

Magnetic materials for nano-scale sensors

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Objectives:

- | | |
|------------------------|---|
| Low power consumption | <input type="checkbox"/> Smaller |
| Enhanced sensitivity | <input type="checkbox"/> lower noise, no domain walls |
| Enhanced functionality | <input type="checkbox"/> novel couplings |

With strong reference to work by:

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Giesle Herzer (VAC/Morgan),
Francesco Stellacci (M.I.T.),

Mike McHenry (C.M.U.),
Manfred Wuttig (U. Md),
Moungi Bawendi (M.I.T.)

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Materials matrix suggested by Workshop

Outline

1) **Functionality:** beyond flux, Hall, MR...

2) **Materials suggestions (beyond single-phase thin films):**

- Not new compositions; new materials:
Nano-structured magnetic materials,
Immiscible materials
- Relevant length scales and
implications for sensor materials

3) **Processing opportunities**

- From Top-down lithography
of nano-magnetic structures
- to Bottom-up, Self assembled nano-magnetics,
immiscible systems; metallic...polymeric
Carbon-based (Fullerenes, nanotubes)

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1) Magnetic material functionality 2) Materials 3) Processing

The obvious:

Induction	<i>dB/dt</i> sensor
flux channel	Shield or focus <i>H</i>
Hall effect	<i>H</i> sensor
MR, GMR, tunnel MR	<i>H</i> sensor
Magneto-elastic	Accelerometer
Magneto-optic	Optical rotation (or inverse?)

Perhaps less obvious:

Magneto caloric
Magneto-electric
Spin-semiconductor hetero-effects

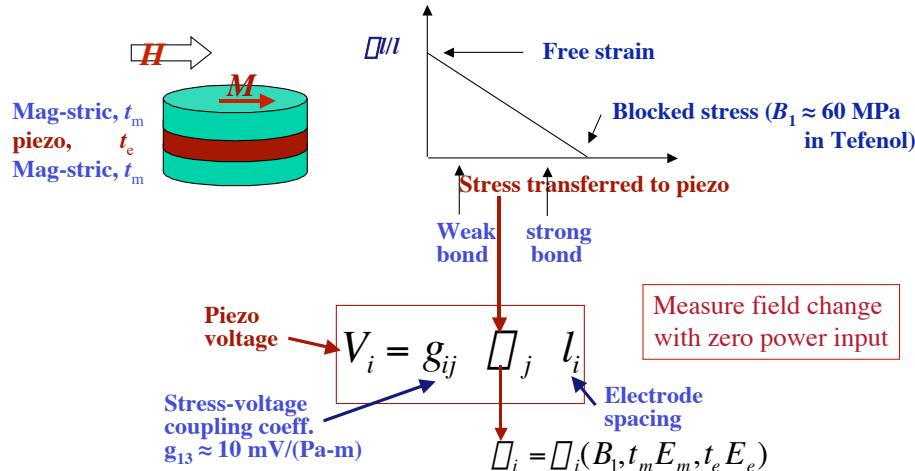
H sensor or accelerometer

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Magnetostriuctive/electro-active sensors



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Prior art: magnetostrictive/electro-active field sensors

Marconi Research d₃₁ sensor
Single magnetic layer,
AC bias field on magnetic layer,
Lynch and Gallantree. GEC Jnl. Of Res. **8**, (1990)

Spinix d₃₁ sensor
Two or more magnetic layers,
Device is **passive**, no power needed,
Y-Q. Li and R. C. O'Handley,
J. Appl. Sens. Tech. **17**, 10 (2000)

Penn State sensor
Two compressively prestressed Terfenol-D layers,
Device is **passive**, no power needed,
J. Ryu, A. V. Carazo, K. Uchino, and H. -E. Kim,
Jpn. J. Appl. Phys. **40**, 4948 (2001)

The top diagram shows a cross-section of the Marconi Research d₃₁ sensor with three layers: Magnetostrictive (top), AC Magnetic bias (middle), and Piezoelectric (bottom). The middle diagram shows a cross-section of the Spinix d₃₁ sensor with two layers: Magnetostrictive (top) and Piezoelectric (bottom), with an output terminal. The bottom diagram shows a cross-section of the Penn State sensor with two layers: Terfenol-D layers (inner) and PZT (outer), with a magnetic field arrow pointing upwards.

