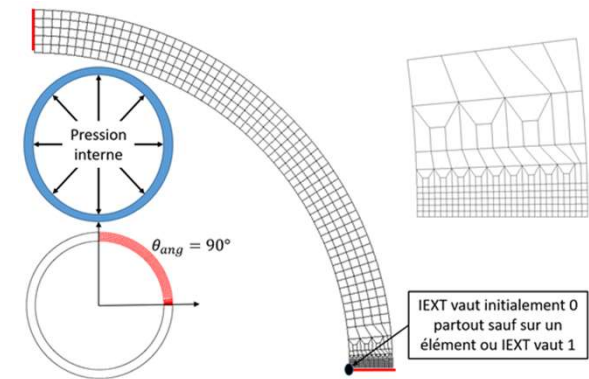
$$G_0(I) = \frac{1}{2} \cdot \sigma_0(I) \cdot u_c(I)$$



KACHANOV's damage law (3/3)

- The local iodine content “I” is assumed to follow I-SCC Miller’s model
 - Based on Fick’s law (diffusion)
 - Assuming that iodine does not diffuse in the zirconium, except in the I-SCC crack or at the tip of a crack, the Fick’s law can be written, in 2D-Rθ, as:

$$\frac{dI}{dt} = \frac{1}{\tau} (I_{EXT} - I)$$
- τ is a period of time characteristic of the diffusion of iodine in the crack and the tip of a crack, in the zirconium.
- External iodine load “I_{EXT}” and the local iodine content “I” are fields of values defined on all the elements of the mesh in the FE simulations.
 - I_{EXT} is 0 everywhere, except in the crack or at the tip of the crack.
 - I_{EXT} is updated after each step of calculation in CAST3M according to the updated position of the I-SCC crack.





Viscoplastic behavior's law (1/3)

- Cladding's viscoplastic model for PCI conditions

- Implemented in MFRONT
- Zy-4 (Annie SONIAK, 1996) and M5 (Annie SONIAK, 2001)
- Primary and secondary orthotropic viscoplastic creep strain combined

- $\dot{\varepsilon}_{eq}^{vp} = V_s + (V_p - V_s) \cdot \exp\left(-\frac{\varepsilon_{eq}^{vp}}{\varepsilon_0}\right)$

- $V_p = V_{p0}(\sigma_{Hill}, T) \cdot V_{pf}(\phi_t)$

$$V_{p0}(\sigma_{Hill}, T) = VP00 \cdot \exp\left(-\frac{Q_{p0}}{T}\right) \cdot \sinh\left(\frac{\sigma_{Hill}}{\sigma_{p0}}\right)$$

$$V_{pf}(\phi_t) = VPF1 \cdot \exp\left(-\frac{\phi_t}{PHI1}\right) + (1 - VPF1) \cdot \exp\left(-\frac{\phi_t}{PHI2}\right)$$

- $V_s = V_{s0}(\sigma_{Hill}, T) \cdot V_{sf}(\phi_t)$

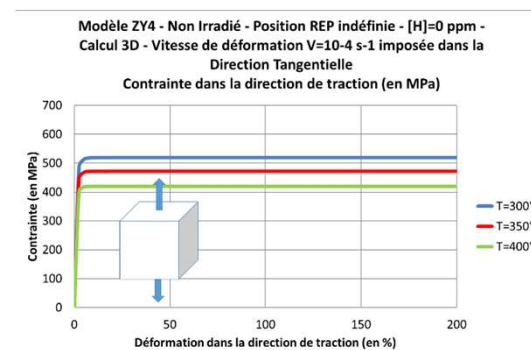
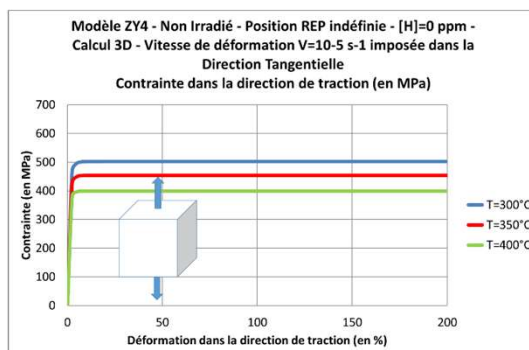
$$V_{s0}(\sigma_{Hill}, T) = VS00 \cdot \exp\left(-\frac{Q_{s0}}{T}\right) \cdot \sinh\left(\frac{\sigma_{Hill}}{\sigma_{s0}}\right)$$

$$V_{sf}(\phi_t) = VSF1 + (1 - VSF1) \cdot \exp\left(-\frac{\phi_t}{PHI1}\right)$$



Viscoplastic behavior's law (2/3)

