Mathematical modelling for all-solid-state battery: (SE|SSE)-Interface

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Mathematical modelling for the next-generation All-solid-state batteries: Nucleation (SE|SSE)$^{(*)}$-interface

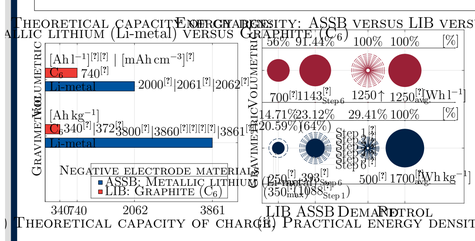
Rechargeable Lithium-ion battery (LIB) is at the heart of every electric vehicle (EV), portable electronic device, and energy storage system [1]. Nowadays, LIBs enable human life more efficient and help to solve global environment issues thanks to EVs' zero emission. However, conventional LIB (c-LIB) is sensible to temperature and pressure, hence, flammable and explosive, which is undesirable. This bottleneck is mainly due to liquid-based electrolyte found in c-LIBs.

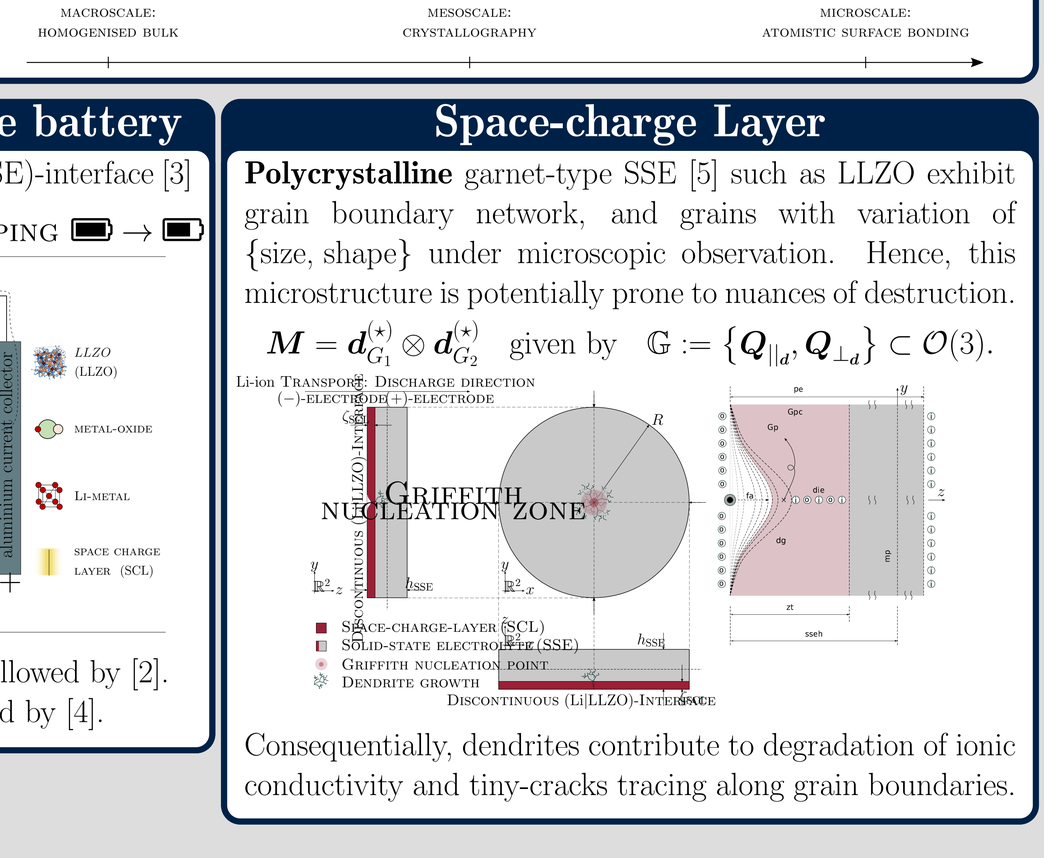
All-solid-state battery (ASSB) is one of promising candidates to overcome bottlenecks of c-LIBs. Thanks to solid-state electrolyte (SSE), ASSB is highly stable towards temperature and pressure. Nevertheless, Li-metal dendrite triggered at (SE|SSE)-interface [5] is the main drawback of ASSB since these dendritic threads extrapolate into SSE grain boundary network, causing crevice, degradation of ionic conductivity, and the probability of short-circuit, which is unfavorable.

