

# Stochastic morphological simulation based on partial statistical description of polycrystals

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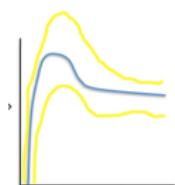
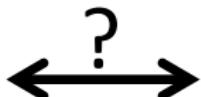
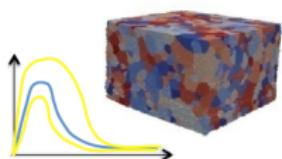


The Johns Hopkins University

Engineering Mechanics Institute (EMI) Conference  
Stanford University, June 18 2015

# Motivation, Objective and Approach

## Structure - Property Relation

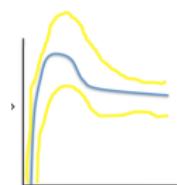
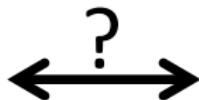
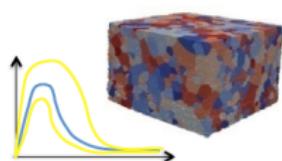


- Grain size, shape, boundary type
  - Crystallographic orientation
  - Phase configuration
- Dissipation mechanism
  - Strength, ductility
  - Thermal, electrical conductivity

Objective: Facilitate the development of a better understanding of the mechanical behavior of polycrystals by means of numerical simulation.

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## Microstructure simulation

Approach: Enable the simulation of polycrystalline microstructures with target:

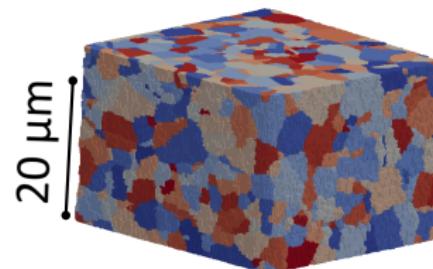
- Distribution of morphological features;
- and correlation structures.

# Parameterization of polycrystalline microstructures

Polycrystalline microstructures are generally represented as digitalized data sets; for instance this sample of nickel super-alloy IN100

Data set from EBSD:

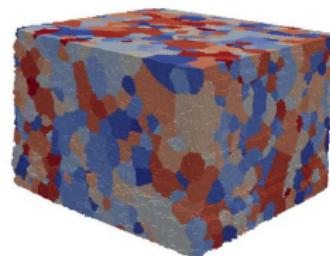
- Data consuming;
- Finite resolution.



4, 444, 713 parameters

## Parameterization by ellipsoidal growth tessellation

- Less than 2% the original amount of data;
- Infinite resolution (in theory).



14, 118 parameters

## Presentation outline

- ① Ellipsoidal growth tessellation: a model for morphologically anisotropic polycrystals.
- ② Tessellation resolution: from numerical to semi-analytical solutions.
- ③ Transformation of the problem: from growth to collision.
- ④ Grain boundary resolution: a semi-analytical approach.
- ⑤ Morphology characterization: using Minkowski valuations.
- ⑥ Stochastic simulation with target morphological features: next steps and objectives.

## Ellipsoidal growth tessellation (EGT) — Definition

- Tessellation on  $\mathbb{R}^d$ : *Countable* set  $m = \{\mathring{\mathcal{C}}_\alpha\}$  of *disjoint* and *space-filling* cells  $\mathring{\mathcal{C}}_\alpha \subset \mathbb{R}^d$ :

$$\mathring{\mathcal{C}}_\alpha \cap \mathring{\mathcal{C}}_\gamma = \emptyset \quad ; \quad \bigcup_{\alpha} \mathring{\mathcal{C}}_\alpha = \mathbb{R}^d \quad ; \quad \#\{\mathring{\mathcal{C}}_\alpha \cap m \mid \mathring{\mathcal{C}}_\alpha \cap \mathcal{B} \neq \emptyset\} < \infty \quad \forall \mathcal{B} \subset \mathbb{R}^d$$

- Multiplicatively weighted anisotropic Voronoi tessellation:

$$\mathcal{C}_\alpha = \{\underline{x} \in \mathbb{R}^d \mid T_\alpha(\underline{x}) < T_\gamma(\underline{x}) \quad \forall \gamma \neq \alpha\}$$

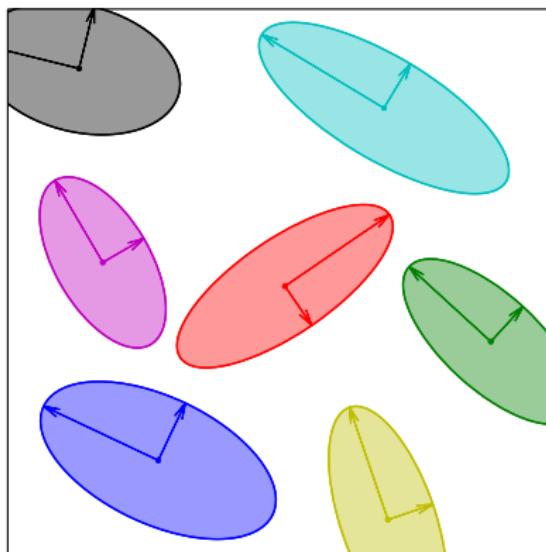
where  $T_\alpha(\underline{x}) \equiv (\underline{x} - \underline{x}^\alpha) \cdot \mathbf{Z}_\alpha \cdot (\underline{x} - \underline{x}^\alpha)$  and

$$\mathbf{Z}_\alpha = \sum_{j=1}^d \frac{\underline{u}_j^\alpha \otimes \underline{u}_j^\alpha}{(v_j^\alpha)^2}$$

while  $\underline{u}_j^\alpha \in \mathbb{R}^d$  and  $v_j^\alpha \in \mathbb{R}^+$  such that every  $\mathbf{Z}_\alpha$  is a second-rank positive definite tensor.

## Ellipsoidal growth tessellation (EGT) — Definition

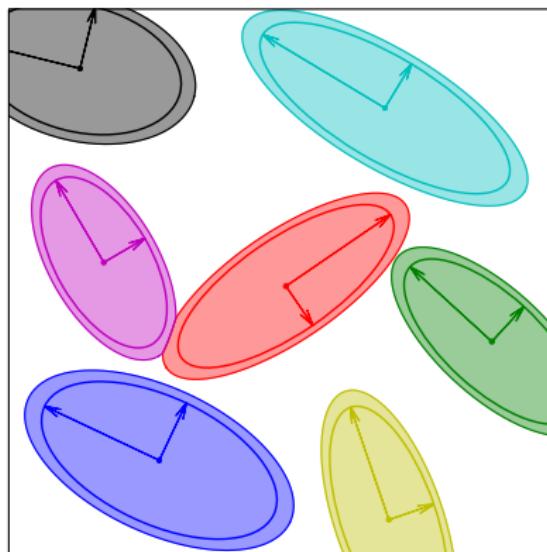
- Grain growth analogy: Every tensor  $\mathbf{Z}_\alpha^{-1/2}$  is a representation of an ellipsoidal velocity profile with principal component  $v_j^\alpha$  along  $\underline{u}_j^\alpha$ .



