

# Bottom-up Ab-initio Multiscale Modeling of Materials with Machine Learning

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**DPMD: potential energy, polarization, and polarizability surfaces are represented by DNNs trained on DFT data → AIMD on steroids**

$$\Phi(\mathbf{R}_1, \mathbf{R}_2, \mathbf{R}_3, \dots, \mathbf{R}_N) = \sum_i \Phi_i, \quad \Phi_i = \mathcal{F}(\{\mathbf{R}_k \in \mathcal{N}_i\})$$

$$V\mathcal{P}(\mathbf{R}_1, \mathbf{R}_2, \mathbf{R}_3, \dots, \mathbf{R}_N) = \sum_i z_i \mathbf{R}_i - 2 \sum_w \mathbf{R}_w + C = \sum_c \mathbf{d}_c + C, \quad \mathbf{R}_w = \mathcal{F}(\{\mathbf{R}_k \in \mathcal{N}_w\})$$

$$\alpha(\mathbf{R}_1, \mathbf{R}_2, \mathbf{R}_3, \dots, \mathbf{R}_N) = V \left( \frac{\partial \mathcal{P}}{\partial \mathcal{E}} \right)_{\mathbf{R}^N} = -2 \sum_w \frac{\partial \mathbf{R}_w}{\partial \mathcal{E}} = \sum_w \alpha_w (\{\mathbf{R}_k \in \mathcal{N}_w\})$$

**DPLR** includes long-range electrostatics with Ewald sum on top of DP

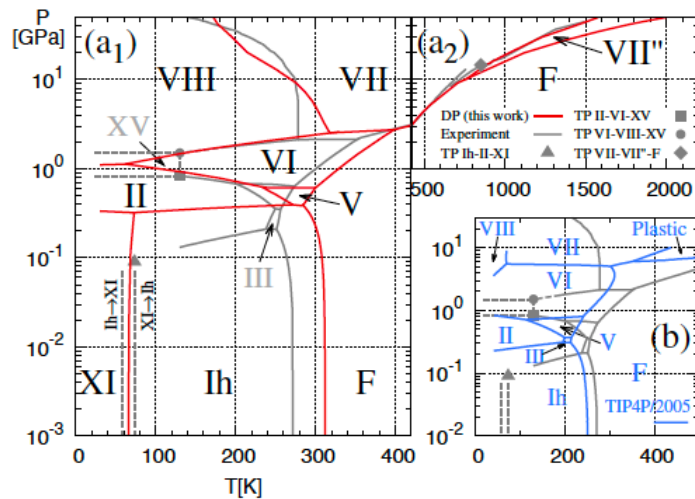
Coupling with an external electric field (static or dynamic)

$$\Phi_{\mathcal{E}(t)} = \Phi_{\mathcal{E}=0} - V\mathcal{P} \cdot \mathcal{E}(t)$$

L. Zhang *et al*, *Phys Rev Lett* **120**, 143001 (2018); L. Zhang *et al*, in *Advances in Neural Information Processing Systems* **31**, 4441 (2018);  
L. Zhang *et al*, *Phys Rev Materials* **3**, 023804 (2019); L. Zhang *et al*, *Phys Rev B* **102**, 041121(R) (2020); G. Sommers *et al*, *PCCP* **22**, 10592 (2020); L. Zhang *et al*, *J Chem Phys* **156**, 124107 (2022); J. Zheng *et al*, “DeePMD kit v2...”, *J Chem Phys* **159**, 054801 (2023)

