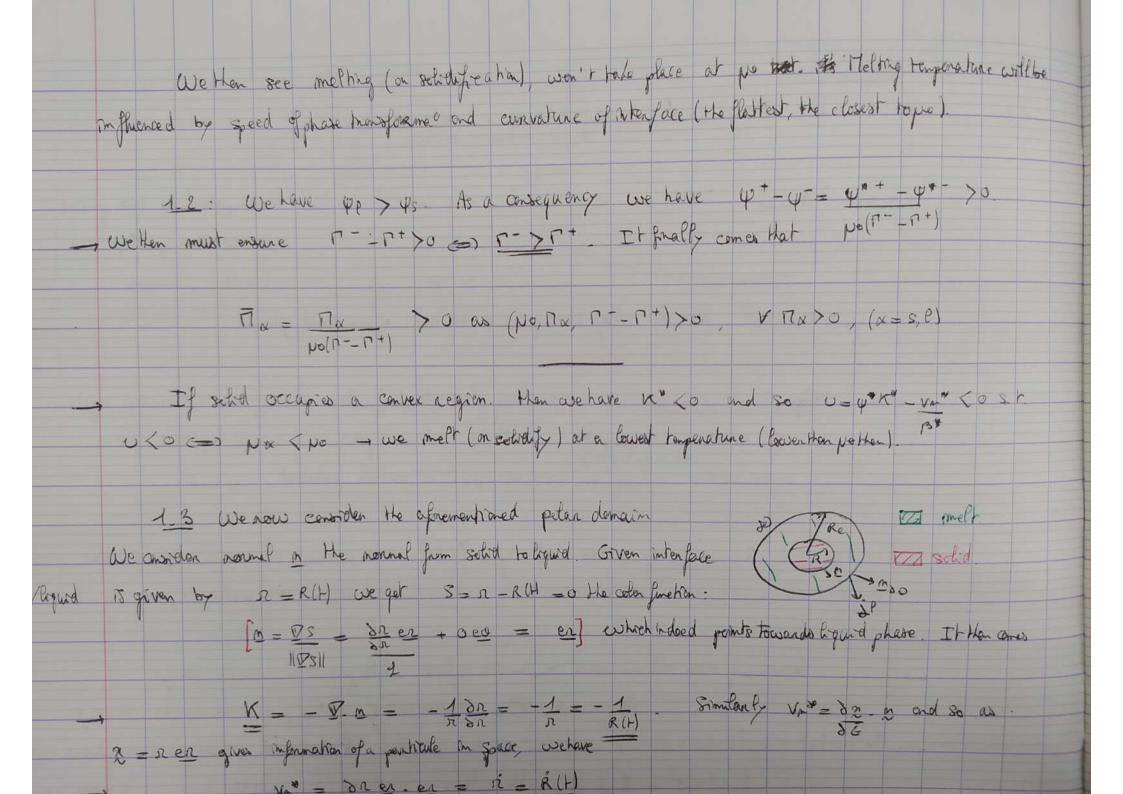
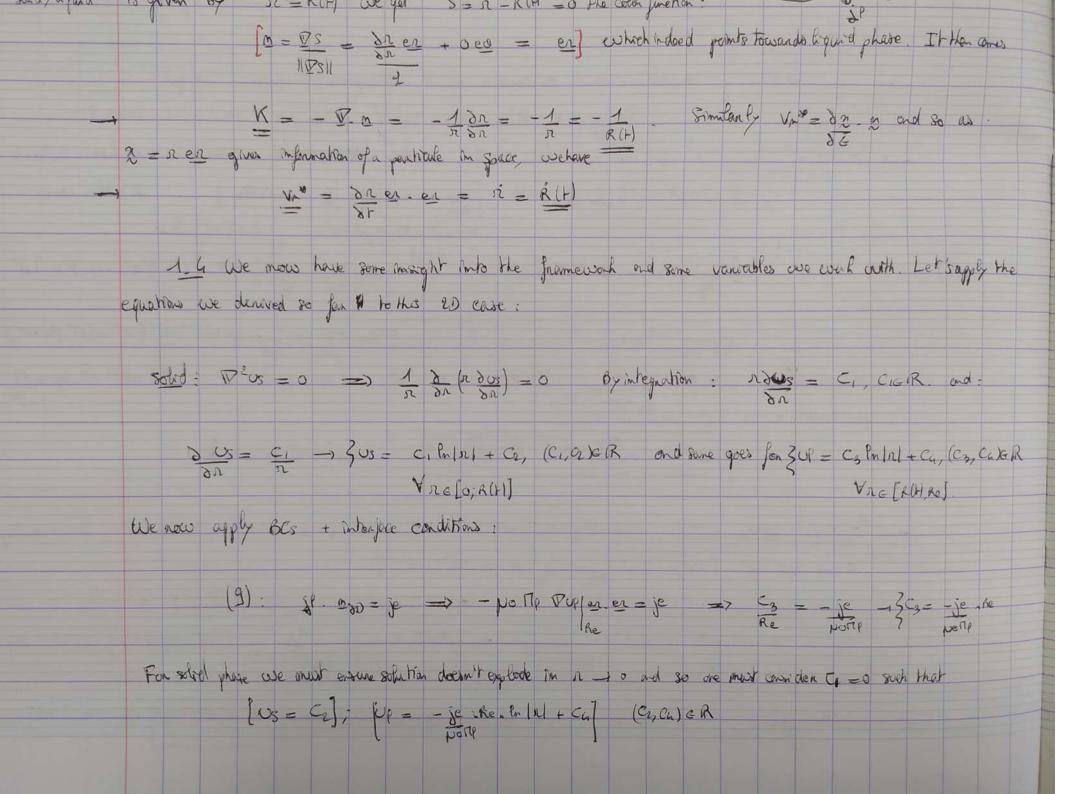
Homework: I. I sothermal solvelification of a liquid mixture Domain 20 relo; Rel x e lo; cal - decomposed in two phases: . solid binary alloy I melted ____ Binarry alloy that presents the two aforementared phases. Equation Alypotheses : mo coupling distraty / diffustity -e diffusion vity-driven system · we consider wi(p) = wi(pe) + po riv as our potential. 1.1: General theory: we don't especially take into account the domain's specifications. We have 1 = 18 U. 15 with 18 n 15 with 1 a the domain occapied by a phase; (a=4,8) Considering mass balance with me inentia effects (slow phase-hausfannes) we get Carry & De such that I De Ua = 0 in 1 a (+) (a=5, P)

Considering mass belance with me mention effects (slow phase-houstones) we get $C\alpha\delta U = t\alpha$. $R^2U\alpha$ such that $(R^2U\alpha = 0)$ in $\Lambda\alpha(t)$ $(\alpha = 3, \ell)$, To this ag of equilibrium one can add state at the interface sold toquid where of the two phases where: [pd vn* = [fd. m with [dw = -p] and [f = -17 Pp] It then comes: => (wt - w | + po u(n+-n-)) vm = (n+ put-n-10-)m Eventually this headates as Vn = - [17+ IVU+ - 17- IVU] me intenfacral mass belance. - Gibbs - Thompson relace neglecting HOI in f. Vn = B* (I wil + 4 " K") if no clashicity. Then knowing w(p) we ger [w] = (w+pr w | pr + pov(r+-r-)) by continuity. = 3* (NOU(1+-1-) + 4* K*) such that $U = \left[\frac{Vn^*}{B^*} - \psi^* K^*\right] \wedge \frac{1}{\mu d(\Gamma^+ - \Gamma^-)} = \left[\psi^* K^* - \frac{Vn^*}{B^*}\right] \mu d(\Gamma^- - \Gamma^+)$ To hook cong rescaled vanishing $\psi = \psi^*$ and $\beta = \mu_0(\Gamma^- - \Gamma^+)\beta^*$ are gets expected as la time $\mu_0 = \psi \kappa^* - \nu_n \kappa$ (generalized) Gibbs.