- For a cell with velocity profile  $\mathbf{Z}_\alpha^{-1/2}$  growing from  $\underline{x}_\alpha$ :
  - ▶  $T_\alpha(\underline{x})$  is the time necessary for the cell to reach  $\underline{x}$ ;
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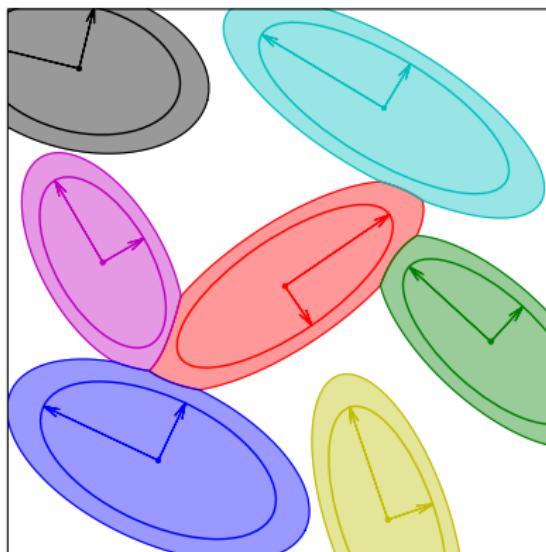
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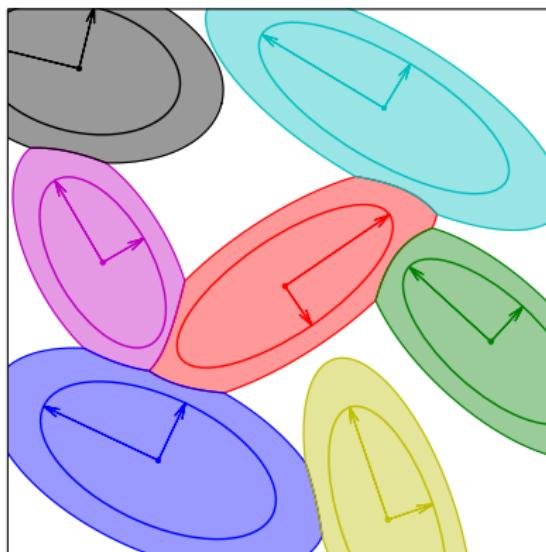
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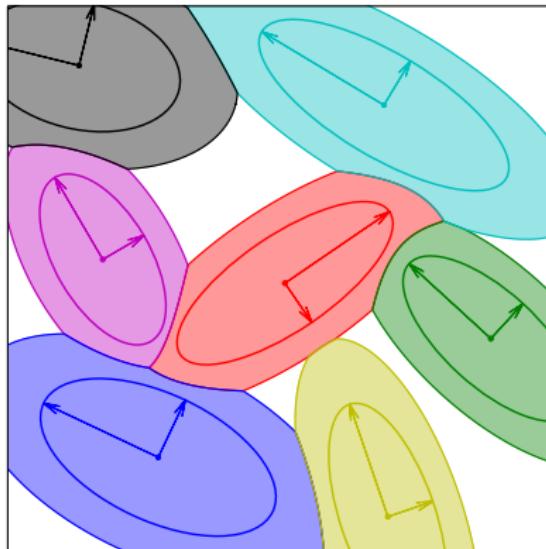
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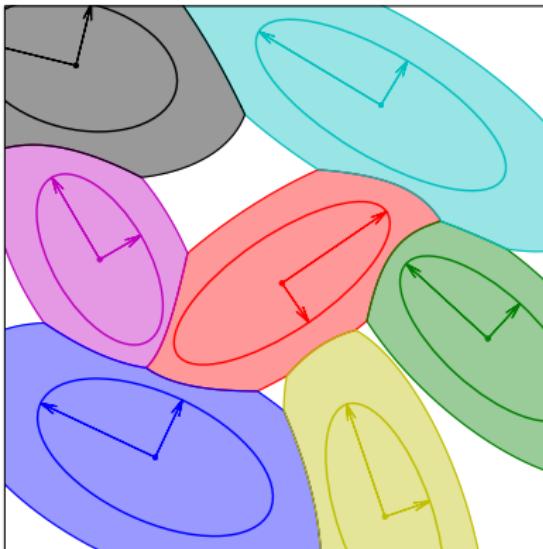
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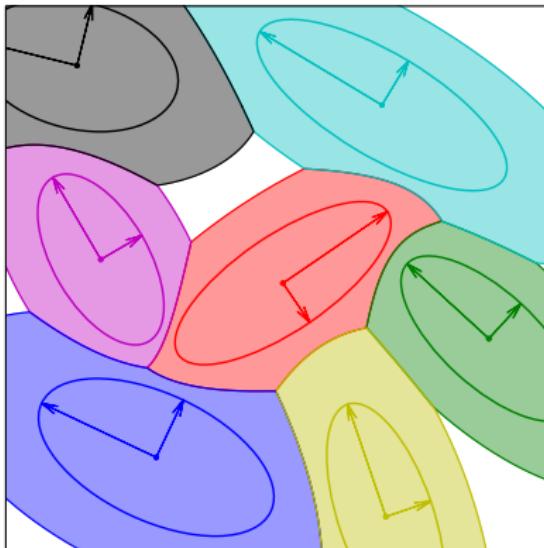
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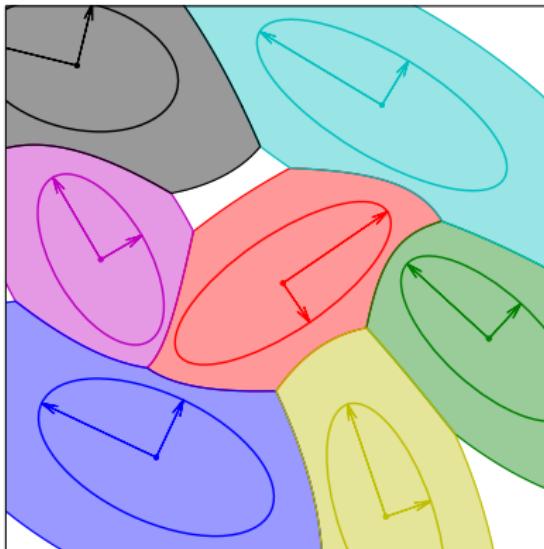
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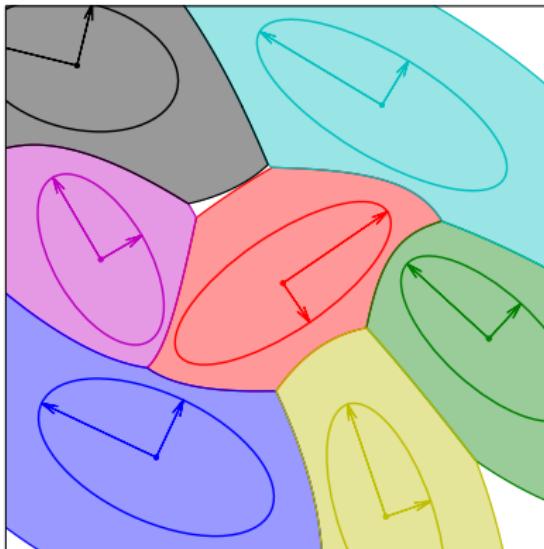
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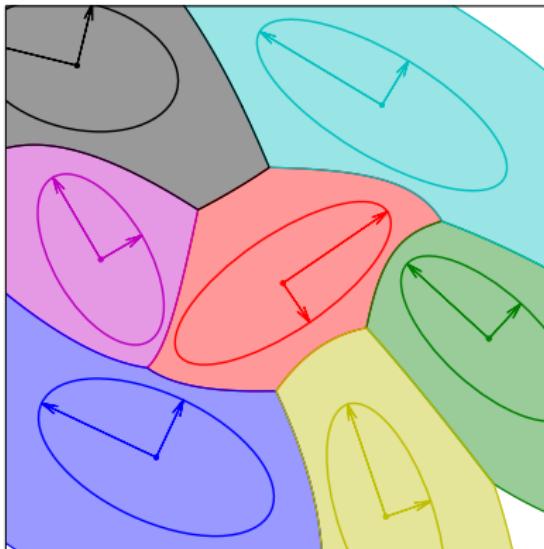
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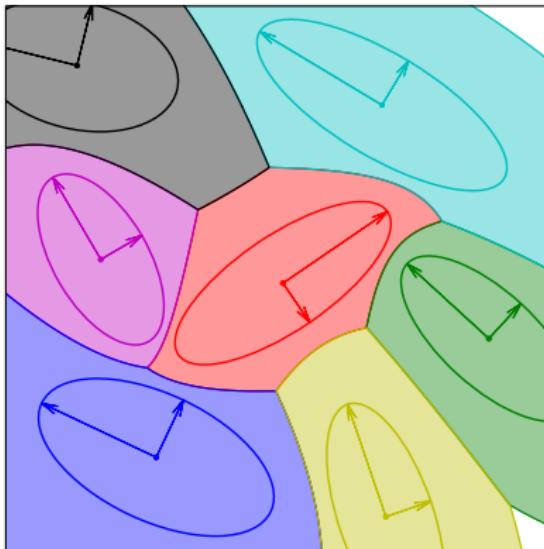
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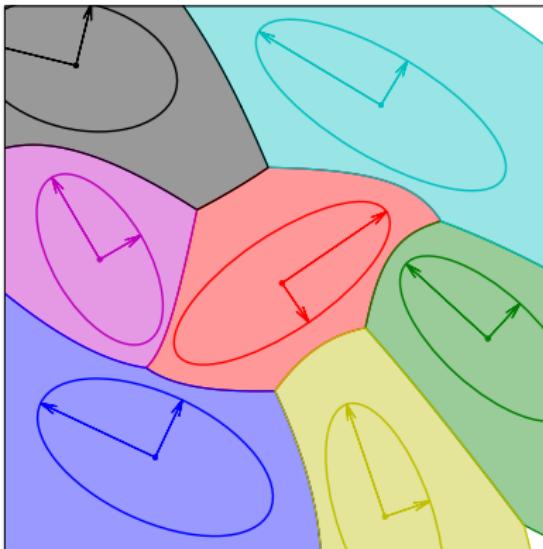
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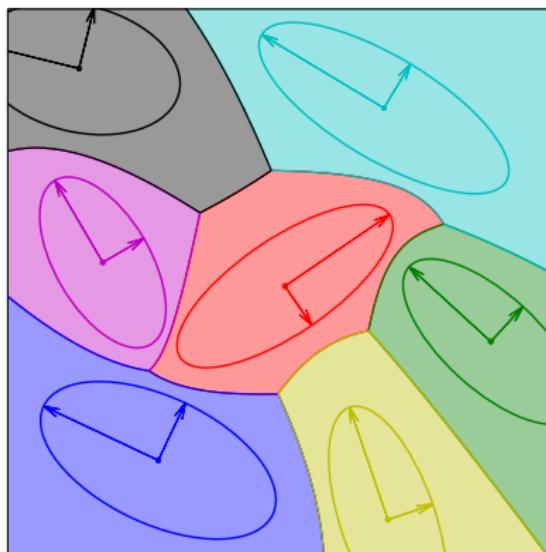
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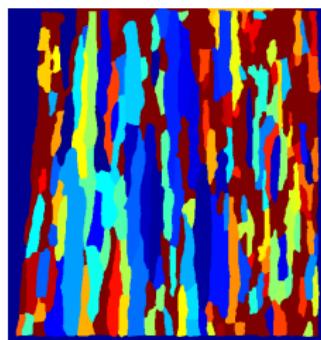