Next-generation All-solid-state battery (ng-ASSB) with a consideration of nucleation criterion defined by

$$a_{\text{Griffith}} := a^* = \arg\min_{a\in\mathbb{R}} \iiint_\Omega f(a,\boldsymbol{u},\theta;\lambda,\mu,\boldsymbol{d}^{(\star)}\otimes\boldsymbol{d}^{(\star)})\,d\Omega - \iint_\Gamma f(a;\gamma)\,d\Gamma\Big|_{\bar{\boldsymbol{u}}}$$

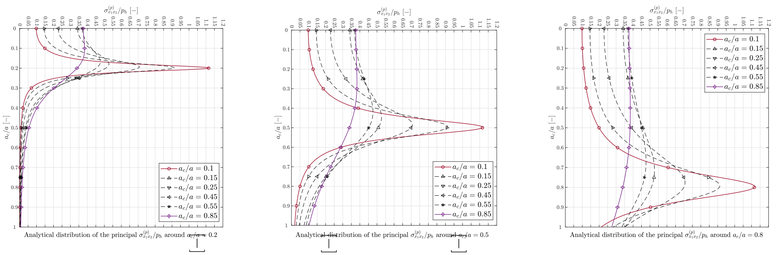
where $\boldsymbol{u}$ displacement field, θ temperature field, a crevice length, λ,μ Lamé constants, $\boldsymbol{d}^{(\star)}\otimes\boldsymbol{d}^{(\star)}$ embedded misorientation structural tensor, and γ cracking-surface energy density, can help to improve ASSB performance.



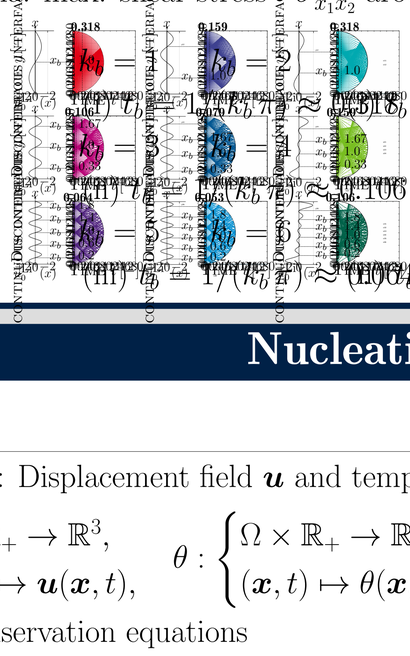


(SE|SSE)-Interface Analysis

Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge layer (SCL) [2] found in ASSBs critically exhibits mechanical and electrochemical instability [3]. This evidence points directly to the fact that the soft metallic li anode is erroneously prone to triggering dendrites, under cycles of electric charge & discharge [5].

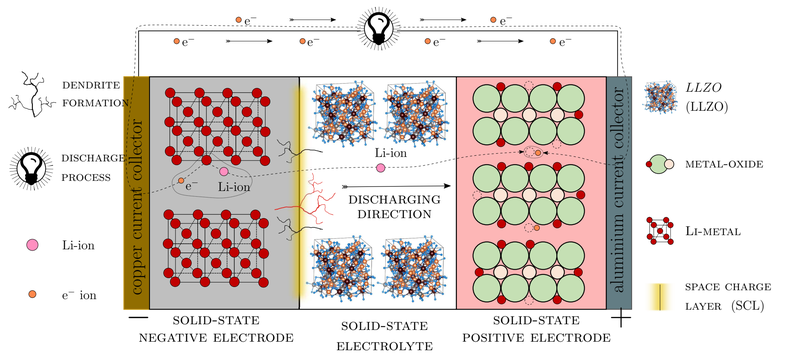


Distribution: anal. max. shear stress ${}^{\mathcal{W}_\mathcal{A}}\sigma^{\Pi}_{x_1x_2}$ around crack tip a_c.



Next-generation All-solid-state battery

Nucleation criterion governs the instable (SE|SSE)-interface [3]



✓ **Thermodynamic consistency** is satisfied, followed by [2].
✓ **Closure** $\bar{\Omega}$ is fulfilled by 15 moments, followed by [4].

Space-charge Layer

Polycrystalline garnet-type SSE [5] such as LLZO exhibit grain boundary network, and grains with variation of {size, shape} under microscopic observation. Hence, this microstructure is potentially prone to nuances of destruction.