- Claddings' viscoplastic models for Reactivity Injection Accident (RIA)
 - Implemented in MFRONT
 - ZY4 (Matthieu LE SAUX, CEA, 2008) and CROCORIA (Elodie BOSSO, EDF, 2015)
 - Orthotropic Lemaître's law
 - $\dot{\varepsilon}_{eq}^{vp} = \left(\frac{\sigma_{Hill}}{K.L(\varepsilon_{eq}^{vp})} \right)^{1/m}$





Viscoplastic behavior's law (3/3)

- Effective stress

- $\sigma_{Hill}^{eff} = \frac{\sigma_{Hill}}{1-D}$

- Viscoplastic models coupled to Kachanov's law

- The effective stress acts on the material

- $\dot{\varepsilon}_{eq}^{vp} = (1 - D) \left(\frac{\sigma_{Hill}^{eff}}{K.L(\varepsilon_{eq}^{vp})} \right)^{1/m}$

- Assuming $\sigma_{Hill}^{eff} \cdot \dot{\varepsilon}_{eq}^{eff} = \sigma_{Hill} \cdot \dot{\varepsilon}_{eq}^{vp}$ it allows to define an effective equivalent strain $\dot{\varepsilon}_{eq}^{eff} = (1-D) \cdot \dot{\varepsilon}_{eq}^{vp}$

- $\dot{\varepsilon}_{eq}^{eff} = (1 - D) \left(\frac{\sigma_{Hill}^{eff}}{K.L(\varepsilon_{eq}^{eff})} \right)^{1/m}$



3. Implementation in MFRONT

Strain rate and cyclic behavior



Implementation

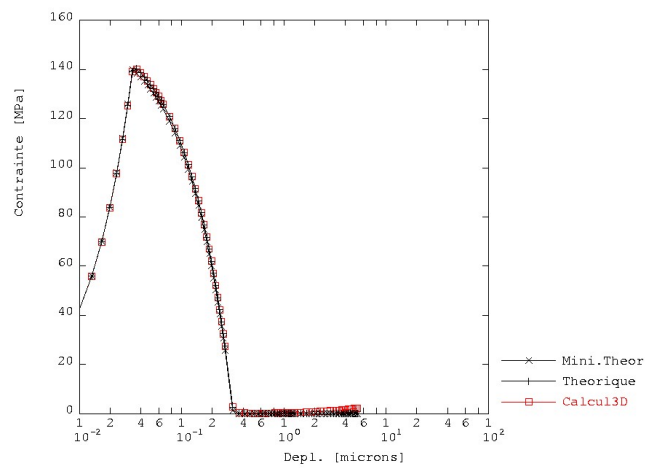
- Damage allowed in the Radial-Axial plane of the cladding only
 - To avoid rotations of the principal stress coordinate system that are un-usefull for our needs
- Solved by Newton-Raphson in MFRONT
 - θ -method with $\theta=0,5$.
 - Jacobian calculated
 - Jacobian validated with the CompareToNumericalJacobian compilation option of MFRONT

$$[J] = \begin{bmatrix} \frac{\partial feel}{\partial \Delta eel} & \frac{\partial feel}{\partial \Delta p} & \frac{\partial feel}{\partial \Delta D} & \frac{\partial feel}{\partial \Delta I} \\ \frac{\partial fvp}{\partial \Delta eel} & \frac{\partial fvp}{\partial \Delta p} & \frac{\partial fvp}{\partial \Delta D} & \frac{\partial fvp}{\partial \Delta I} \\ \frac{\partial fD}{\partial \Delta eel} & \frac{\partial fD}{\partial \Delta p} & \frac{\partial fD}{\partial \Delta D} & \frac{\partial fD}{\partial \Delta I} \\ \frac{\partial fI}{\partial \Delta eel} & \frac{\partial fI}{\partial \Delta p} & \frac{\partial fI}{\partial \Delta D} & \frac{\partial fI}{\partial \Delta I} \end{bmatrix}$$

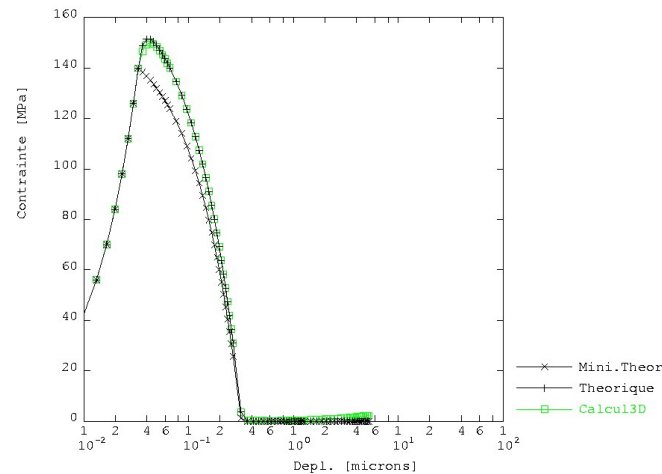


Example on 1-element (1/3)