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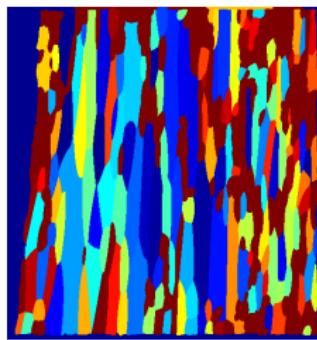
# Ellipsoidal growth tessellation (EGT) — Application

Teferra and Graham-Brady (2015):

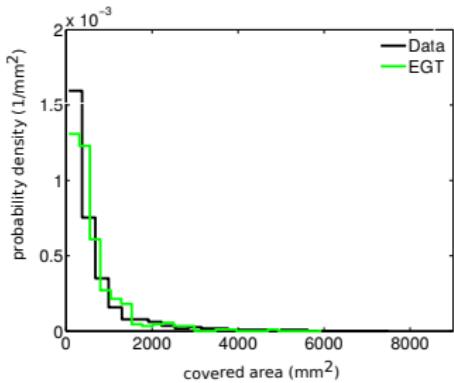
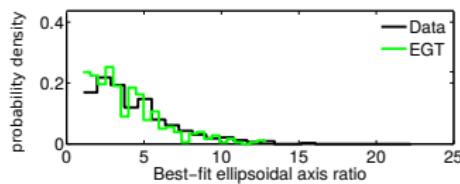
Parameterization of an hot-rolled aluminum microstructure by best-fit EGT.



Experimental data

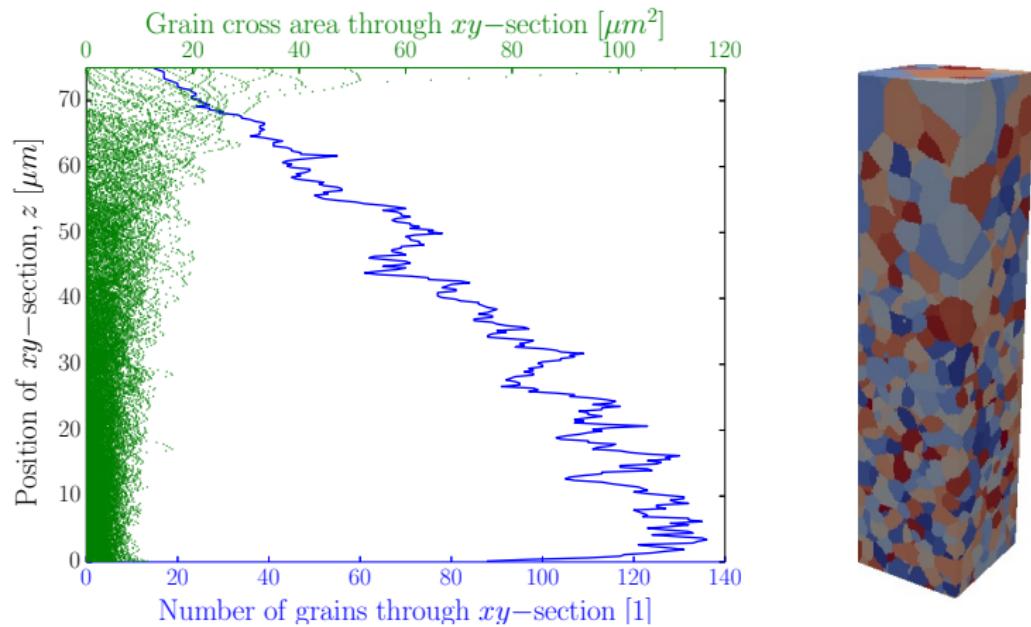


Fitted EGT



# Ellipsoidal growth tessellation (EGT) — Application

Stochastic simulation of a functionally graded structure with target distributions of tessellation parameters.