$\boldsymbol{M} = \boldsymbol{d}^{(\star)}_{G_1}\otimes\boldsymbol{d}^{(\star)}_{G_2}$ given by $\mathbb{G} := \{\boldsymbol{Q}_{\|_d}, \boldsymbol{Q}_{\perp_d}\} \subset \mathcal{O}(3)$.

Consequentially, dendrites contribute to degradation of ionic conductivity and tiny-cracks tracing along grain boundaries.

Nucleation interface: Taking place at the critically dendritic (SE|SSE)-interface

Coupled fields: Displacement field $\boldsymbol{u}$ and temperature field θ; structural tensor $\boldsymbol{M}$

$$\boldsymbol{u}: \begin{cases}\Omega\times\mathbb{R}_+\to\mathbb{R}^3,\\ (\boldsymbol{x},t)\mapsto\boldsymbol{u}(\boldsymbol{x},t),\end{cases} \quad \theta: \begin{cases}\Omega\times\mathbb{R}_+\to\mathbb{R},\\ (\boldsymbol{x},t)\mapsto\theta(\boldsymbol{x},t),\end{cases} \quad \boldsymbol{M}^{\{RR,RE\}}_{i=1,\dots,N}: \begin{cases}\boldsymbol{d}^R_{\text{Grain i}}\otimes\boldsymbol{d}^R_{\text{Grain i}}\\ \boldsymbol{d}^R_{\text{Grain i}}\otimes\boldsymbol{d}^E\end{cases}$$

Governing conservation equations

$$\frac{d}{dt}\int_\Omega(\cdot)\,d\Omega = \int_\Omega(\cdot)^{\text{action}}\,d\Omega + \int_{\partial\Omega}(\cdot)^{\text{action}}\,d\partial\Omega + \int_\Omega(\cdot)^{\text{production (+/-)}}\,d\Omega$$

used to describe balance of mass, conservation of linear momentum, conservation of angular momentum, and conservation of energy with $\rho(\boldsymbol{x},t)$ is mass density per unit volume (puv); $\boldsymbol{b}(\boldsymbol{x},t)$ body force puv; $\boldsymbol{v}(\boldsymbol{x},t)$ velocity; $e(\boldsymbol{x},t)$ internal energy puv; $\boldsymbol{q}(\boldsymbol{x},t)$ heat flux; $r(\boldsymbol{x},t)$ heat source puv; $\boldsymbol{\sigma}$ Cauchy stress and $\boldsymbol{\varepsilon}$ infinitesimal strain. Then, the governing partial differential equation (PDE) for the thermodynamically consistent coupled problem with deformation field $\boldsymbol{u}$ as its variable takes the form

$$\partial_t\boldsymbol{u} + \nabla\cdot\left(\overset{4}{\mathbb{C}}{}^{f_{\text{alocation}}(\lambda,\mu,\boldsymbol{d}^R_{G_i,i=1,\dots,N},\boldsymbol{d}^E;\boldsymbol{x})} : \nabla\boldsymbol{u}^{(s)}\right) + \rho\boldsymbol{b} = -\rho\nabla V_e,$$

where $V_e:\mathbb{R}^3\to\mathbb{R}$ is the electric potential applied globally on ASSB. Due to nature setting of ASSB taking the form (SE|SSE|SE) the electric potential becomes uniform.

Strain energy is based on the deformation of SSE due to dendrite formation at (SE|SSE)-interface

$$\iiint_\Omega f(a,\boldsymbol{u};\lambda,\mu,\boldsymbol{d}\otimes\boldsymbol{d})\,d\Omega$$

Surface energy is analysized based on the open crevice cracking at (SE|SSE)-interface affected by prescribed pressure

$$\iint_\Gamma f(a;\gamma)\,d\Gamma$$

Therefore, the governing problem of dendritic nucleation at (SE|SSE) takes the form

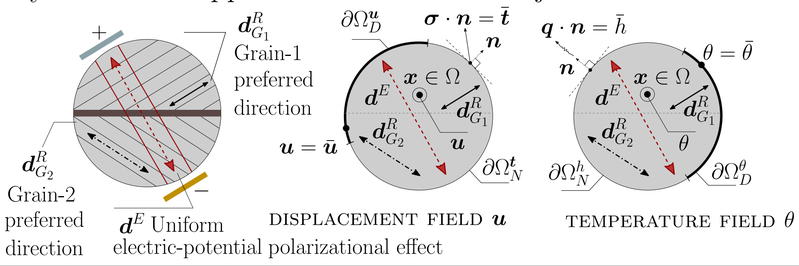
$$\partial_t\boldsymbol{u} + \nabla\cdot\left(\overset{4}{\mathbb{C}}{}^{f_{\text{alocation}}(\lambda,\mu,\boldsymbol{d}^R_{G_i,i=1,\dots,N},\boldsymbol{d}^E;\boldsymbol{x})} : \nabla\boldsymbol{u}^{(s)}\right) + \rho\boldsymbol{b} = -\rho\nabla V_e, \tag{1}$$