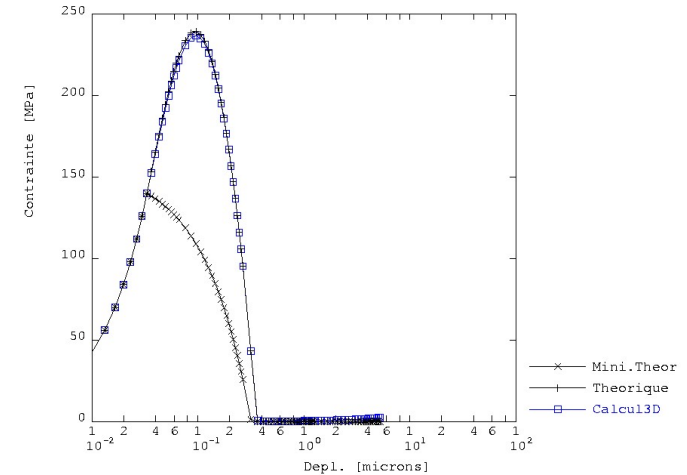
- Viscoplastic model + KACHANOV
- Comparison of the solution calculated with MFRONT compared to an analytical solution
 - One element loaded in constant strain rate in 3D (tested in pure shear too).
- The « Mini.Theor » solution is the theoretical solution obtained with the KACHANOV's law for an infinitely slow applied strain rate.
 - This solution defines an minimal mechanical energy required to lead the material to failure.
 - This solution is recalled on all the figures.



T=320 Celcius, F=100E24 n/m2, V= 1.E-8 1/s



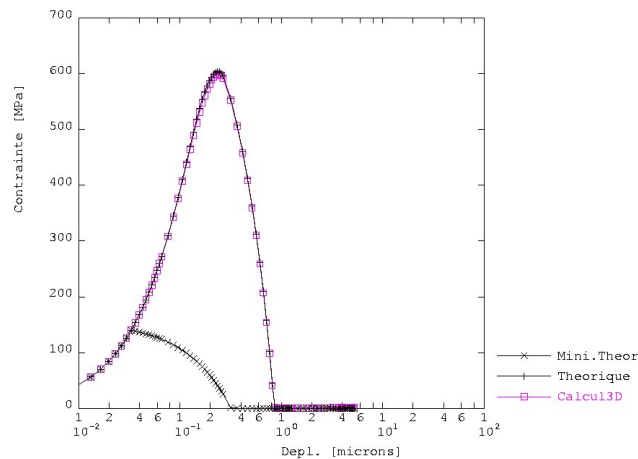
T=320 Celcius, F=100E24 n/m2, V= 1.E-7 1/s



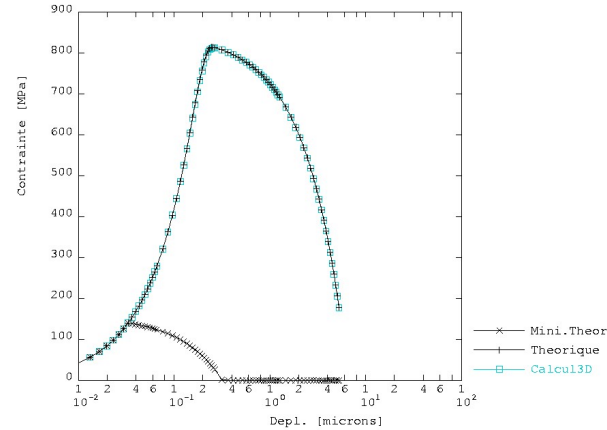
T=320 Celcius, F=100E24 n/m2, V= 1.E-6 1/s

Validation de KACHANOV sur un élément (2/3)

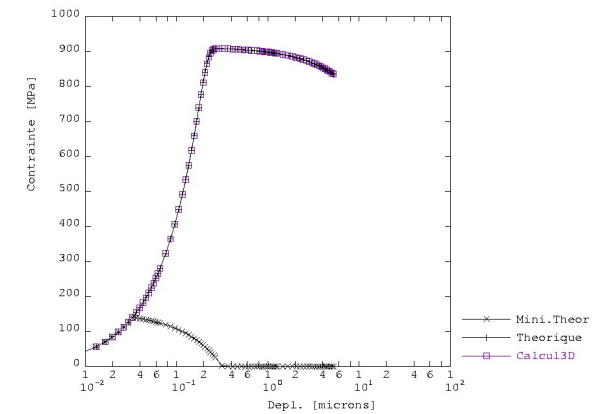
- Viscoplastic model + KACHANOV
- Comparison of the solution calculated with MFRONT compared to an analytical solution
 - One element loaded in constant strain rate in 3D (tested in pure shear too).
- The « Mini.Theor » solution is the theoretical solution obtained with the KACHANOV's law for an infinitely slow applied strain rate.
 - This solution defines an minimal mechanical energy required to lead the material to failure.
 - This solution is recalled on all the figures.