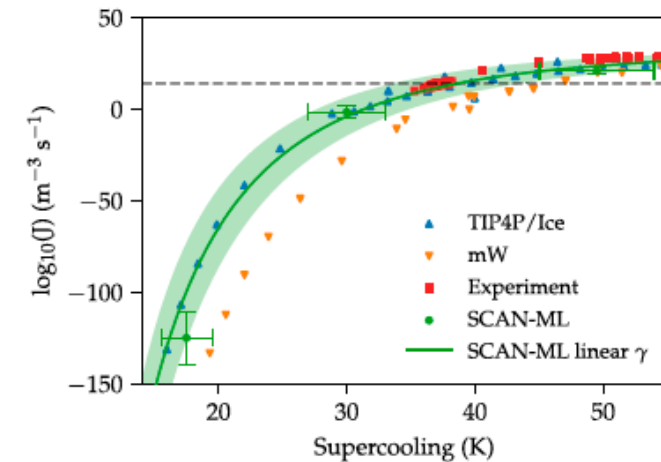
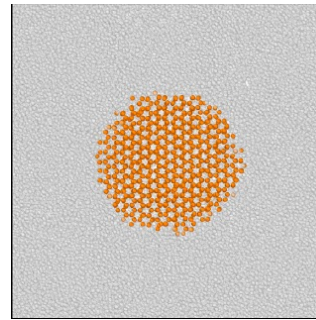
**DPMD made possible AIMD studies with unprecedented size and time scales**

## Water phase diagram



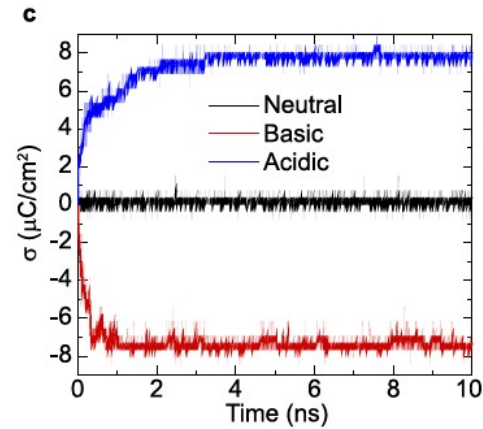
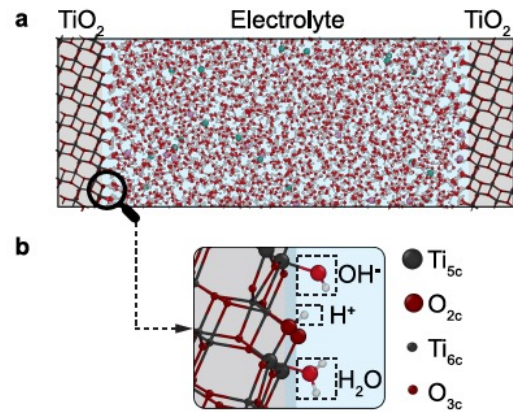
L. Zhang, H. Wang, R.C., and W.E,  
*Phys Rev Lett* **126**, 236001 (2021)

## Homogeneous nucleation of ice



P. Piaggi, J. Weis, A. Panagiotopoulos, P. Debenedetti, RC, *PNAS* (2022)

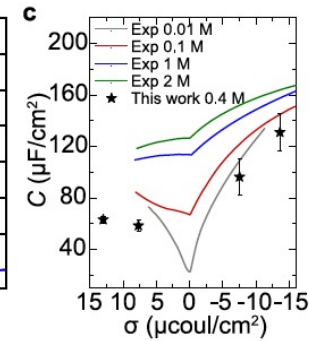
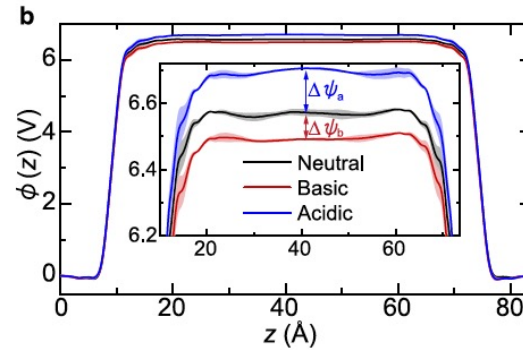
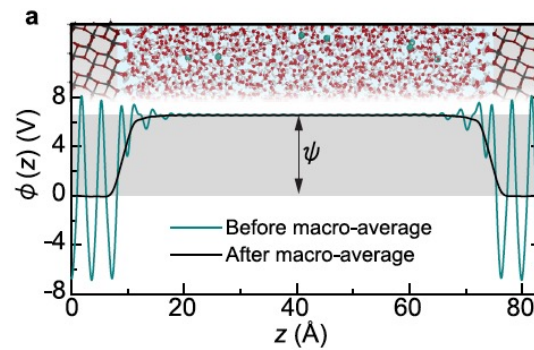
# Electrolyte solutions in contact with TiO<sub>2</sub>



