# Next-generation all-solid-state battery (#ASSB)

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# Mathematical modelling for the next-generation All-solid-state batteries: Nucleation (SE|SSE)<sup>(\*)</sup>-interface

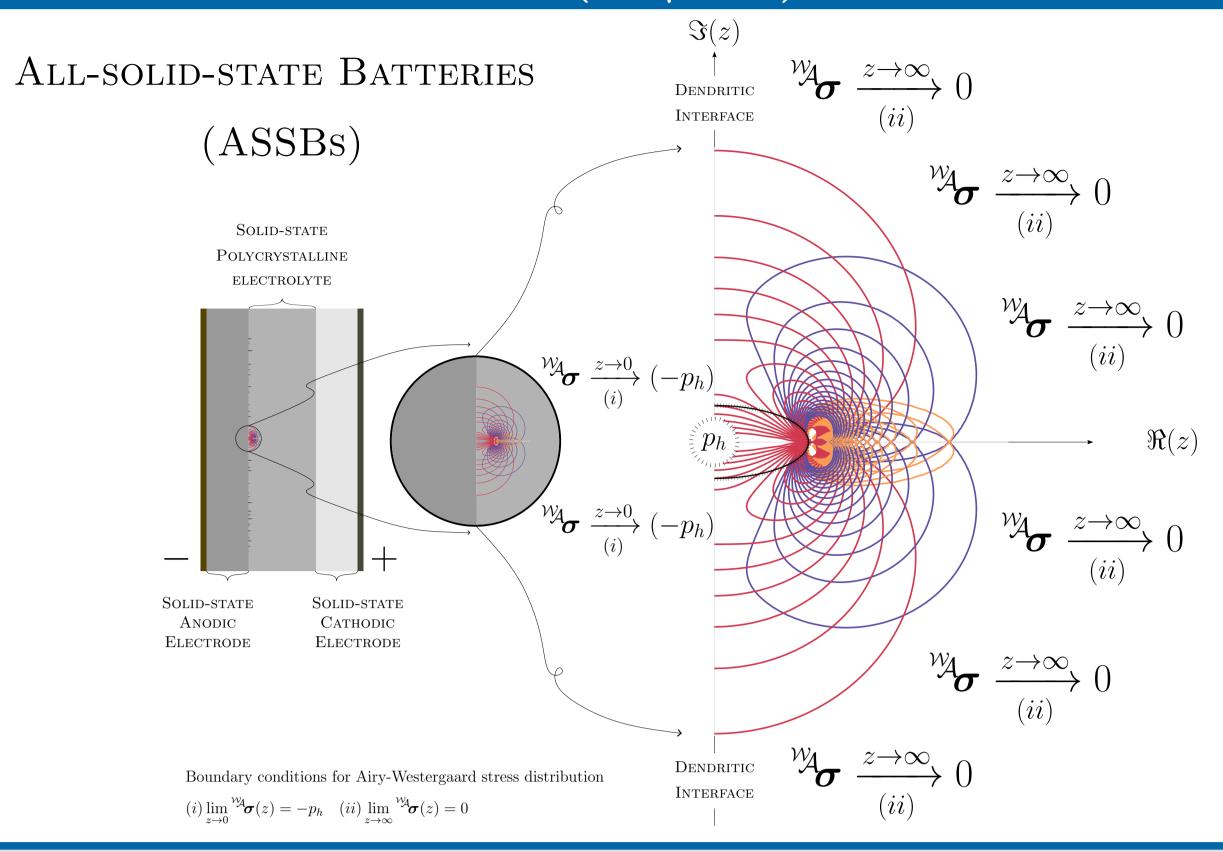
Rechargeable Lithium-ion battery (LIB) is at the heart of every electric vehicle (EV), portable electronic device, and energy storage system [1]. Nowadays, LIBs enable human life more efficient and help to solve global environment issues thanks to EVs' zero emission. However, conventional LIB (c-LIB) is sensible to temperature and pressure, hence, flammable and explosive, which is undesirable. This bottleneck is mainly due to liquid-based electrolyte found in c-LIBs.