Applying Gibbs - Thompson retation to both phases leads to an explicit formulation of Canal Ca constants: OP = YPK* - Vm* => - Je Re Pr(RIM) + C4 = YPK* - Vm*

POTTE POTTE => 3 C4 = 40 N° - Vn° + Jeke - ln(R) = 300 = 50 Re ln(R) + 44 K° N° N° TIP N° TIP - Vn° N° TIP 1.5 By checking continuity at the interface we get that of - (1) = ge Re In (R) + 43 - VR RP We then doduce an equation for RAT: - 43 - RR + RR + 43 = je Re & Em(R)
BS BP I VOTEP 1.5 We we interformed made balance telling Vm = R = - [17p. Vut - 178 vu]. We then have Lyalong SUH, i.e. m. 1 = R R = - [TP , (-je Re . 1)] such that RR = TP FE Re and by integration it comes: RELY = je x TIPRe, t + Cs, CseR = ZR(t) = Zje, TiPRe + 2Cs which provides

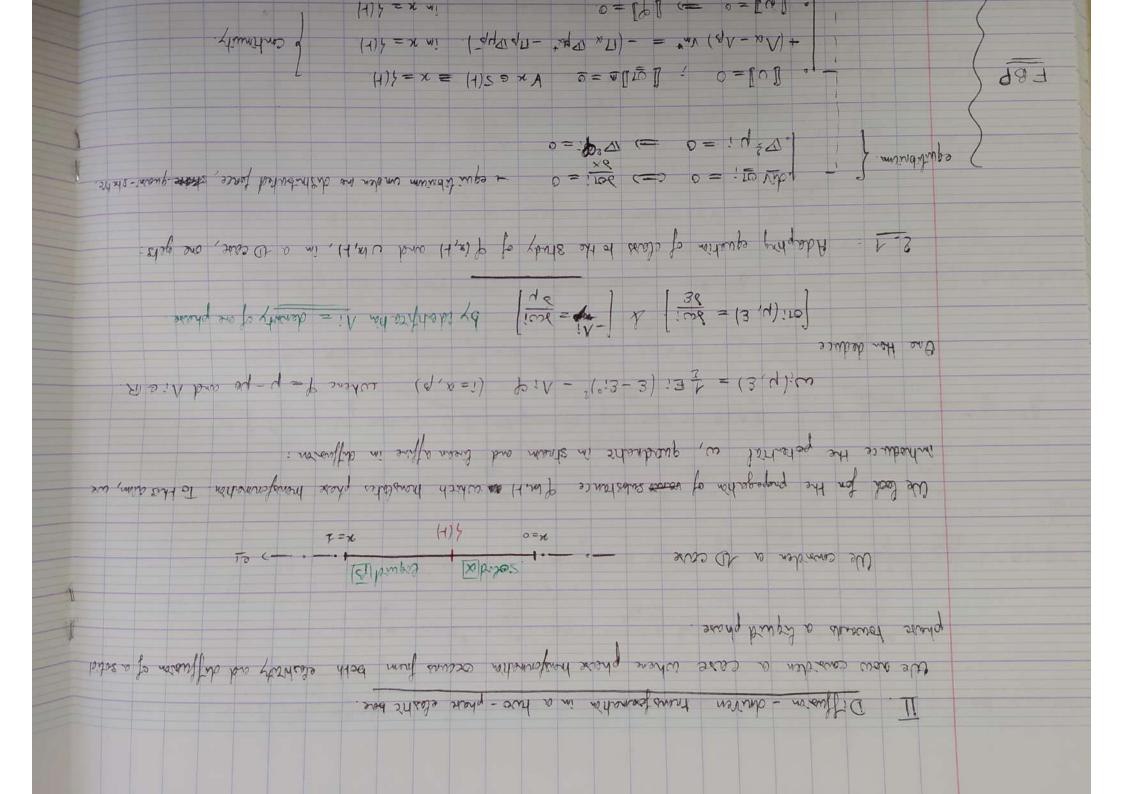
2 portron front along time.

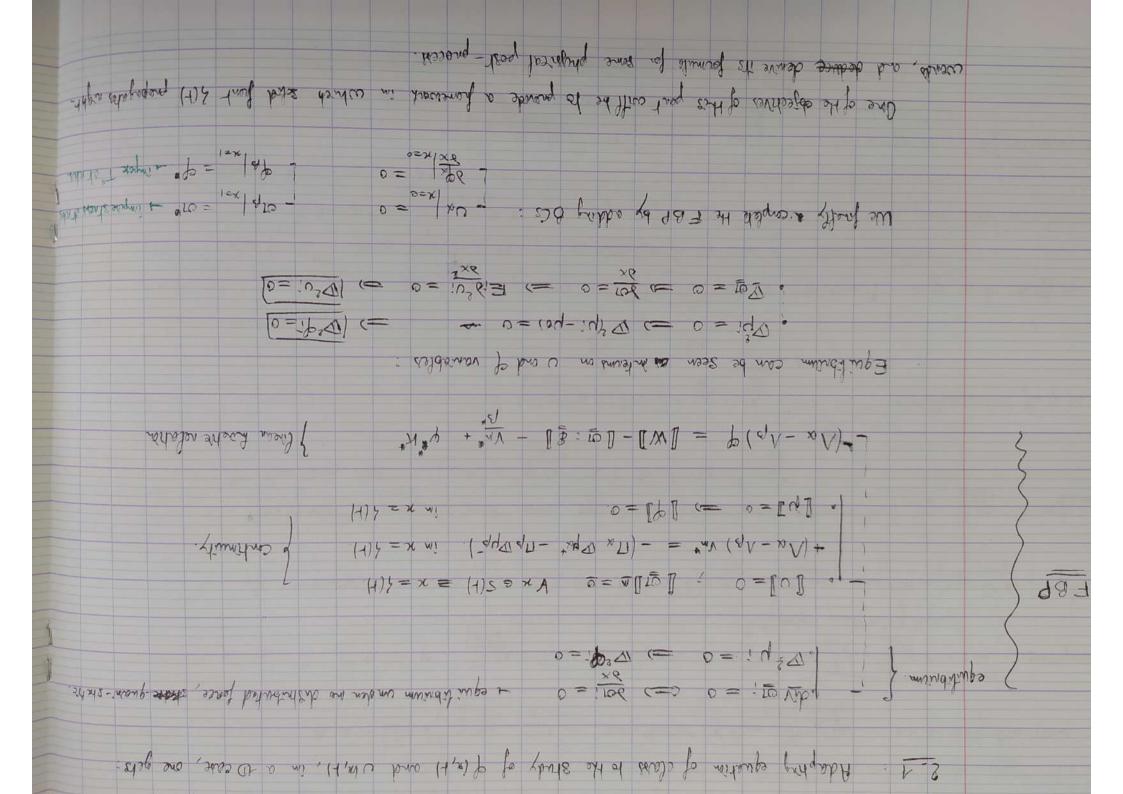
R = - [11] * (- Je Re 1) such that RR = TIP JE Re and by integration it comes: R2H = je x TPRe, + + Cs, CseR = ZR(+) = Zje, TPRe + + 2Cs which provides

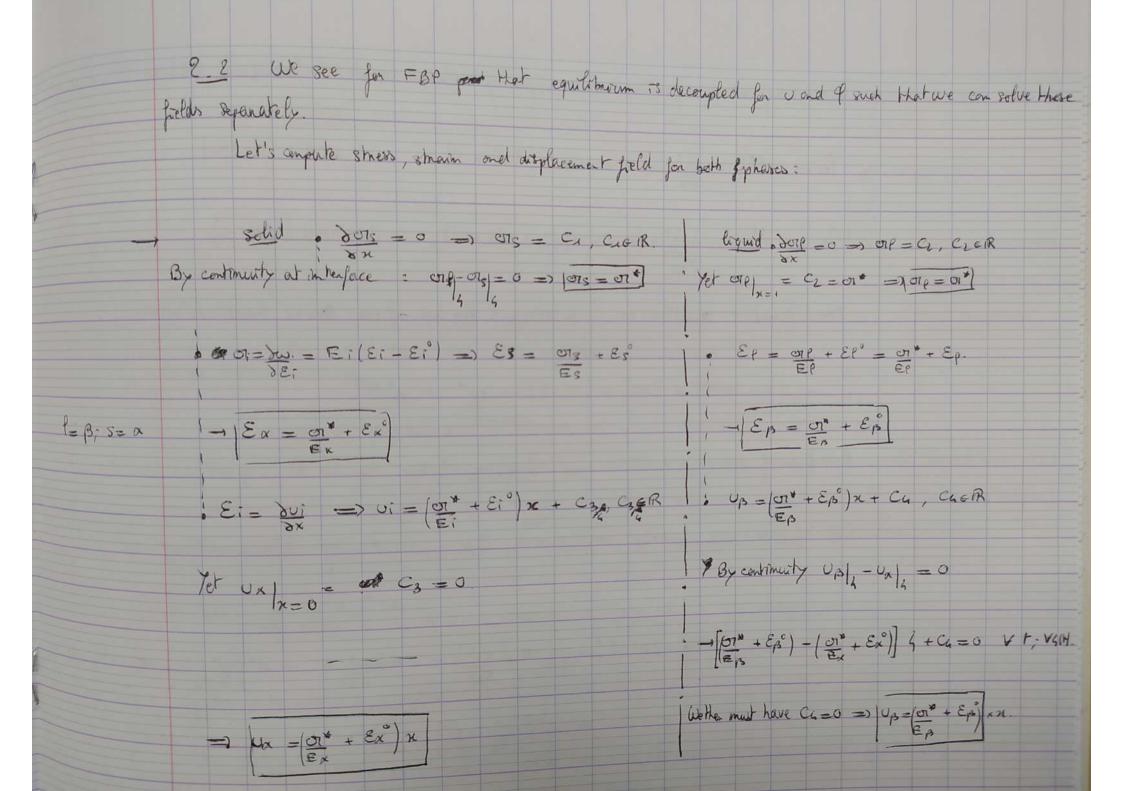
Pe TP

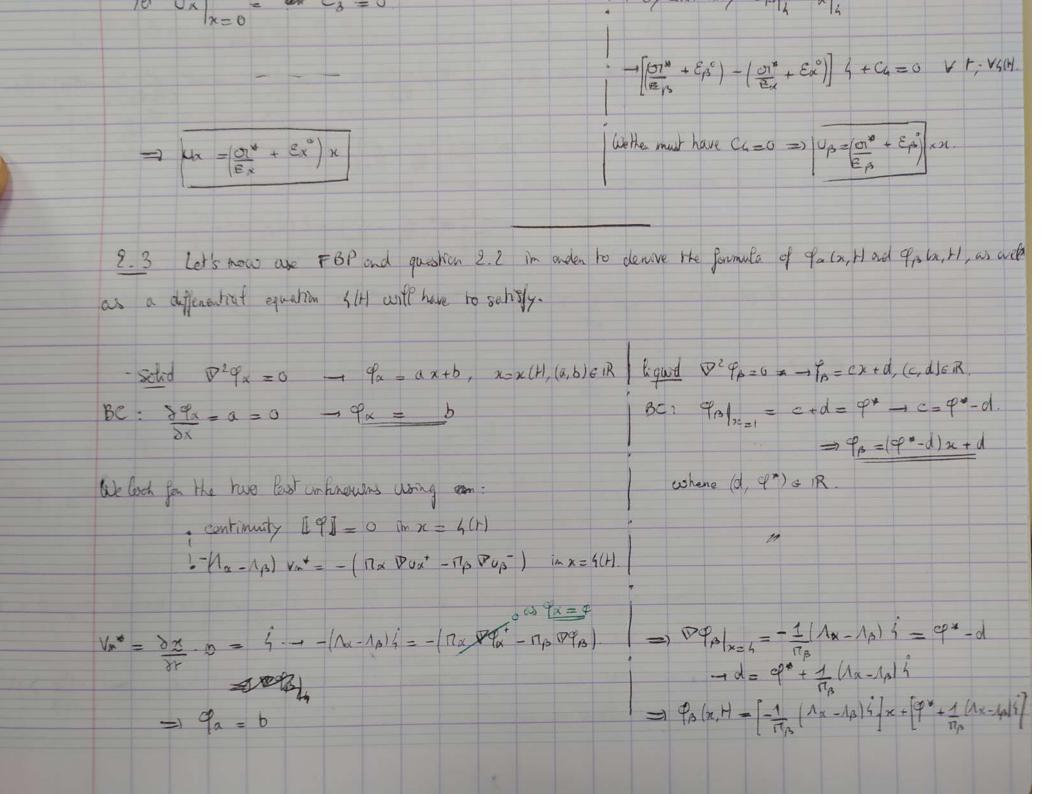
Pe TP

Portion front along time. Yet, we know R(+=0)=0 from (10) such that (2Cs=0 =) Cs=0 =)[R(H= |25e TPRe = +) 1/2] For the solid to grow, we must ensure square root is positive and by definition that 2 je TPRE > 0 => je >0 It then comes that upln; t) = je Re [Pm R - Pm n] + 4P K" - Vn"
pp have up < 0 and so Vra[R; Re] = UP - R we deduce that for je >0, we have up < 0 and so PP-po (0 =) PP-po (0 =) [PP(R;H) < po] Yro[R(H; Re] - for one to solidify we must be at a lower temperature their pe as discussed in the case of a convex curvature. As we continue to propogate (i.e. as set pluid phase