Analysis: *Airy-Westergaard* fcn. used for stress analysis: (i) max. shear stress and (ii) principal stresses

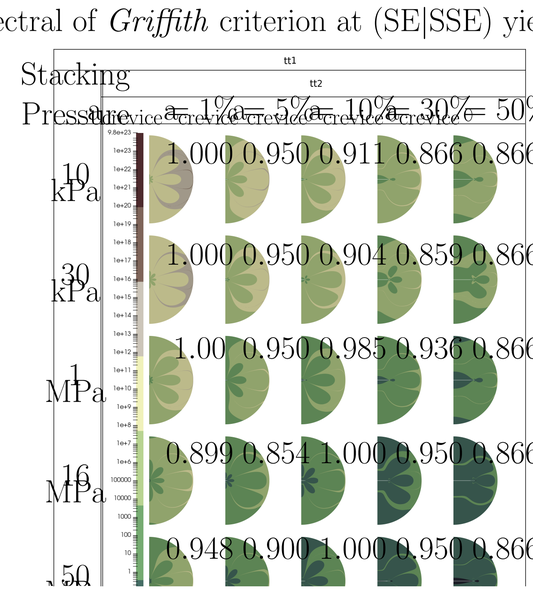
$${}^{\mathcal{W}}\mathcal{A}: \begin{cases}\mathbb{C}\to\mathbb{C},\\ z\mapsto {}^{\mathcal{W}}\mathcal{A}(z) := \Re\left(\iint_\Gamma \mathcal{K}^{(\star)}\,dz\right) + x_2\,\Im\left(\oint_\Gamma \mathcal{K}^{(\star)}\,dz\right),\end{cases} \quad \mathcal{K}^{(\star)}: \begin{cases}\mathbb{C}\to\mathbb{C},\\ z\mapsto \mathcal{K}^{(\star)} := -p_h + p_h/\sqrt{1-a^2/z^2},\end{cases}$$

where a the crevice length, p_h pressure at the opening crevice on dendritic interface, and $\forall\{p_h,a\}\in\mathbb{R}_+$.

Boundary conditions applied on a solid-state object



Numerical spectral of *Griffith* criterion at (SE|SSE) yields (xx-yy-xy)



FEM: Strain energy density

Partial differential equation (PDE)

$$\nabla\cdot\left(\overset{4}{\mathbb{C}}{}^{f^{\mathbb{D}(\Omega)}_{(\lambda,\mu)}}\nabla^{(s)}\boldsymbol{u}\right) + \rho\,\boldsymbol{b} = \boldsymbol{0}$$

Displacement vector field solution

$$u_i \leftarrow \boldsymbol{u} = \boldsymbol{K}^{-1}\boldsymbol{f}$$

Strain tensor

$$\varepsilon_{ij} = \tfrac{1}{2}\left(\partial_{x_j}u_i + \partial_{x_i}u_j\right)$$

Stress tensor

$$\sigma_{ij} = \sum_{k,l}\overset{4}{\mathbb{C}}{}^{f^{\mathbb{D}(\Omega)}_{(\lambda,\mu)}}_{ijkl}\,\varepsilon_{kl}$$

Strain energy density

$$\mathcal{E}_{\text{strain}} := \tfrac{1}{2}\sum_{i,j}\sigma_{ij}\,\varepsilon_{ij}$$

Strain solution takes the following form $\frac{1}{2}\sum_{\alpha=1}^{\mathcal{N}^{\Omega^e}_{\text{node}}}\left(\sum_{L=1}^{\mathcal{N}^{\Omega^{\text{node}}}_{\text{dof}}}N^\alpha_{,\xi_L}\xi_{L,x_k}u^\alpha_k + \sum_{K=1}^{\mathcal{N}^{\Omega^{\text{node}}}_{\text{dof}}}N^\alpha_{,\xi_K}\xi_{K,x_l}u^\alpha_l\right)$ where displacement field $\boldsymbol{u}$ is known, numerically computed from the PDE above.

Griffith-based critical stress