T=320 Celcius, F=100E24 n/m2, V= 1.E-5 1/s



T=320 Celcius, F=100E24 n/m2, V= 1.E-4 1/s



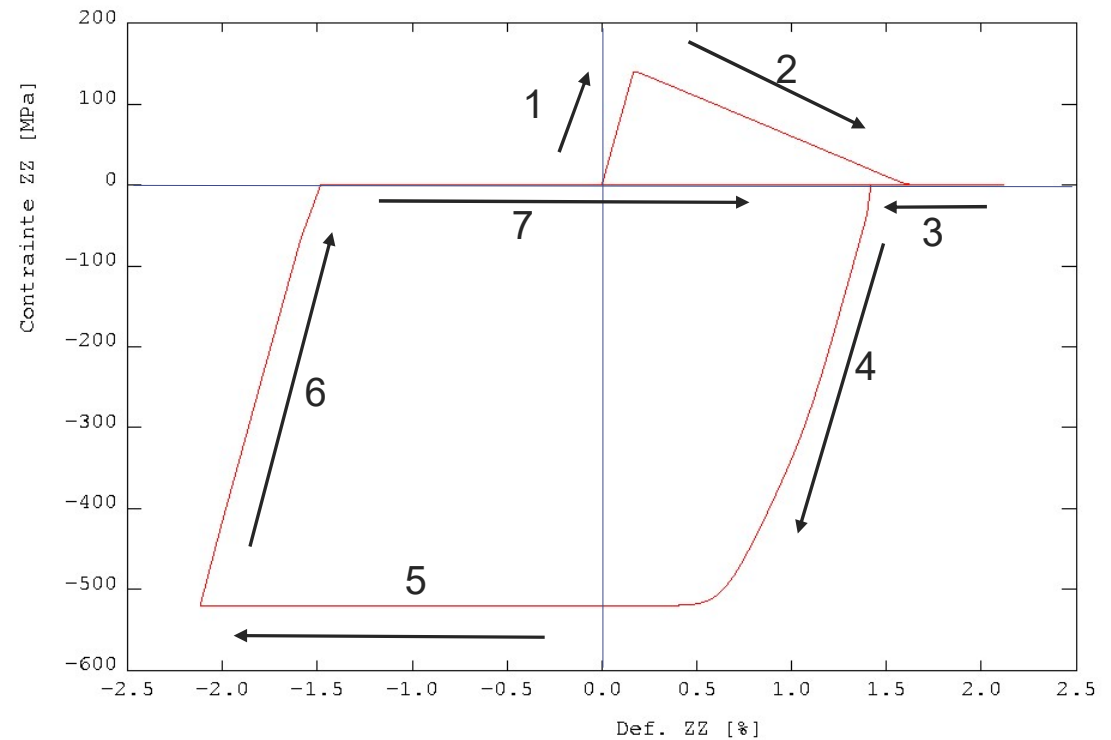
T=320 Celcius, F=100E24 n/m2, V= 1.E-3 1/s

Cyclic behavior(3/3)



■ Example of cyclic behavior

1. Elasto-viscoplastic load in traction
2. Progressive damage to final failure
3. Un-load of the failed element
4. Elastic re-load in compression
1. From the cracking-deformation-to-failure and the viscoplastic déformation (neglectible in this example)
5. Viscoplastic déformation in compression
6. Compressive elastic un-load
7. Elastic load of the failed element in tension



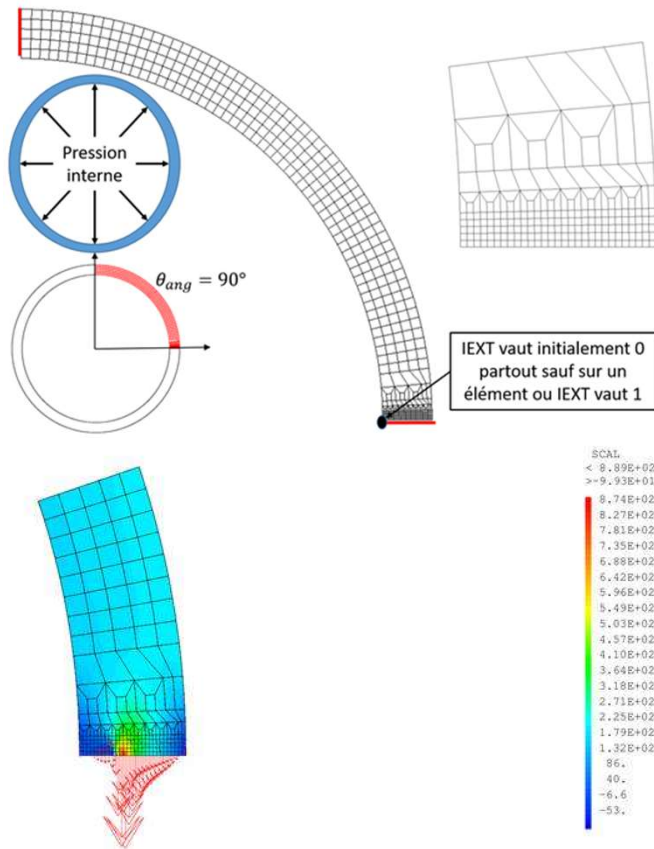
Contrainte - Deformation Totale, T=320 C, F=100E24 n/m2