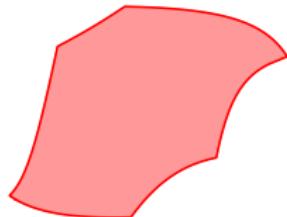


# Toward a semi-analytical approach — Motivation

Morphology characterization:

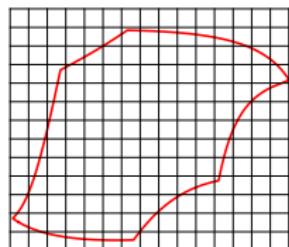
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$$\mathcal{I}_\alpha = \{\underline{x} \in \mathbb{R}^d \mid T_\alpha(\underline{x}) = T_\gamma(\underline{x}) \forall \gamma \neq \alpha\}$$



Drawback of a numerical approach:

- The accuracy of the estimated shape metrics depends on the resolution used to solve the tessellation;
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Question:

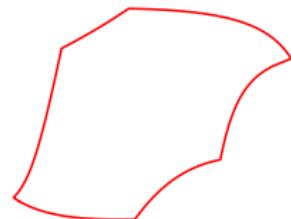
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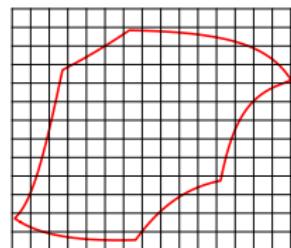
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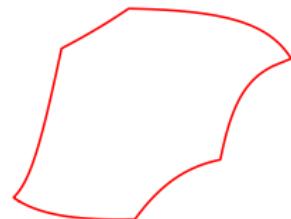
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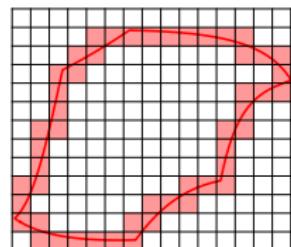
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## Evolving morphology of growing cells

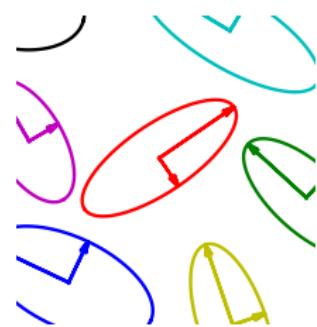
To solve for  $\mathcal{I}_\alpha$ , we introduce the time variable  $\xi$  at which all the points reached by the cell growing from  $\underline{x}_\alpha$  lie in

$$\mathcal{D}_\alpha(\xi) = \{\underline{x} \in \mathbb{R}^d \mid [\underline{x} - \underline{x}_\alpha] \cdot \mathbf{Z}_\alpha \cdot [\underline{x} - \underline{x}_\alpha] = \xi^2\}$$



All the points of the evolving cell boundary  $\mathcal{I}_\alpha(\xi)$  either lie in  $\mathcal{D}_\alpha(\xi)$  or in common curves  $\mathcal{I}_{\alpha\gamma}(\xi)$  given by:

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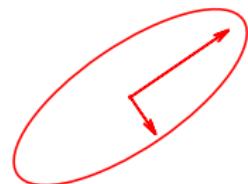


Solving for points in  $\mathcal{I}_{\alpha\gamma}$  requires to solve for a distinct quartic equation at every time  $\xi$ . Problems identifying the correct roots for large values of  $\xi$ .

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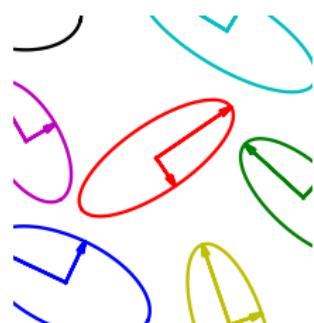
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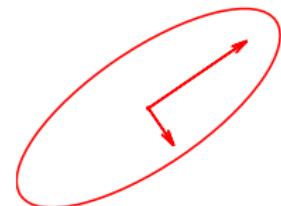


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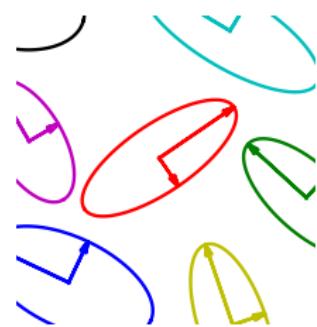
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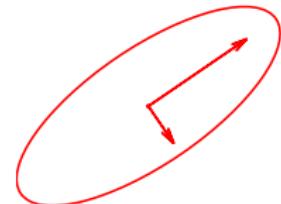


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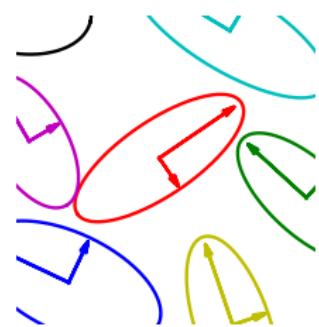
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$$\mathcal{D}_\alpha(\xi) = \{\underline{x} \in \mathbb{R}^d \mid [\underline{x} - \underline{x}_\alpha] \cdot \mathbf{Z}_\alpha \cdot [\underline{x} - \underline{x}_\alpha] = \xi^2\}$$



All the points of the evolving cell boundary  $\mathcal{I}_\alpha(\xi)$  either lie in  $\mathcal{D}_\alpha(\xi)$  or in common curves  $\mathcal{I}_{\alpha\gamma}(\xi)$  given by:

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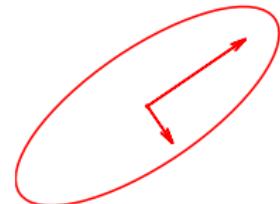


Solving for points in  $\mathcal{I}_{\alpha\gamma}$  requires to solve for a distinct quartic equation at every time  $\xi$ . Problems identifying the correct roots for large values of  $\xi$ .

## Evolving morphology of growing cells

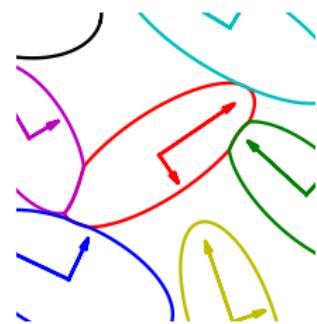
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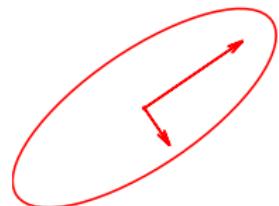


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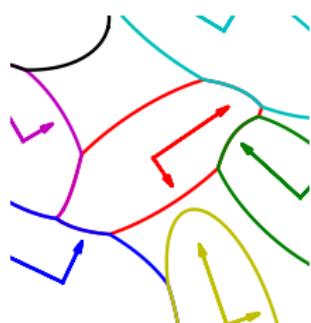
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## Transformation: from growth to collision

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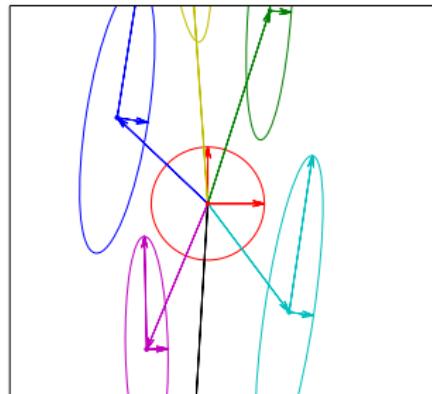
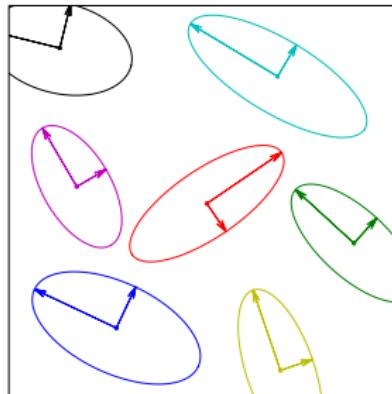
$$\underline{x}' = (1/\xi) \mathbf{R} \cdot \mathbf{Z}_{\alpha}^{1/2} \cdot (\underline{x} - \underline{x}_{\alpha})$$

with  $\mathbf{R} = \sum_{j=1,d} \underline{e}_j \otimes \underline{u}_j^{\alpha}$  so that  $\mathcal{D}_{\gamma}(\xi)$  becomes:

