

# Density Functional Theory: Part 2. Practical concepts & Basic Tutorials with Quantum Espresso

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# Input file: Advanced topics

1. Metals and fractional occupations
2. Structural optimization
3. Low-dimensional systems
4. Spin polarization & magnetism  
(not examined here)

&CONTROL

```
calculation = "relax",  
etot_conv_thr = 1.0D-4  
forc_conv_thr = 1.0D-3
```

```
prefix = "CO",  
pseudo_dir = "./",
```

/

&SYSTEM

```
ibrav = 0
```

```
nat = 2, ntyp = 2,  
nbnd = 6,  
ecutwfc = 50,
```

```
nspin = 2,  
starting_magnetization(1)=0.1
```

```
occupations='smearing',  
degauss = 0.001,
```

/

&ELECTRONS

/

&IONS

/

```
CELL_PARAMETERS {bohr}  
30.0 0.0 0.0  
0.0 12.0 0.0  
0.0 0.0 12.0
```

ATOMIC\_SPECIES

```
O 1.00 O.pw-mt_fhi.UPF
```

```
C 1.00 C.pw-mt_fhi.UPF
```

ATOMIC\_POSITIONS {bohr}

```
C 5.0 0.0 0.0 1 0 0
```

```
O 0.0 0.0 0.0 0 0 0
```

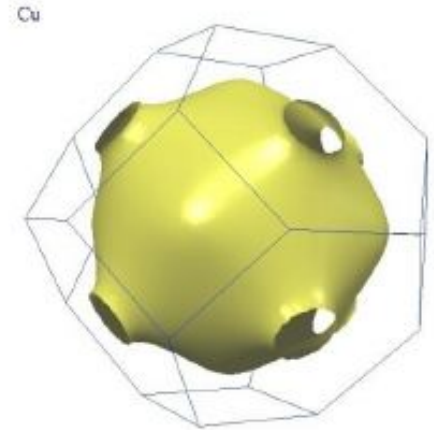
```
K_POINTS {Gamma}
```

# 1. Metals and fractional occupations

- Recall:

$$\langle P \rangle = \frac{\Omega}{(2\pi)^3} \sum_{n \text{ occ} BZ} \int P_n(\mathbf{k}) d^3k$$

- For metals, at  $T=0$ , this corresponds to (for highest band) an integral over all **wave-vectors contained within the Fermi surface**, i.e., for highest band, **sharp discontinuity** in k-space between occupied and unoccupied states...need many k-points to reproduce this accurately.
- Also can lead to **scf convergence problems** because of band-crossings above/below Fermi level.
- Solve by “smearing”.



Fermi Surface of Cu  
[iramis.cea.fr](http://iramis.cea.fr)

# Smearing in Quantum-ESPRESSO

**occupations 'smearing'**

Instruction: use smearing

smearing 'gaussian'  
'methfessel-paxton'  
'marzari-vanderbilt'  
'fermi-dirac'

Type of  
smearing

**degauss**

Smearing width

In Ry typical 0.01

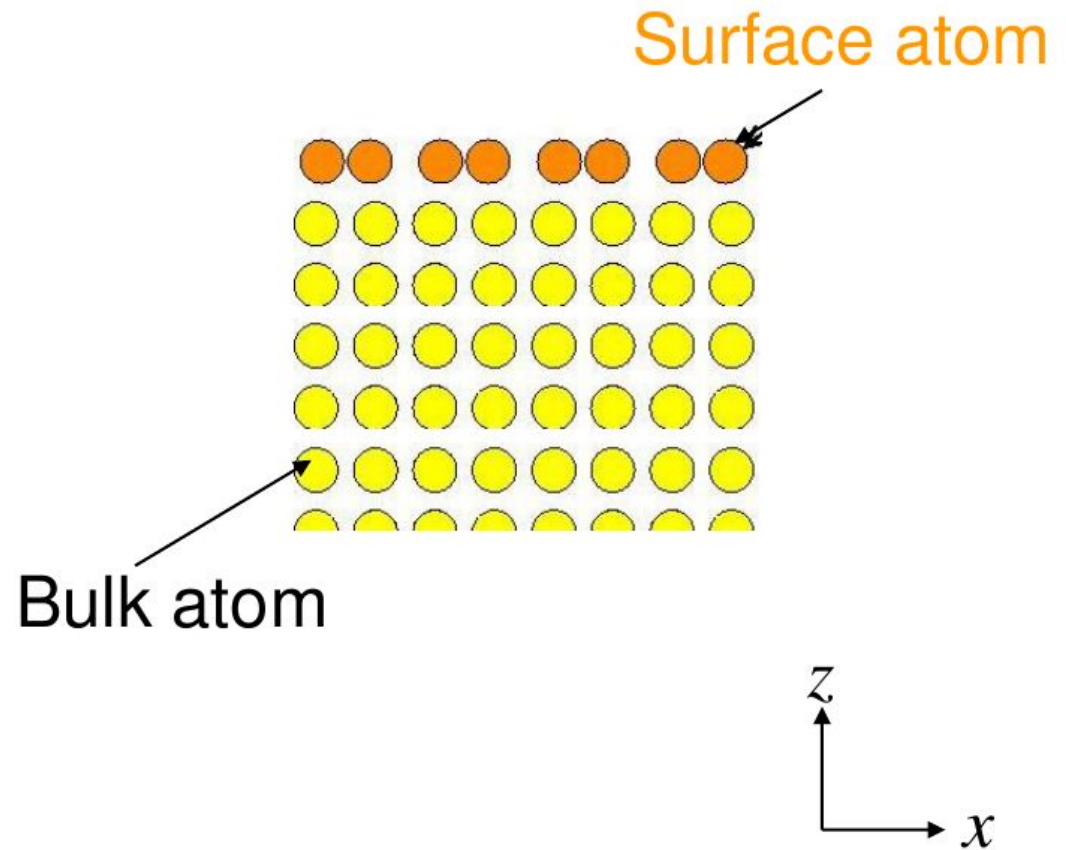
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Methfessel & Paxton, Phys. Rev. B 40, 3616 (1989).  
Marzari & Vanderbilt, Phys Rev. Lett. 82, 3296 (1999).



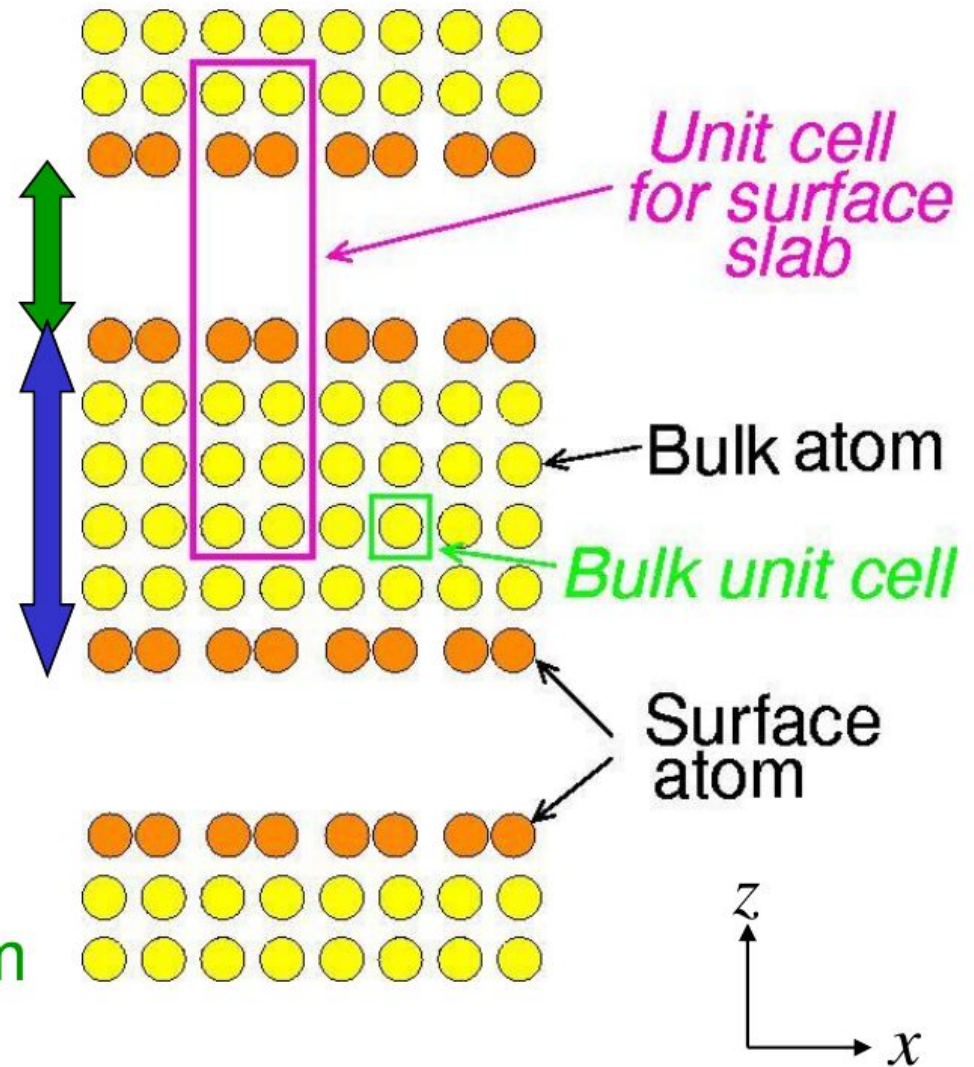
## 2 What if the system is not periodic?

- Example 1: Want to study properties of a system with a **surface**.
- Presence of surface  $\Rightarrow$  No periodicity along  $z$ .



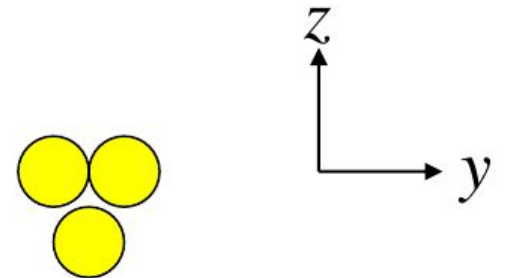
# What if the system is not periodic?

- Example 1: Want to study properties of a system with a **surface**.
- Presence of surface  $\Rightarrow$  No periodicity along  $z$ .
- Use a **supercell**: **artificial periodicity along  $z$**  by repeating slabs separated by **vacuum**.
- Have to check convergence w.r.t. **slab thickness** & **vacuum thickness**.



# What if the system is not periodic?

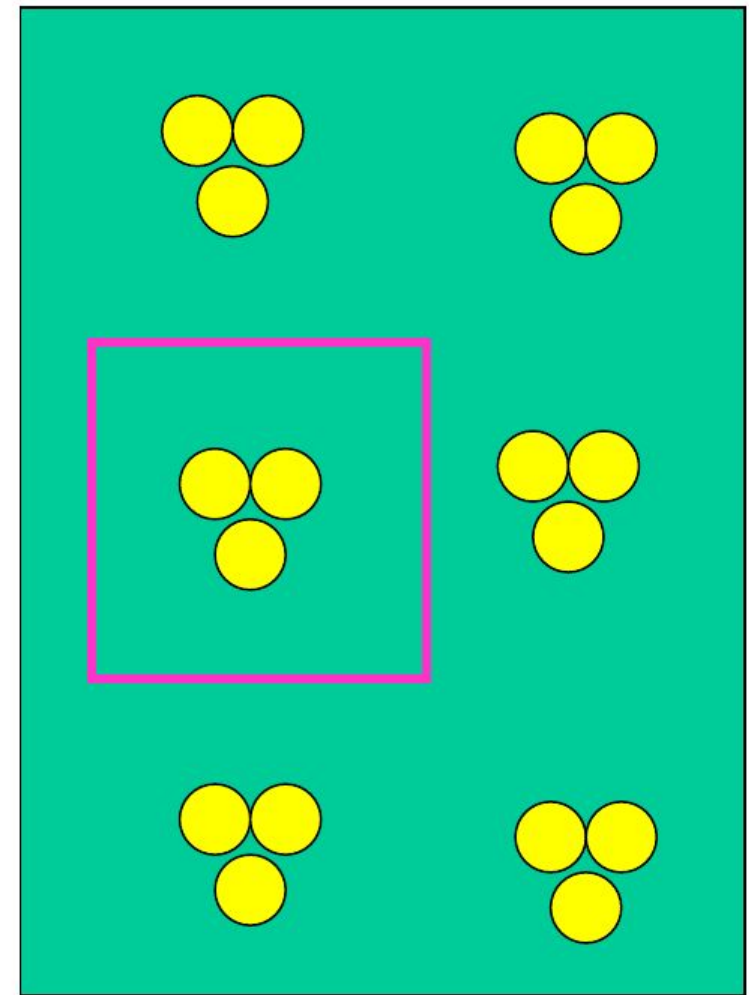
- Example 2: Want to study properties of a nanowire.
- Example 3: Want to study properties of a cluster





# What if the system is not periodic?

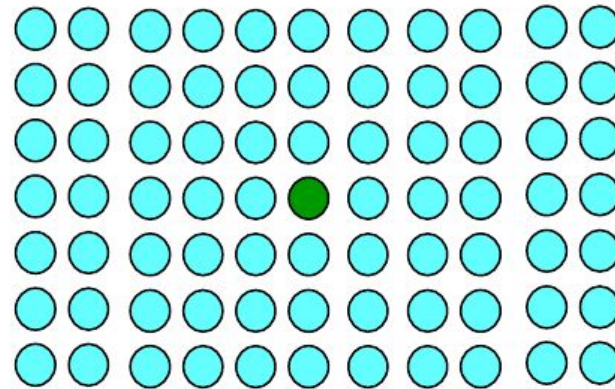
- Example 2: Want to study properties of a nanowire  $\Rightarrow$  introduce **artificial periodicity along  $y$  &  $z$** .
- Example 3: Want to study properties of a cluster  $\Rightarrow$  introduce **artificial periodicity along  $x$ ,  $y$  &  $z$** .





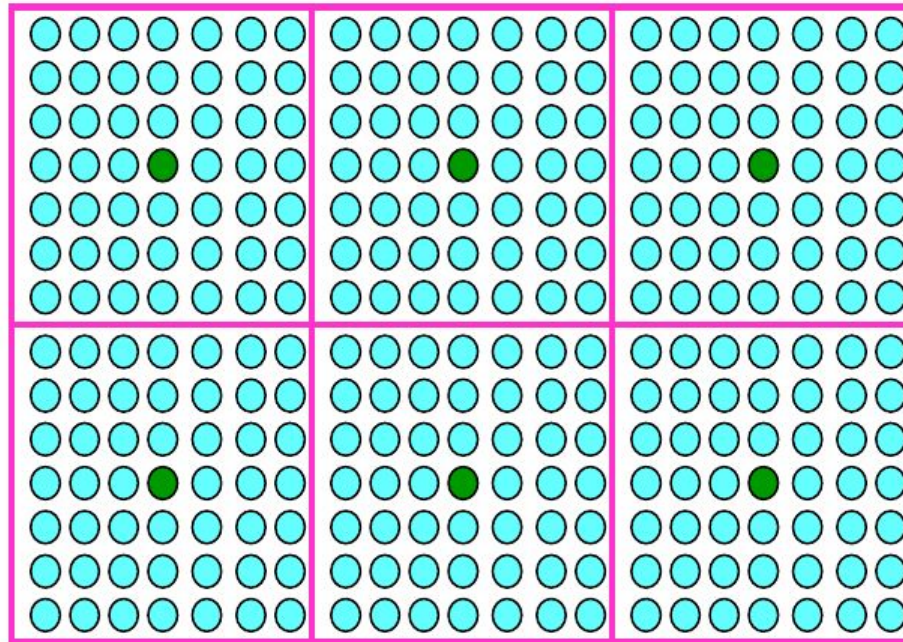
# What if the system is not periodic?

- Example 4: Want to study a system with a defect, e.g., a **vacancy or impurity**:



# What if the system is not periodic?

- Example 4: Want to study a system with a defect, e.g., a **vacancy or impurity**:



# Artificially Periodic Systems $\Rightarrow$ Large Unit Cells

- Note: In all these cases, to minimize the effects of the artificially introduced periodicity, need a large unit cell.



- Long  $\mathbf{a}_1, \mathbf{a}_2, \mathbf{a}_3$  (primitive lattice vectors)



- Short  $\mathbf{b}_1, \mathbf{b}_2, \mathbf{b}_3$  (primitive reciprocal lattice vectors)



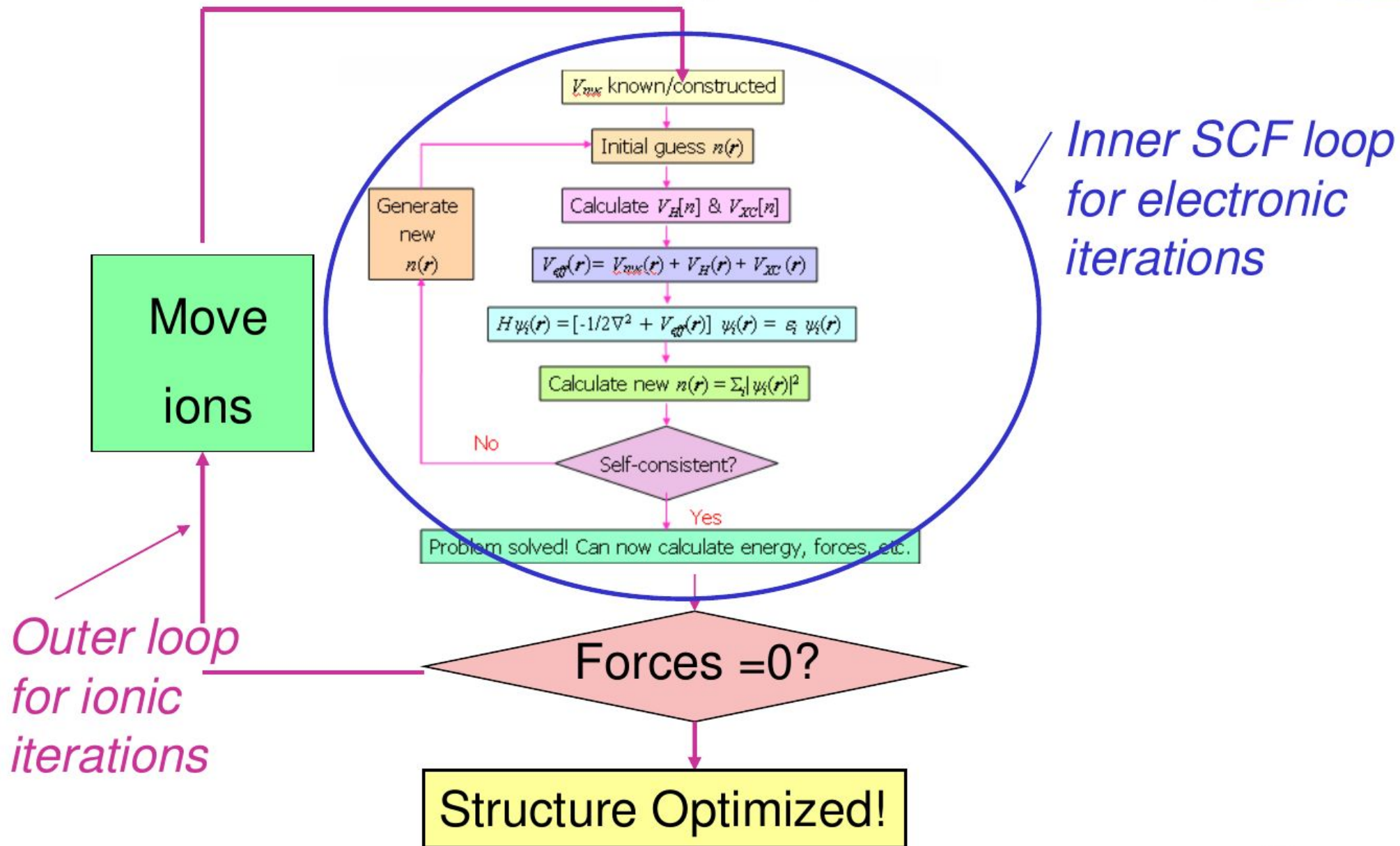
- Many  $\mathbf{G}$ 's will fall within  $E_{cut}$  sphere!

$\Rightarrow$  Many G-vectors (for the same cutoff)

$\Rightarrow$  Fewer k-points (2D grid, 1D grid, or Gamma point only)



### 3 An Outer Loop: Ionic Relaxation





# Forces

- Need for geometry optimization and molecular dynamics.
- Could get as finite differences of total energy - too expensive!
- Use force (Hellmann-Feynman) theorem:
  - Want to calculate the force on ion  $I$ :

$$\mathbf{F}_I = -\frac{d}{d\mathbf{R}_I} \langle \Psi | H | \Psi \rangle$$

- Get three terms:

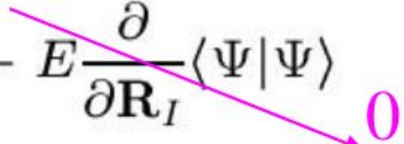
$$\mathbf{F}_I = -\langle \Psi | \frac{\partial H}{\partial \mathbf{R}_I} | \Psi \rangle - \langle \frac{\partial \Psi}{\partial \mathbf{R}_I} | H | \Psi \rangle - \langle \Psi | H | \frac{\partial \Psi}{\partial \mathbf{R}_I} \rangle$$

When  $|\Psi\rangle$  is an eigenstate,  $H|\Psi\rangle = E|\Psi\rangle$

-Substitute this...

# Forces (contd.)

- The **force** is now given by

$$\begin{aligned}\mathbf{F}_I &= -\langle \Psi | \frac{\partial H}{\partial \mathbf{R}_I} | \Psi \rangle - E \langle \frac{\partial \Psi}{\partial \mathbf{R}_I} | \Psi \rangle - E \langle \Psi | \frac{\partial \Psi}{\partial \mathbf{R}_I} \rangle \\ &= -\langle \Psi | \frac{\partial H}{\partial \mathbf{R}_I} | \Psi \rangle - E \frac{\partial}{\partial \mathbf{R}_I} \langle \Psi | \Psi \rangle\end{aligned}$$


- Note that we can now **calculate the force** from a calculation at **ONE configuration alone** – huge savings in time.
- If the basis depends upon ionic positions (not true for plane waves), would have extra terms = **Pulay forces**.
- $|\Psi\rangle$  should be exact eigenstate, i.e., scf well-converged!

Input parameter `tpnfor`

# Geometry Optimization With Forces

- Especially useful for optimizing internal degrees of freedom, surface relaxation, etc.

```
calculation = 'relax'
```

- Choice of algorithms for ionic relaxation, e.g., steepest descent, BFGS.

```
NAMELIST &IONS
```

```
Input parameter ion_dynamics
```