**All-solid-state battery** (ASSB) is one of promising candidates to overcome bottlenecks of c-LIBs. Thanks to solid-state electrolyte (SSE), ASSB is highly stable towards temperature and pressure. Nevertheless, Limetal dendrite triggered at (SE|SSE)-interface [5] is the main drawback of ASSB since these dendritic threads extrapolate into SSE grain boundary network, causing crevice, degradation of ionic conductivity, and the probability of short-circuit, which is unfavorable.

**Next-generation All-solid-state battery** (ng-ASSB) with a consideration of nucleation criterion defined by

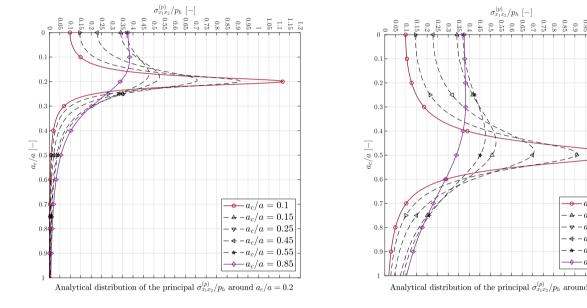
$$a_{ ext{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \left. \iint_{\Omega} f(a, \boldsymbol{u}, \theta; \lambda, \mu, \boldsymbol{d}^R \otimes \boldsymbol{d}^R) \, d\Omega - \iint_{\Gamma} f(a; \gamma) \, d\Gamma \right|_{\boldsymbol{u}^{(s)}}$$

where  $\boldsymbol{u}$  displacement field,  $\theta$  temperature field, a crevice length,  $\lambda, \mu$  Lamé constants,  $\boldsymbol{d}^R \otimes \boldsymbol{d}^R$  embedded misorientation structural tensor, and  $\gamma$  cracking-surface energy density, can help to improve ASSB performance.



## Interface Analysis

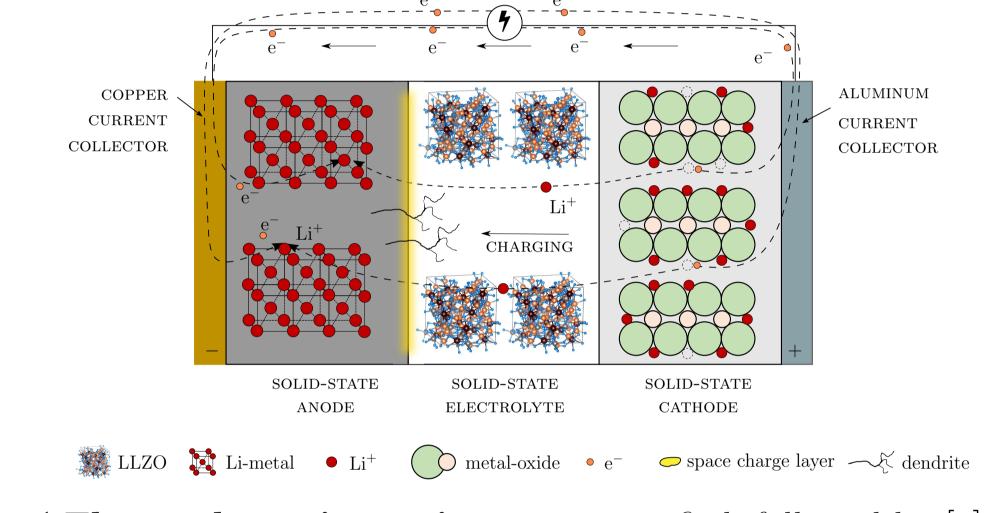
Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge layer (SCL) [2] found in ASSBs critically exhibits mechanical and electrochemical instability [3]. This evidence points directly to the fact that the soft metallic li anode is erroneously prone to triggering dendrites, under cycles of electric charge & discharge [5].



<u>Distribution</u>: ana. max. shear stress  ${}^{\mathcal{W}}\!\!\sigma_{x_1x_2}^{\Pi}$  around crack tip  $a_c$ .

