# VOTCA-XTP EXCITON TRANSPORT SIMULATIONS

USER MANUAL



compiled from: 1.5-dev (e3b8d11)

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#### Disclamer

This manual is not complete. The best way to start using the software is to look at provided tutorials. The reference section is generated automatically from the source code, so please make sure that your software and manual versions match.

#### **Citations**

Development of this software depends on academic research grants. If you are using the package, please cite the following papers

[1] Microscopic simulations of charge transport in disordered organic semiconductors, Victor Rühle, Alexander Lukyanov, Falk May, Manuel Schrader, Thorsten Vehoff, James Kirkpatrick, Björn Baumeier and Denis Andrienko *J. Chem. Theor. Comp.* 7, 3335, 2011

[2] Versatile Object-oriented Toolkit for Coarse-graining Applications Victor Rühle, Christoph Junghans, Alexander Lukyanov, Kurt Kremer and Denis Andrienko J. Chem. Theor. Comp. 5, 3211, 2009

# Development

The core development is currently taking place at the Max Planck Institute for Polymer Research, Mainz, Germany and TU/e Eindhoven.

# Copyright

VOTCA-XTP is free software. The entire package is available under the Apache License. For details, check the LICENSE file in the source code. The VOTCA-XTP source code is available on our homepage, www.votca.org.

# **Contents**

1	Intro	Introduction		
2	The	oretical background	5	
	2.1	Workflow	5	
	2.2	Material morphology	5	
	2.3	Conjugated segments and rigid fragments	6	
	2.4	Neighbor list	8	
	2.5	Reorganization energy	8	
	2.0	2.5.1 Intramolecular reorganization energy	8	
		2.5.2 Outersphere reorganization energy	9	
	2.6		10	
	2.0	e e e e e e e e e e e e e e e e e e e	10	
			10	
			11	
			11 12	
	2.7	0)	14	
	2.7			
			15	
			16	
	2.0		18	
	2.8		19	
			19	
			19	
	2.0		20	
	2.9		20	
	2 4 0		21	
	2.10		21	
			22	
		O T	24	
	2.11		28	
			28	
			29	
			29	
		2.11.4 Spatial correlations of energetic disorder	30	
3	T	st and output Clas	33	
3	3.1	1	ээ 33	
	-		35 35	
	3.2			
	3.3		36	
	3.4		37 20	
	3.5	O Company of the comp	39 11	
	3.6		41	
	3.7	State file	42	

iv CONTENTS

4	Refe	erence		45
	4.1	Progra	ams	45
		4.1.1	xtp_testsuite	45
		4.1.2	xtp_update	45
		4.1.3	xtp_update_exciton	45
		4.1.4	xtp_basisset	45
		4.1.5	xtp_map	46
		4.1.6	xtp_run	46
		4.1.7	xtp_tools	46
		4.1.8	xtp_parallel	46
		4.1.9	xtp_dump	47
	4.2	Calcul	lators	47
		4.2.1	excitoncoupling	47
		4.2.2	gencube	48
		4.2.3	partialcharges	48
		4.2.4	qmanalyze	48
		4.2.5	kmclifetime	49
		4.2.6	kmcmultiple	49
		4.2.7	xeanalyze	50
		4.2.8	xeinternal	50
		4.2.9	xianalyze	50
		4.2.10	xjobwriter	51
		4.2.11	xneighborlist	51
			iimport	51
			pairdump	51
			xrates	51
		4.2.15	sandbox	52
		4.2.16	stateserver	52
			tdump	52
			vaverage	52
			egwbse	53
			iexcitoncl	53
		4.2.21	igwbse	53
		4.2.22	qmape	53
		4.2.23	xedft	54
		4.2.24	xidft	54
		4.2.25	xqmmm	55
		4.2.26	energy2xml	55
		4.2.27	integrals2xml	56
			occupations2xml	56
		4.2.29	pairs2xml	56
		4.2.30	rates2xml	56
			segments2xml	56
			trajectory2pdb	56
	4.3		non options	56
Bil	bliog	raphy		57

# Index

```
charge transfer rate, 19
    bimolecular, 19
    classical, 19
    semiclassical, 20
conjugated segment, 6
current, 29
    local, 29
diabatic states, 14
distributed multipoles, 11
electronic coupling, 14
    DFT, 16
    ZINDO, 18
hopping site, 6
kinetic Monte Carlo, 20
mobility, 29
neighbor list, 8
Pekar factor, 10
reorganization energy, 8
    intramolecular, 8
    outersphere, 9
rigid fragment, 7
site energy, 10
    distributed multipoles, 11
    external field, 10
    internal, 10
    polarization, 12
    spatial correlation, 30
Thole model, 12
transfer integral, see electronic coupling
```

2 INDEX

# Chapter 1

# Introduction

sec:introduction

11

12

13

15

19

20

23

26

Charge carrier dynamics in an organic semiconductor can often be described in terms of charge hopping between localized states. The hopping rates depend on electronic coupling elements, reorganization energies, and site energies, which vary as a function of position and orientation of the molecules. The purpose of the VOTCA-XTP package [1] is to simplify the workflow for charge transport simulations, provide a uniform error-control for the methods, flexible platform for their development, and eventually allow *in silico* prescreening of organic semiconductors for specific applications.

The toolkit is implemented using modular concepts introduced earlier in the Versatile Objectoriented Toolkit for Coarse-graining Applications (VOTCA) [2]. It contains different programs, which execute specific tasks implemented in calculators representing an individual step in the workflow. Figure 1.1 summarizes a typical chain of commands to perform a charge transport simulation: First, the VOTCA code structures are adapted to reading atomistic trajectories, mapping them onto conjugated segments and rigid fragments, and substituting (if needed) rigid fragments with the optimized copies (xtp\_map). The programs xtp\_run and xtp\_parallel (for heavy-duty tasks) are then used to calculate all bimolecular charge hopping rates (via precalculation of all required ingredients). Site energies (or energetic disorder) can be determined as a combination of internal (ionization potentials/electron affinities of single molecules) as well as electrostatic and polarization contributions within the molecular environment. The calculation of electronic coupling elements between conjugated segments from the corresponding molecular orbitals can be performed using a dimer-projection technique based on density-functional theory (DFT). This requires explicit calculations using quantum-chemistry software for which we provide interfaces to Gaussian, Turbomole, and NWChem. Alternatively, the molecular orbital overlap module calculates electronic coupling elements relying on the semi-empirical INDO Hamiltonian and molecular orbitals in the format provided by the Gaussian package.

The kinetic Monte Carlo module reads in the neighbor list, site coordinates, and hopping rates and performs charge dynamics simulations using either periodic boundary conditions or charge sources and sinks.

The toolkit is written as a combination of modular C++ code and scripts. The data transfer be-

tween programs is implemented via a state file (sql database), which is also used to restart simulations. Analysis functions and most of the calculation routines are encapsulated by using the
observer pattern [3] which allows the implementation of new functions as individual modules.
In the following chapter 2, we summarize the theoretical background of the workflow of charge
transport simulations and in particular its individual steps. Chapter 3 describes the structure and
content of input and output files, while a full reference of programs and calculators is available
in chapter 4. For a hands-on tutorial, the reader is referred to the VOTCA-XTP project page at
http://code.google.com/p/votca-xtp/.

#### Input files: conf.gro GROMACS trajectory topol.tpr GROMACS topology map.xml Mapping mapping and energies Converts and partitions atomistic GROMACS trajectory xtp\_map -t topol.tpr -c traj.xtc -s map.xml -f options for calculators state.sql Output files: state.sq sqlite3 database file for Neighbor list Indentifies close molecular pairs between which charge transfer data transfer between modules rates will be calculated xtp\_run -o options.xml -f state.sql -e neighborlist Site energies Calculates electrostatic and polarization contribution to site enerxtp\_run -o options.xml -f state.sql -e emultipole Internal site and reorganization energies Imports internal site energy (IP, EA) and reorganization energies for charging and discharging to state.sql xtp\_run -o options.xml -f state.sql -e einternal ZINDO DFT Transfer integrals Monomers with DFT Transfer integrals with ZINDO Calculate the relevant transport orbitals of monomers Calculate electronic coupling elements for all pairs in the neighbor list xtp\_parallel -o options.xml -f state.sql -e edft -j "write run' xtp\_run -o options.xml -f state.sql -e izindo One can choose between quantum-chemical Transfer integrals with DFT (computationally expensive) or semi-empirical Calculate electronic coupling elements for all pairs in (fast, but not always sufficiently accurate) evaluation of transfer integrals. the neighbor list xtp\_parallel -o options.xml -f state.sql -e idft -j "write run read" Outersphere reorganization energies Contribution to reorganization of surrounding molecules due to polarization. (optional for Marcus rates) xtp\_run -o options.xml -f state.sql -e outersphere Charge transfer rates Calculates rates for charge transfer among all pairs in the neighborlist xtp\_run -o options.xml -f state.sql -e rates Charge dynamics via kMC Hopping of charge carriers simulated via kinetic Monte Carlo xtp\_kmc\_run -o options.xml -f state.sql -e kmcmultiple

Figure 1.1: A practical workflow of charge transport simulations using VOTCA-XTP. The theoretical background of the individual steps is given in chapter 2. Chapter 3 describes the content of input and output files, while a full reference of programs and calculators is available in chapter 4.

Get help and list of options for a calculator: xtp\_run/xtp\_parallel/xtp\_kmc\_run -d neighborlist

Get list of available calculators: xtp\_run/xtp\_parallel/xtp\_kmc\_run -1

# Chapter 2

# . Theoretical background

sec:theory

#### 2.1 Workflow

#### sec:wok

- A typical workflow of charge transport simulations is depicted in figure 2.1. The first step is the simulation of an atomistic morphology, which is then partitioned on hopping sites. The
- coordinates of the hopping sites are used to construct a list of pairs of molecules, or neighbor list.



Figure 2.1: Workflow for microscopic simulations of charge transport.

- 45 For each pair an electronic coupling element, a reorganization energy, a driving force, and even-
- tually the hopping rate are evaluated. The neighbor list and hopping rates define a directed
- 47 graph. The corresponding master equation is solved using the kinetic Monte Carlo method,
- which allows to explicitly monitor the charge dynamics in the system as well as to calculate time
- or ensemble averages of occupation probabilities, charge fluxes, correlation functions, and field-
- 50 dependent mobilities.

# 2.2 Material morphology

sec:morpholo

fig:workflow

There is no generic recipe on how to predict a large-scale atomistically-resolved morphology of

an organic semiconductor. The required methods are system-specific: for ultra-pure crystals, for



Figure 2.2: The concept of conjugated segments and rigid fragments. Dashed lines indicate conjugated segments while colors denote rigid fragments. (a) Hexabenzocoronene: the  $\pi$ -conjugated system is both a rigid fragment and a conjugated segment. (b)  $\mathrm{Alq}_3$ : the Al atom and each ligand are rigid fragments while the whole molecule is a conjugated segment. (c) Polythiophene: each repeat unit is a rigid fragment. A conjugated segment consists of one or more rigid fragments. One molecule can have several conjugated segments.

fig:segment

example, density-functional methods can be used provided the crystal structure is known from experiment. For partially disordered organic semiconductors, however, system sizes much larger than a unit cell are required. Classical molecular dynamics or Monte Carlo techniques are then the methods of choice.

In molecular dynamics, atoms are represented by point masses which interact via empirical potentials prescribed by a force-field. Force-fields are parametrized for a limited set of compounds and their refinement is often required for new molecules. In particular, special attention shall be paid to torsion potentials between successive repeat units of conjugated polymers or between functional groups and the  $\pi$ -conjugated system. First-principles methods can be used to characterize the missing terms of the potential energy function.

Self-assembling materials, such as soluble oligomers, discotic liquid crystals, block copolymers, partially crystalline polymers, etc., are the most complicated to study. The morphology of such systems often has several characteristic length scales and can be kinetically arrested in a thermodynamically non-equilibrium state. For such systems, the time- and length-scales of atomistic simulations might be insufficient to equilibrate or sample desired morphologies. In this case, systematic coarse-graining can be used to enhance sampling [2]. Note that the coarse-grained representation must reflect the structure of the atomistic system and allow for back-mapping to the atomistic resolution.

Here we assume that the morphology is already known, that is we know how the topology and the coordinates of all atoms in the systems at a given time. VOTCA-XTP can read standard GROMACS topology files. Custom definitions of atomistic topology via XML files are also possible. Since the description of the atomistic topology is the first step in the charge transport simulations, it is important to follow simple conventions on how the system is partitioned on molecules, residues, and how atoms are named in the topology. Required input files are described in section atomistic topology.

# 2.3 Conjugated segments and rigid fragments

With the morphology at hand, the next step is partitioning the system on hopping sites, or conjugated segments, and calculating charge transfer rates between them. Physically intuitive arguments can be used for the partitioning, which reflects the localization of the wave function of a charge. For most organic semiconductors, the molecular architecture includes relatively rigid, planar  $\pi$ -conjugated systems, which we will refer to as rigid fragments. A conjugated segment can contain one or more of such rigid fragments, which are linked by bonded degrees of freedom.

The dynamics of these degrees of freedom evolves on timescales much slower than the frequency of the internal promoting mode. In some cases, e.g. glasses, it can be 'frozen' due to non-bonded interactions with the surrounding molecules.

To illustrate the concept of conjugated segments and rigid fragments, three representative molecular architectures are shown in figure 2.2. The first one is a typical discotic liquid crystal, hexabenzocoronene. It consists of a conjugated core to which side chains are attached to aid self-assembly and solution processing. In this case the orbitals localized on side chains do not participate in charge transport and the conjugated  $\pi$ -system is both, a rigid fragment and a conjugated segment. In  $\mathrm{Alq_3}$ , a metal-coordinated compound, a charge carrier is delocalized over all three ligands. Hence, the whole molecule is one conjugated segment. Individual ligands are relatively rigid, while energies of the order of  $k_\mathrm{B}T$  are sufficient to reorient them with respect to each other. Thus the Al atom and the three ligands are rigid fragments. In the case of a conjugated polymer, one molecule can consist of several conjugated segments, while each backbone repeat unit is a rigid fragment. Since the conjugation along the backbone can be broken due to large out-of-plane twists between two repeat units, an empirical criterion, based on the dihedral angle, can be used to partition the backbone on conjugated segments [4]. However, such intuitive partitioning is, to some extent, arbitrary and shall be validated by other methods [5–7].

After partitioning, an additional step is often required to remove bond length fluctuations introduced by molecular dynamics simulations, since they are already integrated out in the derivation of the rate expression. This is achieved by substituting respective molecular fragments with rigid, planar  $\pi$ -systems optimized using first-principles methods. Centers of mass and gyration tensors are used to align rigid fragments, though a custom definition of local axes is also possible. Such a procedure also minimizes discrepancies between the force-field and first-principles-based ground state geometries of conjugated segments, which might be important for calculations of electronic couplings, reorganization energies, and intramolecular driving forces.

To partition the system on hopping sites and substitute rigid fragments with the corresponding ground-state geometries xtp\_map program is used:

```
Mapping the GROMACS trajectory

| xtp_map -t topol.tpr -c traj.xtc -s map.xml -f state.sql
```

It reads in the GROMACS topology (topol.tpr) and trajectory (traj.xtc) files, definitions of conjugated segments and rigid fragments (map.xml) and outputs coordinates of conjugated segments (hopping sites) and rigid fragments (as provided in the MD trajectory and after rigidification) to the state file (state.sql). In order to do this, a mapping file map.xml has to be provided, which specifies the corresponding atoms in the different representations. After this step, all information (frame number, dimensions of the simulation box, etc) are stored in the state file and only this file is used for further calculations.

# Be careful!

sec:xtp\_du33p

VOTCA-XTP requires a wrapped trajectory for mapping the segments and fragments, so all molecules should be whole in the frame.

In order to visually check the mapping one can use either the tdump calculator or the programm xtp\_dump with the calculator trajectory2pdb.

```
Writing a mapped trajectroy with xtp_dump

xtp_dump -f state.sql -e trajectory2pdb
```

It reads in the state file created by xtp\_map and outputs two trajectory files corresponding to the original and rigidified atom coordinates. To check the mapping, it is useful to superimpose the three outputs (original atomistic, atomistic stored in the state file, and rigidified according to ground state geometries), e.g., with VMD.