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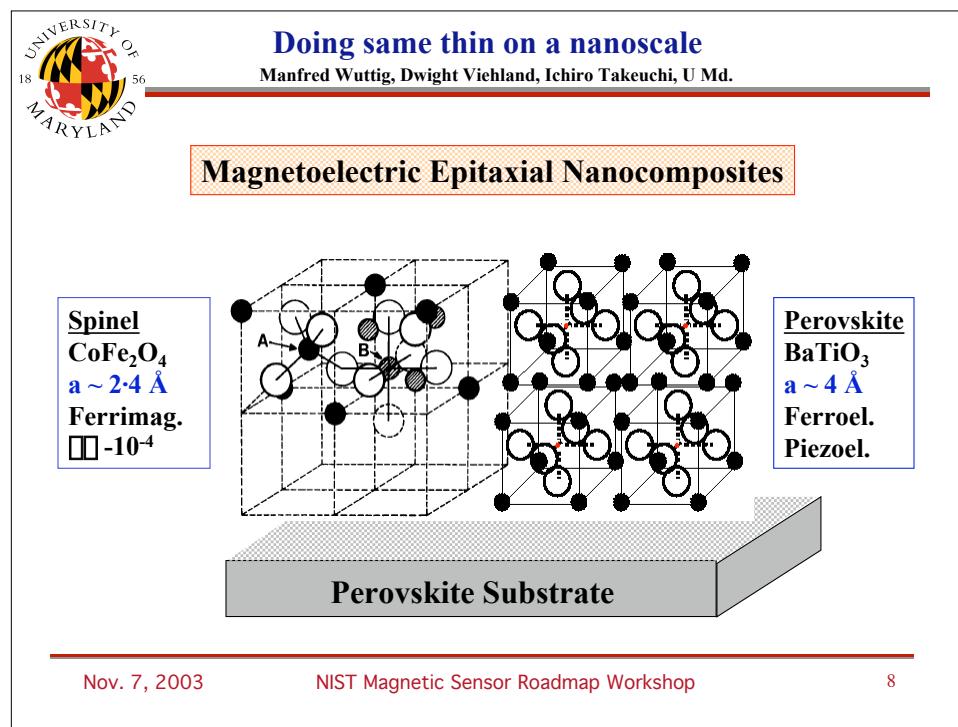
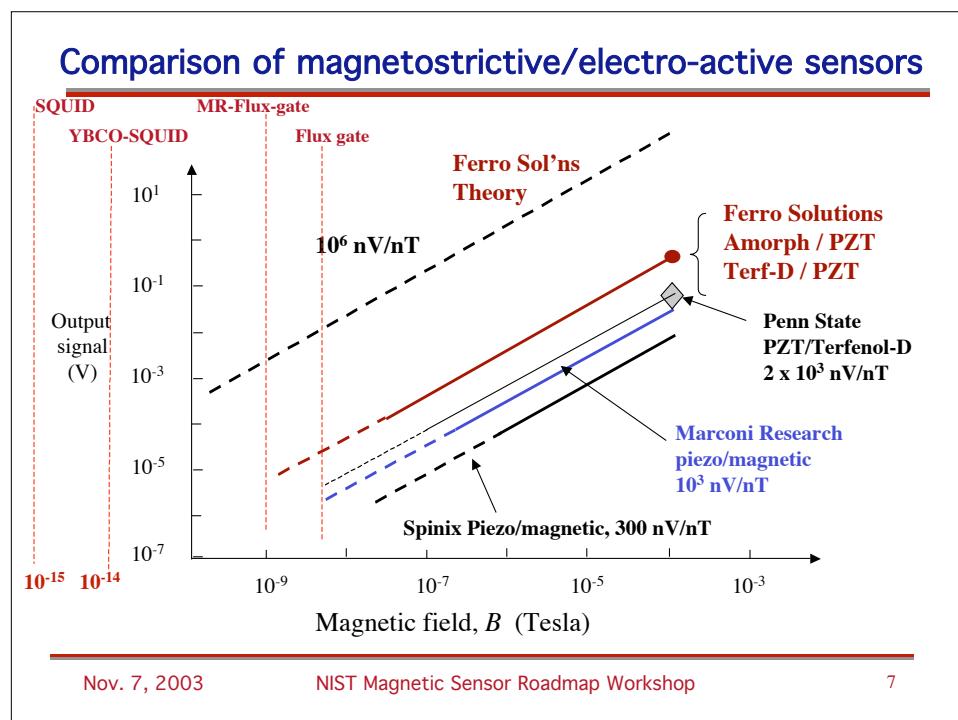
New magnetostrictive/electro-active field sensor

A photograph of the new magnetostrictive/electro-active field sensor, which is a rectangular device with a grid-like internal structure. A dimension line indicates a width of 1.5cm.

Ferro Solutions sensor
Two Terfenol-D layers, PZT
Device is **passive**, no power needed: sensitivity = 600 nV/nT
Can be AC biased: sensitivity = 6,000 nV/nT

J. K. Huang, R. C. O'Handley, Patent pending; SPIE Proc. March, 2003.

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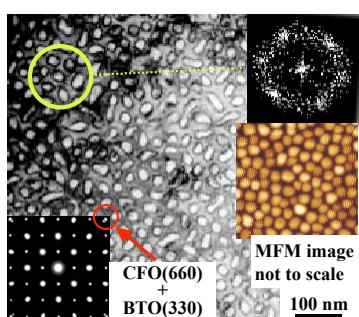
MULTIFERROICS FOR SMART STRUCTURES AND DEVICES

Manfred Wuttig, Dwight Viehland, Ichiro Takeuchi, U. Md

Promising Candidates

$\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$
studied in bulk form 1976

Other systems:
 $\text{CoFe}_2\text{O}_4\text{-FeBiO}_3$
 $\text{CoFe}_2\text{O}_4\text{-PbTiO}_3$



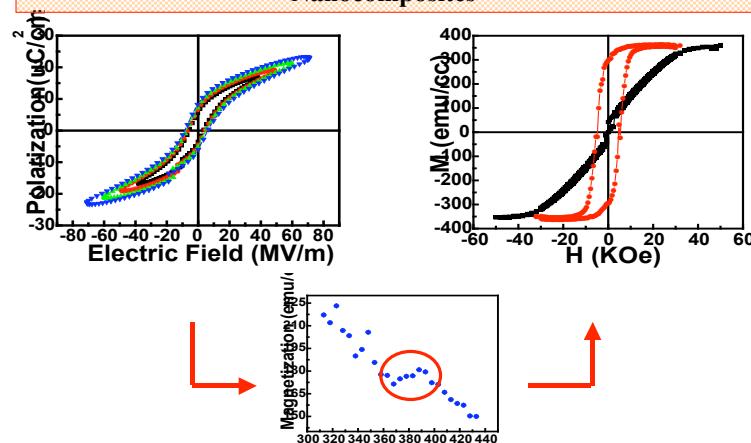
Self-assembled $\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$ hexagonal ME hetero-nano-structure

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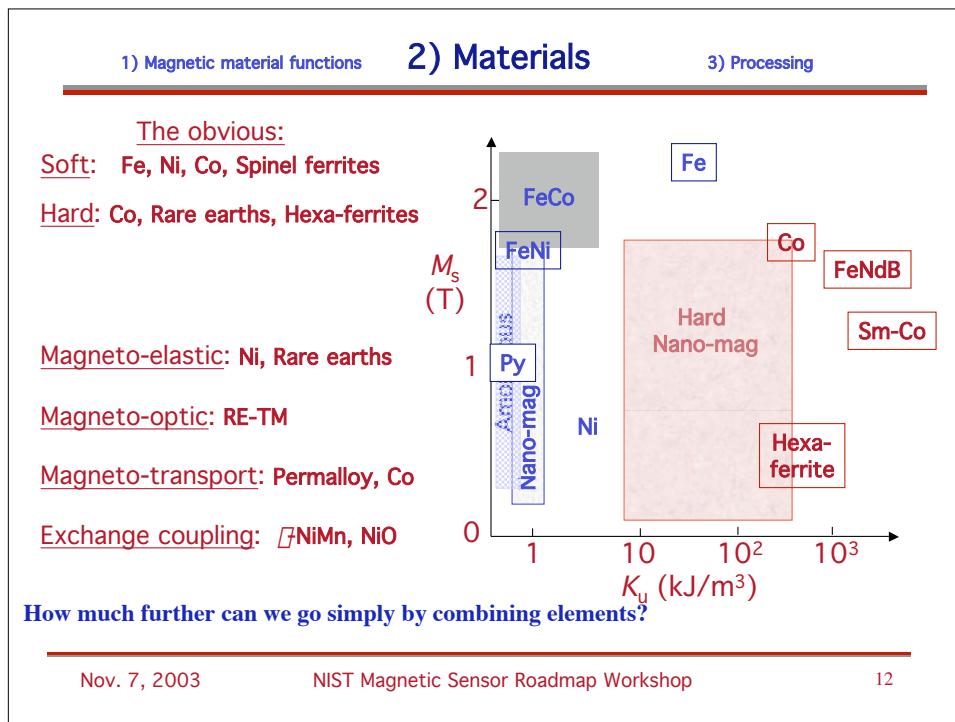
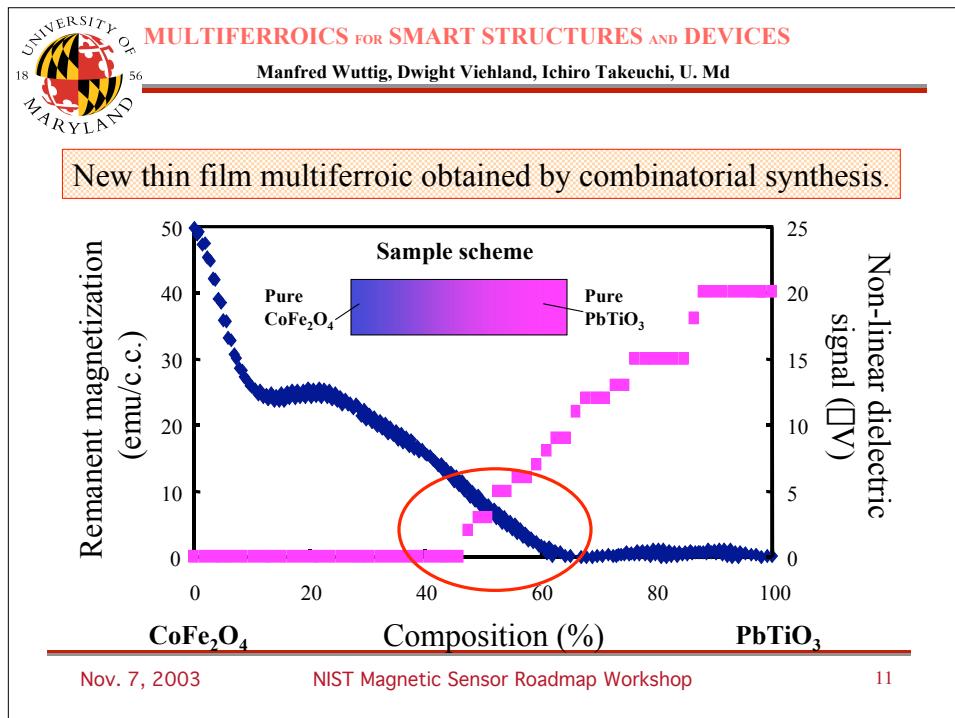
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Magnetoelectricity of Epitaxial $\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$ Nanocomposites

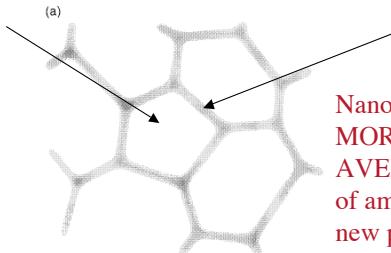


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Magnetic nano-structured materials

Crystalline properties



Amorphous properties

Nanostructured materials are
MORE THAN JUST
AVERAGE PROPERTIES
of amorphous and crystalline materials;
new property fields are accessible...

...and in nano-MAGNETIC materials,
several competing length scales in play
and can be exploited to advantage

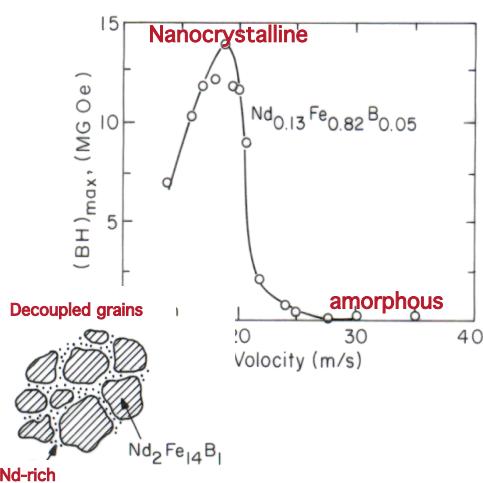
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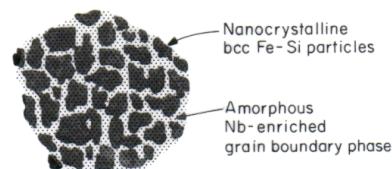
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Nano-magnetic materials - first pass

Rapidly solidified Nd-Fe-B
(Croat et al. '84)



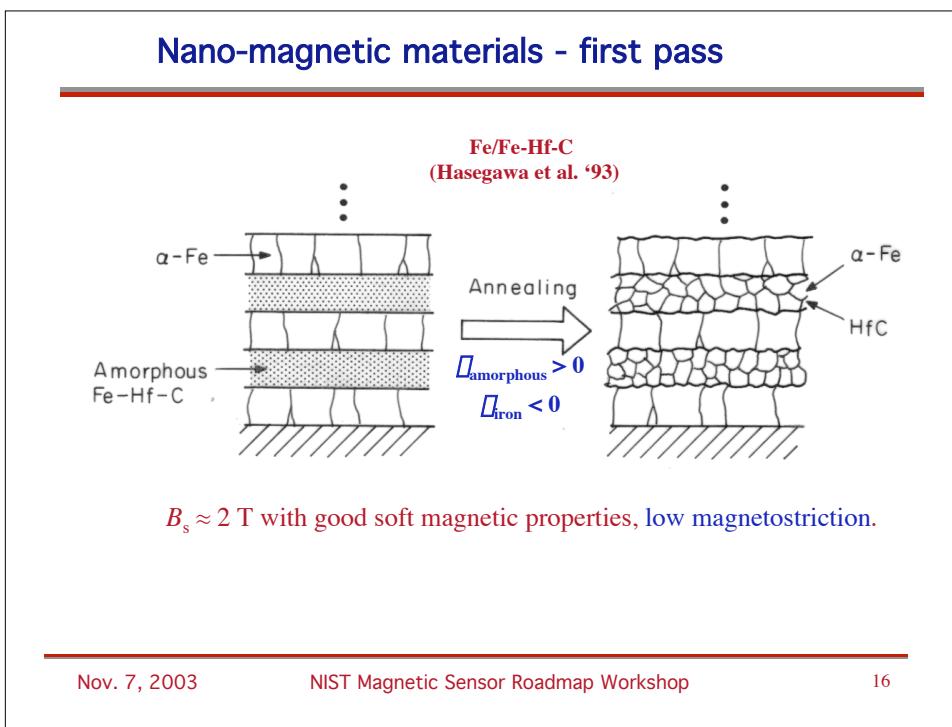
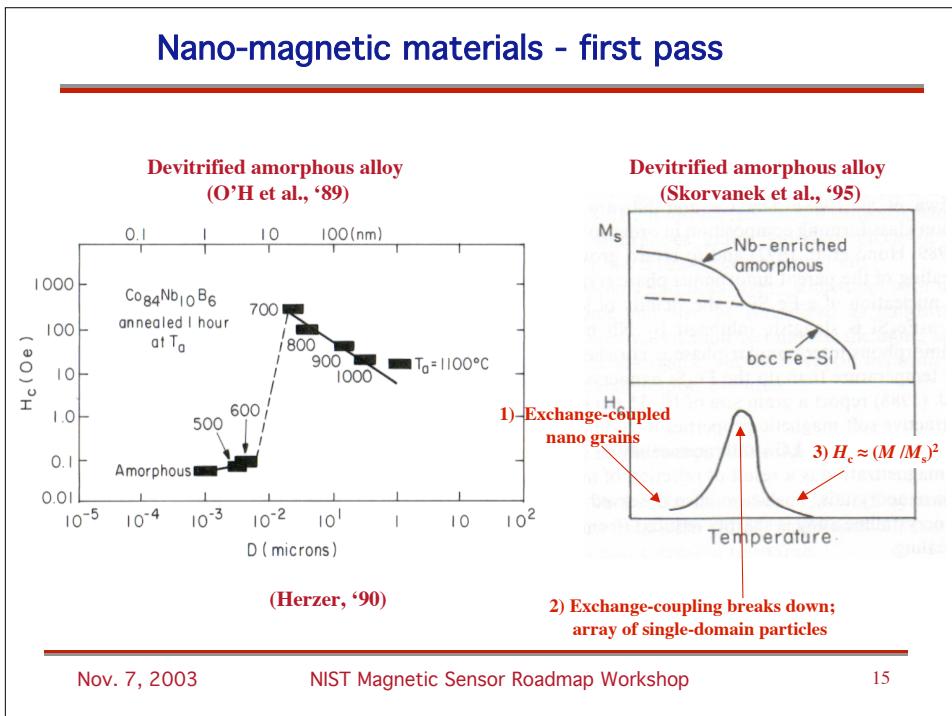
Amorphous $\text{Fe}_{74}\text{Si}_{15}\text{B}_7\text{Cu}_1\text{Nb}_3$
□ "Fe₃Si" nano-crystals
in amorphous matrix
(Yoshizawa, '88)



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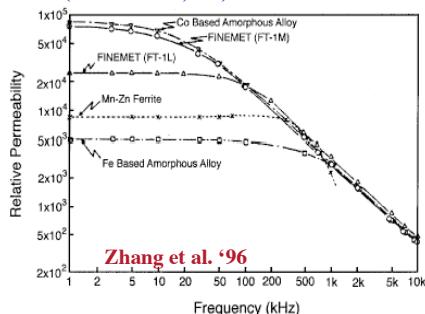
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Nano-magnetic materials - first pass

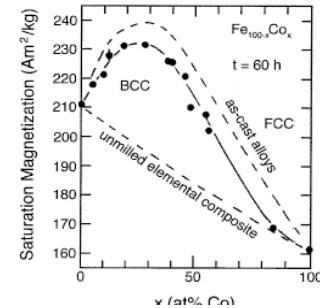
Crystallizing to β -Fe rather than Fe_3Si (DO_3) increases B_s ;

Amorphous $\text{Fe}_{88}\text{Zr}_7\text{B}_4\text{Cu}_1$
 \square β -Fe nanocrystals
 in amorphous matrix
 (Suzuki et al., '91)



Zhang et al. '96

Amorphous $\text{Fe}_{44}\text{Co}_{44}\text{Zr}_7\text{B}_4\text{Cu}_1$
 \square β -FeCo nanocrystals
 in amorphous matrix (HITPERM)
 (Willard et al., '98)



$B_s > 2$ T with permeability up to 4000
 Now in thin film form: HITPERM/ SiO_2
 McHenry group, C.M.U.

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Nano-magnetic material properties - first pass

(Machino et al. '95)

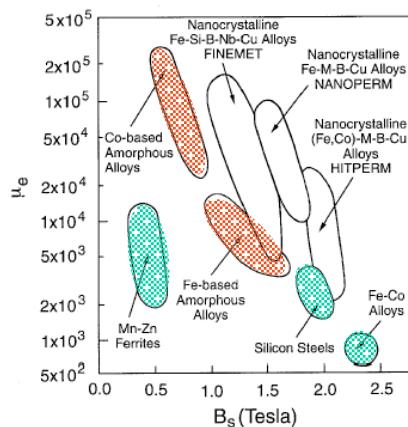


Fig. 1. Relationship between permeability, μ_r (at 1 kHz) and saturation polarization for soft magnetic materials (adapted from Ref. [193]).

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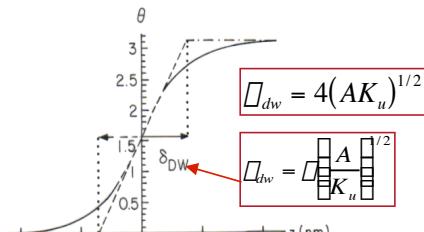
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1) Magnetic material functions 2) length scales 3) Processing

Prior nano-magnetic results due to
Changed *length scales* => Completely new properties

Bloch domain walls (180°) walls in materials



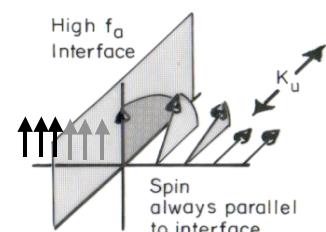
$$\square_{dw} = 4(AK_u)^{1/2}$$

$$\square_{dw} = \frac{\square}{K_u} A^{1/2}$$

	K_u (kJ/m³)	\square_{dw} (nm)	\square_{dw} (mJ/m²)
Ni	4.5	150	0.85
Fe	48	45	2.8
Co	410	15.5	8.1
Fe ₁₄ Nd ₂ B	5,000	4.4	28
Amorphous	1	1,000	0.1

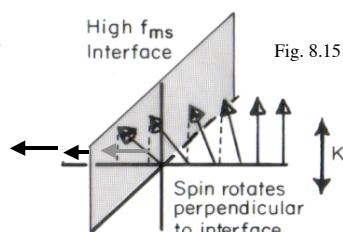
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Exchange lengths



High f_a Interface

Spin always parallel to interface



High f_{ms} Interface

Spin rotates perpendicular to interface

Fig. 8.15

M parallel to interface:

$$l_{ex}^{\parallel} = \frac{\square}{K_u} A^{1/2} = \frac{\square_{dw}}{\square}$$

M perpendicular to interface:

$$l_{ex}^{\perp} = \frac{\square}{K_u + 2\square M^2} A^{1/2}$$

can be $\ll \square_{dw}$

Energy pushing spins back to EA

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Single-domain particles

For what spherical radius is magnetostatic energy smaller than domain wall energy?

$$\frac{1}{3} \mu_0 M_s^2 \frac{4}{3} \pi r^3 \quad \square \square_{dw} \square \square r^2 = 4 \pi r^2 \sqrt{AK_u}$$

$$r_c \square 9 \frac{(AK_u)^{1/2}}{\mu_0 M_s^2}$$

(large K_u)

$r_c = 8 \text{ nm for Co, } 31 \text{ nm for Fe}_{14}\text{Nd}_2\text{B}$

Compare exchange energy cost to magnetostatic energy

$$r_c = \sqrt{\frac{9A}{\mu_0 M_s^2} \frac{2r_c}{a} n} \quad (\text{small } K_u)$$

r_c for Fe (small K_u) $\approx 25 \text{ nm}$
(instead of $\approx 5 \text{ nm}$)

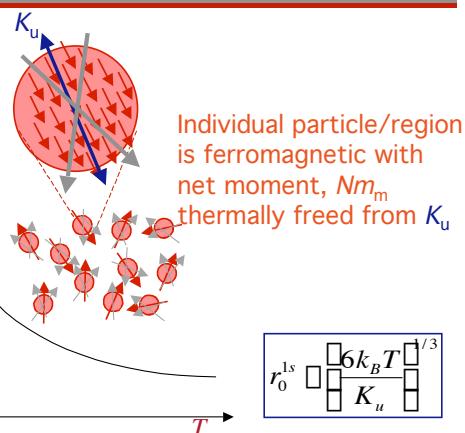
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Superparamagnetism

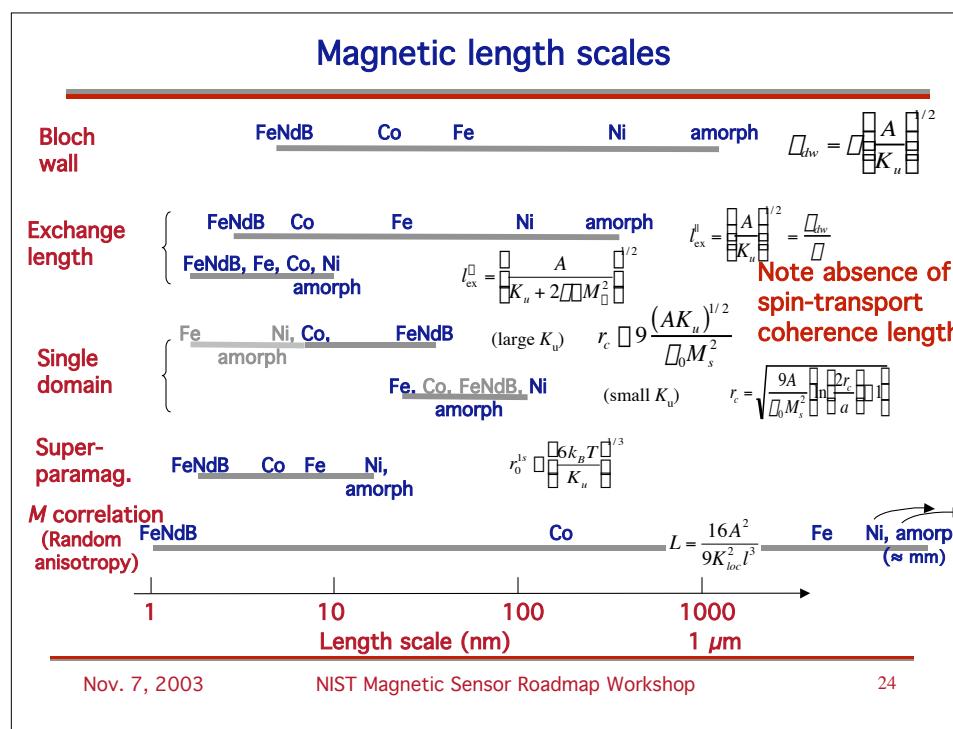
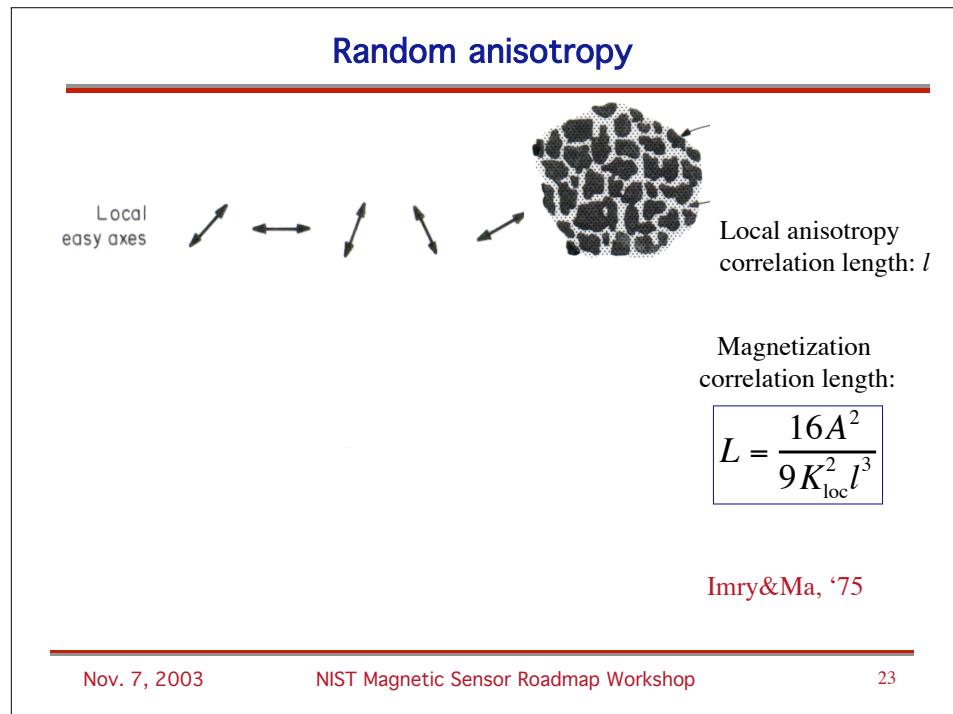
Material is an assembly of super-large (super) magnetic moments, Nm_m , that are NON-interacting (paramagnetic)

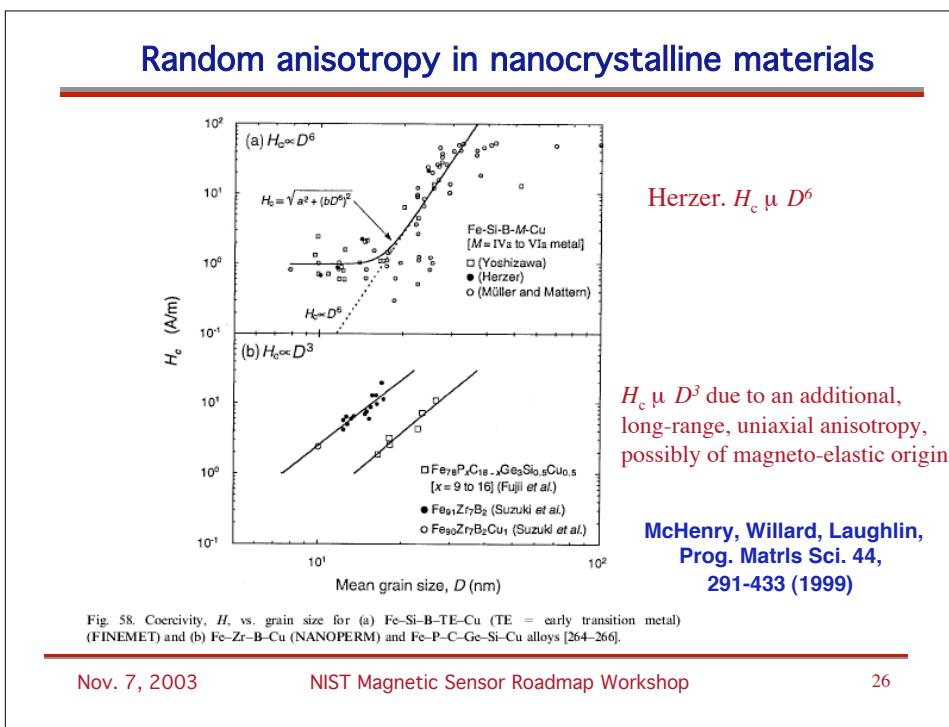
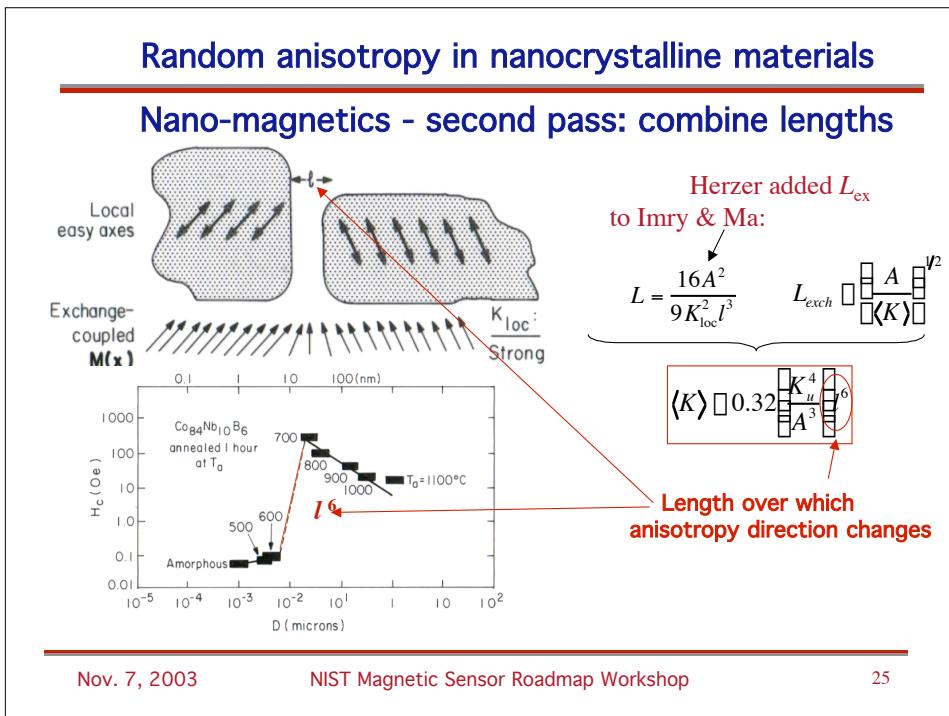