## Next-generation All-solid-state battery

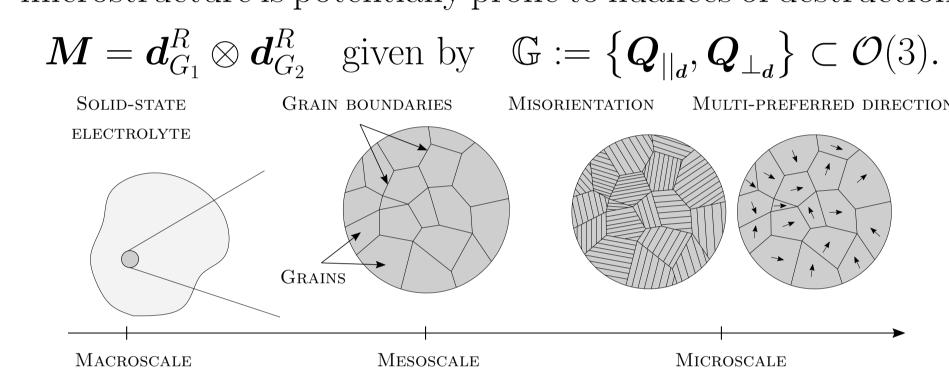
**Nucleation** criterion governs the instable (SE|SSE)-interface [3]



**Thermodynamic consistency** is satisfied, followed by [2]. ✓ Closure  $\bar{\Omega}$  is fulfilled by 15 moments, followed by [4].

# Embedded structural-tensor in SSE

Polycrystalline garnet-type SSE [5] such as LLZO exhibit grain boundary network, and grains with variation of {size, shape} under microscopic observation. Hence, this microstructure is potentially prone to nuances of destruction.



Consequentially, dendrites contribute to degradation of ionic conductivity and tiny-cracks tracing along grain boundaries.

# Nucleation interface: Taking place at the critical dendritic interface

Coupled fields: Displacement field  $\boldsymbol{u}$  and temperature field  $\boldsymbol{\theta}$ ; structural tensor  $\boldsymbol{M}$ 

$$\boldsymbol{u}: \begin{cases} \Omega \times \mathbb{R}_{+} \to \mathbb{R}^{3}, \\ (\boldsymbol{x},t) \mapsto \boldsymbol{u}(\boldsymbol{x},t), \end{cases} \quad \theta: \begin{cases} \Omega \times \mathbb{R}_{+} \to \mathbb{R}, \\ (\boldsymbol{x},t) \mapsto \theta(\boldsymbol{x},t), \end{cases} \quad \boldsymbol{M}_{i=1,\dots,N}^{\{RR,RE\}}: \begin{cases} \boldsymbol{d}_{\text{Grain i}}^{R} \otimes \boldsymbol{d}_{\text{Grain i}}^{R} \\ \boldsymbol{d}_{\text{Grain i}}^{R} \otimes \boldsymbol{d}^{E} \end{cases}$$

Governing conservation equations

$$\frac{d}{dt} \int_{\Omega} (\cdot) \ d\Omega = \int_{\Omega} (\cdot)^{\text{action}} \ d\Omega + \int_{\partial \Omega} (\cdot)^{\text{action}} \ d\partial\Omega + \int_{\Omega} (\cdot)^{\text{production (+/-)}} \ d\Omega$$

used to describe balance of mass, conservation of linear momentum, conservation of angular momentum, and conservation of energy with  $\rho(\boldsymbol{x},t)$  is mass density per unit volume (puv);  $\boldsymbol{b}(\boldsymbol{x},t)$  body force puv;  $\boldsymbol{v}(\boldsymbol{x},t)$  velocity;  $e(\boldsymbol{x},t)$  internal energy puv; q(x,t) heat flux; r(x,t) heat source puv;  $\sigma$  Cauchy stress and  $\varepsilon$  infinitesimal strain. Then, the governing partial differential equation (PDE) of deformation takes the form

$$\partial_t oldsymbol{u} + 
abla \cdot \left( \overset{4}{\mathbb{C}}^{f_{ ext{alocation}}(\lambda, \mu, oldsymbol{d}^R_{G_i, i=1,...,N}, oldsymbol{d}^E; oldsymbol{x})} : 
abla oldsymbol{u}^{(s)} 
ight) + 
ho oldsymbol{b} = -
ho 
abla V_e,$$

where  $V_e: \mathbb{R}^3 \to \mathbb{R}$  is the electric potential applied globally on ASSB. Due to nature setting of ASSB taking the form (SE|SSE|SE) the electric potential is uniform.