0.4 M NaCl + 0.2 M HCl

0.4 M NaCl

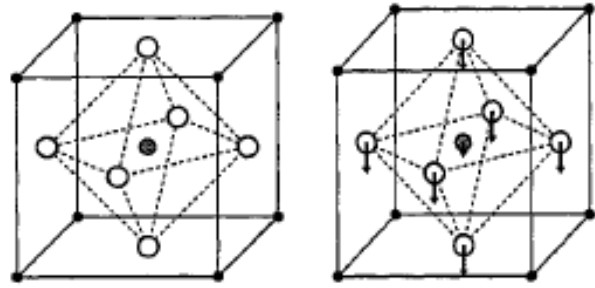
0.4 M NaCl + 0.2 M NaOH



Differential Capacitance

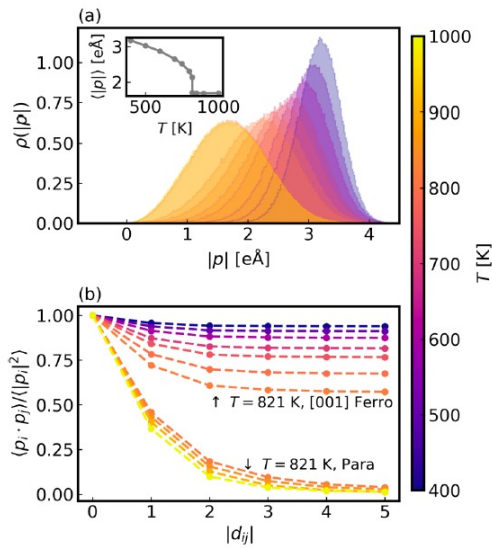
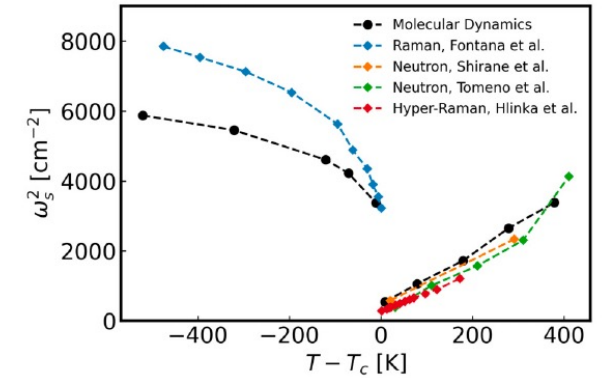
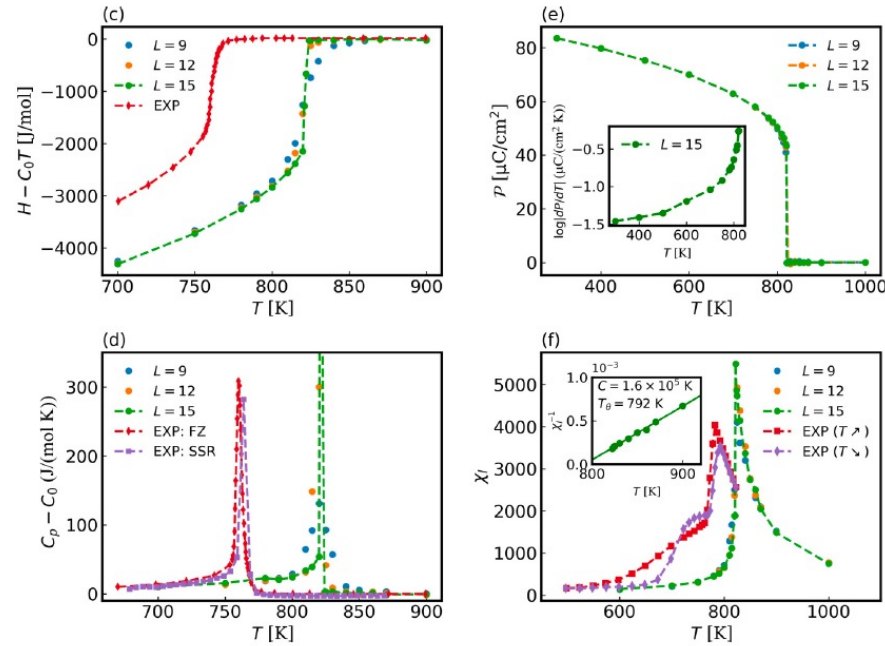
$$C = \frac{d\sigma}{d\psi} \approx \frac{\Delta\sigma}{\Delta\psi}$$