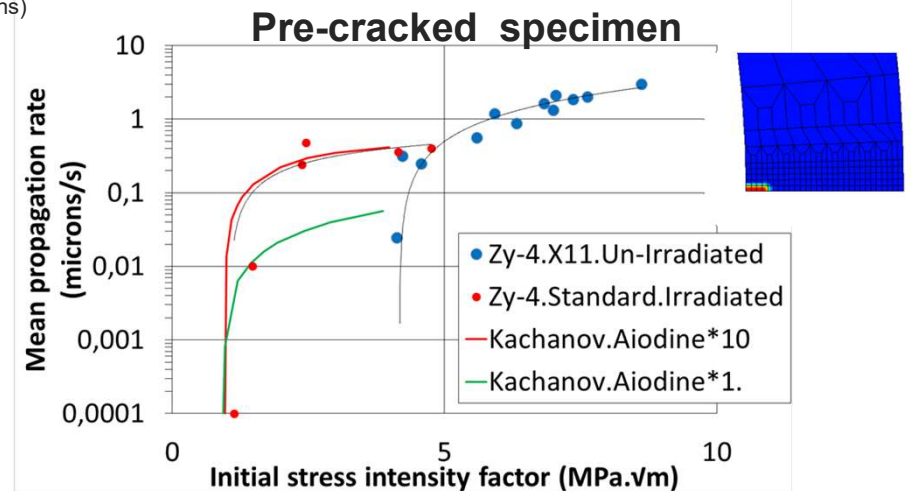
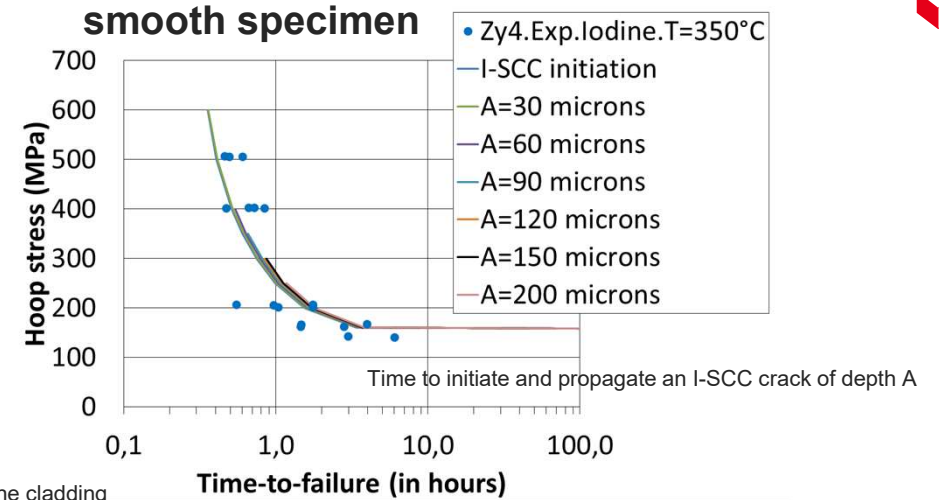
4. Examples of application

Inner Pressure tests, polycrystalline simulations

Inner Pressure Tests



Constant pressure applied inside the cladding
(experiment and simulations)



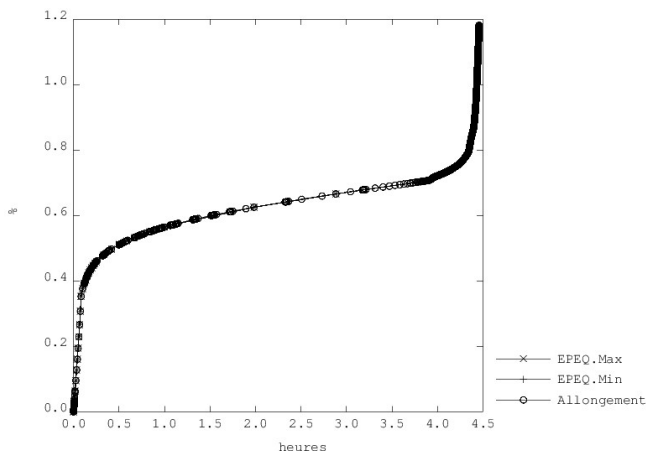
Results presented at OECD-NEA - PCI WORKSHOP – 22-24 june 2016 – Lucca,