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```
Writing a mapped trajectroy with tdump

| xtp_run -f state.sql -o options.xml -e tdump
```

131 132 It also reads in the state file but appends the coordinates to a pdb. file. So make sure to delete old 133 QM.pdb and MD.pdb if you want to create a new imagef

# 2.4 Neighbor list

A list of neighboring conjugated segments, or neighbor list, contains all pairs of conjugated segments for which coupling elements, reorganization energies, site energy differences, and rates are evaluated.

Two segments are added to this list if the distance between centers of mass of any of their rigid fragments is below a certain cutoff. This allows neighbors to be selected on a criterion of minimum distance of approach rather than center of mass distance, which is useful for molecules with anisotropic shapes.

The neighbor list can be generated from the atomistic trajectory by using the neighborlist calculator. This calculator requires a cutoff, which can be specified in the options.xml file. The list is saved to the state.sql file:

```
Generating a neighbor list

| xtp_run -o options.xml -f state.sql -e neighborlist
```

# 2.5 Reorganization energy

The reorganization energy  $\lambda_{ij}$  takes into account the change in nuclear (and dielectric) degrees of freedom as the charge moves from donor i to acceptor j. It has two contributions: intramolecular,  $\lambda_{ij}^{\text{int}}$ , which is due to reorganization of nuclear coordinates of the two molecules forming the charge transfer complex, and intermolecular (outersphere),  $\lambda_{ij}^{\text{out}}$ , which is due to the relaxation of the nuclear coordinates of the environment. In what follows we discuss how these contributions can be calculated.

#### 2.5.1 Intramolecular reorganization energy

If intramolecular vibrational modes of the two molecules are treated classically, the rearrangement of their nuclear coordinates after charge transfer results in the dissipation of the internal reorganization energy,  $\lambda_{ij}^{\text{int}}$ . It can be computed from four points on the potential energy surfaces (PES) of both molecules in neutral and charged states, as indicated in figure 2.3.

Adding the contributions due to discharging of molecule i and charging of molecule j yields [8]

$$\lambda_{ij}^{\rm int} = \lambda_i^{cn} + \lambda_j^{nc} = U_i^{nC} - U_i^{nN} + U_j^{cN} - U_j^{cC} \,. \tag{2.1} \quad \text{equ:lambdas}$$

Here  $U_i^{nC}$  is the internal energy of the neutral molecule i in the geometry of its charged state (small n denotes the state and capital C the geometry). Similarly,  $U_j^{cN}$  is the energy of the charged molecule j in the geometry of its neutral state. Note that the PES of the donor and acceptor are not identical for chemically different compounds or for conformers of the same molecule. In this case  $\lambda_i^{cn} \neq \lambda_j^{cn}$  and  $\lambda_i^{nc} \neq \lambda_j^{nc}$ . Thus  $\lambda_{ij}^{int}$  is a property of the charge transfer complex, and not of a single molecule.

Intramolecular reorganization energies for discharging ( $\lambda^{cn}$ ) and charging ( $\lambda^{nc}$ ) of a molecule need to be determined using quantum-chemistry and given in map.xml. The values are written to the state.sql using the calculator einternal (see also internal energy):



Figure 2.3: Potential energy surfaces of (a) donor and (b) acceptor in charged and neutral states. After the charge of the charge state both molecules relax their nuclear coordinates. If all vibrational modes are treated classically, the total internal reorganization energy and the internal energy difference of the electron transfer reaction are  $\lambda_{ij}^{\rm int} = \lambda_i^{en} + \lambda_j^{nc}$  and  $\Delta E_{ij}^{\rm int} = \Delta U_i - \Delta U_j$ , respectively.

fig:parabola

Intramolecular reorganization energies

xtp\_run -ooptions.xml -f state.sql -eeinternal

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#### 2.5.2 Outersphere reorganization energy

During the charge transfer reaction, also the molecules outside the charge transfer complex reorient and polarize in order to adjust for changes in electric potential, resulting in the outersphere contribution to the reorganization energy.  $\lambda_{ij}^{\text{out}}$  is particularly important if charge transfer occurs in a polarizable environment. Assuming that charge transfer is much slower than electronic polarization but much faster than nuclear rearrangement of the environment,  $\lambda_{ij}^{\text{out}}$  can be calculated from the electric displacement fields created by the charge transfer complex [9]

$$\lambda_{ij}^{\text{out}} = \frac{c_p}{2\epsilon_0} \int_{V^{\text{out}}} dV \left[ \vec{D}_I(\vec{r}) - \vec{D}_F(\vec{r}) \right]^2 , \qquad (2.2) \text{ equ:lambda\_outer}$$

where  $\epsilon_0$  is the the permittivity of free space,  $\vec{D}_{I,F}(\vec{r})$  are the electric displacement fields created by the charge transfer complex in the initial (charge on molecule i) and final (charge transferred to molecule j) states,  $V^{\text{out}}$  is the volume outside the complex, and  $c_p = \frac{1}{\epsilon_{\text{opt}}} - \frac{1}{\epsilon_s}$  is the Pekar factor, which is determined by the low  $(\epsilon_s)$  and high  $(\epsilon_{\text{opt}})$  frequency dielectric permittivities.

Eq. (2.2) can be simplified by assuming spherically symmetric charge distributions on molecules i and j with total charge e. Integration over the volume  $V^{\rm out}$  outside of the two spheres of radii  $R_i$  and  $R_j$  centered on molecules i and j leads to the classical Marcus expression for the outersphere reorganization energy

$$\lambda_{ij}^{\text{out}} = \frac{c_p e^2}{4\pi\epsilon_0} \left( \frac{1}{2R_i} + \frac{1}{2R_j} - \frac{1}{r_{ij}} \right) \,, \tag{2.3} \quad \text{equ:lambda\_outer2}$$

where  $r_{ij}$  is the molecular separation. While eq. (2.3) captures the main physics, e.g. predicts smaller outer-sphere reorganization energies (higher rates) for molecules at smaller separations, it often cannot provide quantitative estimates, since charge distributions are rarely spherically symmetric.

symmetric.

Alternatively, the displacement fields can be constructed using the atomic partial charges. The difference of the displacement fields at the position of an atom  $b_k$  outside the charge transfer complex (molecule  $k \neq i, j$ ) can be expressed as

$$\vec{D}_{I}(\vec{r}_{b_{k}}) - \vec{D}_{F}(\vec{r}_{b_{k}}) = \sum_{a_{i}} \frac{q_{a_{i}}^{c} - q_{a_{i}}^{n}}{4\pi} \frac{(\vec{r}_{b_{k}} - \vec{r}_{a_{i}})}{|\vec{r}_{b_{k}} - \vec{r}_{a_{i}}|^{3}} + \sum_{a_{i}} \frac{q_{a_{j}}^{n} - q_{a_{j}}^{c}}{4\pi} \frac{(\vec{r}_{b_{k}} - \vec{r}_{a_{j}})}{|\vec{r}_{b_{k}} - \vec{r}_{a_{j}}|^{3}},$$
(2.4)

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where  $q_{a_i}^n$  ( $q_{a_i}^c$ ) is the partial charge of atom a of the neutral (charged) molecule i in vacuum. The partial charges of neutral and charged molecules are obtained by fitting their values to reproduce the electrostatic potential of a single molecule (charged or neutral) in vacuum. Assuming a uniform density of atoms, the integration in eq. (2.2) can be rewritten as a density-weighted sum over all atoms excluding those of the charge transfer complex.

The remaining unknown needed to calculate  $\lambda_{ij}^{\text{out}}$  is the Pekar factor,  $c_p$ . In polar solvents  $\epsilon_s \gg \epsilon_{\text{opt}} \sim 1$  and  $c_p$  is of the order of 1. In most organic semiconductors, however, molecular orientations are fixed and therefore the low frequency dielectric permittivity is of the same order of magnitude as  $\epsilon_{\text{opt}}$ . Hence,  $c_p$  is small and its value is very sensitive to differences in the permittivities.

Outersphere reorganization energies for all pairs of molecules in the neighbor list can be computed from the atomistic trajectory by using the eoutersphere calculator.

Two methods can be used to compute  $\lambda_{ij}^{\text{out}}$ . The first method uses the atomistic partial charges of neutral and charged molecules from files specified in map.xml and eq. (2.2). The Pekar factor  $c_p$  and a cutoff radius based on molecular centers of mass have to be specified in the options.xml file.

If this method is computationally prohibitive,  $\lambda_{ij}^{\mathrm{out}}$  can be computed using eq. (2.3), which assumes spherical charge distributions on the molecules. In this case the radii of these spheres are specified in segments.xml, while the Pekar factor  $c_p$  is given in the options.xml file and no cutoff radius is needed.

The outer sphere reorganization energies are saved to the state.sql file:

```
Outersphere reorganization energy

| xtp_run -ooptions.xml -f state.sql -e outersphere
```

# 2.6 Site energies

A charge transfer reaction between molecules i and j is driven by the site energy difference,  $\Delta E_{ij} = E_i - E_j$ . Since the transfer rate,  $\omega_{ij}$ , depends exponentially on  $\Delta E_{ij}$  (see eq. (2.31)) it is important to compute its distribution as accurately as possible. The total site energy difference has contributions due to externally applied electric field, electrostatic interactions, polarization effects, and internal energy differences. In what follows we discuss how to estimate these contributions by making use of first-principles calculations and polarizable force-fields.

## 2.6.1 Externally applied electric field

The contribution to the total site energy difference due to an external electric field  $\vec{F}$  is given by  $\Delta E_{ij}^{\rm ext} = q \vec{F} \cdot \vec{r}_{ij}$ , where  $q = \pm e$  is the charge and  $\vec{r}_{ij} = \vec{r}_i - \vec{r}_j$  is a vector connecting molecules i and j. For typical distances between small molecules, which are of the order of  $1\,\mathrm{nm}$ , and moderate fields of  $F < 10^8\,\mathrm{V/m}$  this term is always smaller than  $0.1\,\mathrm{eV}$ .

#### 2.6.2 Internal energy

The contribution to the site energy difference due to different internal energies (see figure 2.3) can be written as

$$\Delta E_{ij}^{\rm int} = \Delta U_i - \Delta U_j = \left(U_i^{cC} - U_i^{nN}\right) - \left(U_j^{cC} - U_j^{nN}\right) \,, \tag{2.5}$$

where  $U_i^{cC(nN)}$  is the total energy of molecule i in the charged (neutral) state and geometry.  $\Delta U_i$  corresponds to the adiabatic ionization potential (or electron affinity) of molecule i, as shown in figure 2.3. For one-component systems and negligible conformational changes  $\Delta E_{ij}^{\rm int} = 0$ , while it is significant for donor-acceptor systems.

2.6. SITE ENERGIES 11

Internal energies determined using quantum-chemistry need to be specified in map.xml. The values are written to the state.sql using the calculator einternal (see also intramolecular reorganization energy):

```
Internal energies

| xtp_run -o options.xml -f state.sql -e einternal
```

# 2.6.3 Electrostatic interaction energy

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sec:distributed multipole

We represent the molecular charge density by choosing multiple expansion sites ("polar sites") per molecule in such a way as to accurately reproduce the molecular electrostatic potential (ESP), with a set of suitably chosen multipole moments  $\{Q_{lk}^a\}$  (in spherical-tensor notation) allocated to each site. The expression for the electrostatic interaction energy between two molecules A and B in the multi-point expansion includes an implicit sum over expansion sites  $a\epsilon A$  and  $b\epsilon B$ ,

$$U_{AB} = \sum_{a \in A} \sum_{b \in B} \hat{Q}_{l_1 k_1}^a T_{l_1 k_1 l_2 k_2}^{a,b} \hat{Q}_{l_2 k_2}^b \equiv \hat{Q}_{l_1 k_1}^a T_{l_1 k_1 l_2 k_2}^{a,b} \hat{Q}_{l_2 k_2}^b, \tag{2.6}$$

where we have used the Einstein sum convention for the site indices a and b on the right-hand side of the equation, in addition to the sum convention that is in place for the multipole-moment components  $t \equiv l_1k_1$  and  $u \equiv l_2k_2$ . The  $T_{l_1k_1l_2k_2}^{a,b}$  are tensors that mediate the interaction between a multipole component  $l_1k_1$  on site a with the moment  $l_2k_2$  on site b. If we include the molecular environment into a perturbative term W to enter in the single-molecule Hamiltonian, the above expression is exactly the first-order correction to the energy where the quantum-mechanical detail has been absorbed in classical multipole moments.

The are a number of strategies how to arrive at such a collection of *distributed multipoles*. They can be classified according to whether the multipoles are derived (a) from the electrostatic potential generated by the SCF charge density or (b) from a decomposition of the wavefunction itself. Here, we will only draft two of those approaches, CHELPG [10] from category (a) and DMA [11] from category (b).

The CHELPG (CHarges from ELectrostatic Potentials, Grid-based) method relies on performing a least-squares fit of atom-placed charges to reproduce the electrostatic potential as evaluated from the SCF density on a regularly spaced grid [10]. The fitted charges result from minimizing the Lagrangian function [12]

$$z(\{q_i\}) = \sum_{k=1}^{M} \left( \phi(\vec{r}_k) - \sum_{i=1}^{N} \frac{1}{4\pi\varepsilon_0} \frac{q_i}{|\vec{r}_i - \vec{r}_k|} \right) + \lambda \left( q_{\text{mol}} - \sum_{i=1}^{N} q_i \right), \tag{2.7}$$

with M grid points, N atomic sites, the set of atomic partial charges  $\{q_i\}$  and the SCF potential  $\phi$ . The Lagrange multiplier  $\lambda$  constrains the sum of the fitted charges to the molecular charge  $q_{\text{mol}}$ . The main difference from other fitting schemes [13] is the algorithm that selects the positions at which the potential is evaluated (we note that the choice of grid points can have substantial effects especially for bulky molecules). Clearly, the CHELPG method can be (and has been) extended to include higher atomic multipoles. It should be noted, however, how already the inclusion of atomic dipoles hardly improves the parametrization, and can in fact be harmful to its conformational stability.

The Distributed-Multipole-Analysis (DMA) approach [11, 14], developed by A. Stone, operates directly on the quantum-mechanical density matrix, expanded in terms of atom- and bond-centered Gaussian functions  $\chi_{\alpha} = R_{LK}(\vec{x} - \vec{s}_{\alpha}) \exp[-\zeta(\vec{x} - \vec{s}_{\alpha})^2]$ ,

$$\rho(\vec{x}) = \sum_{\alpha,\beta} \rho_{\alpha\beta} \chi_{\alpha}(\vec{x} - \vec{s}_{\alpha}) \chi_{\beta}(\vec{x} - \vec{s}_{\beta}). \tag{2.8}$$

The aim is to compute multipole moments according in a distributed fashion: If we use that the overlap product  $\chi_{\alpha}\chi_{\beta}$  of two Gaussian basis functions yields itself a Gaussian centered at  $\vec{P}=(\zeta_{\alpha}\vec{s}_{\alpha}+\zeta_{\beta}\vec{s}_{\beta})/(\zeta_{\alpha}+\zeta_{\beta})$ , it is possible to proceed in two steps: First, we compute the multipole moments associated with a specific summand in the density matrix, referred to the overlap center  $\vec{P}$ :

$$Q_{LK}[\vec{P}] = -\int R_{LK}(\vec{x} - \vec{P})\rho_{\alpha\beta}\chi_{\alpha}\chi_{\beta}d^3x.$$
 (2.9)

Second, we transfer the resulting  $Q_{lk}[\vec{P}]$  to the position  $\vec{S}$  of a polar site according to the rule [11]

$$Q_{nm}[\vec{S}] = \sum_{l=0}^{L} \sum_{k=-l}^{l} \left[ \binom{n+m}{l+k} \binom{n-m}{l-k} \right]^{1/2} R_{n-l,m-k}(\vec{S} - \vec{P}) \cdot Q_{lk}[\vec{P}].$$
 (2.10)

Note how this requires a rule for the choice of the expansion site to which the multipole moment should be transferred. In the near past [14], the nearest-site algorithm, which allocates the multipole moments to the site closest to the overlap center, was replaced for diffuse functions by an algorithm based on a sxtpth weighting function in conjunction with grid-based integration methods in order to decrease the basis-set dependence of the resulting set of distributed multipoles. One important advantage of the DMA approach over fitting algorithms such as CHELPG or Merz-Kollman (MK) is that higher-order moments can also be derived without too large an am-267 268 The 'mps' file format used by VOTCA for the definition of distributed multipoles (as well as 269 point polarizabilities, see subsequent section) is based on the GDMA punch format of A. Stone's 270 GDMA program [14] (the punch output file can be immediately plugged into VOTCA without 271 any conversions to be applied). Furthermore the log-file of different QM packages (currently 272 Gaussian, Turbomole and NWChem) may be fed into the log2mps tool, which will subsequently generate the appropriate mps-file

```
Read in ESP charges from a QM log file

| xtp_tools -ooptions.xml -e log2mps
```

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#### 2.6.4 Induction energy - the Thole model

If we in addition to the permanent set of multipole moments  $\{Q_t^a\}$  allow for induced moments  $\{\Delta Q_t^a\}$  and penalize their generation with a bilinear form (giving rise to a strictly positive contribution to the energy),

$$U_{\text{int}} = \frac{1}{2} \sum_{A} \Delta Q_t^a \eta_{tt'}^{aa'} \Delta Q_{t'}^{a'}, \tag{2.11}$$

it can be shown that the induction contribution to the site energy evaluates to an expression where all interactions between induced moments have cancelled out, and interactions between permanent and induced moments are scaled down by 1/2 [15]:

$$U_{pu} = \frac{1}{2} \sum_{A} \sum_{B>A} \left[ \Delta Q_t^a T_{tu}^{ab} Q_u^b + \Delta Q_t^b T_{tu}^{ab} Q_u^a \right]. \tag{2.12}$$

This term can be viewed as the second-order (induction) correction to the molecular interaction energy. The sets of  $\{Q_t^a\}$  are solved for self-consistently via

$$\Delta Q_t^a = -\sum_{B \neq A} \alpha_{tt'}^{aa'} T_{t'u}^{a'b} (Q_u^b + \Delta Q_u^b), \tag{2.13} \quad \text{equ:self_consistent_dQ}$$

2.6. SITE ENERGIES

where the polarizability tensors  $\alpha_{tt'}^{aa'}$  are given by the inverse of  $\eta_{tt'}^{aa'}$ .

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With eqs. 2.13 and 2.12 we have at hand expressions that allow us to compute the induction energy contribution to site energies in an iterative manner based on a set of molecular distributed multipoles  $\{Q_t^a\}$  and polarizabilities  $\{\alpha_{tt'}^{aa'}\}$ . We have drafted in the previous section how to obtain the former from a wavefunction decomposition or fitting scheme (GDMA, CHELPG). The  $\{\alpha_{tt'}^{aa'}\}$  can be derived formally (or rather: read off) from a perturbative expansion of the molecular interaction. In this work we make use of the Thole model [16, 17] as a semi-empirical approach to obtain the sought-after point polarizabilities in the local dipole approximation, that is,  $[\alpha_{tt'}^{aa'}] = \alpha_{tt'}^{aa'} \delta_{t\beta} \delta_{t'\beta} \delta_{aa'}$ , where  $\beta \epsilon \{x,y,z\}$  references the dipole-moment component.

The Thole model is based on a modified dipole-dipole interaction, which can be reformulated in terms of the interaction of smeared charge densities. This has been shown to be necessary due to the divergent head-to-tail dipole-dipole interaction that otherwise results at small interseparations on the Å scale [16–18]. Smearing out the charge distribution mimics the nature of the QM wavefunction, which effectively guards against this unphysical polarization catastrophe. Since the point dipoles however only react individually to the external field, any correlation effects as were still accounted for in the  $\{\alpha_{tt'}^{aa'}\}$  are lost, except perhaps those correlations that are due to the mere classical field interaction.

The smearing of the nuclei-centered multipole moments is obtained via a fractional charge density  $\rho_f(\vec{u})$  which should be normalized to unity and fall off rapidly as of a certain radius  $\vec{u} = \vec{u}(\vec{R})$ . The latter is related to the physical distance vector  $\vec{R}$  connecting two interacting sites via a linear scaling factor that takes into account the magnitude of the isotropic site polarizabilities  $\alpha^a$ . This isotropic fractional charge density gives rise to a modified potential

$$\phi(u) = -\frac{1}{4\pi\varepsilon_0} \int\limits_0^u 4\pi u' \rho(u') du' \tag{2.14}$$
equ:mod\_potential

We can relate the multipole interaction tensor  $T_{ij...}$  (this time in Cartesian coordinates) to the fractional charge density in two steps: First, we rewrite the tensor in terms of the scaled distance vector  $\vec{u}$ ,

$$T_{ij...}(\vec{R}) = f(\alpha^a \alpha^b) t_{ij...}(\vec{u}(\vec{R}, \alpha^a \alpha^b)), \tag{2.15}$$

where the specific form of  $f(\alpha^a \alpha^b)$  results from the choice of  $u(\vec{R}, \alpha^a \alpha^b)$ . Second, we demand that the smeared interaction tensor  $t_{ij...}$  is given as usual by the appropriate derivative of the potential in eq. 2.14,

$$t_{ij...}(\vec{u}) = -\partial_{u_i}\partial_{u_i}\dots\phi(\vec{u}). \tag{2.16}$$

It turns out that for a suitable choice of  $\rho_f(\vec{u})$ , the modified interaction tensors can be rewritten in such a way that powers n of the distance  $R=|\vec{R}|$  are damped with a damping function  $\lambda_n(\vec{u}(\vec{R}))$  [19].

There is a large number of fractional charge densities  $\rho_f(\vec{u})$  that have been tested for the purpose of giving best results for the molecular polarizability as well as interaction energies. Note how a great advantage of the Thole model is the exceptional transferability of the atomic polarizabilities to compounds not used for the fitting procedure [17]. In fact, for most organic molecules, a fixed set of atomic polarizabilities ( $\alpha_C = 1.334$ ,  $\alpha_H = 0.496$ ,  $\alpha_N = 1.073$ ,  $\alpha_O = 0.873$ ,  $\alpha_S = 2.926 \text{ Å}^3$ ) based on atomic elements yields satisfactory results.

VOTCA implements the Thole model with an exponentially-decaying fractional charge density

$$\rho(u) = \frac{3a}{4\pi} \exp(-au^3),\tag{2.17}$$

where  $\vec{u}(\vec{R}, \alpha^a \alpha^b) = \vec{R}/(\alpha^a \alpha^b)^{1/6}$  and the smearing exponent a = 0.39 (which can however be changed from the program options), as used in the AMOEBA force field [19].

Even though the Thole model performs very well for many organic compounds with only the above small set of element-based polarizabilities, conjugated molecules may require a more intricate parametrization. The simplest approach is to resort to scaled polarizabilities to match the effective molecular polarizable volume  $V \sim \alpha_x \alpha_y \alpha_z$  as predicted by QM calculations (here  $\alpha_x, \alpha_y, \alpha_z$  are the eigenvalues of the molecular polarizability tensor). The molpol tool assists with this task, it self-consistently calculates the Thole polarizability for an input mps-file and optimizes (if desired) the atomic polarizabilities in the above simple manner.

```
Generate Thole-type polarizabilites for a segment

| xtp_tools -o options.xml -e molpol
```

The electrostatic and induction contribution to the site energy is evaluated by the emultipole calculator. Atomistic partial charges for charged and neutral molecules are taken from mps-files (extended GDMA format) specified in map.xml. Note that, in order to speed up calculations for both methods, a cut-off radius (for the molecular centers of mass) can be given in options.xml. Threaded execution is advised.

```
Electrostatic and induction corrections

| xtp_run -o options.xml -f state.sql -e emultipole
```

Furthermore available are zmultipole, which extends emultipole to allow for an electrostatic buffer layer (loosely related to the z-buffer in OpenGL, hence the name) and anisotropic point polarizabilities. For the interaction energy of charged clusters of any user-defined composition (Frenkel states, CT states, ...), xqmultipole can be used.

```
Interaction energy of charged molecular clusters embedded in a molecular environment

xtp_parallel -ooptions.xml -f state.sql -e xqmultipole
```

2.7 Transfer integrals

The electronic transfer integral element  $J_{ij}$  entering the Marcus rates in eq. (2.31) is defined as

$$J_{ij} = \left\langle \phi_i \left| \hat{H} \right| \phi_j \right\rangle,$$
 (2.18) equ:T1

where  $\phi_i$  and  $\phi_j$  are diabatic wavefunctions, localized on molecule i and j respectively, participating in the charge transfer, and  $\hat{H}$  is the Hamiltonian of the formed dimer. Within the frozencore approximation, the usual choice for the diabatic wavefunctions  $\phi_i$  is the highest occupied molecular orbital (HOMO) in case of hole transport, and the lowest unoccupied molecular orbital (LUMO) in the case of electron transfer, while  $\hat{H}$  is an effective single particle Hamiltonian, e.g. Fock or Kohn-Sham operator of the dimer. As such,  $J_{ij}$  is a measure of the strength of the electronic coupling of the frontier orbitals of monomers mediated by the dimer interactions. Intrinsically, the transfer integral is very sensitive to the molecular arrangement, i.e. the distance and the mutual orientation of the molecules participating in charge transport. Since this arrangement can also be significantly influenced by static and/or dynamic disorder [20–24], it is essential to calculate  $J_{ij}$  explicitly for each hopping pair within a realistic morphology. Considering that the number of dimers for which eq. (2.18) has to be evaluated is proportional to the number of molecules times their coordination number, computationally efficient and at the same time quantitatively reliable schemes are required.

#### 2.7.1 Projection of monomer orbitals on dimer orbitals (DIPRO)

An approach for the determination of the transfer integral that can be used for any single-particle electronic structure method (Hartree-Fock, DFT, or semiempirical methods) is based on the projection of monomer orbitals on a manifold of explicitly calculated dimer orbitals. This dimer projection (DIPRO) technique including an assessment of computational parameters such as the basis set, exchange-correlation functionals, and convergence criteria is presented in detail in ref. [25]. A brief summary of the concept is given below.

We start from an effective Hamiltonian 1

$$\hat{H}^{\text{eff}} = \sum_{i} \epsilon_{i} \hat{a}_{i}^{\dagger} \hat{a}_{i} + \sum_{j \neq i} J_{ij} \hat{a}_{i}^{\dagger} \hat{a}_{j} + c.c. \tag{2.19}$$

where  $\hat{a}_i^{\dagger}$  and  $\hat{a}_i$  are the creation and annihilation operators for a charge carrier located at the molecular site i. The electron site energy is given by  $\epsilon_i$ , while  $J_{ij}$  is the transfer integral between two sites i and j. We label their frontier orbitals (HOMO for hole transfer, LUMO for electron transfer)  $\phi_i$  and  $\phi_j$ , respectively. Assuming that the frontier orbitals of a dimer (adiabatic energy surfaces) result exclusively from the interaction of the frontier orbitals of monomers, and consequently expand them in terms of  $\phi_i$  and  $\phi_j$ . The expansion coefficients,  $\bar{\mathbf{C}}$ , can be determined by solving the secular equation

$$(\mathbf{H} - E\mathbf{S})\bar{\mathbf{C}} = 0 \tag{2.20} \quad \text{equ:dipro_eq2}$$

where **H** and **S** are the Hamiltonian and overlap matrices of the system, respectively. These matrices can be written explicitly as

$$\mathbf{H} = \begin{pmatrix} e_i & H_{ij} \\ H_{ij}^* & e_j \end{pmatrix} \qquad \mathbf{S} = \begin{pmatrix} 1 & S_{ij} \\ S_{ij}^* & 1 \end{pmatrix} \tag{2.21} \quad \text{equidipro\_eq3}$$

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$$\begin{array}{ll} e_{i} = \langle \phi_{i} | \, \hat{H} \, | \phi_{i} \rangle & H_{ij} = \langle \phi_{i} | \, \hat{H} \, | \phi_{j} \rangle \\ e_{j} = \langle \phi_{j} | \, \hat{H} \, | \phi_{j} \rangle & S_{ij} = \langle \phi_{j} | \, \phi_{j} \rangle \end{array} \tag{2.22} \quad \text{equidipro\_eq4}$$

The matrix elements  $e_{i(j)}$ ,  $H_{ij}$ , and  $S_{ij}$  entering eq. (2.21) can be calculated via projections on the dimer orbitals (eigenfunctions of  $\hat{H}$ )  $\left\{\left|\phi_n^{\rm D}\right\rangle\right\}$  by inserting  $\hat{1}=\sum_n\left|\phi_n^{\rm D}\right\rangle\left\langle\phi_n^{\rm D}\right|$  twice. We exemplify this explicitly for  $H_{ij}$  in the following

$$H_{ij} = \sum_{nm} \left\langle \phi_i \mid \phi_n^{\rm D} \right\rangle \left\langle \phi_n^{\rm D} \mid \hat{H} \mid \phi_m^{\rm D} \right\rangle \left\langle \phi_m^{\rm D} \mid \phi_j \right\rangle. \tag{2.23} \quad \text{eq:dipro_eq16}$$

The Hamiltonian is diagonal in its eigenfunctions,  $\langle \phi_n^{\rm D} | \hat{H} | \phi_m^{\rm D} \rangle = E_n \delta_{nm}$ . Collecting the projections of the frontier orbitals  $|\phi_{i(j)}\rangle$  on the *n*-th dimer state  $(\bar{\mathbf{V}}_{(i)})_n = \langle \phi_i | \phi_n^{\rm D} \rangle$  and  $(\bar{\mathbf{V}}_{(j)})_n = \langle \phi_i | \phi_n^{\rm D} \rangle$  respectively, into vectors we obtain

$$H_{ij} = \bar{\mathbf{V}}_{(i)} \mathbf{E} \bar{\mathbf{V}}_{(i)}^{\dagger}.$$
 (2.24) eq:dipro\_eq17

What is left to do is determine these projections  $\bar{\mathbf{V}}_{(k)}$ . In all practical calculations the molecular orbitals are expanded in basis sets of either plane waves or of localized atomic orbitals  $|\varphi_{\alpha}\rangle$ . We will first consider the case that the calculations for the monomers are performed using a counterpoise basis set that is commonly used to deal with the basis set superposition error (BSSE). The basis set of atom-centered orbitals of a monomer is extended to the one of the dimer by adding the respective atomic orbitals at virtual coordinates of the second monomer. We can then write the respective expansions as

$$\left|\phi_{k}\right\rangle = \sum_{\alpha} \lambda_{\alpha}^{(k)} \left|\varphi_{\alpha}\right\rangle \qquad \text{and} \qquad \left|\phi_{n}^{\mathrm{D}}\right\rangle = \sum_{\alpha} D_{\alpha}^{(n)} \left|\varphi_{\alpha}\right\rangle \tag{2.25}$$

we use following notations: a - number,  $\bar{\mathbf{a}}$  - vector,  $\mathbf{A}$  - matrix,  $\hat{A}$  - operator

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where k = i, j. The projections can then be determined within this common basis set as

$$\left(\bar{\mathbf{V}}_{k}\right)_{n} = \left\langle \phi_{k} \mid \phi_{n}^{\mathrm{D}} \right\rangle = \sum_{\alpha} \lambda_{\alpha}^{(k)} \left\langle \alpha \mid \sum_{\beta} D_{\beta}^{(n)} \mid \beta \right\rangle = \bar{\mathbf{\lambda}}_{(k)}^{\dagger} \mathcal{S} \bar{\mathbf{D}}_{(n)} \tag{2.26}$$
eq:dipro\_eq19

where S is the overlap matrix of the atomic basis functions. This allows us to finally write the elements of the Hamiltonian and overlap matrices in eq. (2.21) as:

$$H_{ij} = \bar{\lambda}_{(i)}^{\dagger} \mathcal{S} \mathbf{D} \mathbf{E} \mathbf{D}^{\dagger} \mathcal{S}^{\dagger} \bar{\lambda}_{(j)}$$

$$S_{ij} = \bar{\lambda}_{(i)}^{\dagger} \mathcal{S} \mathbf{D} \mathbf{D}^{\dagger} \mathcal{S}^{\dagger} \bar{\lambda}_{(j)}$$
(2.27) eq:dipro\_eq20

Since the two monomer frontier orbitals that form the basis of this expansion are not orthogonal in general ( $S \neq 1$ ), it is necessary to transform eq. (2.20) into a standard eigenvalue problem of the form

$$\mathbf{H}^{\mathrm{eff}}\mathbf{\bar{C}}^{\mathrm{eff}} = E\mathbf{\bar{C}}^{\mathrm{eff}}$$
 (2.28) eq:dipro\_eq7

to make it correspond to eq. (2.19). According to Löwdin such a transformation can be achieved by

$$\mathbf{H}^{\text{eff}} = \mathbf{S}^{-1/2} \mathbf{H} \mathbf{S}^{-1/2}. \tag{2.29} \quad \text{eq:dipro_eq8}$$

This then yields an effective Hamiltonian matrix in an orthogonal basis, and its entries can directly be identified with the site energies  $\epsilon_i$  and transfer integrals  $J_{ij}$ :

$$\mathbf{H}^{\text{eff}} = \begin{pmatrix} e_i^{\text{eff}} & H_{ij}^{\text{eff}} \\ H_{ij}^{*,\text{eff}} & e_j^{\text{eff}} \end{pmatrix} = \begin{pmatrix} \epsilon_i & J_{ij} \\ J_{ij}^{*} & \epsilon_j \end{pmatrix}$$
(2.30) eq:dipro\_eq11

## 2.7.2 DFT-based transfer integrals using DIPRO

The calculation of one electronic coupling element based on DFT using the DIPRO method requires the overlap matrix of atomic orbitals  $\mathcal{S}$ , the expansion coefficients for monomer  $\bar{\lambda}_{(k)} = \{\lambda_{\alpha}^{(k)}\}$  and dimer orbitals  $\bar{\mathbf{D}}_{(n)} = \{D_{\alpha}^{(n)}\}$ , as well as the orbital energies  $E_n$  of the dimer are required as input. In practical situations, performing self-consistent quantum-chemical calculations for each individual monomer and one for the dimer to obtain this input data is extremely demanding. Several simplifications can be made to reduce the computational effort, such as using non-Counterpoise basis sets for the monomers (thereby decoupling the monomer calculations from the dimer run) and performing only a single SCF step in a dimer calculation starting from an initial guess formed from a superposition of monomer orbitals. This "noCP+noSCF" variant of DIPRO is shown in figure 2.4(a) and recommended for production runs. A detailed comparative study of the different variants can be found in [25].

The code currently contains supports evaluation of transfer integrals from quantum-chemical calculations performed with the Gaussian, Turbomole, and NWChem packages. The interfacing procedure consists of three main steps: generation of input files for monomers and dimers, performing the actual quantum-chemical calculations, and calculating the transfer integrals.

#### Monomer calculations

First, hopping sites and a neighbor list need to be generated from the atomistic topology and trajectory and written to the state.sql file. Then the parallel edft calculator manages the calculation of the monomer properties required for the determination of electronic coupling elements. Specifically, the individual steps it performs are:

1. Creation of a job file containing the list of molecules to be calculated with DFT

```
Writing job file for edft

xtp_parallel -ooptions.xml -f state.sql -e edft -j write
```



Figure 2.4: Schematics of the DIPRO method. (a) General workflow of the projection technique. (b) Strategy of the efficient noCP+noSCF implementation, in which the monomer calculations are performed independently form the dimer configurations (noCP), using the edft calculator. The dimer Hamiltonian is subsequently constructed based on an initial guess formed from monomer orbitals and only diagonalized once (noSCF) before the transfer integral is calculated by projection. This second step is performed by the idft calculator.

fig:dipro\_scheme

#### 2. Running of all jobs in job file

```
Running all edft jobs

| xtp_parallel -o options.xml -f state.sql -e edft -j run
```

which includes

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• creating the input files for the DFT calculation (using the package specified in options.xml) in the directory

```
OR_FILES/package/frame_F/mol_M
```

where F is the index of the frame in the trajectory, M is the index of a molecule in this frame,

- executing the DFT run, and
- after completion of this run, parsing the output (number of electrons, basis set, molecular orbital expansion coefficients), and saving it in compressed form to

OR\_FILES/molecules/frame\_F/molecule\_M.orb

#### Calculating the transfer integrals

After the momomer calculations have been completed successfully, the respective runs for dimers from the neighborlist can be performed using the parallel idft calculator, which manages the

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DFT runs for the hopping pairs and determines the coupling element using DIPRO. Again, several steps are required:

1. Creation of a job file containing the list of pairs to be calculated with DFT

```
Writing job file for idft

| xtp_parallel -o options.xml -f state.sql -e idft -j write
```

2. Running of all jobs in job file

```
Running all idft jobs

| xtp_parallel -o options.xml -f state.sql -e idft -j run
```

which includes

creating the input files (including the merged guess for a noSCF calculation, if requested) for the DFT calculation (using the package specified in options.xml) in the directory

```
OR_FILES/package/frame_F/pair_M_N
```

where M and N are the indices of the molecules in this pair,

- · executing the DFT run, and
- after completion of this run, parsing the output (number of electrons, basis set, molecular orbital expansion coefficients and energies, atomic orbital overlap matrix), and saving the pair information in compressed form to

```
OR_FILES/pairs/frame_F/pair_M_N.orb
```

- loading the monomer orbitals from the previously saved \*.orb files.
- calculating the coupling elements and write them to the job file
- 3. Reading the coupling elements from the job file and saving them to the state.sql file

```
Saving idft results from job file to state.sql

xtp_parallel -o options.xml -f state.sql -e idft -j read
```

## 2.7.3 ZINDO-based transfer integrals using MOO

An approximate method based on Zerner's Intermediate Neglect of Differential Overlap (ZINDO) has been described in Ref. [26]. This semiempirical method is substantially faster than first-principles approaches, since it avoids the self-consistent calculations on each individual monomer and dimer. This allows to construct the matrix elements of the ZINDO Hamiltonian of the dimer from the weighted overlap of molecular orbitals of the two monomers. Together with the introduction of rigid segments, only a single self-consistent calculation on one isolated conjugated segment is required. All relevant molecular overlaps can then be constructed from the obtained molecular orbitals.

The main advantage of the molecular orbital overlap (MOO) library is *fast* evaluation of electronic coupling elements. Note that MOO is based on the ZINDO Hamiltonian which has limited applicability. The general advice is to first compare the accuracy of the MOO method to the DFT-based calculations.

MOO can be used both in a standalone mode and as an izindo calculator of VOTCA-XTP.

Since MOO constructs the Fock operator of a dimer from the molecular orbitals of monomers by translating and rotating the orbitals of rigid fragments, the optimized geometry of all conjugated segments and the coefficients of the molecular orbitals are required as its input in addition to the state file (state.sql) with the neighbor list. Coordinates are stored in geometry.xyz

files with four columns, first being the atom type and the next three atom coordinates. This is a standard xyz format without a header. Note that the atom order in the geometry.xyz files can be different from that of the mapping files. The correspondence between the two is established in the map.xml file.



#### Be careful!

Izindo requires the specification of orbitals for hole and electron transport in map.xml. They are the HOMO and LUMO respectively and can be retrieved from the log file from which the zindo.orb file is generated. The number of alpha electrons is the HOMO, the LUMO is HOMO+1

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The calculated transfer integrals are immediately saved to the state.sql file.