# The ferroelectric phase transition in $\text{PbTiO}_3$ (PTO)



$T > T_c$

$T < T_c$

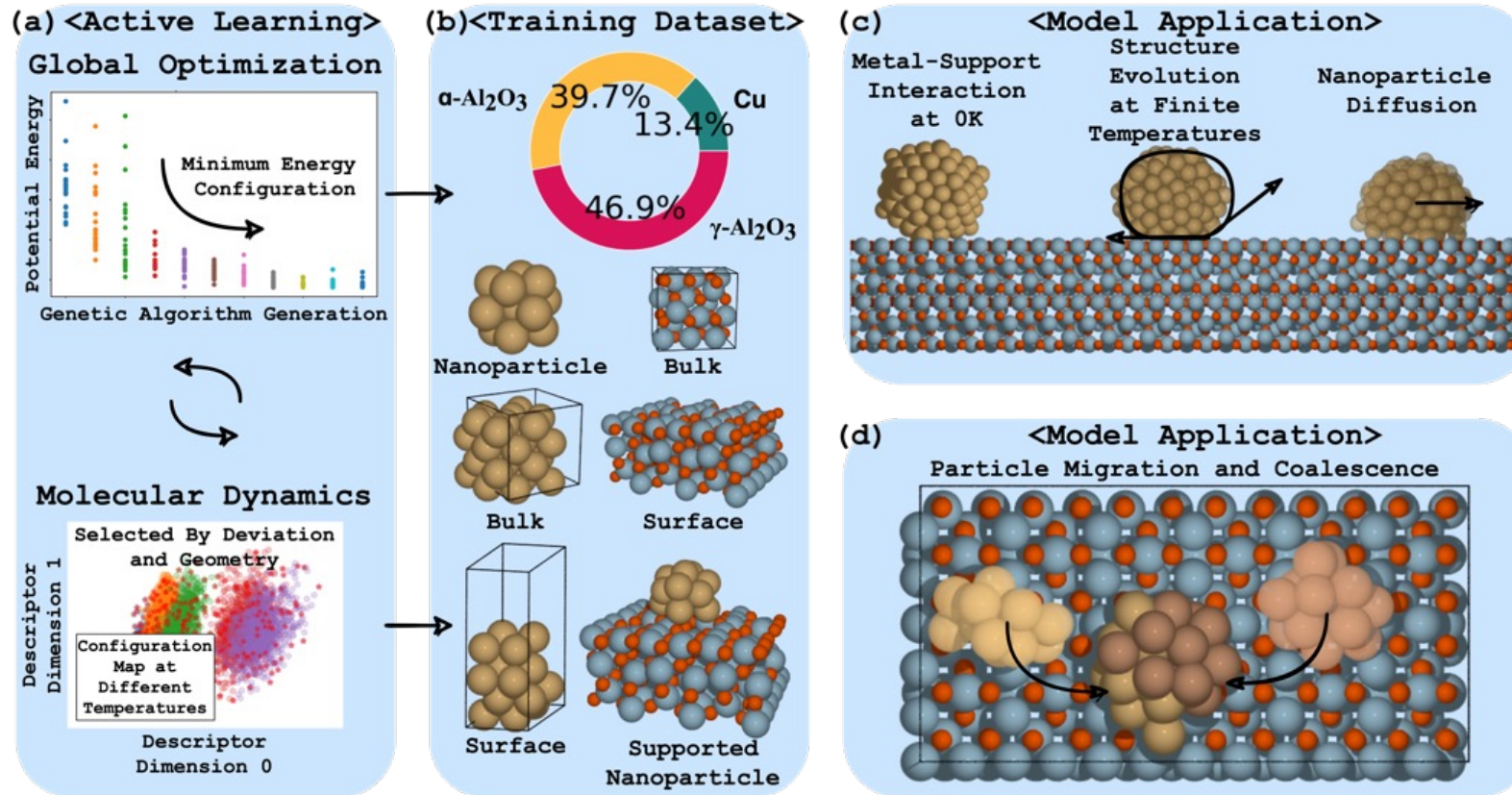


The dominant order-disorder character of the transition

P. Xie, Y. Chen, W. E. R. C., PRB **111**, 094113 (2025)



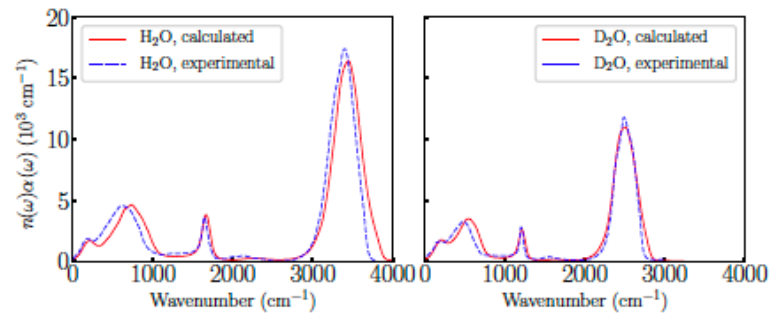
# Cu nanoparticles on $\text{Al}_2\text{O}_3$ surfaces



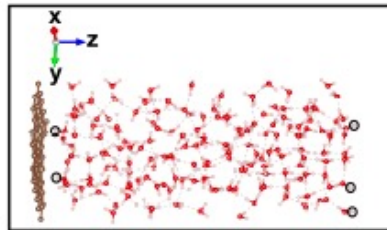
The DP model accuracy:  $E_{\text{rmse}}$  of 0.004 eV/atom,  $F_{\text{rmse}}$  of 0.057 eV/Å upon training on ~150k structures representing  $\text{Cu}_n$  on  $\gamma\text{-Al}_2\text{O}_3(100)$ ,  $\gamma\text{-Al}_2\text{O}_3(110)$ , and  $\alpha\text{-Al}_2\text{O}_3(0001)$ .

# Vibrational spectroscopies (Infrared, Raman, SFG)

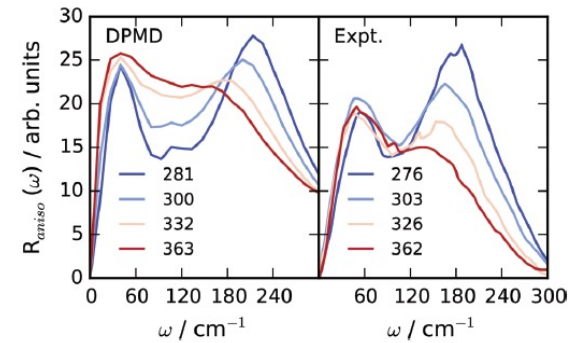
## Infrared absorption of liquid water



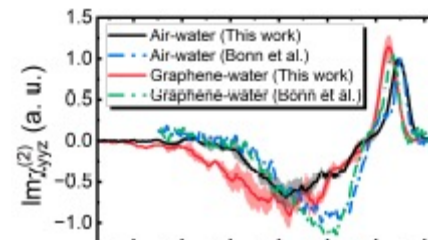
L. Zhang et al., PRB (2020)