24/11/2025

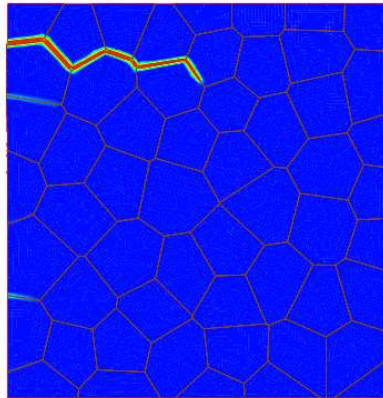
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Polycrystalline simulations

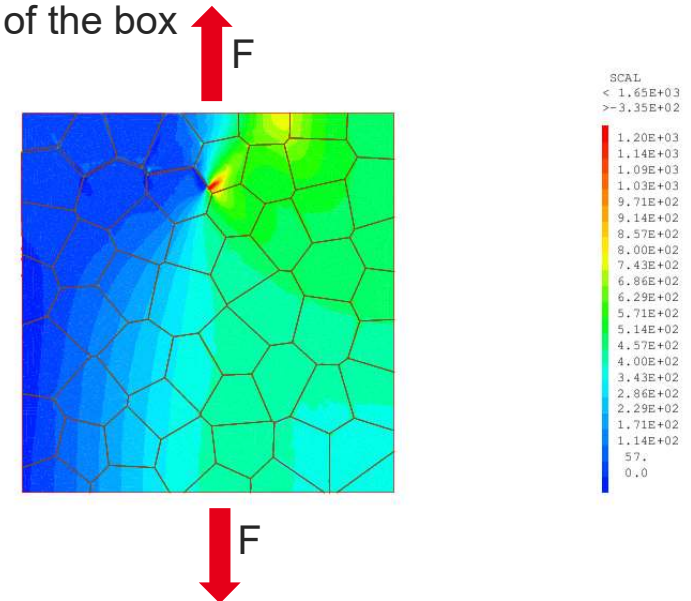
- To simulate the intergranular initiation of an I-SCC crack.
 - The KACHANOV-MILLER model has been implemented in the Grain-Boundaries of a polycrystalline mesh
 - Orthotropic viscoplastic models in the grains.
 - Constant resulting force F applied at the upper and lower boundaries of the box