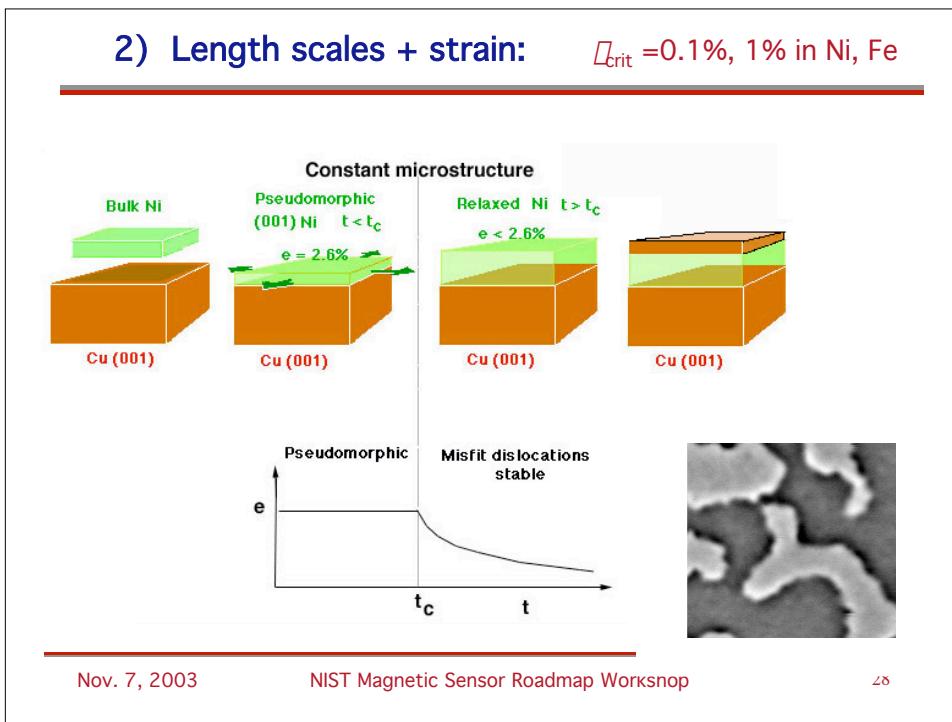
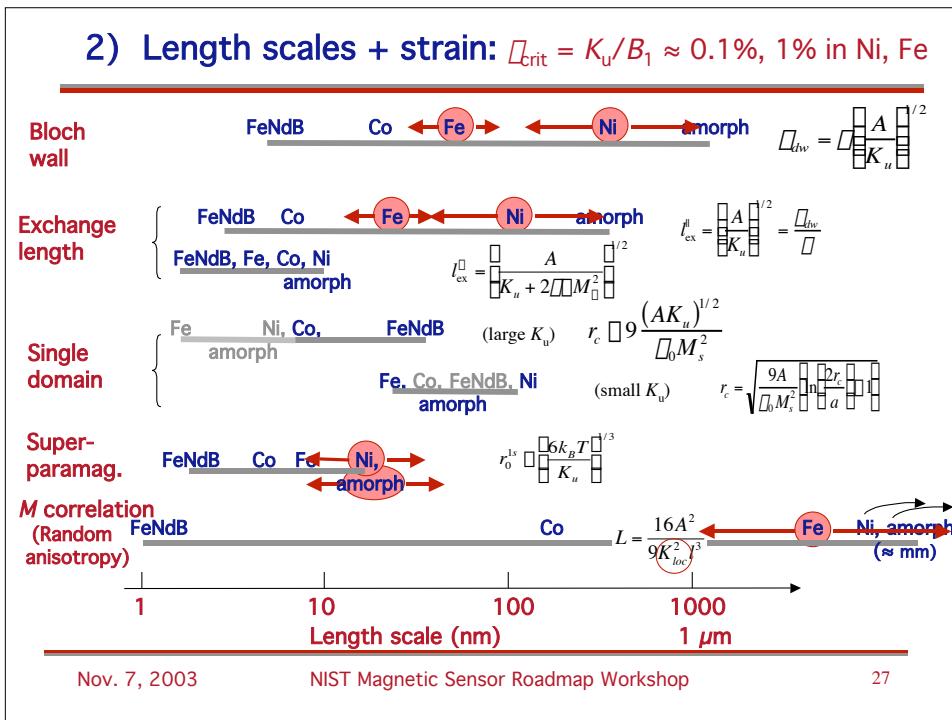
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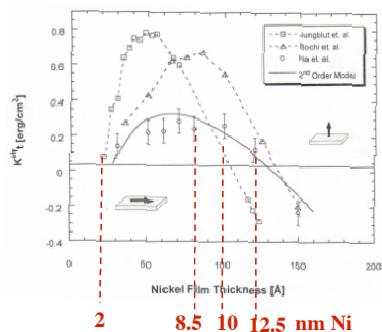
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2) Length scales + strain: $\Delta_{\text{crit}} = 0.1\%, 1\% \text{ in Ni, Fe}$



Films strongly perpendicular 3 - 10 nm, mixed above 10 nm

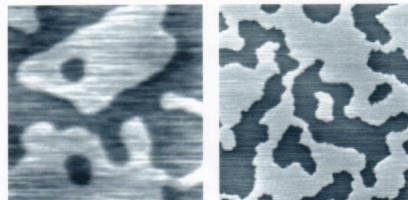


Fig. 1. Domain structure of Cu/2 nm Ni/Cu over a 12 micron square.



Fig. 2. Domain structure of Cu/8.5nm Ni/Cu over a 12 micron square.



Fig. 3. Domain structure of Cu/10nm Ni/Cu over a 12 micron square.

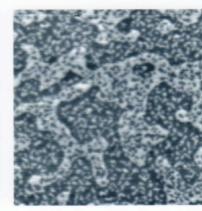


Fig. 4. Domain structure of Cu/12.5nm Ni/Cu over a 12 micron square.

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