Next-generation all-solid-state battery (#ASSB)

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Mathematical modelling for the next-generation All-solid-state batteries: Nucleation (SE|SSE)^(*)-interface

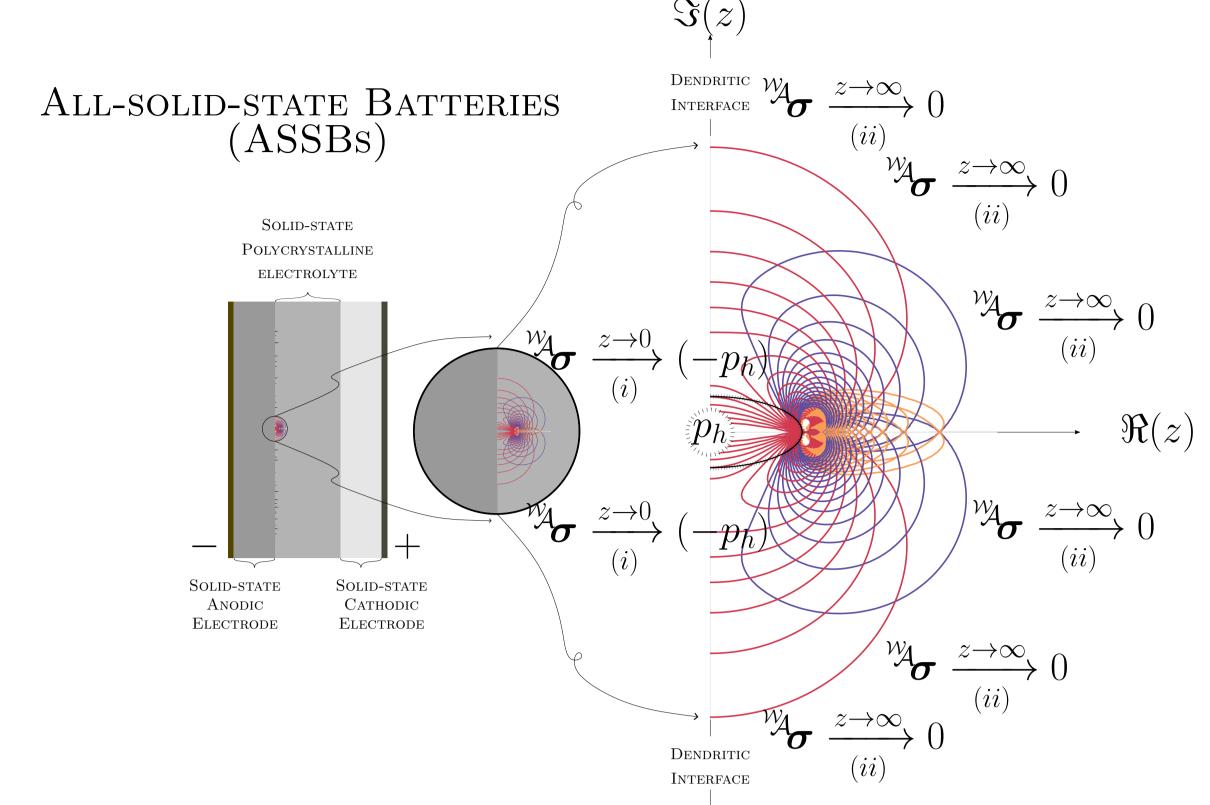
Rechargeable Lithium-ion battery (LIB) is at the heart of every electric vehicle (EV), portable electronic device, and energy storage system [1]. Nowadays, LIBs enable human life more efficient and help to solve global environment issues thanks to EVs' zero emission. However, conventional LIB (c-LIB) is sensible to temperature and pressure, hence, flammable and explosive, which is undesirable. This bottleneck is mainly due to liquid-based electrolyte found in c-LIBs.

All-solid-state battery (ASSB) is one of promising candidates to overcome bottlenecks of c-LIBs. Thanks to solid-state electrolyte (SSE), ASSB is highly stable towards temperature and pressure. Nevertheless, Limetal dendrite triggered at (SE|SSE)-interface is the main drawback of ASSB since these dendritic threads extrapolate into SSE grain boundary network, causing crevice, degradation of ionic conductivity, and the probability of short-circuit, which is unfavorable.