Vertical elongation of the box as a function of the time



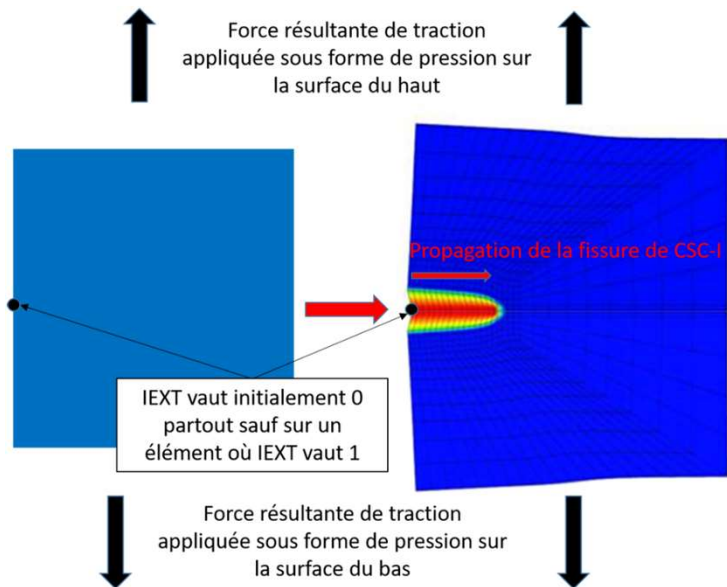
I-SCC intergranular crack at 4h27mn



Local vertical stress at 4h27mn

Example on a plate and a tube

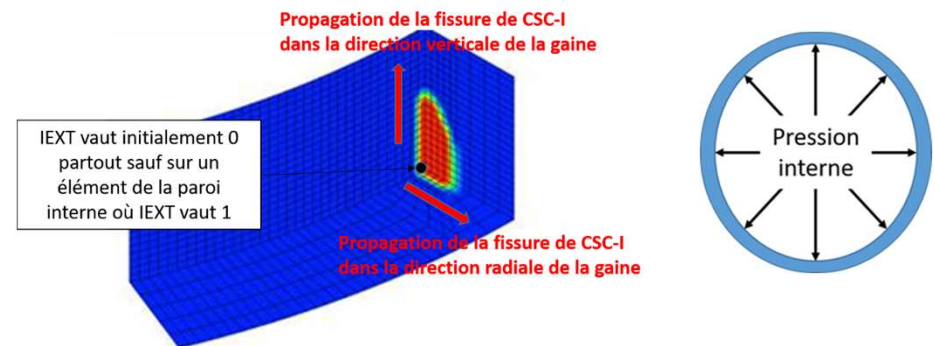
■ Plate in tension a 2D



■ Portion of cladding in inner pressure in 3D

■ Miller's diffusion law generalised to 3D

$$\dot{I} = \left(\frac{n_R}{\tau_R} + \frac{n_Z}{\tau_Z} \right) (I_{EXT} - I)$$





4. **Conclusions**

Perspectives

Conclusions and perspectives

- KACHANOV implemented in MFRONT
 - KACHANOV was implemented in MISTRAL before
 - MISTRAL is a catalogue of behavior's law. It pre-allocates 99 parameters and 99 inner variables in the computer's memory whatever the behavior's law of the catalogue
 - KACHANOV works faster with MFRONT
- Coupled to a simplified diffusion of iodine to simulate the initiation and the propagation of an I-SCC crack
 - At a macroscopic scale (in the frame of homogeneous and continuous media)
 - At a polycrystalline scale (to simulate the intergranular initiation)
- Perspectives
 - Replacing the MILLER's simplified diffusion model by a physical diffusion model.

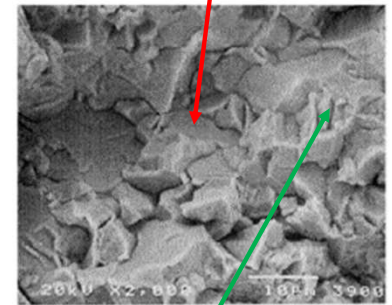
$$\frac{\partial I}{\partial t} = -D_{Zr} \cdot \left(\nabla I + \frac{IQ^*}{RT^2} \nabla T - \frac{\bar{IV}}{RT} \nabla P \right)$$

- The implementation in MFRONT makes it possible to simulate the transgranular propagation at the polycrystalline scale
 - Using the model, coupled with an orthotropic model, in each grains

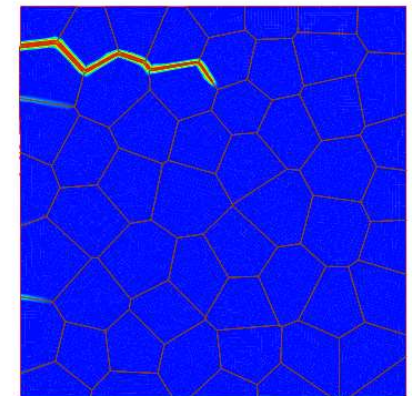
$$\sigma_D = \sqrt{\underset{\text{Quasi-clivage}}{\langle \sigma_{\theta\theta} \rangle^2} + \underset{\text{Fluting}}{\beta (\sigma_{R\theta}^2 + \sigma_{Z\theta}^2)}}$$

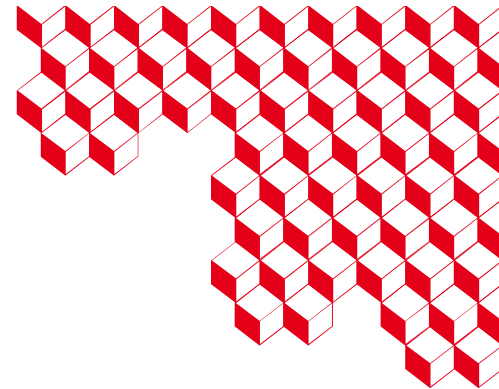


Quasi-clivage



Fluting





Thank you

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Laboratory tests (example)