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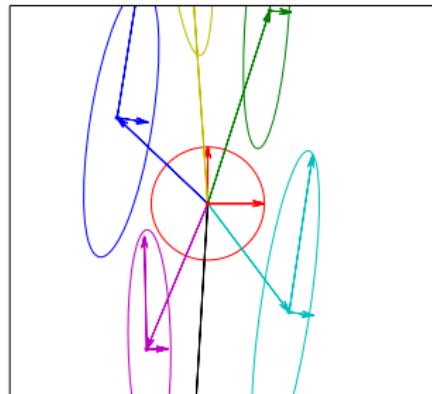
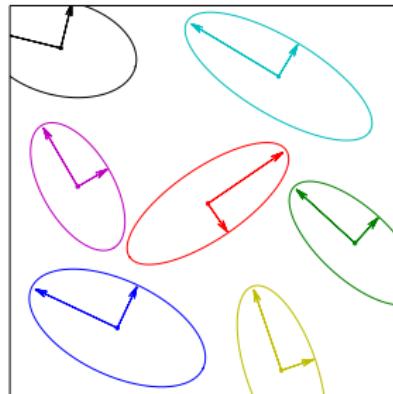
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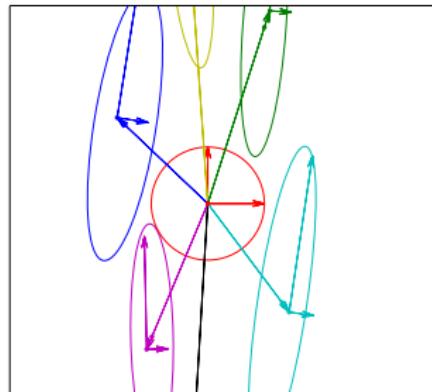
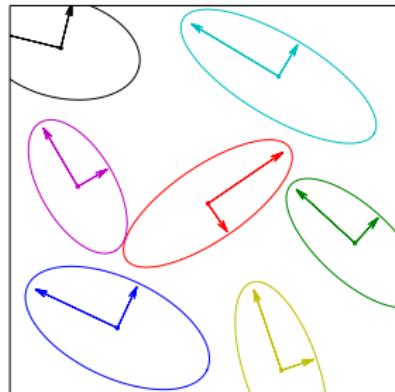
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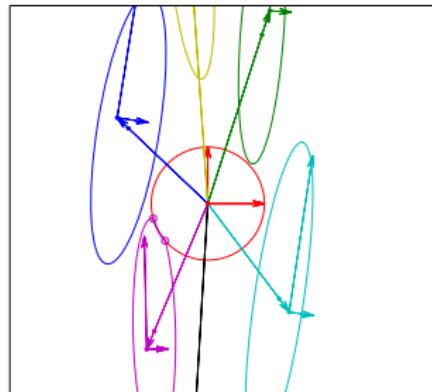
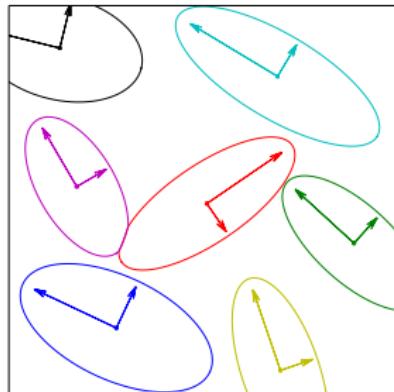
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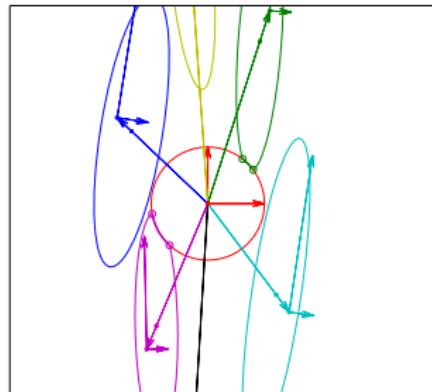
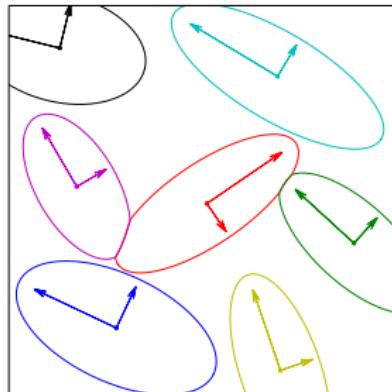
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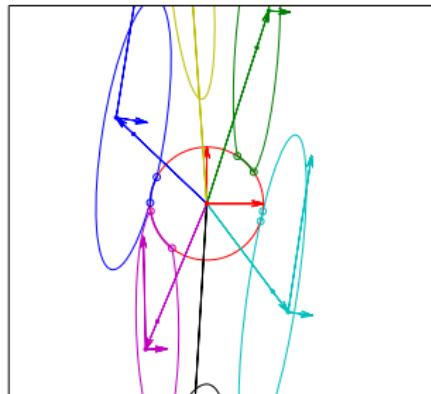
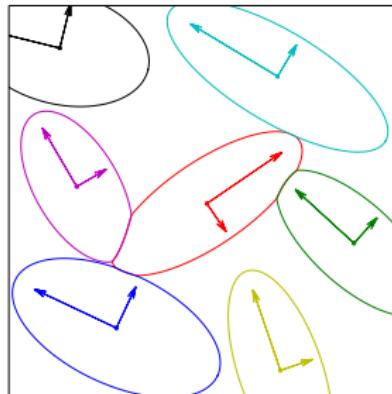
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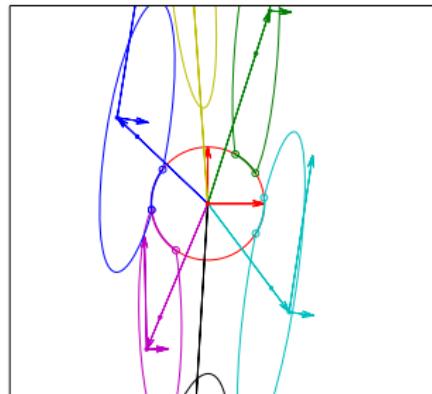
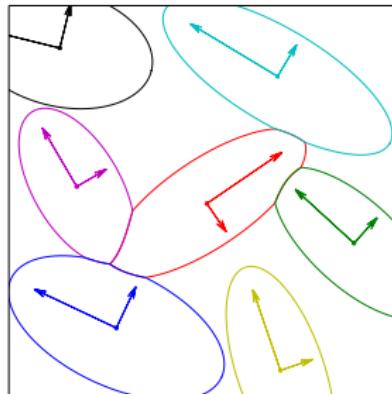
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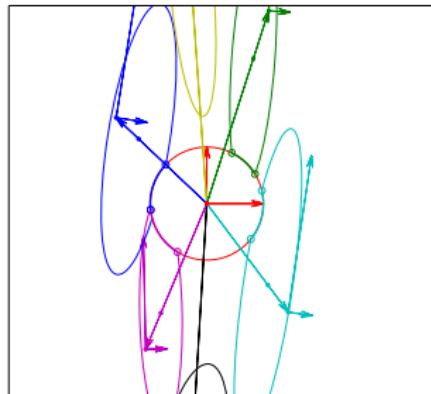
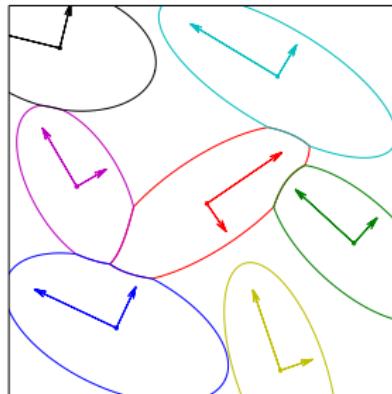
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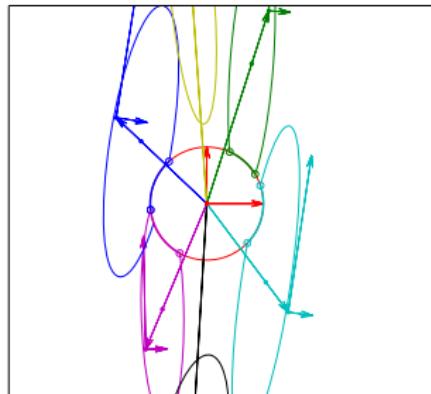
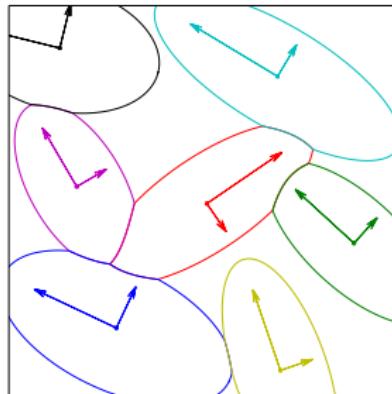
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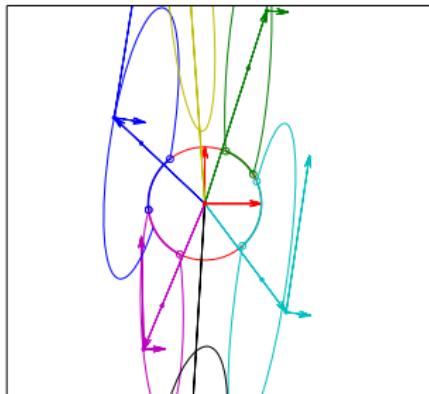
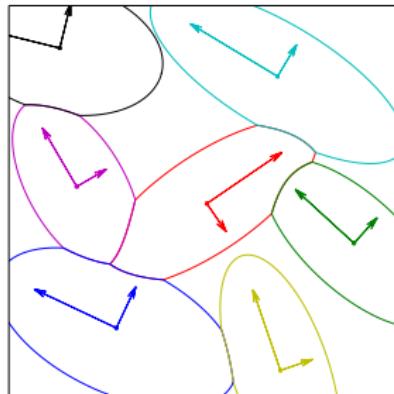
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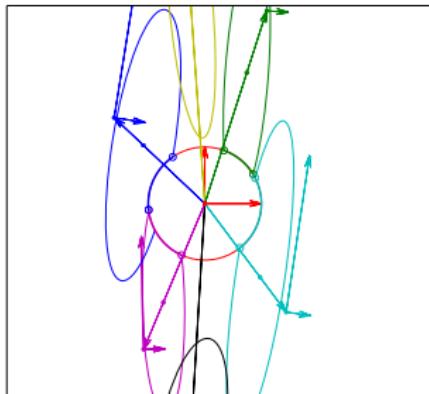
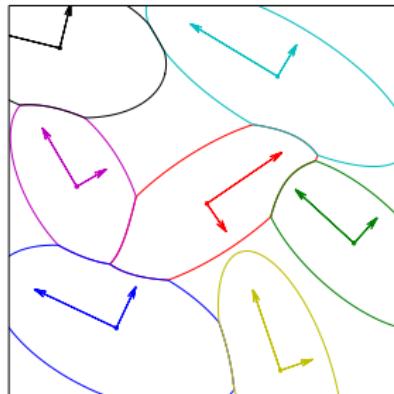
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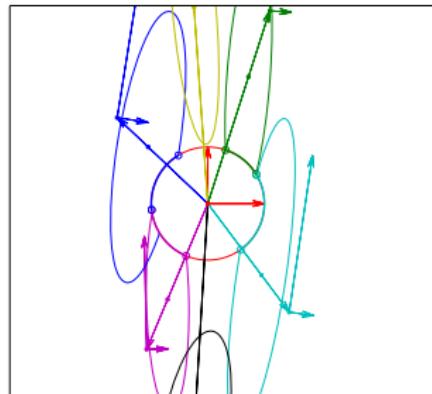
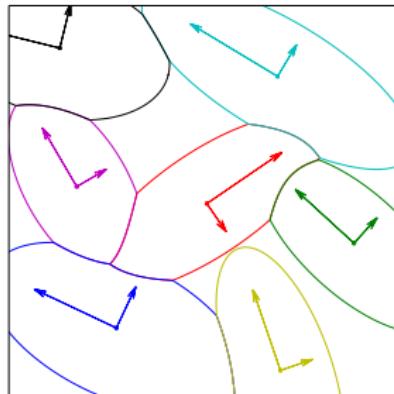
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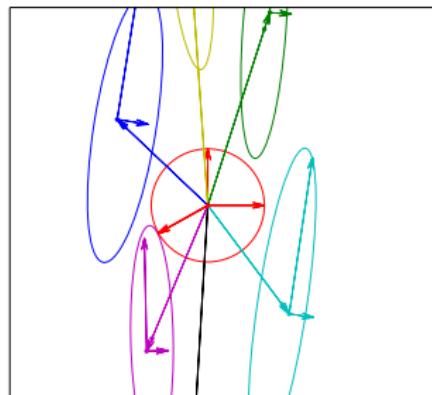
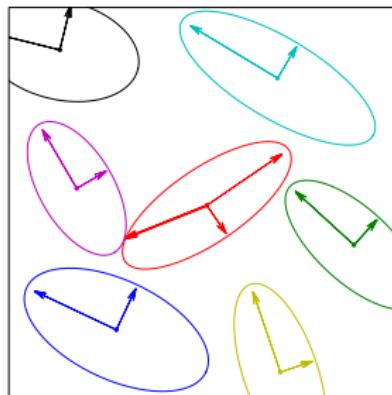
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## Transformation: from growth to collision