Next-generation All-solid-state battery (ng-ASSB) with a consideration of nucleation criterion defined by

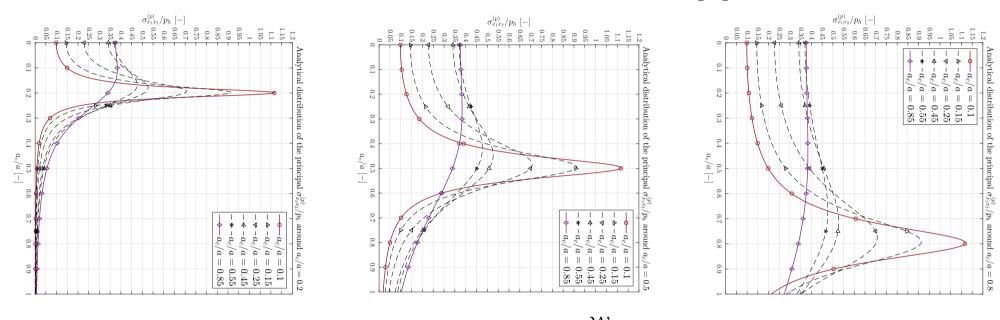
$$a_{ ext{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \left. \iint_{\Omega} f(a, \boldsymbol{u}; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) \, d\Omega - \left. \iint_{\Gamma} f(a; \gamma) \, d\Gamma \right|_{\boldsymbol{u}^{(i)}}$$

where, can help to improve ASSB performance. can help to improve ASSB performance. can help to improve ASSB performance.



Interface

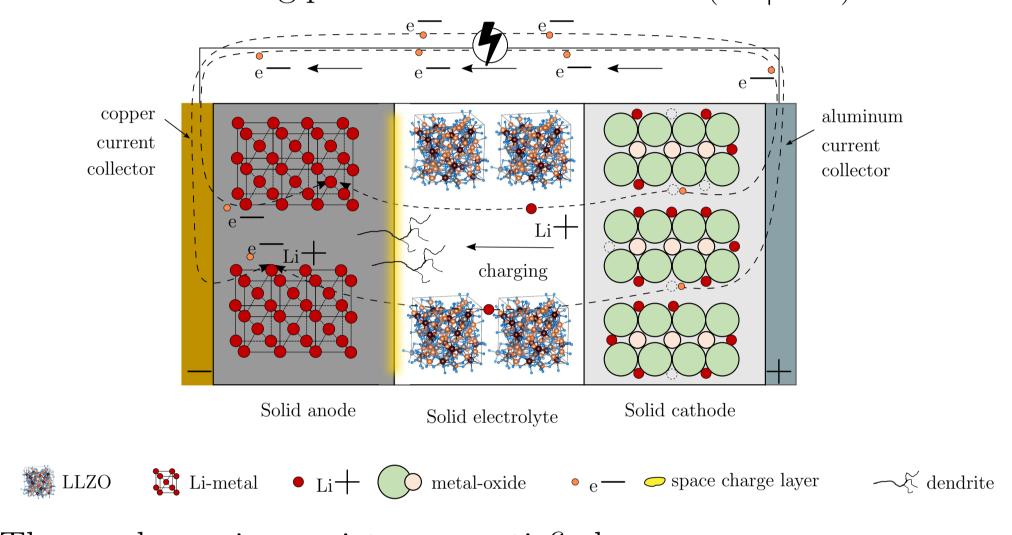
Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge layer (SCL) [2] found in ASSBs critically exhibits mechanical and electrochemical instability [3]. This evidence points directly to the fact that the soft metallic li anode is erroneously prone to triggering dendrites, under cycles of electric charge & discharge [4].



<u>Distribution</u>: ana. max. shear stress ${}^{\mathcal{W}}\!\!\sigma_{x_1x_2}^{\Pi}$ around crack tip a_c .