## Raman spectra of H-bonding modes in liquid water

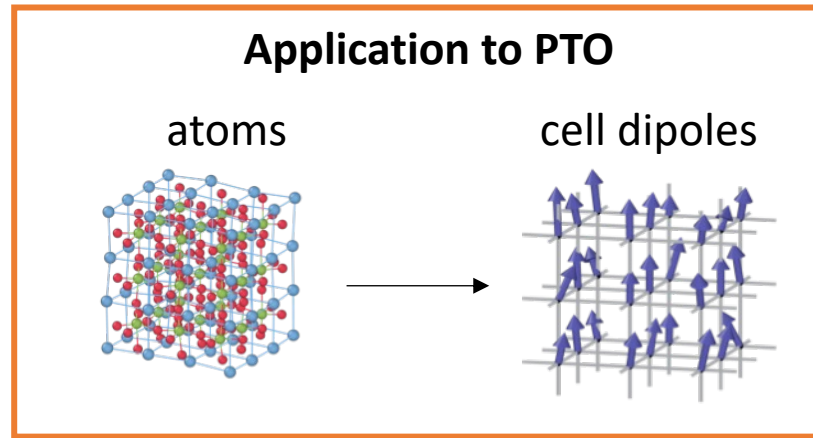


G. Sommers et al., PCCP (2020)



Y. Wang et al., arXiv (2025)

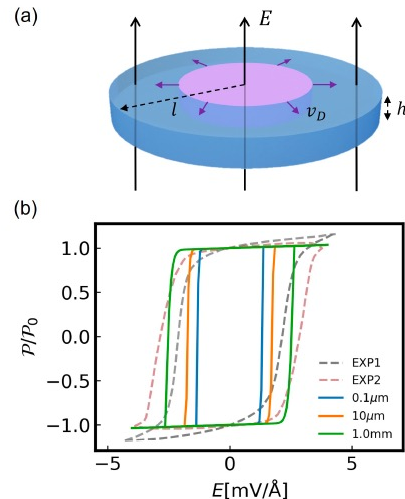
# From all-atom to collective variable dynamics: Ab-initio Generalized Langevin Equation (AIGLE)



$$M\ddot{X}(t) = -\nabla G(X) + F(t) + M \int_0^t ds K(s) \dot{X}(t-s) + R(t)$$

Effective Hamiltonian model or free energy model learned from DPMD

ML is used to learn colored noise; 2FDT is assumed



## Crude 1D model of hysteresis loop

Curvature of domain wall as well as edge, thickness ( $h$ ), and quenched disorder (O vacancies) effects are neglected.

Loops are calculated with AIGLE for different radii  $l$ , with switching field period  $t_p = 100 \mu s$ , and