Strain **energy** is based on the deformation of SSE due to dendrite formation at (SE|SSE)-interface

Surface energy is analysized based on the open crevice cracking at (SE|SSE)interface affected by prescribed pressure

$$\iiint_{\Omega} f(a, \boldsymbol{u}; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) d\Omega$$

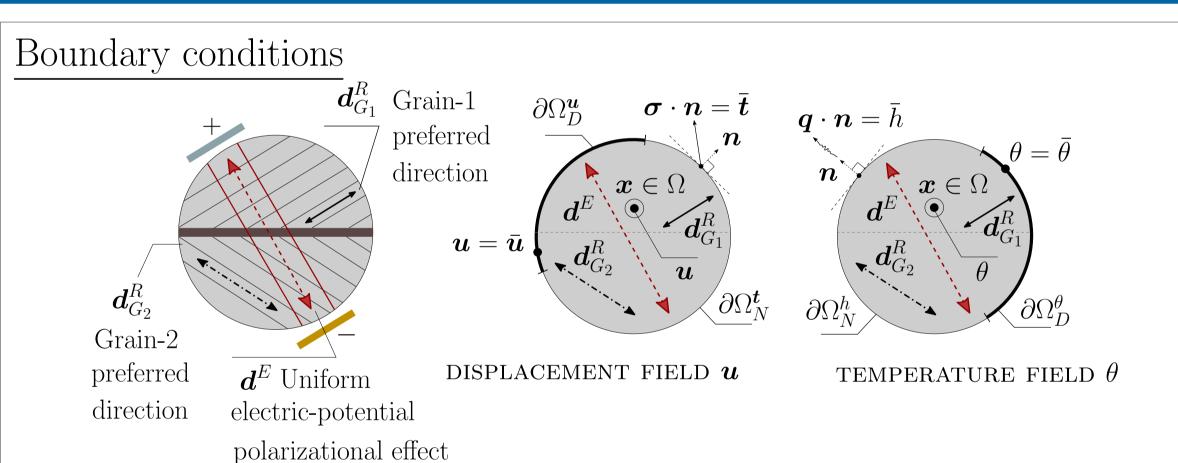
$$\iint_{\Gamma} f(a; \gamma) d\Gamma$$

Therefore, the governing problem of dendritic nucleation at (SE|SSE) takes the form

$$\partial_{t}\boldsymbol{u} + \nabla \cdot \left( \mathbb{C}^{f_{\text{alocation}}(\lambda, \mu, \boldsymbol{d}_{G_{i}, i=1,\dots,N}^{R}, \boldsymbol{d}^{E}; \boldsymbol{x})} : \nabla \boldsymbol{u}^{(s)} \right) + \rho \boldsymbol{b} = -\rho \nabla V_{e}, \tag{1}$$

s.t. 
$$a_{\text{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \iiint_{\Omega} f(a, \boldsymbol{u}, \theta; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) d\Omega - \iint_{\Gamma} f(a; \gamma) d\Gamma \Big|_{\bar{\boldsymbol{u}}}$$
 (2)

where deformation  $\bar{\boldsymbol{u}}$  is (i) based on (1), and then (ii) for Griffith-analysis in (2).



Numerical spectral of Griffith criterion in x-direction at (SE|SSE) yields (a) Case 1 (b) Case 2 (c) Case 3 (d) Case 4 (e) Case 5  $a_{\text{crevice}} = 0.5$  $a_{\text{crevice}} = 0.3$  $a_{\text{crevice}} = 0.1$  $a_{\text{crevice}} = 0.01$  $a_{\text{crevice}} = 0.05$ 

where a sample of 5 cases with various prescribed crevice length is studied.