Next-generation All-solid-state battery

Nucleation taking place at critical dendritic (SE|SSE)-interface

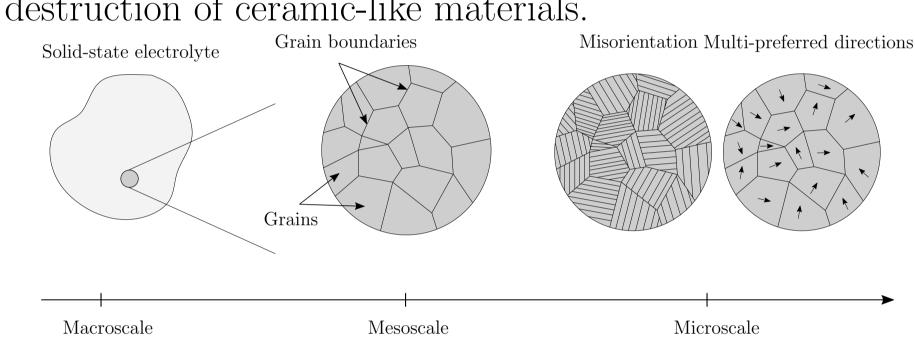


Thermodynamic consistency: satisfied. Closure problem: fulfille by 15 moments.

direction

Embedded structural-tensor SSE

Polycrystalline garnet-typed SSE such as LLZO exhibit a network of grain boundaries, and grains with various sizes and shapes under microscopic observation. Therefore, this type of microstructure is potentially prone to nuance destruction of ceramic-like materials.



Consequentially, dendrites contribute to degradation of ionic conductivity and cracks via tracing along grain boundaries.

Nucleation interface: Taking place at the critical dendritic interface

Coupled fields: Displacement vector field and temperature scalar field

$$\boldsymbol{u}: \begin{cases} \Omega \times \mathbb{R}_{+} \to \mathbb{R}^{3}, \\ (\boldsymbol{x},t) \mapsto \boldsymbol{u}(\boldsymbol{x},t), \end{cases} \quad \theta: \begin{cases} \Omega \times \mathbb{R}_{+} \to \mathbb{R}, \\ (\boldsymbol{x},t) \mapsto \theta(\boldsymbol{x},t), \end{cases} \quad \theta: \begin{cases} \Omega \times \mathbb{R}_{+} \to \mathbb{R}, \\ (\boldsymbol{x},t) \mapsto \theta(\boldsymbol{x},t), \end{cases}$$

Governing conservation equations

$$\frac{d}{dt} \int_{\Omega} (\cdot) \ d\Omega = \int_{\Omega} (\cdot)^{\text{action}} \ d\Omega + \int_{\partial \Omega} (\cdot)^{\text{action}} \ d\partial\Omega + \int_{\Omega} (\cdot)^{\text{production/source/sink}} \ d\Omega$$

 $\rho(\boldsymbol{x},t)$ is mass density per unit volume (puv); $\boldsymbol{b}(\boldsymbol{x},t)$ body force puv; $\boldsymbol{v}(\boldsymbol{x},t)$ velocity; $e(\boldsymbol{x},t)$ internal energy puv; $\boldsymbol{q}(\boldsymbol{x},t)$ heat flux; $r(\boldsymbol{x},t)$ heat source puv; $\boldsymbol{\sigma}$ Cauchy stress and $\boldsymbol{\varepsilon}$ infinitesimal strain. Helmholtz energy functional

$$a_{\text{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \left. \iint_{\Omega} f(a, \boldsymbol{u}; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) \, d\Omega - \left. \iint_{\Gamma} f(a; \gamma) \, d\Gamma \right|_{\boldsymbol{u}^{(s)}}$$

Governing PDE

$$a_{\text{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \left. \iint_{\Omega} f(a, \boldsymbol{u}; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) \, d\Omega - \left. \iint_{\Gamma} f(a; \gamma) \, d\Gamma \right|_{\boldsymbol{u}^{(s)}}$$

Strain energy: Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge

 $\iiint_{\Omega} f(a,oldsymbol{u};\lambda,\mu,oldsymbol{d}\otimesoldsymbol{d})\,d\Omega$