When two ellipses  $\mathcal{D}_\alpha(\xi)$  and  $\mathcal{D}_\gamma(\xi)$  come into contact, so does  $\mathcal{D}'_\gamma(\xi)$  with the centered unit circle  $\mathcal{D}'_\alpha$ :



We note  $x'_{\alpha\gamma}$  the point at which  $\mathcal{D}'_\gamma(\xi)$  first comes into contact with  $\mathcal{D}'_\alpha$ . To solve for  $x'_{\alpha\gamma}$  we find first the time at which the contact occurs:

$$\xi_{\alpha\gamma} = \min\{\xi \in \mathbb{R}^+ \mid \mathcal{D}'_\alpha \cap \mathcal{D}'_\gamma(\xi) \neq \emptyset\}$$

which is equivalent to solving for the smallest real root of a 6-th order polynomial. The contact point  $x'_{\alpha\gamma}$  is then calculated using  $\xi_{\alpha\gamma}$ .

## Condition for $\xi$ to be of the form $\xi(\theta')$

Every point  $\underline{x}$  of the grain boundary  $\mathcal{I}_\alpha$  satisfies the equation

$$\underline{x} = \underline{x}_\alpha + \xi(\underline{x}', \underline{x}) \mathbf{Z}_\alpha^{-1/2} \cdot \mathbf{R}^T \cdot \underline{x}' \text{ with } \underline{x}' \cdot \underline{x}' = 1$$

where  $\xi(\underline{x}', \underline{x})$  is the time at which contact happens at  $\underline{x}'$  between  $\mathcal{D}_\alpha(\xi)$  and the neighboring ellipse  $\mathcal{D}_\zeta(\xi)$ .

For a 2D cell  $\mathcal{C}_\alpha$  *radially convex* at  $\underline{x}_\alpha$ , the relation  $\xi(\underline{x}', \underline{x})$  simplifies to  $\xi(\theta')$  where  $\underline{x}' = \underline{e}_1 \cos \theta' + \underline{e}_2 \sin \theta'$ .