schilliftees) we have at one point a total schillifteation . Remove solute (ie si liquid - solut), we expect to lower even more temperature Then ROTI = he and to It

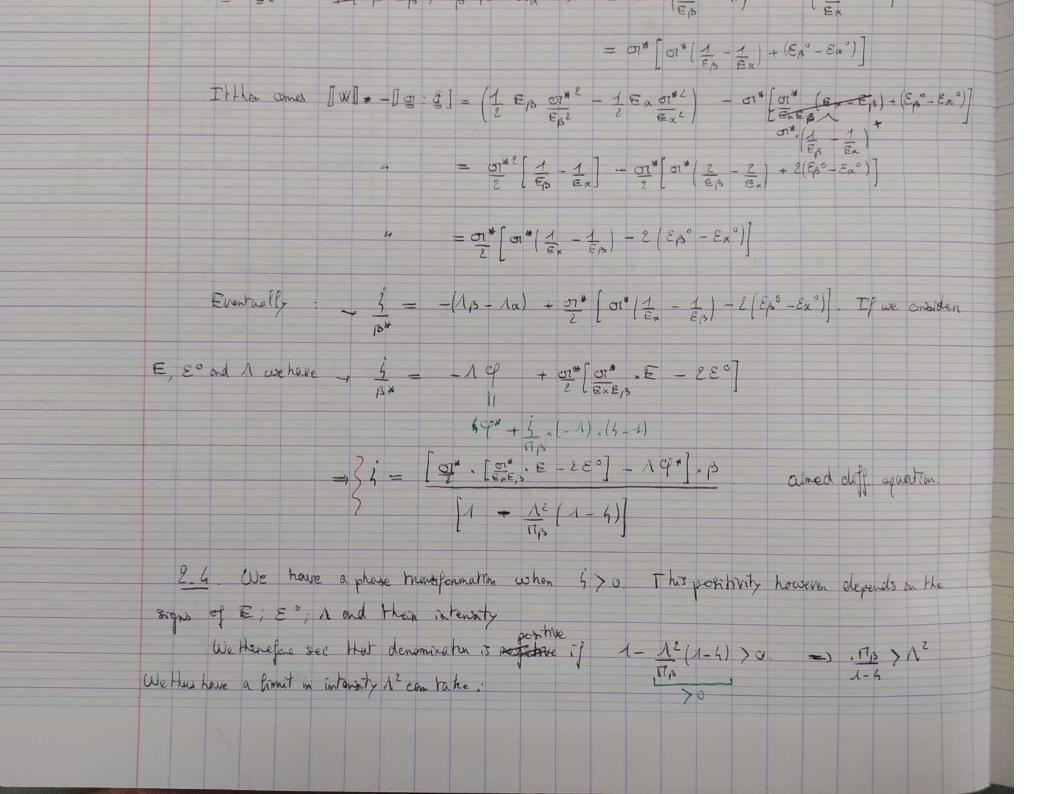


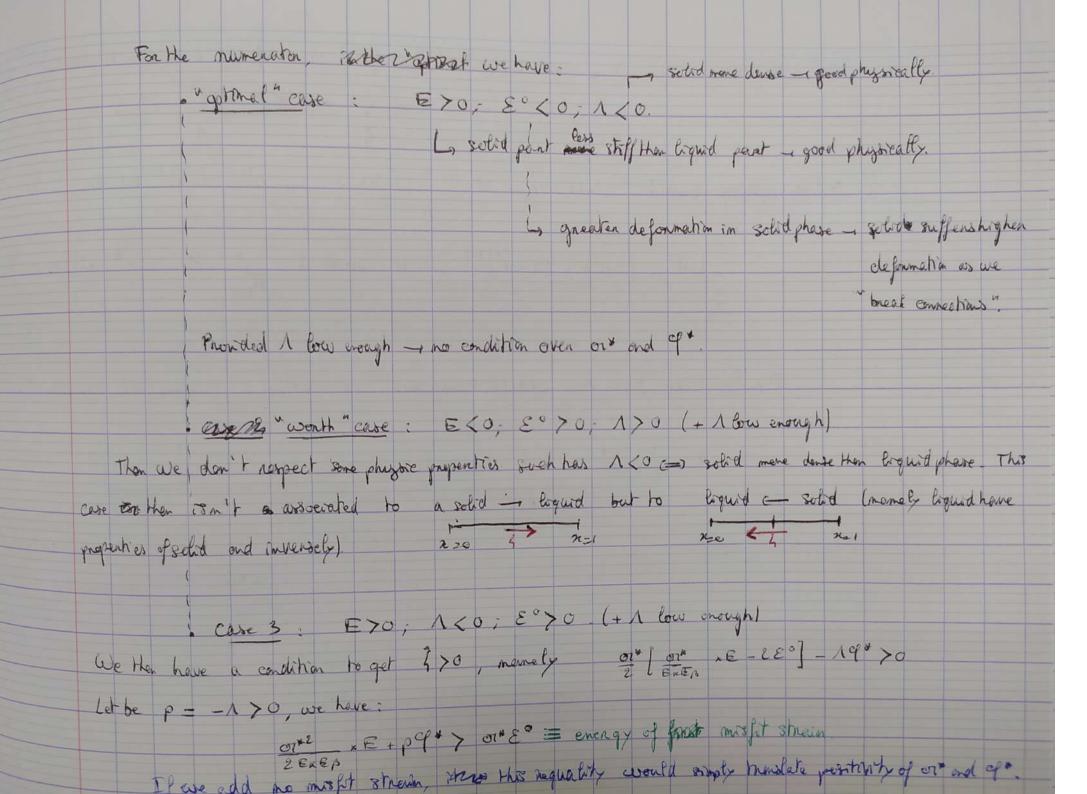






Recomitting these equation and applying continuity at interface (i.e. \$91(x=4(H)=0) trans $\begin{cases} \varphi_{x} = \varphi^{*} + \frac{i(H * (\Lambda_{x} - \Lambda_{\beta}) * (3(H - 1))}{\pi_{\beta}}, & \forall x (H \in [0; 4(H)]) \end{cases}$ $\begin{cases} \varphi_{x} = \varphi^{*} + \frac{i(H)}{\pi_{\beta}} * (\Lambda_{x} - \Lambda_{\beta}) * (x(H - 1)), & \forall x (H \in [4(H); 1]) \end{cases}$ The differential equation on 4 is obtained by use of the Brean knetic relation $-(\Lambda \alpha - \Lambda \beta) \mathcal{C} = [W] - [\mathcal{C}] : \mathcal{E}] - V_{n}^{*} + \mathcal{C}^{*}$ $= -\mathcal{C}(\beta^{*}).$ $- \left[\sigma : \mathcal{E} \right] = \sigma \beta \left[\mathcal{E}_{\beta} - \mathcal{E}_{\beta} \right] \sigma \beta \mathcal{E}_{\beta} - \sigma \alpha \mathcal{E}_{\alpha} = \sigma^* \cdot \left(\frac{\sigma^*}{\mathcal{E}_{\beta}} + \mathcal{E}_{\alpha}^{\circ} \right) - \sigma^* \left(\frac{\sigma^*}{\mathcal{E}_{\alpha}} + \mathcal{E}_{\alpha}^{\circ} \right).$ $= O(* \left[O(* \left(\frac{1}{\epsilon_0} - \frac{1}{\epsilon_0}\right) + (\epsilon_0^{\circ} - \epsilon_0^{\circ})\right]$ Then omes $[W]_{*} - [g]_{*} = (1 + g)_{*} = (1 + g)_{*}$ $= \underbrace{\sigma_1}^* \left[\underbrace{\sigma_1}^* \left(\underbrace{1}_{\varepsilon_K} - \underbrace{1}_{\varepsilon_B} \right) - 2 \left(\underbrace{\varepsilon_B}^\circ - \underbrace{\varepsilon_A}^\circ \right) \right]$





(+ 1 low chough) We then have a condition to get \$>0, memely on [on RE-280] - 190 >0. Let be p = -1 >0, we have: If we add no misht strein, its requality would simply hunder perturby of or and of. Note: multiple different combinations of signs and emplitude of E, 1, & exist, but they are not always physical, manely they don't respect the fact we study subdification at the expense of a trouved, i.e settle solely To hindsight we can however see that:

The lower of - the more of is eager to be positive => lower "temperature at the I the higher ext - => as we puff at ahs. 2.5 By integration of outs. equation we derived for 4 (+): $\begin{bmatrix} 1 - \Lambda^2 + \Lambda^2 & 3 \end{bmatrix} & = C, C = \sigma_1^* \begin{bmatrix} \sigma_1^* & - \\ \hline \eta_B & \eta_B \end{bmatrix} & = C, C = \sigma_1^* \begin{bmatrix} \sigma_1^* & - \\ \hline \Xi_{*} \varepsilon_B \end{bmatrix} & = R.