```
Transfer integrals from izindo
```

xtp\_run -ooptions.xml -f state.sql -e izindo

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# 2.8 Charge transfer rate

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ec:rate:

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Charge transfer rates can be postulated based on intuitive physical considerations, as it is done in the Gaussian disorder models [20, 27–29]. Alternatively, charge transfer theories can be used to evaluate rates from quantum chemical calculations [1, 8, 25, 30–32]. In spite of being significantly more computationally demanding, the latter approach allows to link the chemical and electronic structure, as well as the morphology, to charge dynamics.

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sec:rate\_classica

The high temperature limit of classical charge transfer theory [33, 34] is often used as a tradeoff between theoretical rigor and computational complexity. It captures key parameters which influence charge transport while at the same time providing an analytical expression for the rate. Within this limit, the transfer rate for a charge to hop from a site i to a site j reads

$$\omega_{ij} = \frac{2\pi}{\hbar} \frac{J_{ij}^2}{\sqrt{4\pi \lambda_{ij} k_{\rm B} T}} \exp\left[-\frac{(\Delta E_{ij} - \lambda_{ij})^2}{4\lambda_{ij} k_{\rm B} T}\right],\tag{2.31}$$

where T is the temperature,  $\lambda_{ij} = \lambda_{ij}^{\text{int}} + \lambda_{ij}^{\text{out}}$  is the reorganization energy, which is a sum of intraand inter-molecular (outersphere) contributions,  $\Delta E_{ij}$  is the site-energy difference, or driving force, and  $J_{ij}$  is the electronic coupling element, or transfer integral.

## 484 **Z.8.**

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#### 2.8.2 Semi-classical bimolecular rate

Classical charge transfer rate

The main assumptions in eq. (2.31) are non-adiabaticity (small electronic coupling and charge transfer between two diabatic, non-interacting states), and harmonic promoting modes, which are treated classically. At ambient conditions, however, the intramolecular promoting mode, which roughly corresponds to C-C bond stretching, has a vibrational energy of  $\hbar\omega\approx 0.2\,\mathrm{eV}\gg k_\mathrm{B}T$  and should be treated quantum-mechanically. The outer-sphere (slow) mode has much lower vibrational energy than the intramolecular promoting mode, and therefore can be treated classically. The weak interaction between molecules also implies that each molecule has its own, practically independent, set of quantum mechanical degrees of freedom.

A more general, quantum-classical expression for a bimolecular multi-channel rate is derived in the Supporting Information of ref. [1] and has the following form

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$$\omega_{ij} = \frac{2\pi}{\hbar} \frac{|J_{ij}|^2}{\sqrt{4\pi\lambda_{ij}^{\text{out}}k_{\text{B}}T}} \sum_{l',m'=0}^{\infty} |\langle \chi_{i0}^c | \chi_{il'}^n \rangle|^2 |\langle \chi_{j0}^n | \chi_{jm'}^c \rangle|^2 \exp\left\{-\frac{\left[\Delta E_{ij} - \hbar(l'\omega_i^n + m'\omega_j^c) - \lambda_{ij}^{\text{out}}\right]^2}{4\lambda_{ij}^{\text{out}}k_{\text{B}}T}\right\}. \tag{2.32}$$

If the curvatures of intramolecular PES of charged and neutral states of a molecule are different, that is  $\omega_i^c \neq \omega_i^n$ , the corresponding reorganization energies,  $\lambda_i^{cn} = \frac{1}{2}[\omega_i^n(q_i^n-q_i^c)]^2$  and  $\lambda_i^{nc} = \frac{1}{2}[\omega_i^c(q_i^n-q_i^c)]^2$ , will also differ. In this case the Franck-Condon (FC) factors for discharging of molecule i read [35]

$$|\langle \chi_{i0}^{c} | \chi_{il'}^{n} \rangle|^{2} = \frac{2}{2^{l'} l'!} \frac{\sqrt{\omega_{i}^{c} \omega_{i}^{n}}}{(\omega_{i}^{c} + \omega_{i}^{n})} \exp\left(-|s_{i}|\right) \left[ \sum_{\substack{k=0\\k \text{ even}}}^{l'} \binom{l'}{k} \left(\frac{2\omega_{i}^{c}}{\omega_{i}^{c} + \omega_{i}^{n}}\right)^{k/2} \frac{k!}{(k/2)!} H_{l'-k} \left(\frac{s_{i}}{\sqrt{2S_{i}^{cn}}}\right) \right]^{2}, \tag{2.33}$$

where  $H_n(x)$  is a Hermite polynomial,  $s_i = 2\sqrt{\lambda_i^{nc}\lambda_i^{cn}}/\hbar(\omega_i^c + \omega_i^n)$ , and  $S_i^{cn} = \lambda_i^{cn}/\hbar\omega_i^c$ . The FC factors for charging of molecule j can be obtained by substituting  $(s_i, S_i^{cn}, \omega_i^c)$  with  $(-s_j, S_j^{nc}, \omega_j^n)$ . In order to evaluate the FC factors, the internal reorganization energy  $\lambda_i^{cn}$  can be computed from the intramolecular PES.

#### 2.8.3 Semi-classical rate

One can also use the quantum-classical rate with a common set of vibrational coordinates [9]

$$\omega_{ij} = \frac{2\pi}{\hbar} \frac{|J_{ij}|^2}{\sqrt{4\pi\lambda_{ij}^{\text{out}}k_{\text{B}}T}} \sum_{N=0}^{\infty} \frac{1}{N!} \left(\frac{\lambda_{ij}^{\text{int}}}{\hbar\omega^{\text{int}}}\right)^N \exp\left(-\frac{\lambda_{ij}^{\text{int}}}{\hbar\omega^{\text{int}}}\right) \exp\left\{-\frac{\left[\Delta E_{ij} - \hbar N\omega^{\text{int}} - \lambda_{ij}^{\text{out}}\right]^2}{4\lambda_{ij}^{\text{out}}k_{\text{B}}T}\right\}. \tag{2.34}$$

Numerical estimates show that if  $\lambda_{ij}^{\rm int} \approx \lambda_{ij}^{\rm out}$  and  $|\Delta E_{ij}| \ll \lambda_{ij}^{\rm out}$  the rates are similar to those of eq. (2.31). In general, there is no robust method to compute  $\lambda_{ij}^{\rm out}$  [36] and both reorganization energies are often assumed to be of the same order of magnitude. In this case the second condition also holds, unless there are large differences in electron affinities or ionization potentials of neighboring molecules, e.g. in donor-acceptor blends.

To calculate rates of the type specified in options.xml for all pairs in the neighbor list and to save them into the state.sql file, run the rates calculator. Note that all required ingredients (reorganization energies, transfer integrals, and site energies have to be calculated before).

```
Calculation of transfer rates

| xtp_run -ooptions.xml -f state.sql -e rates
```

# 2.9 Master equation

Having determined the list of conjugated segments (hopping sites) and charge transfer rates between them, the next task is to solve the master equation which describes the time evolution of the system

$$\frac{\partial P_{\alpha}}{\partial t} = \sum_{\beta} P_{\beta} \Omega_{\beta \alpha} - \sum_{\beta} P_{\alpha} \Omega_{\alpha \beta}, \tag{2.35}$$

where  $P_{\alpha}$  is the probability of the system to be in a state  $\alpha$  at time t and  $\Omega_{\alpha\beta}$  is the transition rate from state  $\alpha$  to state  $\beta$ . A state  $\alpha$  is specified by a set of site occupations,  $\{\alpha_i\}$ , where  $\alpha_i = 1(0)$  for an occupied (unoccupied) site i, and the matrix  $\hat{\Omega}$  can be constructed from rates  $\omega_{ij}$ .

The solution of eq. (2.35) is be obtained by using kinetic Monte Carlo (KMC) methods. KMC

The solution of eq. (2.35) is be obtained by using kinetic Monte Carlo (KMC) methods. KMC explicitly simulates the dynamics of charge carriers by constructing a Markov chain in state space and can find both stationary and transient solutions of the master equation. The main advantage of KMC is that only states with a direct link to the current state need to be considered at each step. Since these can be constructed solely from current site occupations, extensions to multiple charge carriers (without the mean-field approximation), site-occupation dependent rates (needed for the explicit treatment of Coulomb interactions), and different types of interacting particles and processes, are straightforward. To optimize memory usage and efficiency, a combination of the variable step size method [37] and the first reaction method is implemented.

To obtain the dynamics of charges using KMC, the program xtp\_kmc\_run executes a specific calculator after reading its options (charge carrier type, runtime, numer of carriers etc.) from options.xml.

```
KMC for a single carrier in periodic boundary conditions

| xtp_kmc_run -o options.xml -f state.sql -e kmcsingle
```

```
KMC for multiple carriers of the same type in periodic boundary conditions

| xtp_kmc_run -o options.xml -f state.sql -e kmcmultiple
```

## 2.9.1 Extrapolation to nondispersive mobilities

Predictions of charge-carrier mobilities in partially disordered semiconductors rely on charge transport simulations in systems which are only several nanometers thick. As a result, simulated charge transport might be dispersive for materials with large energetic disorder [38, 39] and simulated mobilities are system-size dependent. In time-of-flight (TOF) experiments, however, a typical sample thickness is in the micrometer range and transport is often nondispersive. In order to link simulation and experiment, one needs to extract the nondispersive mobility from simulations of small systems, where charge transport is dispersive at room temperature.

Such extrapolation is possible if the temperature dependence of the nondispersive mobility is known in a wide temperature range. For example, one can use analytical results derived for one-dimensional models [40–42]. The mobility-temperature dependence can then be parametrized by simulating charge transport at elevated temperatures, for which transport is nondispersive even for small system sizes. This dependence can then be used to extrapolate to the nondispersive mobility at room temperature [43].

For Alq<sub>3</sub>, the charge carrier mobility of a periodic system of 512 molecules was shown to be more than three orders of magnitude higher than the nondispersive mobility of an infinitely large system [43]. Furthermore, it was shown that the transition between the dispersive and nondispersive transport has a logarithmic dependence on the number of hopping sites N. Hence, a brute-force increase of the system size cannot resolve the problem for compounds with large energetic disorder  $\sigma$ , since N increases exponentially with  $\sigma^2$ .

#### 2.10 Stochastic Networks

The VOTCA package contains the functionality of generating large, amorphous charge transport networks ( $\sim 10^6$  molecules). This is done with a combined coarse-grained and stochastic approach. VOTCA::CSG is used to generate a coarse-grained morphology. The stochastic modeling of VOTCA::CTP allows to make a charge transfer network out of this morphology by reproduc-

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ing the neighbor list (connectivity), transfer integrals, correlated site energies. An overview is given in Figure 2.5.

Througout this section we will use two state files. One is the state file state\_ref.sql of the smaller reference system that can be generated as explained in the previous sections. The second one is the state file state\_cg.sql of the coarse-grained system, or the stochastic network, that can be parametrized as explained in this section.

When using the stochastic functionalities, please cite the corresponding work:

- 1. B. Baumeier, O. Stenzel, C. Poelking, D. Andrienko, and V. Schmidt: Stochastic modeling of molecular charge transport networks. Phys. Rev. B 86, 184202 (2012)
- 2. P. Kordt and D. Andrienko: Modeling of Spatially Correlated Energetic Disorder in Organic Semiconductors. Journal of Chemical Theory and Computation 12, 36–40 (2016)
  - 3. P. Kordt, J. J. M. van der Holst, M. Al Helwi, W. Kowalsky, F. May, A. Badinski, C. Lennartz, and D. Andrienko: Modeling of Organic Light Emitting Diodes: From Molecular to Device Properties. Advanced Functional Materials 25, 1955–1971 (2015)

#### 2.10.1 Coarse-grained morphology

The first step is to generate a coarse-grained morphology. In this example, it is done by mapping a DPBIC molecule (which consists of 103 atoms) to a single point, its center of mass and by using the iterative Boltzmann inversion (IBI) method. Starting point is a smaller reference morphology, generated with GROMACS. Using the command

977 g rdf -f traj.xtc -s topol.tpr

you can extract the radial distribution function g(r) of your reference topolgy, outputted into the file rdf.xvg. This file, together with table.xvg,grompp.mdp, topol.top, index.ndx and confout.gro form your reference data.

For VOTCA::CSG you need a **setting.xml** file:

```
582
583
    <!-- example for a non-bonded interaction entry -->
584
    <non-bonded>
585
     <!-- name of the interaction -->
     <name>IR-IR</name>
587
     <!-- types involved in this interaction -->
588
     <type1>IR</type1>
589
     <type2>IR</type2>
590
      <!-- dimension + grid spacing of tables for calculations -->
591
      <min>0.5</min>
592
     < max > 5.0 < / max >
593
     <step>0.01</step>
594
      <inverse>
595
       <!-- target distribution (rdf), just give gromacs rdf.xvg -->
596
       <target>rdf.xvg</target>
       <!-- update cycles -->
598
       <do_potential>1</do_potential>
       <!-- additional post processing of dU before added to potential -->
       <post_update>scale smooth</post_update>
601
       <post_update_options>
602
              <scale>0.5</scale> <!--Scale the potential before updating it -->
603
604
                     <iterations>2</iterations>
605
            </smooth>
606
       </post_update_options>
607
       <!-- additional post processing of U after dU added to potential -->
```

```
<post_add></post_add>
609
      <!-- name of the table for gromacs run -->
610
611
      <gromacs>
       table_IR_IR.xvg
612
      </gromacs>
613
614
     </inverse>
    </non-bonded>
615
616
    <!-- general options for inverse script -->
617
    <inverse>
618
     <!-- 300*0.00831451 gromacs units -->
619
     <kBT>2.49435300</kBT>
620
     <initial_configuration>maindir</initial_configuration>
621
     <!-- use gromacs as simulation program -->
622
     program>gromacs
     <!-- gromacs specific options -->
624
     <gromacs>
625
        <!-- trash so many frames at the beginning -->
626
        <equi_time>500</equi_time>
627
        <!-- grid for table*.xvg !-->
628
        <table_bins>0.001</table_bins>
629
        <!-- cut the potential at this value (gromacs bug) -->
630
        <pot_max>1000000</pot_max>
631
        <!-- extend the tables to this value -->
632
        <table_end>6.0</table_end>
633
     </gromacs>
634
     <!-- these files are copied for each new run -->
635
     <filelist>grompp.mdp topol.top index.ndx table.xvg</filelist>
636
     <!-- do so many iterations -->
637
     <iterations_max>500</iterations_max>
638
     <!-- Try to clean a bit -->
639
     <cleanlist>traj.xtc</cleanlist>
640
     <!-- ibm: inverse boltzmann imc: inverse monte carlo -->
641
     <method>ibi</method>
     <!-- write log to this file -->
     <log_file>inverse.log</log_file>
     <!-- write restart step to this file -->
     <restart_file>restart_points.log</restart_file>
646
     <!-- imc specific stuff -->
647
    </inverse>
648
   </cg>
649
```

You run IBI using the command

651

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csg\_inverse -options settings.xml

IBI intents to find a pontential U(r) that reproduces your radial distribution function. It is stored in the file **table\_IR\_IR.xvg** in our example.

With the interaction potential at hand, a large topology can be generated using molecular dynamics simulations for the coarse grained model. Starting point is a box with equally distributed points, with each point representing one molecule and with the number of points chosen such that the density of the reference system is reproduced. A small python script can generate the conf.gro to start from, here shown to obtain a  $50 \times 50 \times 120~\mathrm{nm}^3$  starting morphology.

```
660
661
662 from pylab import *
663 import numpy as np
664
665 lenX = 50
666 lenY = 50
```

```
lenZ = 120
   originalV = 4704.339
   originalN = 4000
   spacing = (originalV/originalN)**(1./3.)
671
  molecule = "DPBIC"
672
  resname = "IRI"
673
   atomname = "IR'
674
675
   newV = lenX*lenY*lenZ
676
   newN = int(newV/originalV*originalN)
677
678
        = int(lenX/spacing)+1
679
        = int(lenY/spacing)+1
680
   nY
        = int(lenZ/spacing)+1
681
   print "max. molecules in X direction: "+str(nX)
   print "max. molecules in Y direction: "+str(nY)
  print "max. molecules in Z direction: "+str(nZ)
  print "total number of molecules: "+str(newN)
686
687
  file = open("box.gro", "w")
688
  file.write(molecule+"\n")
689
  file.write(str(newN)+"\n")
690
691
692 atomnumber = 1
  for iX in range(nX):
693
       for iY in range(nY):
694
           for iZ in range(nZ):
695
               if(atomnumber > newN):
696
                    break
697
                posX = spacing*iX
698
                posY = spacing*iY
699
                posZ = spacing*iZ
700
                print >> file, "%5d%-5s%5s%5d%8.3f%8.3f%8.4f%8.4f%8.4f" % \
701
702
                (1, resname, atomname, 1, posX, posY, posZ, 0, 0, 0)
                atomnumber += 1
703
704
705
   file.write(" "+str(lenX)+" "+str(lenY)+" "+str(lenZ))
706
   file.close()
707
708
   print "Note: for some obscure reason VMD will not be able to read this file\
709
   properly unless you open it once in vi and save it."
719
```

Open the box.gro in vi and save it (:wq), afterwards you can have a look at it in VDM. Run your MD simulations using the mdrun command. In the end you can compare the radial distribution functions of your reference and coarse-grained system, as shown in figure 2.6(a) as an example.

#### 2.10.2 Charge transport network

To generate a charge transport network you first need a reference system with neighbor list, site energies and transfer integrals calculated and stored in a **state.sql** state file. The procedure for all these three properties is always the same: first analyze the reference data, and second import the analyzation files and reproduce the properties.

#### Neighbor list

 In the atomistic reference system molecules are connected if their two closest segments are below a certain cut-off radius. This finer picture of segments does not exist in the coarse-grained system, where each molecule is represented by a point. To mimick the neighbor list, the probability of two molecules to be connected is analyzed as a function of their center-of-mass distance. This can be done by using the *panalyze* calculator

```
Analyze the pair connectivity (neighborlist) in the reference system

| xtp_run -o options.xml -f state_ref.sql -e panalyze
```

with the options defined as follows:

#### options\_analyze.xml

The only parameter needed is the spacial resolution, i.e., the bin size for calculating the probabilities. The *panalyze* calculator outputs a file **panalyze.distanceprobability.out** with the respective probabilities. Now this file has to be imported into the coarse-grained state file

```
Import the reference pair connectivity (neighborlist) and reproduce it in stochastic network

| xtp_run -o options.xml -f state_cg.sql -e neighborlist
```

using the following options:

#### options\_import.xml

For testing purposes, you can run the *panalyze* calculator on your coarse-grained state file and compare the probability function to the reference. An example is shown in figure 2.6(b). You can also also look at the file **panalyze.distanceprobability.out** for both state files, which has the distribution of coordination numbers (number of neighbors) and its average in.

#### Site energies

Site energies in amorphous organic semiconductors are roughly Gaussian distributed, with the width of the Gaussian,  $\sigma$ , called the energetic disorder. However, there are correlations between sites if they are close enough to each other. The aim in this section is therefore to reproduce the correlated energetic landscape. The first step is to get a spatial correlation function as well as the mean energy and the energetic disorder from your reference state file:

```
Analyze the energy distribution and correlation in the reference system

xtp_run -o options.xml -f state_ref.sql -e eanalyze
```

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with the following options:

#### options\_analyze.xml

```
769
   <options>
770
            <eanalyze>
                     <resolution_sites>0.05</resolution_sites>
772
                     <resolution_pairs>0.05</resolution_pairs>
773
                     <re>olution_space>0.3</resolution_space>
774
                     <states>1,-1</states> <!-- +1 for hole transport, -1 for electron transport --
775
                     <distancemode>centreofmass</distancemode>
776
            </eanalyze>
777
   </options>
778
```

The first three parameters determine bin sizes, then you can choose to look at hole and/or electron energy. The keyword *centreofmass* means, that the correlation function is calculated as a function of the centre-of-mass distance of molecules and not as a function of their nearest segments. For the stochastic simulations you always have to use the *centreofmass* mode!

The output files of this calculator that we need are **eanalyze.sitecorr\_e.out** (for electrons) and **eanalyze.sitecorr\_h.out** (for holes). In the second line of this file, you find mean and sigma of the energy distribution, as well as the mean of the static energies (without induction):

```
788 # EANALYZE: SPATIAL SITE-ENERGY CORRELATION
789 # AVG -0.4412655 STD 0.1739638 MIN_R 0.8365040 MAX_R 14.4771496 AVGESTATIC
790 -0.4730655
```

These values have to be inserted manually into the options file for importing to the coarse-grained system (see below). Apart from that, the file contains the spatial correlation function.

You generate energies following this distribution and correlation by using the eimport calculator

```
Import the energy distribution and correlation and reproduce it in stochastic network

| xtp_run -o options.xml -f state_ref.sql -e eimport
```

with the options:

#### options\_import.xml

```
801
   <options>
802
            <eimport>
803
                     <probabilityfile_h>reference/eanalyze.sitecorr_h.out</probabilityfile_h>
804
                     <sigma_h>0.1763163</sigma_h>
805
                     <avgestatic_h>-0.5913265</avgestatic_h>
806
                     obabilityfile_e>reference/eanalyze.sitecorr_e.out/probabilityfile_e>
                     <sigma_e>0.1739638</sigma_e>
808
809
                     <avgestatic_e>-0.4730655</avgestatic_e>
                     <cutoff>8.5</cutoff>
810
                     <seed>1</seed>
811
            </eimport>
812
   </options>
813
```

The *cutoff* keyword can be used to read in the correlation function only up to a certain distance, which can be useful if larger distances yield unphysical results.

#### Transfer Integrals

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The last ingredient reproduced by the stochastic approach are transfer integrals J. The idea is that  $\log_{10}(J^2/\,\mathrm{eV}^2)$  is roughly Gaussian distributed, with mean and error of the distribution varying with distance (see figure 2.6 (d)). Use the calculator

```
Analyze the distance-depend distribution of transfer integrals in the reference system

| xtp_run -o options.xml -f state_ref.sql -e ianalyze
```

with options

#### options\_analyze.xml

That will generate the files **ianalyze.ispatial\_e.out** and **ianalyze.ispatial\_h.out**, which contain means and errors as a function of centre-of-mass distance.

Now use the *iimport* calculator to generate transfer integrals in the coarse grained state file, following the same statistics.

```
Import distance dependent distribution of transfer integrals and reproduce in stochastic network

| xtp_run -ooptions.xml -f state_cg.sql -e iimport
```

#### options\_import.xml

```
844
    <options>
845
            <iimport>
846
                     <TI_tag></TI_tag>
847
                     <TI_file></TI_file>
                     <idft_jobs_file></idft_jobs_file>
                     <probabilityfile_h>reference/ianalyze.ispatial_h.out</probabilityfile_h>
850
                     <probabilityfile_e>reference/ianalyze.ispatial_e.out</probabilityfile_e>
851
            </iimport>
852
   </options>
853
```

#### 855 einternal

Run the einternal calculator, just as you do it for the reference system.

#### 857 Rates

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If you followed the steps is this section, you have everything at hand to calculate charge transfer
 rates for the coarse grained system from the stochastic ingredients:

```
Calculate rates in the stocchastic network

| xtp_run -o options.xml -f state_cg.sql -e rates
```

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Options are the same as for the reference file. You can check the result by comparing rates from your reference to the coarse-grained system, see figure 2.6(e) for an example. The resulting charge transport network can be used for kinetic Monte Carlo simulations with VOTCA. If everything goes well, mobilities for both systems should agree, as shown in figure 2.6(f).

# 2.11 Macroscopic observables

Spatial distributions of charge and current densities can provide a better insight in the microscopic mechanisms of charge transport. If O is an observable which has a value  $O_{\alpha}$  in a state  $\alpha$ , its ensemble average at time t is a sum over all states weighted by the probability  $P_{\alpha}$  to be in a state  $\alpha$  at time t

$$\langle O \rangle = \sum_{\alpha} O_{\alpha} P_{\alpha}. \tag{2.36}$$
 equ:ensemble

If O does not explicitly depend on time, the time evolution of  $\langle O \rangle$  can be calculated as

$$\frac{d\langle O\rangle}{dt} = \sum_{\alpha,\beta} \left[ P_{\beta} \Omega_{\beta\alpha} - P_{\alpha} \Omega_{\alpha\beta} \right] O_{\alpha} = \sum_{\alpha,\beta} P_{\beta} \Omega_{\beta\alpha} \left[ O_{\alpha} - O_{\beta} \right]. \tag{2.37}$$

If averages are obtained from KMC trajectories,  $P_{\alpha} = s_{\alpha}/s$ , where  $s_{\alpha}$  is the number of Markov chains ending in the state  $\alpha$  after time t, and s is the total number of chains.

Alternatively, one can calculate time averages by analyzing a single Markov chain. If the total occupation time of the state  $\alpha$  is  $\tau_{\alpha}$  then

$$\overline{O} = \frac{1}{\tau} \sum_{\alpha} O_{\alpha} \tau_{\alpha} \,, \tag{2.38}$$

where  $au = \sum_{lpha} au_{lpha}$  is the total time used for time averaging.

For ergodic systems and sufficient sampling times, ensemble and time averages should give identical results. In many cases, the averaging procedure reflects a specific experimental technique. For example, an ensemble average over several KMC trajectories with different starting conditions corresponds to averaging over injected charge carriers in a time-of-flight experiment. In what follows, we focus on the single charge carrier (low concentration of charges) case.

#### 2.11.1 Charge density

For a specific type of particles, the microscopic charge density of a site i is proportional to the occupation probability of the site,  $p_i$ 

$$\rho_i = e p_i / V_i \,, \tag{2.39}$$

where, for an irregular lattice, the effective volume  $V_i$  can be obtained from a Voronoi tessellation of space. For reasonably uniform lattices (uniform site densities) this volume is almost independent of the site and a constant volume per cite,  $V_i = V/N$ , can be assumed. In the macroscopic limit, the charge density can be calculated using a sxtpthing kernel function, i.e. a distance-weighted average over multiple sites. Site occupations  $p_i$  can be obtained from eq. (2.36) or eq. (2.38) by using the occupation of site i in state  $\alpha$  as an observable.

If the system is in thermodynamic equilibrium, that is without sources or sinks and without circular currents (and therefore no net flux) a condition, known as detailed balance, holds

$$p_j \omega_{ji} = p_i \omega_{ij},$$
 (2.40) equidetailed\_balance

It can be used to test whether the system is ergodic or not by correlating  $\log p_i$  and the site energy  $E_i$ . Indeed, if  $\lambda_{ij} = \lambda_{ji}$  the ratios of forward and backward rates are determined solely by the energetic disorder,  $\omega_{ji}/\omega_{ij} = \exp(-\Delta E_{ij}/k_{\rm B}T)$  (see eq. (2.31)).

#### **2.11.2** Current

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If the position of the charge,  $\vec{r}$ , is an observable, the time evolution of its average  $\langle \vec{r} \rangle$  is the total current in the system

$$\vec{J} = e \left\langle \vec{v} \right\rangle = e \frac{d \left\langle \vec{r} \right\rangle}{dt} = e \sum_{i,j} p_j \omega_{ji} (\vec{r}_i - \vec{r}_j). \tag{2.41}$$

Symmetrizing this expression we obtain

$$\vec{J} = \frac{1}{2}e\sum_{i,j} \left(p_j\omega_{ji} - p_i\omega_{ij}\right)\vec{r}_{ij},\tag{2.42}$$

where  $\vec{r}_{ij} = \vec{r}_i - \vec{r}_j$ . Symmetrization ensures equal flux splitting between neighboring sites and absence of local average fluxes in equilibrium. It allows to define a local current through site i as

$$\vec{J}_i = \frac{1}{2}e\sum_j \left(p_j\omega_{ji} - p_i\omega_{ij}\right)\vec{r}_{ij}. \tag{2.43}$$

A large value of the local current indicates that the site contributes considerably to the total current. A collection of such sites thus represents most favorable charge pathways [44].

## 2.11.3 Mobility and diffusion constant

For a single particle, e.g. a charge or an exciton, a zero-field mobility can be determined by studying particle diffusion in the absence of external fields. Using the particle displacement squared,  $\Delta r_i^2$ , as an observable we obtain

$$2dD_{\gamma\delta} = \frac{d\left\langle \Delta r_{i,\gamma} \Delta r_{i,\delta} \right\rangle}{dt} = \sum_{\substack{i,j\\i\neq j}} p_j \omega_{ji} \left( \Delta r_{i,\gamma} \Delta r_{i,\delta} - \Delta r_{j,\gamma} \Delta r_{j,\delta} \right) = \sum_{\substack{i,j\\i\neq j}} p_j \omega_{ji} \left( r_{i,\gamma} r_{i,\delta} - r_{j,\gamma} r_{j,\delta} \right) \,. \tag{2.44}$$

Here  $\vec{r_i}$  is the coordinate of the site i,  $D_{\gamma\delta}$  is the diffusion tensor,  $\gamma, \delta = x, y, z$ , and d = 3 is the system dimension. Using the Einstein relation,

$$D_{\gamma\delta} = k_{\rm B}T\mu_{\gamma\delta} \,, \tag{2.45}$$

one can, in principle, obtain the zero-field mobility tensor  $\mu_{\gamma\delta}$ . Eq. (2.44), however, does not take into account the use of periodic boundary conditions when simulating charge dynamics. In this case, the simulated occupation probabilities can be compared to the solution of the Smoluchowski equation with periodic boundary conditions (see the supporting information for details). Alternatively, one can directly analyze time-evolution of the KMC trajectory and obtain the diffusion tensor from a linear fit to the mean square displacement,  $\overline{\Delta r_{i,\gamma}\Delta r_{i,\delta}}=2dD_{\gamma\delta}t$ . The charge carrier mobility tensor,  $\hat{\mu}$ , for any value of the external field can be determined either from the average charge velocity defined in eq. (2.41)

$$\langle \vec{v} \rangle = \sum_{i,j} p_j \omega_{ji} (\vec{r}_i - \vec{r}_j) = \hat{\mu} \vec{F} , \qquad (2.46)$$

or directly from the KMC trajectory. In the latter case the velocity is calculated from the unwrapped (if periodic boundary conditions are used) charge displacement vector divided by the total simulation time. Projecting this velocity on the direction of the field  $\vec{F}$  yields the charge carrier mobility in this particular direction. In order to improve statistics, mobilities can be averaged over several KMC trajectories and MD snapshots.

## 2.11.4 Spatial correlations of energetic disorder

Long-range, e.g. electrostatic and polarization, interactions often result in spatially correlated disorder [45], which affects the onset of the mobility-field (Poole-Frenkel) dependence [40, 46, 47]. To quantify the degree of correlation, one can calculate the spatial correlation function of  $E_i$  and  $E_j$  at a distance  $r_{ij}$ 

$$C(r_{ij}) = \frac{\langle (E_i - \langle E \rangle) (E_j - \langle E \rangle) \rangle}{\langle (E_i - \langle E \rangle)^2 \rangle}, \tag{2.47}$$

where  $\langle E \rangle$  is the average site energy.  $C(r_{ij})$  is zero if  $E_i$  and  $E_j$  are uncorrelated and 1 if they are fully correlated. For a system of randomly oriented point dipoles, the correlation function decays as 1/r at large distances [48].

For systems with spatial correlations, variations in site energy differences,  $\Delta E_{ij}$ , of pairs of molecules from the neighbor list are smaller than variations in site energies,  $E_{i}$ , of all individual molecules. Since only neighbor list pairs affect transport, the distribution of  $\Delta E_{ij}$  rather than that of individual site energies,  $E_{i}$ , should be used to characterize energetic disorder.

Note that the eanalyze calculator takes into account *all* contributions to the site energies

Analyze distribution and correlations of site energeies

| xtp\_run -o options.xml -f state.sql -e eanalyze

934 935

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MORPH	ORPHOLOGY				
	EXTRACT	REPRODUCE			
RDF and coarse grain potential	GROMACS edg_rdf -f traj.xtc -s topol.tpr	VOTCA::CSG  csg_inverse -options settings.xml			
morphology		GROMACS mdrun			

RATES					
	EXTRACT	REPRODUCE			
neighbor list (pairs)	xtp_run -e <b>panalyze</b> -o options.xml -f state_reference.sql	xtp_run -e <b>neighborlist</b> -o options.xml -f state_cg.sql			
site energies	xtp_run -e <b>eanalyze</b> -o options.xml -f state_reference.sql	xtp_run -e <b>eimport</b> -o options.xml -f state_cg.sql			
transfer integrals	xtp_run -e <b>ianalyze</b> -o options.xml -f state_reference.sql	xtp_run -e <b>iimport</b> -o options.xml -f state_cg.sql			
rates	not used directly	xtp_run -e <b>rates</b> -o options.xml -f state_cg.sql			

Figure 2.5: Stochastic Model in VOTCA. Overview of the different steps for generating stochastic charge transport networks in VOTCA. The Molecular Dynamics software GROMACS allows to analyze the radial distribution function of a morphology, which is then used by VOTCA::CSG to generate a coarse-grained potential that reproduces this distribution function. This potential can then be used for coarse-grained simulations in GROMCAS. For calculating rates in the coarse-grained morphology, first the relavant parameters are extracted (panalyze, eanalyze, ianalyze) from the reference morphology and and then reproduced in the coarsed-coarse grained morphology (neighborlist, eimport, iimport). With all these at hand, the rates calculator can be used in the coarse-grained morphology.

fig:overview\_stochastic



Figure 2.6: Comparison of the atomistic  $(17 \times 17 \times 17 \, \mathrm{nm}^3)$  and coarse-grained  $(50 \times 50 \times 120 \, \mathrm{nm}^3)$  models. (a) Radial distribution function, g(r). (b) Probability of two sites to be connected (added to the neighbor list) as a function of their separation. (c) Spatial site energy autocorrelation function,  $\kappa(r)$ ; Inset: Site energy distribution. (d) Mean m and width  $\sigma$  of a distribution of the logarithm of electronic couplings,  $\log_{10}(J^2/\,\mathrm{eV}^2)$ , for molecules at a fixed separation r. (e) Rate distributions. (f) Mobility as a function of hole density, plotted for four different electric fields.

fig:stochastic

# 。Chapter 3

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# Input and output files

# 3.1 Atomistic topology

If you are using GROMACS for generating atomistic configurations, it is possible to directly use the topology file provided by GROMACS (topology.tpr). In this case the GROMACS residue and atom names should be used to specify the coarse-grained topology and conjugated segments. A custom topology can also be defined using an XML file. Moreover, it s possible to partially overwrite the information provided in, for example, GROMACS topology file. We will illustrate how to create a custom topology file using DCV2T. The structure of DCV2T, together with atom type definitions, is shown in fig. 3.1. DCV2T has two thiophene (THI) and two dicyanovinyl (NIT) residues. The pdb file which contains residue types, residue numbering, atom names, atom types, and atom coordinates is shown in listing 3.1.



Figure 3.1: (a) DCV2T with atoms labelled according to residue\_number:residue\_name:atom\_name. There are four residues and two residue types: thiophene (THI) and dicyanovinyl (NIT). The corresponding pdb file is shown in listing 3.1. Atom numbering is used to split conjugated segments on rigid fragments and to link atomistic ((b) from GROMACS topology) and quantum descriptions (c).

fig:dcv2t

Listing 3.1: pdb file of DCV2T.

					Listing 5.	ı. pub iii	e or DC v	41.		
948 949	HETATM	1	N1 NIT	1	2.388	8.533	11.066	1.00	4.14	N
950	HETATM	2	CN1 NIT	1	1.984	9.553	10.718	1.00	2.54	C
951	HETATM	3	N2 NIT	1	-1.138	10.872	10.087	1.00	3.24	N
952	HETATM	4	CN2 NIT	1	0.003	10.871	10.213	1.00	2.37	C
953	HETATM	5	CC1 NIT	1	1.441	10.824	10.327	1.00	1.91	С
954	HETATM	6	C1 NIT	1	2.193	11.939	10.071	1.00	1.61	С
955	HETATM	7	HN1 NIT	1	1.715	12.710	9.872	1.00	1.97	Н
956	HETATM	8	S1 THI	2	4.758	10.743	10.130	1.00	1.52	S
957	HETATM	9	CA1 THI	2	3.613	12.024	9.948	1.00	1.22	C
958	HETATM	10	CA2 THI	2	6.099	11.836	9.997	1.00	1.30	С
959	HETATM	11	CB1 THI		4.251	13.243	9.782	1.00	1.39	C
960	HETATM	12	CB2 THI		5.658	13.131	9.818	1.00	1.45	C
961	HETATM	13	HC1 THI		3.800	14.047	9.660	1.00	1.66	Н
962	HETATM	14	HC2 THI		6.230	13.860	9.731	1.00	1.74	Н
963	HETATM	15	S1 THI		8.803	12.414	9.882	1.00	1.38	S
964	HETATM	16	CA1 THI		7.456	11.347	10.094	1.00	1.37	C
965	HETATM	17	CA2 THI		9.940	11.122	10.152	1.00	1.42	C
966	HETATM	18	CB1 THI		7.873	10.048	10.355	1.00	1.73	C
967	HETATM	19	CB2 THI		9.267	9.926	10.399	1.00	1.82	C
968	HETATM	20	HC1 THI		7.288	9.335	10.487	1.00	2.05	Н
969	HETATM	21	HC2 THI		9.704	9.123	10.576	1.00	2.21	Н
970	HETATM	22	N1 NIT		11.235	14.572	9.094	1.00	3.08	N
971	HETATM	23	CN1 NIT		11.665	13.566	9.441	1.00	2.04	C
972	HETATM	24	N2 NIT		14.733	12.005	10.009	1.00	2.17	N
973	HETATM	25	CN2 NIT		13.590	12.149	9.933	1.00	1.77	C
974	HETATM	26	CC1 NIT		12.156	12.282	9.861	1.00	1.71	C
975	HETATM	27	C1 NIT		11.363	11.220	10.154	1.00	1.59	C
9 <del>76</del>	HETATM	28	HN1 NIT	4	11.813	10.440	10.389	1.00	1.89	Н

3.2. MAPPING FILE 35

tab:map

Table 3.1: Description of the XML mapping file (map.xml).

topology	Definitions of molecules, segments, and fragments.
molecules	Container for all molecules.
molecule	Mapping of a single molecule.
name	Name of the molecule in the coarse-grained model.
ident	Name (identification) of the molecule in the all-atom representation. This must match the molecule name in the atomistic representation.
segments	Partitioning of the molecule on conjugated segments.
segment	Description of a conjugated segment.
name	Name of a conjugated segment in a molecule.
fragments	Container for all fragments in a segment.
fragment	Description of a rigid fragment.
name	Name of the rigid fragment in a conjugated segment
mdatoms	List of all atoms belonging to the rigid fragment in the format residue number:residue name:atom name.
qmatoms	List of atoms of the rigid fragment in its ground state geometry, atom number:atom type.
weights	Weights are used to determine the fragment center. The order should be the same as in the mdatoms and qmatoms definitions. If the mass of a nucleus in atomic mass units is used, the center of the rigid fragment will be its center of mass.
localframe	Three atoms which define a local frame for each rigid fragment.

# 3.2 Mapping file

ec:xmlmap

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The mapping file (referred here as map.xml) is used by the program xtp\_map to convert an atomistic trajectory to a trajectory with conjugated segments and rigid fragments. This trajectory is stored in a state file and contains positions, names, types of atoms belonging to rigid fragments. The description of the mapping options is given in table 3.1. An example of map.xml for a DCV2T molecule is shown in listing 3.2.

The file map.xml contains the whole electrostatic information about the molecules as well as the structural information. The toolpdb2map creates a map.xml from a pdb file and is a good starting point for further refinement.

986 list:map

Listing 3.2: Examle of map.xml for DCV2T. Each rigid fragment (coarse-grained bead) is defined by a list of atoms. Atom numbers, names, and residue names should correspond to those used in GROMACS topology (see the corresponing listing 3.1 of the pdb file).

```
<topology> <!-- this file is used to conver an atomistic trajectory to conjugated segments -->
990
         <name>DCV2T-MOL</name> <!-- name of the conjugated molecule --->
991
        <mdname>Protein</mdname> <!-- name of the confugate a motivate --- name of the same as the name given at the end of topol.top-->
992
993
         <segments>
994
995
              <name>DCV</name> <!-- name of the conjugated segment within the molecule -->
<qmcoords>QC_FILES/DCV2T.xyz</qmcoords> <!-- QM coordinates of the conjugated segment --->
996
997
998
                    <!-- IZINDO INPUT
999
              <basisset>INDO</basisset>
1000
              <orbitals>QC_FILES/DCV2T.orb</orbitals>
              <torbital_h>50</torbital_h><!-- Number of the HOMO Orbital (e.g. alpha electrons, can be found in the log-file belonging to DCV2T.orb) --->
1002
1003
1004
                   <!-- EMULTIPOLE INPUT
1005
              <multipoles_n>MP_FILES/DCV2T.mps</multipoles_n><!-- Multipole file for neutral state --</pre>
1006
              1009
1010
                    qm e.g. DFT or GWBSE calculations-->
1011
1012
                       <!-- EINTERNAL INPUT -->
```

```
<U_cC_nN_h>0.0</U_cC_nN_h> <!-- Site energy
1014
                  <U_nC_nN_h>0.1</U_nC_nN_h> <!-- Reorg. discharge --->
                 <U_cN_cC_h>0.1</U_cN_cC_h> <!-- Reorg. charge
1017
                      <!-- MD OM MP Mapping -->
1018
                 <fragments>
1019
                  <fragment>
1020
                   <name>NI1</name> <!-- name of the rigid fragment within the segment --->
1021
                   <!-- list of atoms in the fragment resnum:resname:atomname
                   <mdatoms>1:NIT:N1 1:NIT:CN1 1:NIT:N2 1:NIT:CN2 1:NIT:CC1 1:NIT:C1 1:NIT:HN1//mdatoms>
1023
                  <!-- corresponding ground state geometry atomnumber:atomtype read from .xyz file-->
<qmatoms> 20:N 19:C 14:N 13:C 12:C 11:C 23:H </qmatom
<!-- corresponding group state geometry multipoles read from .mps files -->
<mpoles> 20:N 19:C 14:N 13:C 12:C 11:C 23:H </mpoles>
<!-- weights to determine the fragment center (here CoM is used) -->
<meights> 14 12 14 12 12 12 12 1 </meights</pre>
1024
                                                                                                                  23:H </gmatoms>
1025
1026
                                                                                                                  23:H </mpoles>
1027
                 <weights 14 12 14 12 12 12 1 </weights>
<!-- three atoms: define a cartesian local frame, two atoms: fragment is assumed to be rotationally invariant around the axis, one atom: fragment is assumed isotropic -->
1029
1030
1031
                   <localframe> 20 19 14 </localframe>
1032
                 <!-- Optional parameters (if not set <localframe> is used): used when atom labels in the .mps
1033
                          and .xyz file differ or more sites in the .mps file are used, so refers to <mpoles>
                   <localframe_mps> 20 19 14 </localframe_mps>
1035
                  <!-- Optional parameters (if not set <localframe> is used): weights to determine the
    fragment center (here CoM is used), used when atom labels in the .mps and .xyz file
    differ or additional sites in the .mps file are used -->
<weights_mps>    14    12    14    12    12    12    1    </
    weights_mps>
1036
1037
1038
                 <weights_mps>
1039
                 weights_mps>
<!-- Optional flag: says if a site is virtual or not, (virtual=1, real=0)--->
<virtual_mps> 0 0 0 0 0 0
1040
1041
1042
1043
                       virtual mps>
                </fragment>
1044
1045
1046
               <fragment>
                   <name>TH1</name>
1047
1048
                   <mdatoms>2:THI:S1 2:THI:CA1 2:THI:CA2 2:THI:CB1 2:THI:CB2 2:THI:HC1 2:THI:HC2/mdatoms>
1049
                   <qmatoms> 7:S 8:C 6:C
<mpoles> 7:S 8:C 6:C
                                                                           9:C 10:C 24:H 25:H </qmatoms>
9:C 10:C 24:H 25:H </mpoles>
1050
                                     32
                                                 12
                                                               12
                                                                             12
                                                                                                                       1 </weights>
                   <weights>
                                                                                          12
1051
                   <localframe> 7 8 6 </localframe>
1052
               </fragment>
1054
1055
               <fragment>
1056
                   <name>TH2</name>
                   <mdatoms>3:THI:S1 3:THI:CA1 3:THI:CA2 3:THI:CB1 3:THI:CB2 3:THI:HC1 3:THI:HC2</mdatoms>
1057
                   <qmatoms> 3:S 4:C 2:C
<weights> 32 12 12
                                                                                                                27:H </qmatoms>
1 </weights>
                                                                       5:C 1:C 26:H
1058
1059
                   <localframe> 3 4 2 </localframe>
1060
               </fragment>
1061
1062
1063
               <fragment>
                   <name>NI2</name>
1064
1065
                   <mdatoms>4:NIT:N1 4:NIT:CN1 4:NIT:N2
                                                                                                                28:H </qmatoms>
                                               21:C
                                                                                                      15:C
                   <qmatoms> 22:N
                                                            18:N
                                                                       17:C 16:C
1066
                   <mpoles>
                                                                                                                  28:H </mpoles>
1 </weights>
1067
                                 22:N
                                              21:C
                                                            18:N
                                                                         17:C
                                                                                       16:C
                                                                                                     15:C
                                                                          12
                                                                                         12
                                                 12
                                                                                                       12
1068
                   <weights>
                                     14
                                                              14
                   <localframe> 22 21 18 </localframe>
1069
               </fragment>
1070
               </fragments>
1071
           </segment>
          </segments>
1073
       </molecule>
1074
1075
       </molecules>
       </topology>
18<del>79</del>
```

# 3.3 Molecular orbitals

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If the semi-empirical method is used to calculate electronic coupling elements, molecular orbitals of all molecules must be supplied. They can be generated using Gaussian program. The Gaussian input file for DCV2T is shown in listing 3.3. Provided with this input, Gaussian will generate fort. 7 file which contains the molecular orbitals of a DCV2T. This file can be renamed to DCV2T. orb. Note that the order of the atoms in the input file and the order of coefficients should always match. Therefore, the coordinate part of the input file must be supplied together with the orbitals. We will assume the coordinates, in the format atom\_type: x y z, is saved

to the DCV2T.xyz file.



# 🔑 Be careful!

Izindo requires the specification of orbitals for hole and electron transport in map.xml. They are the HOMO and LUMO respectively and can be retrieved from the log file from which the DCV2T.orb file is generated. The number of alpha electrons is the HOMO, the LUMO is HOMO+1

1087 list:zindo\_orbitals

Listing 3.3: Gaussian input file get\_orbitals.com used for generating molecular orbitals. The first line contains the name of the check file, the second the requested RAM. int=zindos requests the method ZINDO, punch=mo states that the molecular orbitals ought to be written to the fort.7 file, nosymm forbids use of symmetry and is necessary to ensure correct position of orbitals with respect to the provided coordinates. The two integer numbers correspond to the charge and multiplicity of the system: 01 corresponds to a neutral system with a multiplicity of one. They are followed by the types and coordinates of all atoms in the molecule.

```
%chk=DCV2T.chk
1089
    %mem=100Mb
1090
1091
    #p int=zindos punch=mo nosymm
1092
1093
    DCV2T molecular orbitals
1094
1095
    0 1
               -1.44650
                               2.12185
                                               0.00135
               -2.43098
                               0.58936
                                               -0.00048
1097
                                               -0.00146
               -1.59065
                               -0.51859
1098
               -0.21222
                               -0.22233
                                               -0.00095
               0.07761
                                                0.00040
    C
                                1.13376
1100
1101
    S
                2.87651
                                0.79316
                                                0.00148
                                                0.00235
1102
                3.86099
                                2.32565
    C
               3.02066
                               3.43359
                                                0.00231
1103
               1.64223
                               3.13733
                                                0.00162
                1.35240
                                1.78125
                                                0.00114
1105
               -3.85350
                               0.52245
                                               -0.00081
1106
               -4.79569
                               1.52479
                                               -0.00008
               -6.18500
                                1.18622
                                               -0.00117
    C
1108
1109
               -4.47544
                                2.91565
                                                0.00081
               5.28350
                               2.39256
                                                0.00296
1110
                                                0.00327
               6.22569
                                1.39020
1111
    C
1112
                7.61500
                                1.72876
                                                0.00432
               5.90542
                               -0.00064
                                                0.00333
1113
                                               -0.00195
1114
    N
               -7.32389
                               0.89743
    N
               -4.21872
                               4.06274
                                                0.00142
1115
               8.75389
                                                0.00510
    N
                               2.01754
1116
1117
    N
               5.64864
                               -1.14772
                                                0.00361
1118
               -1.98064
                               -1.52966
                                               -0.00256
                0.55785
                               -0.98374
                                               -0.00169
1119 H
                3.41065
                                4.44466
                                                0.00272
1120 H
    Н
                0.87216
                                3.89874
                                                0.00147
1121
               -4.24640
                               -0.49192
                                                -0.00188
1122
    Η
                5.67641
                                3.40692
                                                0.00337
1123
```

# 3.4 Monomer calculations for DFT transfer integrals

list:edft\_gaussian\_xml

Listing 3.4: Example package.xml file for the Gaussian package required in the options of the edft calculator for the monomer calculations as preparation for the determination of transfer integrals using DIPRO.

```
1126
1127 <package>
1128 <name>gaussian</name>
1129 <executable>g09</executable>
1130 <checkpoint></checkpoint>
```

```
<scratch></scratch>
1131
1132
       <charge>0</charge>
1133
      <spin>1</spin>
      <options># pop=minimal pbepbe/6-311g** scf=tight punch=mo nosymm test/options>
1135
      <memory>1Gb</memory>
1136
      <threads>2</threads>
1137
1138
      <cleanup></cleanup>
1139
    </package>
1149
```

list:edft turbomole xm

Listing 3.5: Example package.xml file for the Turbomole package required in the options of the edft calculator for the monomer calculations as preparation for the determination of transfer integrals using DIPRO

```
1142
     <package>
1143
        <name>turbomole</name>
1144
        <executable>ridft</executable>
1145
        <scratch>/tmp</scratch>
1147
        <options>
1148
     TITLE
1149
     a coord
1150
1151
     no
1152
     b all def-TZVP
1153
1154
     eht
1155
     У
     0
1157
1158
    dft
1159
     on
1160
     func
1161
     pbe
1162
     grid
1163
     m3
1164
1165
1166
     ri
     m 300
1169
     scf
1170
1171
     conv
1172
     iter
1173
1174
1175
     marij
1176
1177
1178
        </options>
1179
1180
        <cleanup></cleanup>
1181
     </package>
1183
```

list:edft\_nwchem\_xm

Listing 3.6: Example package.xml file for the NWChem package required in the options of the edft calculator for the monomer calculations as preparation for the determination of transfer integrals using

```
<package>
1185
      <name>nwchem</name>
1186
       <executable>nwchem</executable>
1187
       <checkpoint></checkpoint>
1188
       <scratch>/tmp/nwchem</scratch>
1189
      <charge>0</charge>
1190
      <spin>1</spin>
1191
       <threads>1</threads>
       <memory></memory>
1194
       <options>
    start
1195
    basis
1196
    * library 6-311gss
1197
    end
1198
    memory 1500 mb
1199
1200
    dft
1201
    xc xpbe96 cpbe96
     direct
     iterations 100
1204
1205
     noprint "final vectors analysis"
    end
1206
    task dft
1207
    </options>
1208
       <cleanup></cleanup>
1209
    </package>
1319
```

# 3.5 Pair calculations for DFT transfer integrals

list:idft\_gaussian\_xml

Listing 3.7: Example package.xml file for the Gaussian package required in the options of the idft calculator for the pair calculations and the determination of transfer integrals using DIPRO.

```
1213
    <package>
1214
      <name>gaussian
1215
      <executable>q09</executable>
1216
      <checkpoint></checkpoint>
1217
      <scratch></scratch>
1218
1219
      <charge>0</charge>
      <spin>1</spin>
      <options># pop=minimal pbepbe/6-311g** nosymm IOp(3/33=1,3/36=-1) punch=mo guess=cards scf=
1222
      <memory>1Gb</memory>
1223
      <threads>1</threads>
1224
1225
      <cleanup></cleanup>
1226
    </package>
1228
```

list:idft\_turbomole\_xml

Listing 3.8: Example package.xml file for the Turbomole package required in the options of the idft calculator for the pair calculations and the determination of transfer integrals using DIPRO.

```
1229 cyackage> c
```

```
<options>
1235
     $intsdebug cao
     a coord
1238
1239
    no
    b all def-TZVP
1240
1241
     eht
1242
     У
1243
     0
1244
    У
1245
     dft
1246
1247
1248
     func
1249
     pbe
     grid
1250
     m3
1251
1252
1253 ri
1254 on
1255 m 300
1256
   scf
1257
1258 CONV
1259
     7
1260 iter
1261 1
1262 diis
     3
1263
     damp
1264
     0.00
1265
1266
1267
1268
1269
     marij
1270
1271
       </options>
1272
1273
       <cleanup></cleanup>
1274
     </package>
1275
```

list:idft\_nwchem\_xml

Listing 3.9: Example package.xml file for the NWChem package required in the options of the idft calculator for the pair calculations and the determination of transfer integrals using DIPRO.

```
1277
    <package>
1278
      <name>nwchem</name>
1279
      <executable>nwchem</executable>
1280
      <checkpoint></checkpoint>
1281
      <scratch>/tmp/nwchem</scratch>
1282
      <charge>0</charge>
1283
      <spin>1</spin>
1284
      <memory></memory>
1285
      <threads>1</threads>
1286
      <options>
1287
   start
1288
1289 basis
    * library 6-311gss
```

```
memory 1500 mb
1292
1293
    dft
1294
     print "ao overlap"
1295
     xc xpbe96 cpbe96
1296
1297
     direct
     iterations 1
1298
     convergence nodamping nodiis
1299
     noprint "final vectors analysis"
1300
     vectors input system.movecs
1301
    end
1302
    task dft
1303
    </options>
1304
       <cleanup></cleanup>
1305
    </package>
1389
```

# 3.6 DFT transfer integrals

list:TI\_xml

Listing 3.10: Example TI.xml file created as the output of a DIPRO calculation. Due to slightly different implementations, the orbitals indices refer to monomer indices in a Gaussian run but to indices in the merged dimer guess in a Turbomole run.

