

# Density Functional Theory: Practical concepts & Basic Tutorials with Quantum Espresso

**Dr. Conor Hogan**

Istituto di Struttura della Materia (ISM-CNR) and Dip. Fisica Univ. Roma "Tor Vergata"

[conor.hogan@ism.cnr.it](mailto:conor.hogan@ism.cnr.it) or [conor.d.hogan@gmail.com](mailto:conor.d.hogan@gmail.com) Corridoio C0, C002 bis, x4894

# quantum-ESPRESSO



QE is a powerful density functional theory code that uses planewaves and pseudopotentials.

A wide range of **Documentation** is available at

[quantum-espresso.org](http://quantum-espresso.org)

including links to video lectures and tutorials, links to pseudopotentials and external PP databases, etc.

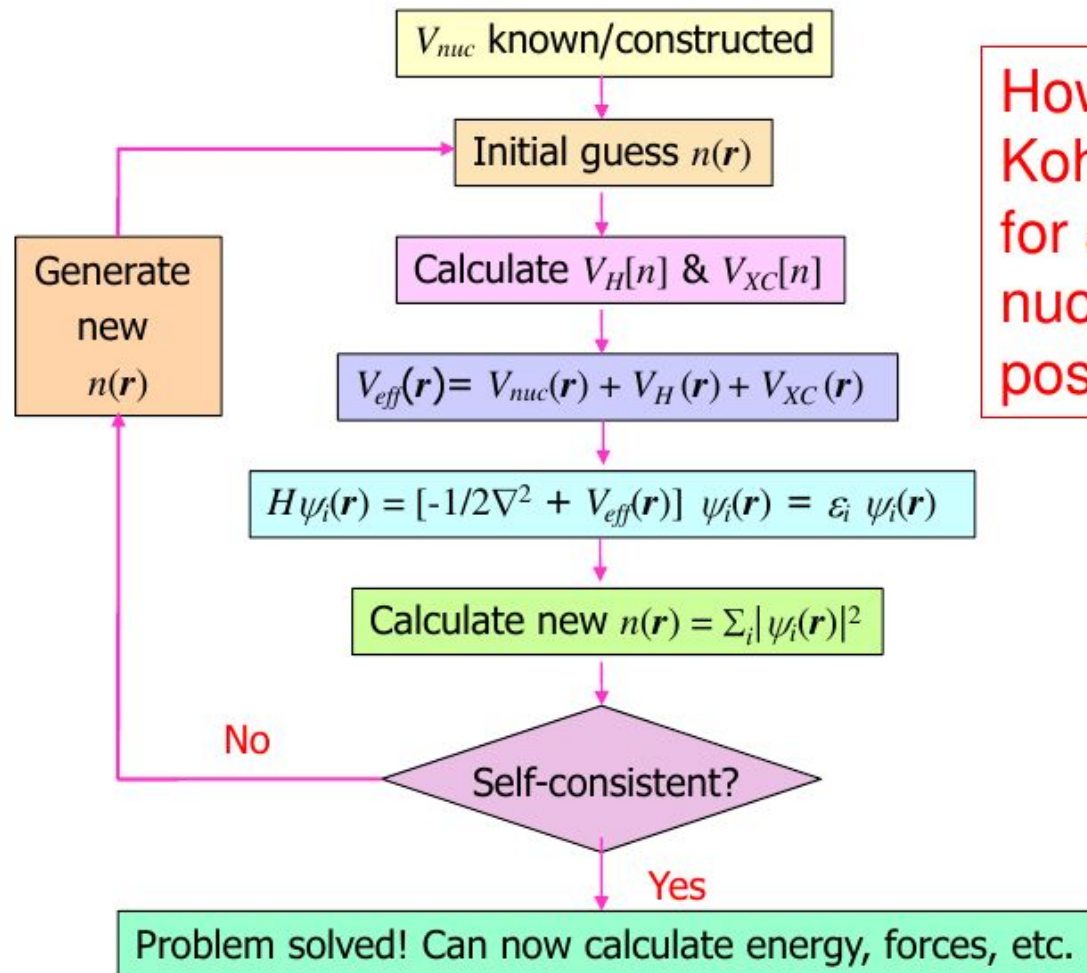
**Input file variables** are explained in [quantum-espresso.org/Doc/INPUT\\_PW.html](http://quantum-espresso.org/Doc/INPUT_PW.html)

also found in the distributed code in QEDIR/**Doc/INPUT\_PW.txt**

## ▣ Executables

- ▣ pw.x      main code
- ▣ pp.x      data analysis
- ