PRISMS-PF Application Formulation: alloySolidification_uniform

This example application implements a simple model to simulate solidification of a binary alloy A-B in the dilute limit with component B acting as a solute in a matrix of A. The implemented model was introduced by Karma [1] in 2001. In this model, latent heat is assumed to diffuse much faster than impurities and, therefore, the temperature field is considered to be fixed by external conditions. In contrast to alloySolidification, this application considers solidification under uniform temperature and no diffusion in the solid. In the default settings of the application, the simulation starts with a circular solid in the corner of a square system. The evolution of the system is calculated for an initial scaled supersaturation value, Ω . As this seed grows, three variables are tracked, an order parameter, ϕ , that denotes whether the material a liquid ($\phi = -1$) or solid ($\phi = 1$), the solute concentration, c, and an auxiliary term, ξ .

1 Model

The free energy of the system can be written as the functional [2]

$$\mathcal{F}[\phi, c, T] = \int_{\Omega} \left[\frac{\sigma}{2} |\nabla \phi|^2 + Hf(\phi, T_M) + f_{AB}(\phi, c, T) \right] dV, \tag{1}$$

where σ is penalty coefficient for phase gradients. The term $f(\phi, T_M)$ is a symmetric double-well term evaluated at the melting temperature of the pure material A, T_M . The constant H is the height of the well and $f_{AB}(\phi, c, T)$ accounts for the relative stability of the liquid and solid phases at different temperatures, according to the phase diagram. The double-well term has the standard form given by

$$f(\phi, T_M) = -\phi^2/2 + \phi^4/4. \tag{2}$$

For a dilute binary alloy, f_{AB} can be written as

$$f_{AB}(\phi, c, T) = f^{A}(T_{M}) - (T - T_{M})s(\phi) + \frac{RT_{M}}{v_{o}}(c \ln c - c) + \epsilon(\phi)c, \tag{3}$$

where $f^A(T_M)$ is the free energy density of pure A at its melting point, v_0 is the molar volume of A and R is the gas constant. The functions $s(\phi)$ and $\epsilon(\phi)$ are interpolation functions for the entropy and internal energy of the solid and liquid phases, respectively. The general form of the coupled governing equations for the ϕ and c is

$$\frac{\partial \phi}{\partial t} = -K_{\phi} \frac{\delta \mathcal{F}}{\delta \phi} \tag{4}$$

and

$$\frac{\partial c}{\partial t} = \nabla \cdot \left(M(\phi, c) \frac{\delta \mathcal{F}}{\delta c} - \vec{j}_{at} \right), \tag{5}$$

where K_{ϕ} is a kinetic constant, $M(\phi, c)$ is the mobility of solute atoms and \vec{j}_{at} is a nonvariational anti-trapping solute current required to correct for spurious effects that arise from considering an interface thickness much larger than the physical solid-liquid interface. in the dilute limit, the solidus and liquidus lines of the T vs c phase diagram are defined by the equations

$$T_l = T_M - |m|c_l \tag{6}$$

and

$$T_s = T_M - \frac{|m|}{k}c_s,\tag{7}$$

where m is the liquidus slope and $k = c_s/c_l$ is the partition coefficient, which relates the equilibrium concentrations, c_l and c_s , of the liquid and solid, respectively.

Considering solidification at a temperature $T_0 < T_M$, we define $c_l^0 = c(T_0)$ as the equilibrium liquid concentration and kc_l^0 as the equilibrium solid concentration. After nondimensionalization, the coupled governing equations for the ϕ and c can be written as [1]

$$\tau \frac{\partial \phi}{\partial t} = \xi(\phi, c), \tag{8}$$

where

$$\xi(\phi, c) = -\frac{\partial f}{\partial \phi} - \frac{\lambda}{1 - k} g'(\phi)(e^u - 1) + W^2 \nabla^2 \phi \tag{9}$$

and

$$\frac{\partial c}{\partial t} = \nabla \cdot \vec{j},\tag{10}$$

where

$$\vec{j} = -Dcq(\phi)\nabla u - aWc_l^0(1-k)e^u\frac{\partial\phi}{\partial t}\frac{\nabla\phi}{|\nabla\phi|},$$
(11)

and

$$u(c,\phi) = \ln\left(\frac{2c/c_l^0}{1 + k - (1-k)h(\phi)}\right).$$
 (12)

The constant λ is defined as $\lambda = a_1 W/d_0$, where d_0 is the microscopic capillary length. The constants W and τ are the unit length and time, respectively and D is the diffusivity in the liquid. The value of D is set to $D = a_2 W^2 \lambda/\tau$ so that interface kinetics are eliminated from the velocity-dependent Gibbs-Thompson condition. The values of a, a_1 , a_2 are given in Section 2. The functions $g'(\phi)$, $h(\phi)$ and $q(\phi)$ are given by

$$g'(\phi) = (1 - \phi^2)^2, \tag{13}$$

$$q(\phi) = \frac{1 - \phi}{1 + k - (1 - k)h(\phi)},\tag{14}$$

and

$$h(\phi) = \phi. \tag{15}$$

Crystalline anisotropy is introduced by generalizing Eqs. (8) and (9) to

$$\tau(\theta)\frac{\partial\phi}{\partial t} = \xi(\phi, c),\tag{16}$$

where

$$\xi(\phi,c) = -f'(\phi) - \frac{\lambda}{1-k}g'(\phi)(e^u - 1) + \nabla \cdot \left[W(\theta)^2 \nabla \phi\right] - \frac{\partial}{\partial x} \left[W(\theta)W'(\theta)\frac{\partial \phi}{\partial y}\right] + \frac{\partial}{\partial y} \left[W(\theta)W'(\theta)\frac{\partial \phi}{\partial x}\right], \tag{17}$$

where θ is the angle between the outward normal of the solid-liquid interface and the positive x axis. The constants W and τ now depend on this angle and are given by $W(\theta) = Wa_s(\theta)$ and $\tau(\theta) = \tau a_s(\theta)^2$, respectively, where $a_s(\theta) = 1 + \epsilon_4 \cos(4\theta)$.

The initial condition is set by placing a solid seed in an undercooled system with a uniform scaled supersaturation value, $\Omega = (c_l^0 - c_{\infty})/[c_l^0(1-k)]$.

2 Model Constants

Symbol	Value	Description
c_l^0	1	Reference concentration (equilibrium liquid concentration at T_0)
k	0.15	Partition coefficient
W	1	Unit length
τ	1	Unit time
d_0/W	0.277	Microscopic capillary length (with respect to W)
a	$1/(2\sqrt{2})$	Antitraping term constant
a_1	0.8839	Coefficient a_1
a_2	0.6267	Coefficient a_2
λ	a_1W/d_0	Parameter λ
D	$a_2\lambda W^2/\tau$	Solute diffusivity in the liquid
ϵ_4	0.02	Anisotropy strength
Ω	0.55	Scaled supersaturation

3 Time Discretization

Considering forward Euler explicit time stepping, we have the time discretized kinetics equations:

$$\phi^{n+1} = \phi^n + \frac{\xi^n}{\tau^n} \Delta t,\tag{18}$$

$$c^{n+1} = c^n + \Delta t \left\{ Dc^n q(\phi^n) \nabla(u^n) + aWc_l^0 (1 - k) (e^u)^n \left(\frac{\partial \phi}{\partial t} \right)^n \frac{\nabla \phi^n}{|\nabla \phi^n|} \right\}$$
 (19)

and

$$\xi(\phi, c) = -f'(\phi^n) - \frac{\lambda}{1 - k} g'(\phi^n) [(e^u)^n - 1]$$

$$+ \nabla \cdot [W(\theta^n)^2 \nabla \phi^n] - \frac{\partial}{\partial x} \left[W(\theta^n) W'(\theta^n) \frac{\partial \phi^n}{\partial y} \right] + \frac{\partial}{\partial y} \left[W(\theta^n) W'(\theta^n) \frac{\partial \phi^n}{\partial x} \right].$$

$$(20)$$

4 Weak Formulation

The weak form of the time-discretized equations for ϕ , c, and, ξ is

$$\int_{\Omega} \omega \phi^{n+1} \ dV = \int_{\Omega} \omega \underbrace{\left(\phi^n + \frac{\xi^n}{\tau(\theta^n)} \Delta t\right)}_{r_+} \ dV, \tag{21}$$

$$\int_{\Omega} \omega c^{n+1} \ dV = \int_{\Omega} \omega \underbrace{c^n}_{r_n} \ dV$$

$$+ \int_{\Omega} \nabla \omega \cdot \underbrace{\left[-\Delta t D \left(q(\phi^{n}) \nabla c^{n} + \frac{(1-k)q(\phi^{n})c^{n} \nabla (\phi^{n})}{1+k-(1-k)\phi^{n}} \right) - \Delta t a W c_{l}^{0} (1-k)(e^{u})^{n} \left(\frac{\partial \phi}{\partial t} \right)^{n} \frac{\nabla \phi^{n}}{|\nabla \phi^{n}|} \right]}_{r_{c}x} dV, \tag{22}$$

and

$$\int_{\Omega} \omega \xi^{n+1} \ dV = \int_{\Omega} \omega r_{\xi} \ dV + \int_{\Omega} \nabla \omega r_{\xi x} \ dV \tag{23}$$

where

$$r_{\xi} = -f'(\phi^n) - \frac{\lambda}{1 - k} g'(\phi^n) [(e^u)^n - 1]$$
(24)

and

$$r_{\xi x} = -\left[W(\theta^n)^2 \frac{\partial \phi^n}{\partial x} - W(\theta^n) W'(\theta^n) \frac{\partial \phi^n}{\partial y}\right] \hat{x}$$

$$-\left[W(\theta^n)^2 \frac{\partial \phi^n}{\partial y} + W(\theta^n) W'(\theta^n) \frac{\partial \phi^n}{\partial x}\right] \hat{y}$$
(25)

References

- [1] A. Karma, Phase-Field Formulation for Quantitative Modeling of Alloy Solidification, *Phys. Rev. Lett.* 87, 115701 (2001).
- [2] B. Echabarria, R. Folch, A. Karma, and M. Plapp, Quantitative phase-field model of alloy solidification, *Phys. Rev. E* **70**, 061904 (2004).