```
1309
     <pair name="pair_100_155">
1310
1311
         <parameters>
            <HOMO_A>162</HOMO_A>
            <NoccA>1</NoccA>
            <LUMO_A>164</LUMO_A>
1315
            <NvirtA>1</NvirtA>
            <HOMO_B>161</HOMO_B>
1316
            <NoccB>1</NoccB>
1317
            <LUMO B>163</LUMO B>
1318
            <NvirtB>1</NvirtB>
1319
         </parameters>
1320
          <transport name="hole">
1321
              <channel name="single">
1322
                   <J>1.546400416750696E-003</J>
1323
                   <e_A>-6.30726450715697</e_A>
1324
                   <e_B>-6.36775613794166</e_B>
1325
              </channel>
1326
              <channel name="multi">
1327
                  <molecule name="A">
1328
                      <e_HOMOm0>-6.30726450715697</e_HOMOm0>
1329
                  </molecule>
1330
                  <molecule name="B">
1331
                      <e_HOMOm0>-6.36775613794166</e_HOMOm0>
1332
                  </molecule>
                      <dimer name="integrals">
                            <T_00>1.546400416750696E-003</T_00>
1335
                            <J_sq_degen>2.391354248926727E-006</J_sq_degen>
                            <J_sq_boltz>2.391354248926727E-006</J_sq_boltz>
1337
                      </dimer>
1338
              </channel>
1339
          </transport>
1340
          <transport name="electron">
1341
              <channel name="single">
1342
                   <J>-2.797473760331286E-003</J>
1343
                   <e_A>-4.50318366770689</e_A>
```

```
<e B>-4.53143397059021/e B>
1345
              </channel>
1346
              <channel name="multi">
1347
                      <molecule name="A">
                            <e_LUMOp0>-4.50318366770689</e_LUMOp0>
1349
                      </molecule>
1350
                      <molecule name="B">
1351
                            <e LUMOp0>-4.53143397059021/e LUMOp0>
1352
                      </molecule>
1353
                      <dimer name="integrals">
1354
                            <T_00>-2.797473760331286E-003</T_00>
1355
                            <J_sq_degen>7.825859439742066E-006</J_sq_degen>
1356
                            <J_sq_boltz>7.825859439742066E-006</J_sq_boltz>
1357
                      </dimer>
              </channel>
          </transport>
1360
     </pair>
1361
```

#### State file 3.7

```
1363
    All data structures are saved to the state.sql file in sqlite3 format, see http://www.sqlite.org/.
1364
    They are available in form of tables in the state.sql file as can be seen by the command
1365
    sqlite3 state.sql " .tables "
    An example of such a table are molecules. The full table can be displayed using the command
1367
    (similar for the other tables)
1368
    sqlite3 state.sql " SELECT * FROM molecules "
1369
    The meaning of all the entries in the table can be displayed by a command like
    sqlite3 state.sql " .SCHEMA molecules "
1371
    The first and second entry are integers for internal and regular id of the molecule and the third
1372
    entry is the name. A single field from the table like the name of the molecule can be displayed by
1373
    a command like
1374
    sqlite3 state.sql " SELECT name FROM molecules "
1375
    Besides molecules, the following tables are stored in the state.sql:
1376
    conjseg_properties:
1377
    Conjugated segments are stored with id, name and x,y,z coordinates of the center of mass in nm.
1378
1379
```

Reorganization energies for charging or discharging a conjugated segment are stored together 1380 with the coulomb energy and any other user defined energy contribution (in eV) and occupation probabilities. 1382

1383

The pairs from the neighborlist are stored with the pair id, the id of the first and second segment, 1384 the rate from the first to the second, the rate from the second to the first (both in  $s^{-1}$ ) and the x,y,z coordinates in nm of the distance between the first and the second segment.

pairintegrals: 1387

Transfer integrals for all pairs are stored in the following way: The pair id, the number for count-1388 ing possible different electronic overlaps (e.g if only the frontier orbitals are taken into account this is always zero, while an effective value is stored in addition to the different overlaps of e.g. 1390 HOMO-1 and HOMO-1 if more frontier orbitals are taken into account) and the integral in eV. 1391 pairproperties:

1392

The outer sphere reorganization energy of all pairs is stored by an id, the pair id, a string lambda\_outer and the energy in eV. 1394

conjseqs: 1395

Conjugated segments are saved in the following way: The id, the name, the type, the molecule 1396 id, the time frame, the x,y,z coordinates in nm and the occupation probability.

3.7. STATE FILE 43

```
conjseg_properties:
1398
    Properties of the conjugated segments like reorganization energies for charging or discharging a
    charge unit or the coulomb contribution to the site energy are stored by: id, conjugated segment
    id, a string like lambda_intra_charging, lambda_intra_discharging or energy_coulomb
    and a corresponding value in eV.
1402
    The tables rigidfrag_properties, rigidfrags and frames offer information about rigid
1403
    fragments and time frames including periodic boundary conditions.
1404
    The data in the state.sql file can also be modified by the user. Here is an example how to
    modify the transfer integral between the conjugated segments number one and two assuming
1406
    that they are in the neighborlist. Their pair id can be found by the command
1407
    pair_ID='sqlite3state.sql"SELECT _id FROM pairs WHERE conjseg1=1 AND conjseg2=2"'
    The old value of the transfer integral can be deleted using
    sqlite3 state.sql "DELETE FROM pair_integrals WHERE pair=$pair_ID"
1410
    Finally the new transfer integral J can be written to the state.sql file by the command
1411
    sqlite3 state.sql "INSERT INTO pair_integrals (pair,num,J) VALUES ($pair_ID,0,$J)"
1412
    Here the num=0 indicates that only the effective transfer integrals is written to the file, while other
1413
    values of num would correspond to overlap between other orbitals than the frontier orbitals.
1414
    In a similar way the coulomb contribution to the site energy of the first conjugated segment can
1415
    be overwritten by first getting its id
    c_ID=`sqlite3 state.sql "SELECT _id from conjseg_properties where conjseg=1 AND
1417
    key =\"energy_coulomb\""
1418
    Then deleting the old value
1419
    sqlite3 state.sql "DELETE FROM from conjseg_properties WHERE _id=$c_ID"
    Then the new coulomb energy E can be written to this id
142
    sqlite3 state.sql "INSERT INTO conjseg_properties (_id,conjseg,key,value)
1422
    VALUES ($c_ID,1,\"energy_coulomb\",$E)"
1423
    Finally the resulting coulomb contribution to all conjugated segments can be displayed by
    sqlite3 state.sql "SELECT * from conjseg_properties WHERE key=\"energy_coulomb\""
```

1426

# Chapter 4

# Reference

4.1.4 xtp\_basisset

sissetfiles optional arguments:

```
4.1
                     Programs
         1429
             Programs execute specific tasks (calculators).
             4.1.1
                    xtp_testsuite
         1431
   prog:xtp_testsuite
             Performs tests en suite + optional arguments:
         1432
                   -h, --help show this help message and exit
         1433
                   -e [ [ ...]], --execute [ [ ...]] Tests to perform, accepts regex (def=".*")
                   -1, --listonly List all tests available, then quit.
         1435
                   -x , --xml Test-suite file (def="$VOTCASHARE/xtp/xml/testsuite.xml")
         1436
                         --source Test source input directory (def="source")
                   -td , --testdirectory Test run directory (def="suite")
         1438
                   -t , --target Directory where to store targets (def="targets")
         1439
                   -r , --reference Folder with reference data to compare to (def="reference")
         1440
         144
                   -g, --generate Generate reference from targets (def=False)
                   -cmp, --compareonly Only compare existing targets to reference (def=False)
         1442
                   -v, --verbose The wordy version (def=False)
         1443
                   -sh, --showoutput Display VOTCA::XTP exec. output (def=False)
         1444
                   -c, --clean To clean or not to clean test dir. (def=False)
                   -m , --mailto Mail the result. (def=False)
         1446
             4.1.2 xtp_update
             Updates the state file + optional arguments:
                   -h, --help show this help message and exit
         1449
                   -f SQLFILE, --file SQLFILE State file to update.
         1450
                    xtp_update_exciton
             4.1.3
         1451
prog:xtp_update_excito
             Updates the state file for singlets and triplets + optional arguments:
         1452
                   -h, --help show this help message and exit
         1453
         1454
                   -f SQLFILE, --file SQLFILE State file to update.
```

xtp\_update, version 1.5-dev gitid: e3b8d11 Creates votca xml basissetfiles from NWCHEM ba-

1500

1501

1502

```
-h, --help show this help message and exit
   1458
            -f NWCHEM, --inputnw NWCHEM NWchem file containing the basisset.
   1459
            -o OUTPUTFILE, --outputvotca OUTPUTFILE Path of votca outputfile
       4.1.5 xtp_map
       Generates QM|MD topology
   1462
            -h [ --help ] display this help and exit
            -v [ --verbose ] be loud and noisy
   1464
            -t [ --topology ] arg topology
   1465
            -c [ --coordinates ] arg coordinates or trajectory
             -s [ --segments ] arg definition of segments and fragments
            -f [ --file ] arg state file
   1468
       4.1.6 xtp_run
   1469
prog:xtp run
       Runs excitation/charge transport calculators
  1470
            -h [ --help ] display this help and exit
   1471
            -v [ --verbose ] be loud and noisy
            -o [ --options ] arg calculator options
   1473
            -f [ --file ] arg sqlight state file, *.sql
   1474
            -i [ --first-frame ] arg (=1) start from this frame
   1475
            -n [ --nframes ] arg (=1) number of frames to process
   1476
            -t [ --nthreads ] arg (=1) number of threads to create
   1477
            -s [ --save ] arg (=1) whether or not to save changes to state file
   1478
            -e [ --execute ] arg List of calculators separated by ',' or ' '
   1470
            -1 [ --list ] Lists all available calculators -d [ --description ] arg Short description of
            a calculator
   1481
       4.1.7 xtp_tools
   1482
       Runs excitation/charge transport tools
  1483
            -h [ --help ] display this help and exit
   1484
            -v [ --verbose ] be loud and noisy
   1485
                [ --nthreads ] arg (=1) number of threads to create
   1486
            -o [ --options ] arg calculator options Tools:
   1487
            -e [ --execute ] arg List of tools separated by ',' or ' '
   1488
            -1 [ --list ] Lists all available tools -d [--description ] arg Short description of a tool
   1489
       4.1.8 xtp_parallel
   1490
       Runs job-based heavy-duty calculators
             -h [ --help ] display this help and exit
   1492
             -v [ --verbose ] be loud and noisy
   1493
            -o [ --options ] arg calculator options
   1494
            -f [ --file ] arg sqlite state file, *.sql
            -i [ --first-frame ] arg (=1) start from this frame
   1496
            -n [ --nframes ] arg (=1) number of frames to process
   1497
            -t [ --nthreads ] arg (=1) number of threads to create
                [ --save ] arg (=1) whether or not to save changes to state file
```

[ --restart ] arg restart pattern: 'host(pc1:234) stat(FAILED)'

-m [ --maxjobs ] arg (=-1) maximum number of jobs to process (-1=inf)

-j [ --jobs ] arg (=run) task(s) to perform: input, run, import

-c [ --cache ] arg (=8) assigns jobs in blocks of this size

```
-e [ --execute ] arg List of calculators separated by ',' or ' '
-l [ --list ] Lists all available calculators -d [ --description ] arg Short description of a calculator
```

# 4.1.9 xtp\_dump

Extracts information from the state file -h [ --help ] display this help and exit 1509 -v [ --verbose ] be loud and noisy 1510 -o [ --options ] arg calculator options 1511 -f [ --file ] arg sqlight state file, \*.sql --first-frame ] arg (=1) start from this frame 1513 -n [ --nframes ] arg (=1) number of frames to process 1514 -t [ --nthreads ] arg (=1) number of threads to create 1515 -s [ --save ] arg (=1) whether or not to save changes to state file Extractors: -e [ --extract ] arg List of extractors separated by ',' or ' 1517 -1 [ --list ] Lists all available extractors -d [ --description ] arg Short description of 1518 an extractor 1519

#### 4.2 Calculators

sæf:calculator

1521

1522

1523

1525

1526 list:cald

1507

Calculator is a piece of code which computes specific system properties, such as site energies, transfer integrals, etc. xtp\_run, xtp\_kmc\_run are wrapper programs which executes such calculators. The generic syntax is

```
1524 xtp_run -e "calc1, calc2, ..." -o options.xml
```

File options.xml lists all options needed to run a specific calculator. The format of this file is explained in listing 4.1. A complete list of calculators is given in the calculators reference section.

Listing 4.1: A part of the options.xml file with options for the calculator\_name {1,2} calculators.

```
1527
     <calculator_name1>
1528
                <option1>value1</option1>
1529
                 <option2>value2</option2>
1530
    </calculator_name1>
1532
1533
    <calculator name2>
1534
                <option1>value1</option1>
1535
                <option2>value2</option2>
1536
1537
    </calculator_name2>
1538
1538
```

A list of all calculators and their short descriptions can be obtain using

```
1542 xtp_run--list
```

A detailed description of all options of a specific calculator(s) is available via