FEM: Strain energy density Partial differential equation (PDE)  $abla \cdot \left( \overset{4}{\mathbb{C}} f^{\mathbb{D}(\Omega)}_{(\lambda,\mu)} \, 
abla^{(s)} oldsymbol{u} 
ight) + 
ho \, oldsymbol{b} = oldsymbol{0}$ Displacement vector field solution  $oldsymbol{u_i} \leftarrow oldsymbol{u} = oldsymbol{K}^{-1} oldsymbol{f}$ Strain tensor  $arepsilon_{ij} = rac{1}{2} \left( \partial_{x_j} u_i + \partial_{x_i} u_j 
ight)$ Stress tensor  $\sigma_{ij} = \sum_{k,l} \overset{4}{\mathbb{C}}_{(\lambda,\mu)}^{f_{(\lambda,\mu)}^{\mathbb{D}(\Omega)}} \, arepsilon_{kl}$ Strain energy density  $\mathcal{E}_{ ext{strain}} := rac{1}{2} \sum_{i,j} \sigma_{ij} \, arepsilon_{ij}$ Strain solution takes the following form  $\frac{1}{2} \sum_{\alpha=1}^{\mathcal{N}_{\text{node}}^{\Omega^e}} \left( \sum_{L=1}^{\mathcal{N}_{\text{dof}}^{\Omega^{\text{node}}}} N_{,\xi_L}^{\alpha} \xi_{L,x_k} \boldsymbol{u}_k^{\alpha} + \sum_{K=1}^{\mathcal{N}_{\text{dof}}^{\Omega^{\text{node}}}} N_{,\xi_K}^{\alpha} \xi_{K,x_l} \boldsymbol{u}_l^{\alpha} \right)$ 

Analysis: Airy-Westergaard function used for stress analysis: (i) max. shear stress and (ii) principal stresses

$$\mathcal{W}_{\mathcal{A}}: \begin{cases} \mathbb{C} \to \mathbb{C}, \\ z \mapsto \mathcal{W}_{\mathcal{A}}(z) := \Re(\iint_{\Gamma} \mathcal{K}^{(\star)} dz) + x_2 \Im(\oint_{\Gamma} \mathcal{K}^{(\star)} dz), \end{cases} \mathcal{K}^{(\star)}: \begin{cases} \mathbb{C} \to \mathbb{C}, \\ z \mapsto \mathcal{K}^{(\star)} := -p_h + p_h/\sqrt{1 - a^2/z^2}, \end{cases}$$

where a the crevice length,  $p_h$  pressure at the opening crevice on dendritic interface, and  $\forall \{p_h, a\} \in \mathbb{R}_+$ .

<u>Numerics</u>  $\rightarrow$  <u>FEM</u>: element matrix  $\mathbf{K}^e$  approx. by *Gauss quadrature*; indices imply 4 + 2 = 6 for-loop:  $K_{ik}^{e^{\alpha\beta}} = \int_{\Omega^{\epsilon}} \left( \mathcal{L}_{1}^{\alpha} \, \mathbb{C}_{i1k1}^{fGL}(\boldsymbol{x}) \, \mathcal{R}_{1}^{\beta} + \mathcal{L}_{1}^{\alpha} \, \mathbb{C}_{i1k2}^{fGL}(\boldsymbol{x}) \, \mathcal{R}_{2}^{\beta} + \mathcal{L}_{2}^{\alpha} \, \mathbb{C}_{i2k1}^{fGL}(\boldsymbol{x}) \, \mathcal{R}_{1}^{\beta} + \mathcal{L}_{2}^{\alpha} \, \mathbb{C}_{i2k2}^{fGL}(\boldsymbol{x}) \, \mathcal{R}_{2}^{\beta} \right) \det(\boldsymbol{J}) \, d\Omega^{\xi}$ 

where  $\mathcal{L}_{i}^{\alpha}$  and  $\mathcal{R}_{l}^{\beta}$  are gradients of basis functions at node  $\alpha^{th}$  and  $\beta^{th}$ , respectively.

#### Contact

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