Inner Pressure tests with iodine vapor

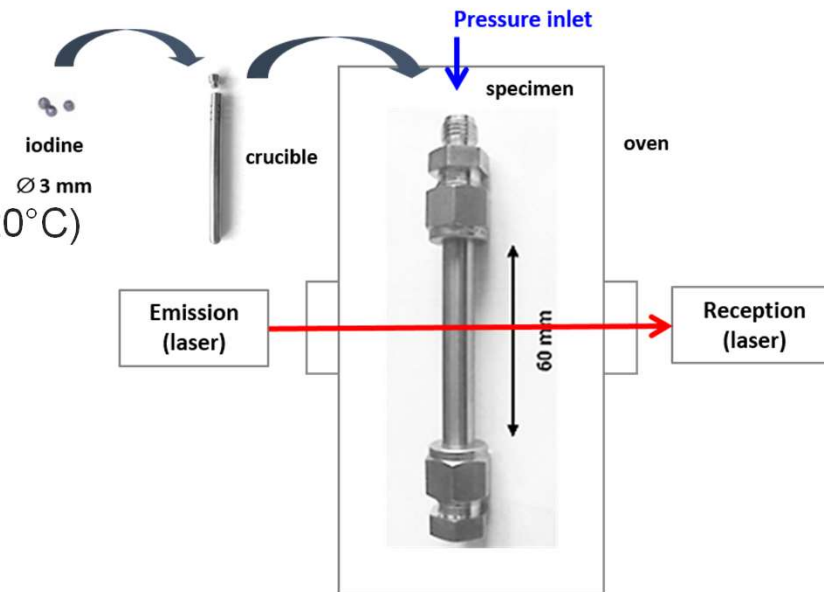
- Inner Pressure P experimentally controlled
- At PWR's temperature of the inner wall of the cladding (320°C-420°C)
- p_{I_2} is very big in these tests and p_{O_2} is not controlled !!!
- Difficult to go lower in iodine content

$$p_{I_2} \approx 3 \text{ bars} = 3.10^5 \text{ Pa}$$

$$\frac{n_{I_2}}{n_{Argon}} \approx 1 \%$$

p_{O_2} evaluated to 3000 Pa (Jezequel, 2017)

$$\frac{n_{O_2}}{n_{Argon}} \approx 0,01 \%$$

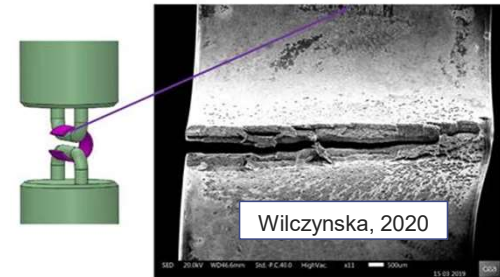
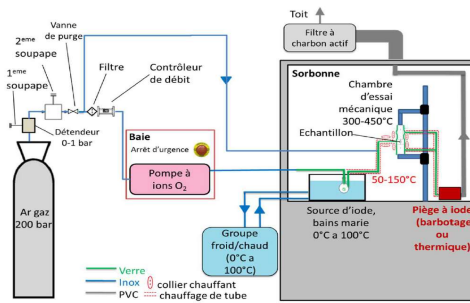


$$\sigma_{\theta\theta} = P \cdot \frac{(R_{out} + R_{in})}{2 \cdot e}$$

$$P \approx 300 \text{ bars}$$

$$\sigma_{\theta\theta} \sim 300 \text{ MPa}$$

- To better simulate the PWR physico-chemical environment in laboratory



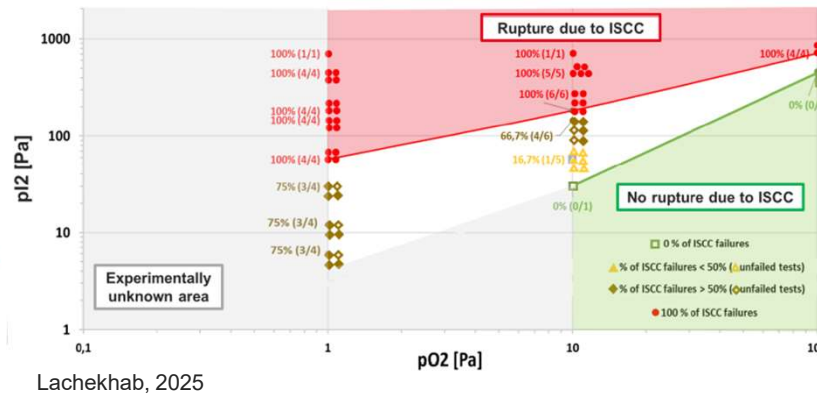
Wilczynska, 2020

- Device in quartz to avoid chemical interactions with iodine
- Low pO_2 controlled (above 1-10 Pa)
 - Possibility to go lower under study
- Low pI_2 controlled (above 1-10 Pa)
 - Possible to go lower
- Temperature controlled

$$pI_2 \approx 10 \text{ Pa}$$
$$pO_2 \approx 10 \text{ Pa}$$

$$\frac{n_{I_2}}{n_{Argon}} \approx 0,01 \%$$

$$\frac{n_{O_2}}{n_{Argon}} \approx 0,01 \%$$



Lachekhab, 2025



PWR simulations (ALCYONE)

- The polycrystalline scale is better to study PCI/I-SCC failures in PWR ?
 - No experimental device to study that scale, under vapor iodine, today → Simulations instead ?