```
1544 xtp_run --desc calc1,calc2,...
```

#### 4.2.1 excitoncoupling

calc:excitoncoupling

Exciton couplings from serialized orbital files

option	default	unit	description
--------	---------	------	-------------

classical		
output	excitoncoupling	Output file
bsecoupling_options		
orbitalsA	A.orb	Serialized orbitals file
orbitalsB	B.orb	Serialized orbitals file
orbitalsAB	AB.orb	Serialized orbitals file

Return to the description of excitoncoupling.

# 4.2.2 gencube

Tool to generate cube files from .orb file

option	default	unit	description
output	state.cube		Output file
input	system.orb		Input file
padding	6.5		How far the grid should start from the molecule
xsteps	25		Gridpoints in x-direction
ysteps	25		Gridpoints in y-direction
zsteps	25		Gridpoints in z-direction
state	1		State to generate cube file for
spin			Singlet or Triplet
type	ground		qp:quasiparticle,ground:groundstate,transition:transitionstate,excited/egs:excitedstate density/density excited-ground state
mode	new		new: generate new cube file, substract: substract to cube files specified below
infile1			Cubefile to substract infile2 from
infile2			Cubefile to substract from infile1

Return to the description of gencube.

# 4.2.3 partialcharges

Tool to derive partial charges from QM results stores in serialized file

option	default	unit	description
output	Moleculecharge		Output file either .mps or .pdb
input	molecule.orb		Serialized file
esp_options			options for the method

Return to the description of partialcharges.

# 4.2.4 qmanalyze

Analysis tool for QM results stores in serialized file

option	default	unit	description
output	qmanalyze.out		Output file
BSE			additonal info about BSE results
input	molecule.orb		Serialized file

Return to the description of qmanalyze.

#### 1557 calc:kmclifetime 1558

1559

# 4.2.5 kmclifetime

Kinetic Monte Carlo simulations of multiple singletsin periodic boundary conditions with finite lifetime

option	default	unit	description
insertions	0	integer	How often charges are allowed to decay
maxrealtime	1E10	hours	Maximum realtime after which the run should abort
seed	0	integer	Integer to initialise the random number generator
numberofcharges	0	integer	Number of excitons in the simulation box
injection	*		Name pattern that specifies on which sites injection is possible. Before injecting on a site it is checked whether the column 'name' in the table 'segments' of the state file matches this pattern. Use the wildcard '*' to inject on any site.
lifetime	0		xml file with the lifetime of an exciton on each segment
carriertype	0		Options: singlet,triplet. Specifies the carrier type of the transport under consideration.
temperature	300	Kelvin	Temperature in Kelvin. Will only be relevant if rates are calculated by KMC and not taken from the state file.
rates	statefile		Options: statefile/calculate. statefile: use the rates for charge transfer specified in the state file; calculate: use transfer integrals, site energies and reorganisation energies specified in the state file as well as temperature specified here to calculate rates before starting the KMC simulation.
carrierenergy			Write the energy of the first carrier to a file
run		bool	Switch energyoutput on
outputfile			file to write energies to
alpha			exponential averageing, set to 1 to retrieve insantaneous energies
outputsteps			Specifies the interval between energy outputs

Return to the description of kmclifetime.



# 4.2.6 kmcmultiple

Kinetic Monte Carlo simulations of multiple holes or electrons in periodic boundary conditions

option	default	unit	description
runtime		seconds or in- teger	Simulated time in seconds (if a number smaller than 100 is given) or number of KMC steps (if a number larger than 100 is given)
outputtime	1E-8	seconds	Time difference between outputs into the trajectory file. Set to 0 if you wish to have no trajectory written out.
trajectoryfile	trajectory.csv		Name of the trajectory file
seed	123	integer	Integer to initialise the random number generator
injection	*		Name pattern that specifies on which sites injection is possible. Before injecting on a site it is checked whether the column 'name' in the table 'segments' of the state file matches this pattern. Use the wildcard '*' to inject on any site.
injectionmethod	random		Options: random/equilibrated. random: injection sites are selected randomly (generally the recommended option); equilibrated: sites are chosen such that the expected energy per carrier is matched, possibly speeding up convergence
numberofcharges	1	integer	Number of electrons/holes in the simulation box

fieldX	0	V/m	x component of the external electric field
fieldY	0	V/m	y component of the external electric field
fieldZ	1E6	V/m	z component of the external electric field
carriertype	electron		Options: electron/hole/singlet/triplet. Specifies the carrier type of the transport under consideration.
temperature	300	Kelvin	Temperature in Kelvin. Will only be relevant if rates are calculated by KMC and not taken from the state file.
explicitcoulomb	0		Options: 0/1/2. 0: no explit Coulomb interaction; 1: explicit Coulomb interaction using partial charges from SQL file, 2: explicit 'raw' Coulomb interaction using charges of +/-1. Note that rates from the state file will not be used if Coulomb interaction is switched on (option 1 or 2) but rates will be calculated within KMC.
rates	statefile		Options: statefile/calculate. statefile: use the rates for charge transfer specified in the state file; calculate: use transfer integrals, site energies and reorganisation energies specified in the state file as well as temperature and electric field specified here to calculate rates before starting the KMC simulation. In case of explicit Coulomb interaction this option is set to 'calculate' automatically. If you use rates from the state file make sure that the electric field specified here matches the one that was used for calculating the rates in the state file.

Return to the description of kmcmultiple.

# 4.2.7 xeanalyze

calc:xeanalyze

Histogram and correlation function of site energies and pair energy differences

option	default	unit	description
resolution_sites		eV	Bin size for site energy histogram
resolution_pairs		eV	Bin size for pair energy histogram
resolution_space		eV	Bin size for site energy correlation
states			?

Return to the description of xeanalyze.

#### 1567 calc:xeinternal

# 4.2.8 xeinternal

Reads in site and reorganosation energies and writes them to the state file

option	default	unit	description
energiesXML			XML input file with vacuum site, reorganization (charging, discharging) energies

Return to the description of xeinternal.

# 4.2.9 xianalyze

calc:xianalyz

Evaluates a histogram of a logarithm of squared couplings

option	default	unit	description
resolution_logJ2			Bin size of histogram log(J2)
resolution_space		nm	Bin size for r in log(J2(r))
states			States for which to calculate the histogram. Example: 1 -1

<sup>1572</sup> Return to the description of xianalyze.

# 4.2.10 xjobwriter

calc:xjobwrite

Writes list of jobs for a parallel execusion

option	default	unit	description
keys			job type
single_id			Segment ID as argument for mps.single
kmc_cutoff		nm	Pair-interaction cut-off as argument for mps.kmc

<sup>1575</sup> Return to the description of xjobwriter.

# 4.2.11 xneighborlist

calc:xneighborlist

Determines neighbouring pairs for exciton transport

option	default	unit	description
segments			
type			
cutoff			
constant			
exciton_cutoff			

Return to the description of xneighborlist.

# 4.2.12 iimport

1579 calc:iimpo

Imports electronic couplings from xml of xtp-dipro using folders of pairdump

option	default	unit	description
idft_jobs_file			idft jobs file
probabilityfile_h	ianalyze.ispatia h.out		For coarse grained simulations provide here the distance dependent means and sigmas of hole transfer integrals. This file can be created using the ianalyze calculator.
probabilityfile_e	ianalyze.ispatia e.out		For coarse grained simulations provide here the distance dependent means and sigmas of electron transfer integrals. This file can be created using the ianalyze calculator.

Return to the description of iimport.

# 4.2.13 pairdump

calc:pairdump

Coordinates of molecules and pairs from the neighbor list

option	default	unit	description
molecules			If **true** outputs single molecules, otherwise only pairs

Return to the description of pairdump.

#### **4.2.14** xrates

1585 e:xrates 1586

Hopping rates using classical or semi-classical expression

option	default	unit	description
--------	---------	------	-------------

field			Field in x y z direction
temperature		K	Temperature for rates
method			Method chosen to compute rates. Can either be **marcus** or **jortner**. The first is the high temperature limit of Marcus theory, the second is the rate proposed by Jortner and Bixon
nmaxvib	20		If the method of choice is **jortner**, the maximal number of excited vibrations on the molecules has to be specified as an integer for the summation
omegavib	0.2	eV	If the method of choice is **jortner**, the vibration frequency of the quantum mode has to be given in units of eV. The default value is close to the CC bond-stretch at 0.2eV

Return to the description of xrates.

# 1588

# 4.2.15 sandbox

Sandbox to test xtp classes

option	default	unit	description
ID			Not in use

1590 Return to the description of sandbox.

#### 1591 calc:stateserve

#### 4.2.16 stateserver

Export SQLite file to human readable format

option	default	unit	description
out			Output file name
pdb			PDB coordinate file name
keys			Sections to write to readable format (topology, segments, pairs, coordinates)

Return to the description of stateserver.

# 4.2.17 tdump

alc:tdump

Coarse-grained and back-mapped (using rigid fragments) trajectories

option	default	unit	description
md	MD.pdb		Name of the coarse-grained trajectory
qm	QM.pdb		Name of the trajectory with back-substituted rigid fragments
frames	1		Number of frames to output

Return to the description of tdump.

# 4.2.18 vaverage



Computes site-centered velocity averages from site occupancies

option	default	unit	description
carriers			Carrier types for which to compute velocity averages
tabulate			Tabulate 'atoms' or 'segments'

Return to the description of vaverage.

# 4.2.19 egwbse

1600 calc:egwbs

Executes qm calculations for individual molecules

option	default	unit	description
tasks			
job_file			
gwbse_options			
dftpackage			
esp_options			

Return to the description of egwbse.

# 1603 calc:iexcitoncl

# 4.2.20 iexcitoncl

Classical coupling of transition charges

option	default	unit	description
mapping			
job_file			
emp_file			
induce			
spin			

<sup>1605</sup> Return to the description of iexcitoncl.

# 4.2.21 igwbse



QM calculator for pairs

option	default	unit	description
job_file			
tasks			
dftpackage readjobfile			
singlets			
triplets			

Return to the description of igwbse.

# 4.2.22 qmape



QM/MM with the Thole MM model and ewald long range interaction

option	default	unit	description
pdb_check			PDB file of polar sites
gwbse			Specify if GW/BSE excited state calculation ist needed
gwbse_options			GW/BSE options file
state			Number of excited state, which is to be calculated
type			Character of the excited state to be calculated
filter			Filter with which to find the excited state after each calculation
oscilla- tor_strength			Oscillator strength filter, only states with higher oscillator strength are considered

charge_transfer			Charge transfer filter , only states with charge transfer above threshold are consdered
qmmmconvg			convergence criteria for the QM/MM
dR	0.001	nm	RMS of coordinates
dQ	0.001	e	RMS of charges
dE_QM	0.0001	eV	Energy change of the QM region
dE_MM	0.0001	eV	Energy change of the MM region
max_iter	10		Number of iterations
ewald			
mapping			
mps_table			
pdb_check			
coulombmethod			
method			
cutoff			
shape			
polarmethod			
method			
induce			
cutoff			
tasks			
calculate_fields			
polarize_fg			
evalu-			
ate_energy			
coarsegrain			
cg_background			
cg_foreground			
cg_radius			
cg_anisotropic			
convergence			
energy			
kfactor			
rfactor			

<sup>1611</sup> Return to the description of qmape.

#### 1612 calc:xedft

# 4.2.23 xedft

<sup>1613</sup> A wrapper for first principles based single site calculations

option	default	unit	description
tasks	input,run,parse		What to run
store	orbitals		What to store

Return to the description of xedft.



# 4.2.24 xidft

Projection method for electronic couplings. Requires edft otput

option	default	unit	description
tasks	input,run,parse,		What to do
store	orbitals,overlap		What to store
degeneracy	0	eV	Criterium for the degeneracy of two levels
levels	3		Output between HOMO,, HOMO-levels; LUMO,, LUMO+levels
trim	2		Use trim*occupied of virtual orbitals

Return to the description of xidft.

# 4.2.25 xqmmm

#### 1618 calc:xqmmn

 $\ensuremath{\mathsf{QM}}/\ensuremath{\mathsf{MM}}$  with the Thole MM model for excited states

option	default	unit	description
pdb_check			PDB file of polar sites
write_chk	dipoles.xyz		XYZ file with dipoles split onto point charges
format_chk	xyz		format, gaussian or xyz
split_dpl	1		'0' do not split dipoles onto point charges, '1' do split
dpl_spacing	1e-3	nm	Spacing to be used when splitting dipole onto point charges: d = q * a
dftpackage			DFT package to use for the QM region
gwbse			Specify if GW/BSE excited state calculation ist needed
gwbse_options			GW/BSE options file
state			Number of excited state, which is to be calculated
type			Character of the excited state to be calculated
filter			Filter with which to find the excited state after each calculation
oscilla- tor_strength			Oscillator strength filter, only states with higher oscillator strength are considered
charge_transfer			Charge transfer filter , only states with charge transfer above threshold are consdered
qmmmconvg			convergence criteria for the QM/MM
dR	0.001	nm	RMS of coordinates
dQ	0.001	e	RMS of charges
dE_QM	0.0001	eV	Energy change of the QM region
dE_MM	0.0001	eV	Energy change of the MM region
max_iter	10		Number of iterations
coulombmethod			Options for the MM embedding
method	cut-off		Method for evaluation of electrostatics
cutoff1			Cut-off for the polarizable MM1 shell
cutoff2			Cut-off for the static MM2 shell
tholemodel			Parameters for teh Thole model
induce			'1' - induce '0' - no induction
induce_intra_pair			'1' - include mutual interaction of induced dipoles in the QM region. '0' - do not
exp_damp	0.39		Sharpness parameter
scaling			Bond scaling factors
convergence			Convergence parameters for the MM1 (polarizable) region
wSOR_N			Mixing factor for the succesive overrelaxation algorithm for a neutral QM region
wSOR_C			Mixing factor for the succesive overrelaxation algorithm for a charged QM region
max_iter	512		Maximal number of iterations to converge induced dipoles
tolerance			Maximum RMS change allowed in induced dipoles

Return to the description of xamm.

# 4.2.26 energy2xml



Write out energies from SQL file

option	default	unit	description
option	aciaait	ariic	description

Return to the description of energy2xml.

# 4.2.27 integrals2xml

calc:integrals2xr

1624

Write out transfer integrals from SQL file

option   default   unit   description				
	option	detault	unit	description

Return to the description of integrals 2xml.

# 4.2.28 occupations2xml

calc:occupations2xml 1628

1627

Write out site occupation probabilities from SQL file

ion   default   unit   description
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Return to the description of occupations 2xml.

# 4.2.29 pairs2xml

calc:pairs2xm 1631

Write out neighbourlist from SQL file

. •	1 1 4 1.	1	1
option	detault	unit	description

Return to the description of pairs 2xml.

#### 1633 4.2.30 rates2xml

alc:rates2

Write out charge transfer rates from SQL file

option	default	unit	description

Return to the description of rates2xml.

#### 4.2.31 segments2xml

alc:segments2xm 1637

Write out segment data from SQL file

Return to the description of segments2xml.

# 4.2.32 trajectory2pdb

calc:trajectory2pdl

1639

Generate PDB files for the mapped MD/QM topology

option	default	unit	description
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Return to the description of trajectory2pdb.

# 4.3 Common options

ref:option

name Description of the option	
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# **Bibliography**

ruhle microscopic 2011

```
[2] V. RÃijhle et al., J. Chem. Theory Comput. 5, 3211 (2009), bibtex: ruhle_versatile_2009. ii, 3, 6
     ruhle versatile 2009
                      [3] E. Gamma, R. Helm, and R. E. Johnson, Design Patterns. Elements of Reusable Object-Oriented Software., 1st ed., reprint.
       mma_design_1995
                          ed. (Addison-Wesley Longman, Amsterdam, ADDRESS, 1995), bibtex: gamma_design_1995. 3
                      [4] V. RÃijhle, J. Kirkpatrick, and D. Andrienko, J. Chem. Phys. 132, 134103 (2010), bibtex: ruhle_multiscale_2010. 7
    ruhle_multiscale_2010
                      [5] N. VukmiroviÄĞ and L.-W. Wang, J. Chem. Phys. 128, 121102 (2008), bibtex: vukmirovic_charge_2008. 7
   vukmirovic charge 2008
                      [6] N. VukmiroviÄĞ and L.-W. Wang, Nano Letters 9, 3996 (2009), bibtex: vukmirovic_charge_2009.
   vukmirovic_charge_2009
                      [7] D. P. McMahon and A. Troisi, Chem. Phys. Lett. 480, 210 (2009), bibtex: mcmahon_ad_2009. 7
      mcmahon_ad_2009
                      [8] J.-L. BrÁl'das, D. Beljonne, V. Coropceanu, and J. Cornil, Chem. Rev. 104, 4971 (2004), bibtex: bredas_charge-
   las charge-transfer 2004
                          transfer_2004. 8, 19
                      [9] F. May et al., J. Mater. Chem. 21, 9538 (2011), bibtex: may_relationship_2011. 9, 20
    may_relationship_2011
breneman_determining_1990 [10] C. M. Breneman and K. B. Wiberg, J. Comput. Chem. 11, 361 (1990), bibtex: breneman_determining_1990. 11
    stone_distributed_1985 [11] A. Stone and M. Alderton, Molecular Physics 56, 1047 (1985), bibtex: stone_distributed_1985. 11, 12
     chirlian_atomic_1987 [12] L. E. Chirlian and M. M. Francl, J. Comput. Chem. 8, 894 (1987), bibtex: chirlian_atomic_1987. 11
     singh_approach_1984 [13] U. C. Singh and P. A. Kollman, J. Comput. Chem. 5, 129 (1984), bibtex: singh_approach_1984. 11
    stone_distributed_2005 [14] A. J. Stone, J. Chem. Theory Comput. 1, 1128 (2005), bibtex: stone_distributed_2005. 11, 12
      stone_theory_1997 [15] A. J. Stone, The Theory of intermolecular forces (Clarendon Press, Oxford, 1997), bibtex: stone_theory_1997. 12
     thole_molecular_1981 [16] B. Thole, Chem. Phys. 59, 341 (1981), bibtex: thole_molecular_1981. 13
van_duijnen_molecular_1998 [17] P. T. van Duijnen and M. Swart, J. Phys. Chem. A 102, 2399 (1998), bibtex: van_duijnen_molecular_1998-1. 13
    applequist_atom_1972 [18] J. Applequist, J. R. Carl, and K.-K. Fung, J. Am. Chem. Soc. 94, 2952 (1972), bibtex: applequist_atom_1972. 13
     ren_polarizable_2003 [19] P. Ren and J. W. Ponder, J. Phys. Chem. B 107, 5933 (2003), bibtex: ren_polarizable_2003. 13
    baessler_charge_1993 [20] H. Baessler, Phys. Status Solidi B 175, 15 (1993), bibtex: baessler_charge_1993. 14, 19
troisi_charge-transport_2006 [21] A. Troisi and G. Orlandi, Phys. Rev. Lett. 96, (2006), bibtex: troisi_charge-transport_2006.
      troisi_charge_2009 [22] A. Troisi, D. L. Cheung, and D. Andrienko, Phys. Rev. Lett. 102, 116602 (2009), bibtex: troisi_charge_2009.
                     [23] D. P. McMahon and A. Troisi, ChemPhysChem 11, 2067 (2010), bibtex: mcmahon_organic_2010.
      vehoff_charge_2010 [24] T. Vehoff et al., J. Phys. Chem. C 114, 10592 (2010), bibtex: vehoff_charge_2010. 14
neier_density-functional_2010 [25] B. Baumeier, J. Kirkpatrick, and D. Andrienko, Phys. Chem. Chem. Phys. 12, 11103 (2010), bibtex: baumeier_density-
                           functional_2010. 15, 16, 19
kirkpatrick_approximate_2008 [26] J. Kirkpatrick, Int. J. Quantum Chem. 108, 51 (2008), bibtex: kirkpatrick_approximate_2008. 18
```

[1] V. RÃijhle et al., J. Chem. Theory Comput. 7, 3335 (2011), bibtex: ruhle\_microscopic\_2011. ii, 3, 19

walker\_electrical\_2002 [27] A. B. Walker, A. Kambili, and S. J. Martin, J. Phys-Condens. Mat. 14, 9825 (2002), bibtex: walker\_electrical\_2002. 19

- borsenberger\_charge\_1991 [28] P. M. Borsenberger, L. Pautmeier, and H. BÃd'ssler, J. Chem. Phys. 94, 5447 (1991), bibtex: borsenberger\_charge\_1991.
  - pasveer\_unified\_2005 [29] W. F. Pasveer et al., Phys. Rev. Lett. 94, 206601 (2005), bibtex: pasveer\_unified\_2005. 19
  - bredas\_molecular\_2009 [30] J.-L. BrÃl'das, J. E. Norton, J. Cornil, and V. Coropceanu, Accounts Chem. Res. 42, 1691 (2009), bibtex: bredas\_molecular\_2009. 19
- coropceanu\_charge\_2007 [31] V. Coropceanu et al., Chem. Rev. 107, 926 (2007), bibtex: coropceanu\_charge\_2007.
- nelson\_modeling\_2009 [32] J. Nelson, J. J. Kwiatkowski, J. Kirkpatrick, and J. M. Frost, Accounts Chem. Res. 42, 1768 (2009), bibtex: nelson\_modeling\_2009. 19
- marcus\_electron\_1993 [33] R. A. Marcus, Rev. Mod. Phys. 65, 599 (1993), bibtex: marcus\_electron\_1993. 19
- hutchison\_hopping\_2005 [34] G. R. Hutchison, M. A. Ratner, and T. J. Marks, J. Am. Chem. Soc. 127, 2339 (2005), bibtex: hutchison\_hopping\_2005.
  - chang\_new\_2005 [35] J.-L. Chang, J. Mol. Spectrosc. 232, 102 (2005), bibtex: chang\_new\_2005. 20
- hoffman\_reorganization\_1996 [36] B. M. Hoffman and M. A. Ratner, Inorg. Chim. Acta. 243, 233 (1996), bibtex: hoffman\_reorganization\_1996. 20
  - bottz\_new\_1975 [37] A. B. Bortz, M. H. Kalos, and J. L. Lebowitz, J. Comput. Phys. 17, 10 (1975), bibtex: bortz\_new\_1975. 21
  - scher\_anomalous\_1975 [38] H. Scher and E. Montroll, Phys. Rev. B 12, 2455 (1975), bibtex: scher\_anomalous\_1975. 21
  - borsenberger\_role\_1993 [39] P. M. Borsenberger, E. H. Magin, M. D. VanAuweraer, and F. C. D. Schryver, Phys. Status Solidi A 140, 9 (1993), bibtex: borsenberger\_role\_1993. 21
    - derrida\_velocity\_1983 [40] B. Derrida, J. Stat. Phys. 31, 433 (1983), bibtex: derrida\_velocity\_1983. 21, 30
- cordes\_one-dimensional\_2001 [41] H. Cordes et al., Phys. Rev. B 63, 094201 (2001), bibtex: cordes\_one-dimensional\_2001.
  - seki\_electric\_2001 [42] K. Seki and M. Tachiya, Phys. Rev. B 65, 014305 (2001), bibtex: seki\_electric\_2001. 21
  - lukyanov\_extracting\_2010 [43] A. Lukyanov and D. Andrienko, Phys. Rev. B 82, 193202 (2010), bibtex: lukyanov\_extracting\_2010. 21
- van\_der\_holst\_modeling\_2009 [44] J. J. M. van der Holst et al., Phys. Rev. B 79, 085203 (2009), bibtex: van\_der\_holst\_modeling\_2009. 29
- dunlap\_charge-dipole\_1996 [45] D. Dunlap, P. Parris, and V. Kenkre, Phys. Rev. Lett. 77, 542 (1996), bibtex: dunlap\_charge-dipole\_1996. 30
  - novikov\_essential\_1998 [46] S. V. Novikov et al., Phys. Rev. Lett. 81, 4472 (1998), bibtex: novikov\_essential\_1998. 30
  - nagata\_atomistic\_2008 [47] Y. Nagata and C. Lennartz, J. Chem. Phys. 129, 034709 (2008), bibtex: nagata\_atomistic\_2008. 30
  - novikov\_cluster\_1995 [48] S. V. Novikov and A. V. Vannikov, J. Phys. Chem. 99, 14573 (1995), bibtex: novikov\_cluster\_1995. 30