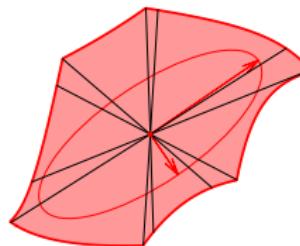
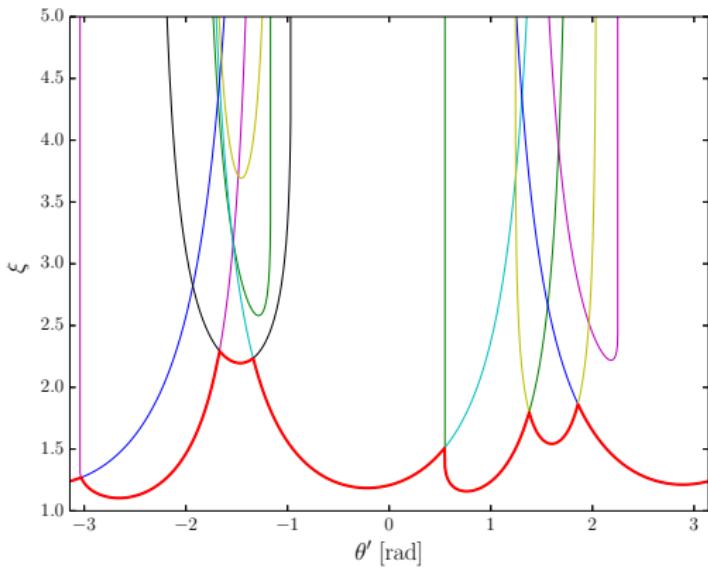
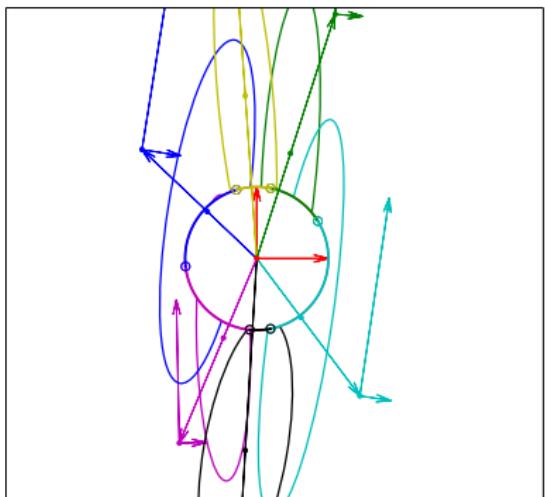


Figure :  $\mathcal{C}_\alpha$  radially convex at  $\underline{x}_\alpha$

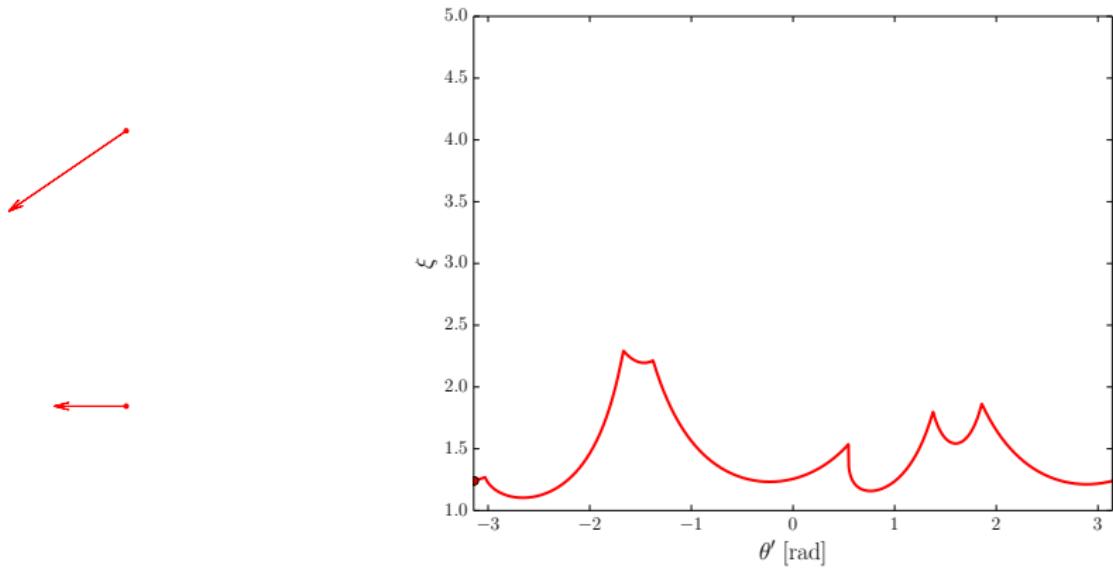
The boundary  $\mathcal{I}_\alpha$  of a cell *radially convex* at  $\underline{x}_\alpha$  can then be reconstructed from a unit circle using the relation  $\xi(\theta')$ .

## Resolution of the $\xi(\theta')$ relation

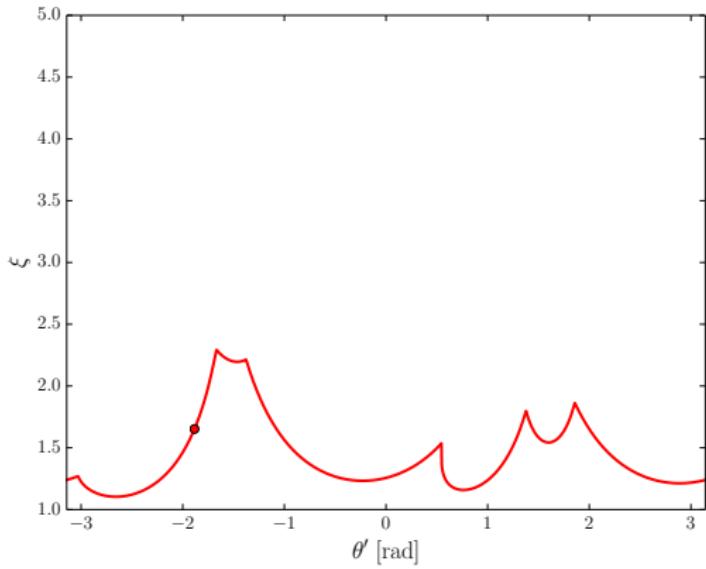
Knowing the velocity at which an ellipse  $\mathcal{D}'_\gamma(\xi)$  travels through  $\mathcal{D}_\alpha$ , we can solve for  $\xi(\theta')$  piece by piece, each part corresponding to an arc of the unit circle (or a distinct common curve of the un-transformed boundary):



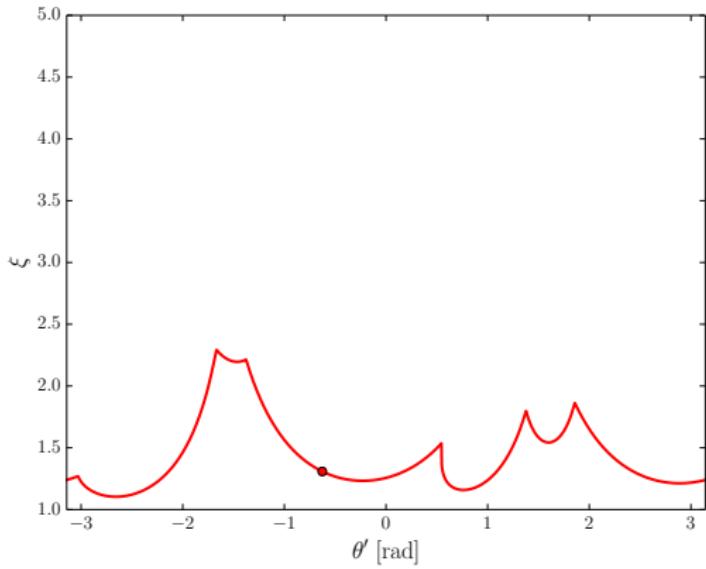
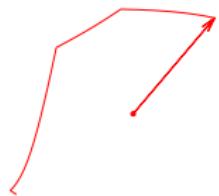
# Resolution of the un-transformed grain boundary