$ Let be $A = \Lambda^2$, we have $= \begin{array}{c} X = (1-A) + A4 & = \begin{array}{c} X = A & \text{ind} & 80 \\ X \times 1 & = \begin{array}{c} X & A4 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A & X & X \end{array} & \begin{array}{c} X & 2 & = \\ \hline A &$ [(1-A) + A 4] i = C

Replacing X by X = (1-A) + A4 => [(1-A) + A4] = 2AC++ D', D'eR $= \frac{31}{4} = \frac{(2ACV + D')^{1/2} - (1-A)}{A}$ one so replacing A and IC: 4(+=0) = D'1/2 - (1-A)=0 =) D' = (1-A)^2 and 80; $\frac{2}{3}(H) = \left[2ACT + (1-A)^{2}\right]^{1/2} - (1-A) \quad \text{with } A = \Lambda^{2} \quad \text{and } C = \left[\frac{\sigma}{2}\left[\frac{\sigma}{2}\left[\frac{\sigma}{2}\right]^{2}\right]^{2}\right] \times \beta^{2}$ Frally, we have bur entirely transformed once 4(H = 1, such that then t = + + and so: $A = [2ACT^{*} + (1-A)^{2}]^{-1/2} - (1-A) \Rightarrow t^{*} = [A + (1-A)]^{2} - (1-A)^{2}$ 2+" = 1-(1-A)2 ben entirely manuformed from EAC B phase to x one