Surface energy: Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place

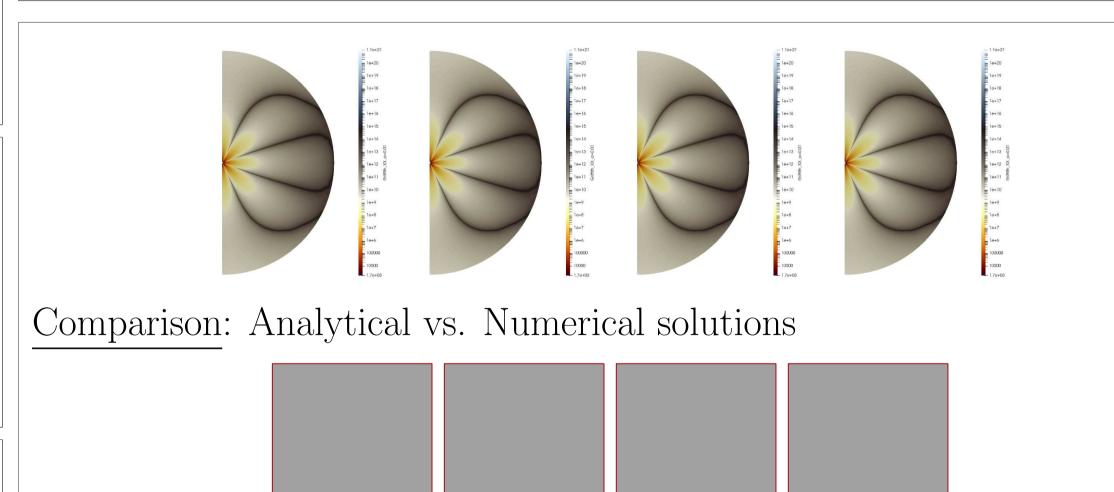
$$\iint f(a;\gamma) \, d\Gamma$$

Therefore

$$\rho \, \partial_{t^2}^2 \boldsymbol{u}^{(s)} + \nabla \cdot \left(\overset{4}{\mathbb{C}} f_{(\lambda,\mu)}^{\mathbb{D}(\Omega)} : \nabla \boldsymbol{u}^{(s)} \right) + \rho \nabla V_e = \boldsymbol{0},$$
s.t. $a_{\text{Griffith}} := a^* = \arg\min_{a \in \mathbb{R}} \left. \iint_{\Omega} f(a, \boldsymbol{u}; \lambda, \mu, \boldsymbol{d} \otimes \boldsymbol{d}) \, d\Omega - \int_{\Gamma} f(a; \gamma) \, d\Gamma \right|_{\boldsymbol{u}^{(s)}}$

Boundary condition settings $\frac{d_{G_1}^R \text{ Grain-1}}{d_{G_2}^R \text{ Grain-1}} \partial \Omega_D^u \qquad \sigma \cdot \boldsymbol{n} = \bar{\boldsymbol{t}} \qquad \boldsymbol{q} \cdot \boldsymbol{n} = \bar{\boldsymbol{h}} \qquad \boldsymbol{\theta} = \bar{\boldsymbol{\theta}}$ The preferred direction and the preferred direction are the preferred direc

Uniform DISPLACEMENT FIELD REMPERATURE FIELD electric-potential polarizational effect



Closure problem: Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge Interface between solid electrode and solid-state electrolyte (SE|SSE) taking place at space charge Interface between solid electrode and solid-state electrolyte

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References

[1] T.Vo, Modeling the swelling phenomena of li-ion batt. cells based on a numerical chemo-mech. coupled approach. MA, Robert Bosch Battery Systems GmbH, 2018.

[2] S.Braun, C. Yada and A. Latz, Thermodynamically consistent model for Space-Charge-Layer formation in a solid electrolyte. Ir. Phys. Chem., 119, 22281-22288, 2015.

[3] **C.Hüter**, S.Fu, M.Finsterbusch, E.Figgemeier, L.Wells, and R.Spatschek, *Electrode-electrolyte interface stability in solid state electrolyte system: influence of coating thickness under varying residual stresses*. AIMS Materials Science, 4(4):867-877, 2017.

[4] **S.Kim**, J.S.Kim, L.Miara, Y.Wang, S.K.Jung, S.Y.Park, Z.Song, H.King, M.Badding, J.M.Chang, V.Roev, G.Yoon, R.Kim, J.H.Kim, K.Yoon, D.Im, and K.Kang, High-energy and durable li metal batt. using garnet-type solid electrolytes with tailored li-metal compatibility. Nature Communications, 13(1):1883, 2022.