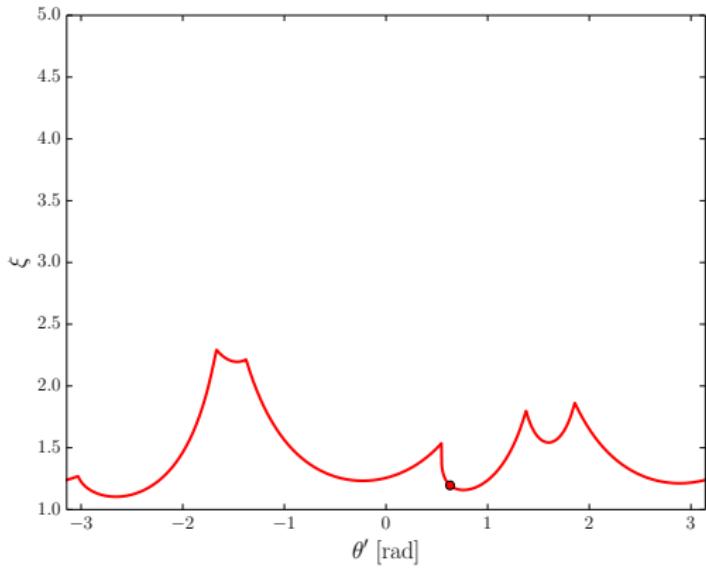
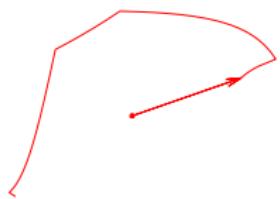
# Resolution of the un-transformed grain boundary



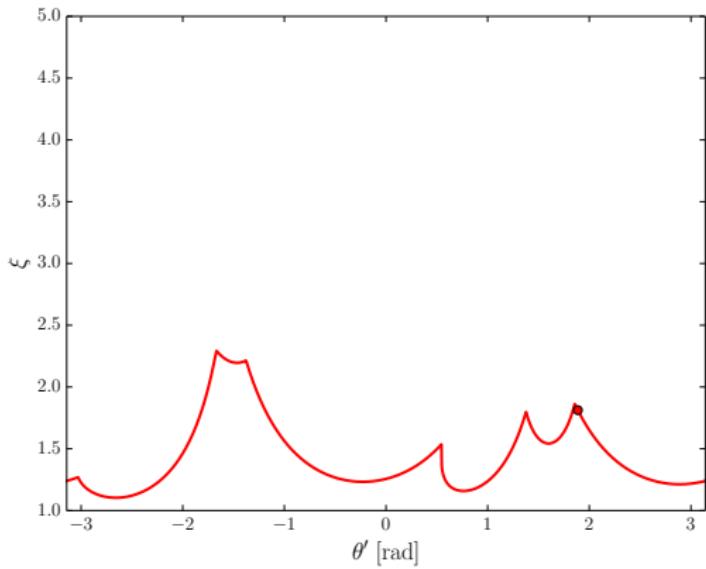
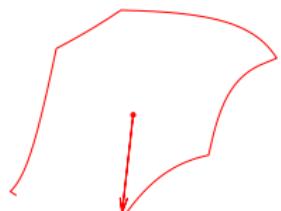
# Resolution of the un-transformed grain boundary



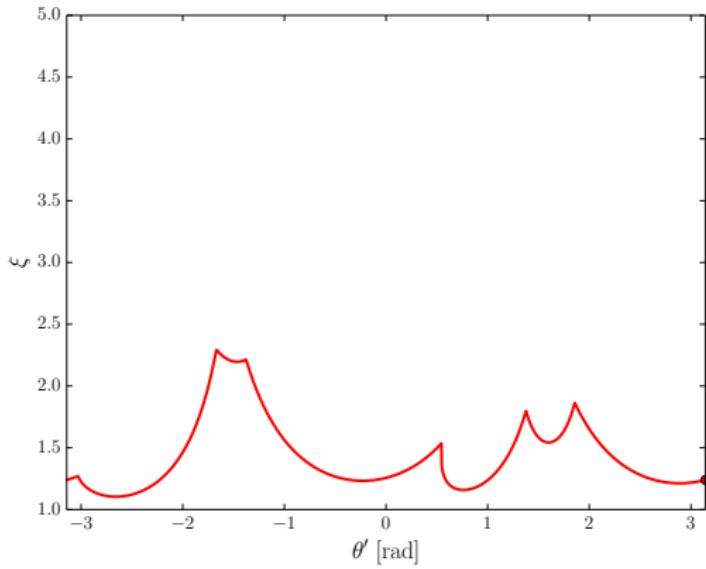
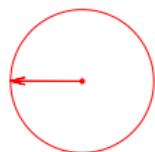
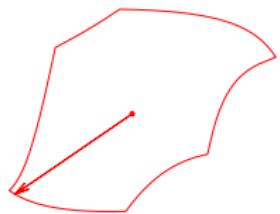
# Resolution of the un-transformed grain boundary



## Resolution of the un-transformed grain boundary

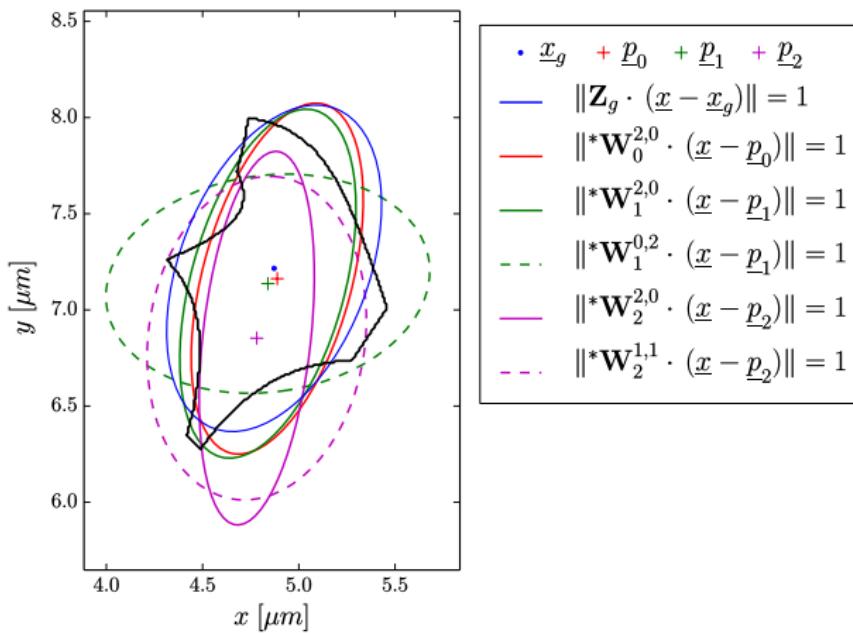


# Resolution of the un-transformed grain boundary



## Morphology characterization — Results

Minkowski valuations are defined as curvature integrals on the boundary of sets. They allow to quantify the different sorts of anisotropy of a grain.



## Next steps and objectives

- Extend the framework to cells  $\mathcal{C}_\alpha$  which are *not* radially convex at the nucleation point  $\underline{x}_\alpha$ ;

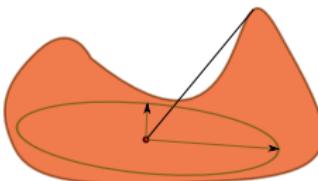


Figure :  $\mathcal{C}_\alpha$  not radially convex at  $\underline{x}_\alpha$

- Develop a simulation strategy of EGT parameters  $\{(\underline{x}_\alpha, \mathbf{Z}_\alpha) | \alpha = 1, \dots, n_\alpha\}$  for some target Minkowski valuation distributions and correlators;
- Extend the framework to three-dimensional tessellation models.

# Questions/Comments

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