remnant polarization  $P_0 = 72 \mu C/cm^2$

EXP1:  $t_p = 100 \mu s$ , unknown  $l$ ,  $P_0 = 94 \mu C/cm^2$

EXP2: unknown  $t_p$ ,  $l \approx 144 \mu m$ ,  $P_0 = 96 \mu C/cm^2$

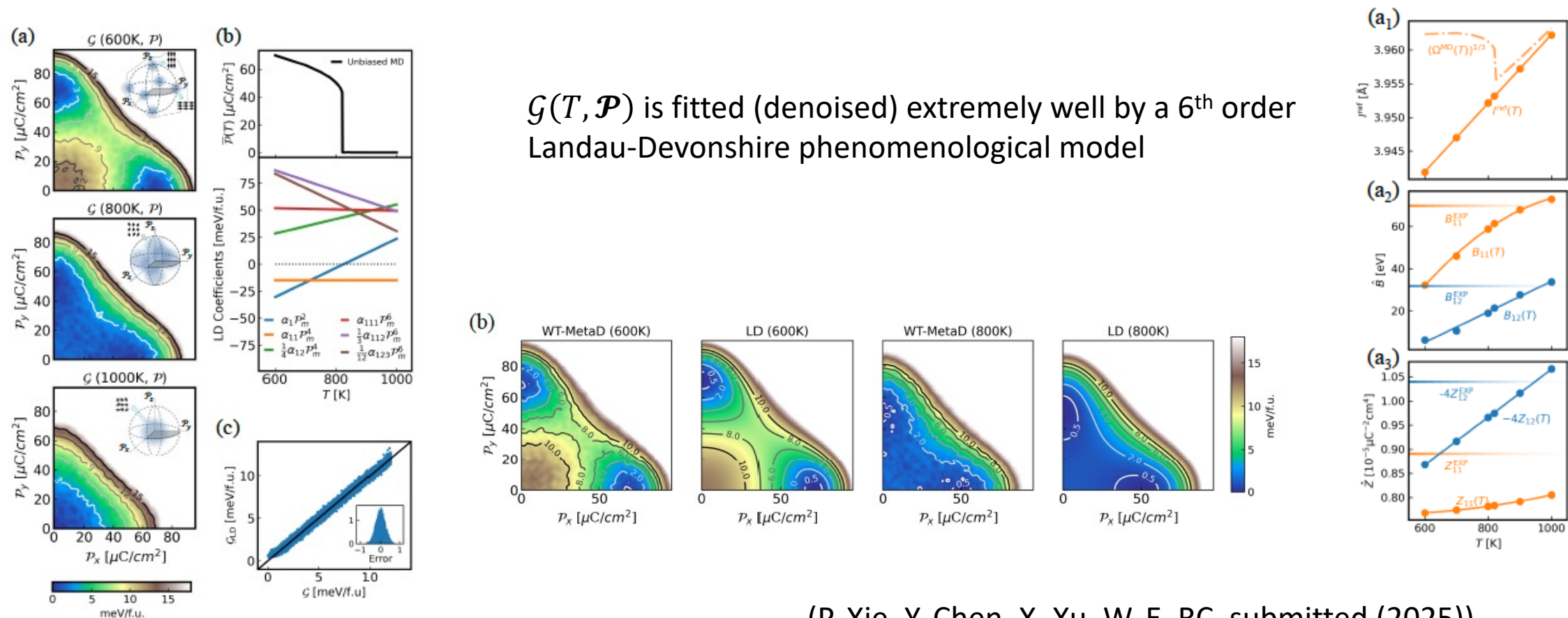


# From atoms to fields (Ginzburg-Landau models)

$$\mathcal{F}(T, \mathcal{P}, \boldsymbol{\eta}) = \mathcal{G}(T, \mathcal{P}) + \frac{N}{2} (\boldsymbol{\eta}^T - \mathbf{Q}^T \hat{\mathbf{Z}}^T) \hat{\mathbf{B}} (\boldsymbol{\eta} - \hat{\mathbf{Z}} \mathbf{Q}) \quad \mathbf{Q}^T \equiv (\mathcal{P}_x^2, \mathcal{P}_y^2, \mathcal{P}_z^2)$$

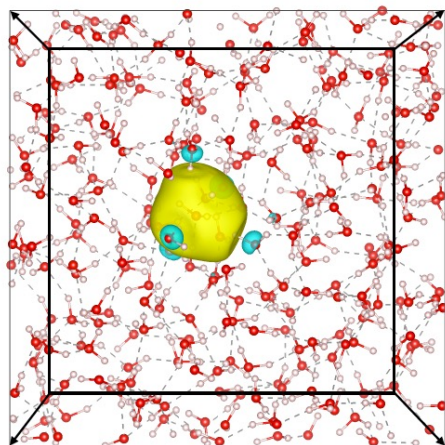
$\mathcal{G}(T, \mathcal{P})$  is obtained with WT-MetaD and the coupling matrices  $\hat{\mathbf{B}}$  and  $\hat{\mathbf{Z}}$  are obtained with linear regression

$\mathcal{G}(T, \mathcal{P})$  is fitted (denoised) extremely well by a 6<sup>th</sup> order Landau-Devonshire phenomenological model



(P. Xie, Y. Chen, X. Xu, W. E, RC, submitted (2025))

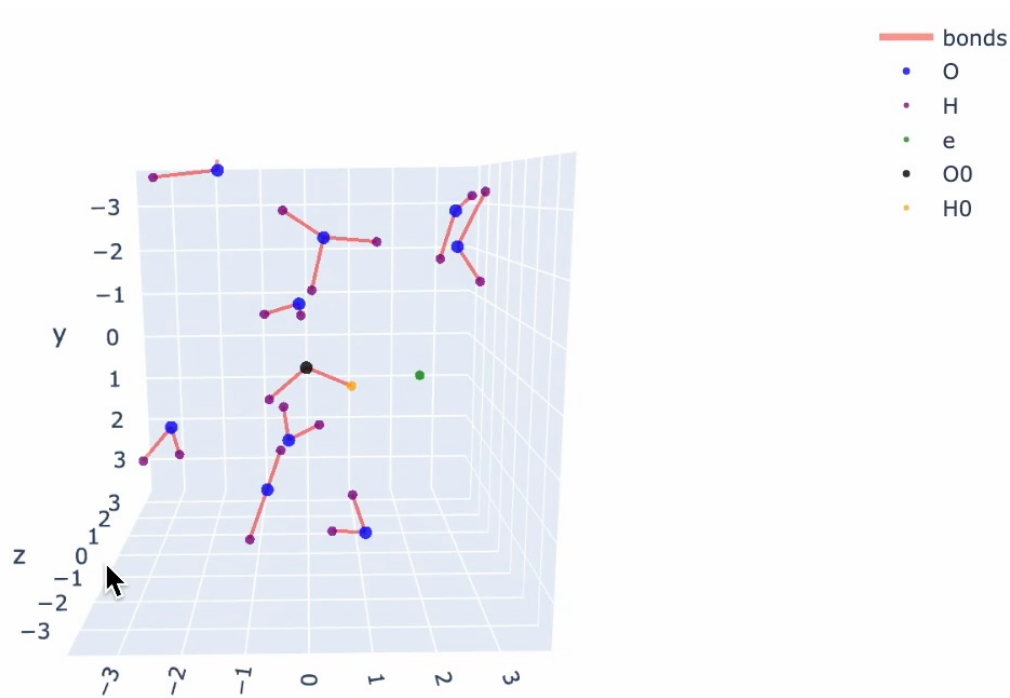
# The solvated electron: a challenge for ML coarse-graining



R. Gao, Y. Li, R.C., *PCCP* **26**, 23080 (2024)

Learning the reaction  $e^- + H_3O^+ \rightarrow H_2O + H^\cdot$  is even more challenging!

R. Gao et al., in preparation (2025)



# Concluding Remarks

- ML and DNN representations boost the power of AIMD, opening the way for ab-initio multiscale models in which the parameters are derived non-empirically from quantum mechanics
- It is essential to construct ML models that are solidly based on physics
- Multiscale ab-initio models will help to understand complex processes and to design materials and devices
- New physics may emerge from the simulations but that requires human intelligence as opposed to AI
- Accuracy of the models will improve with more accurate quantum mechanical reference models and transfer learning techniques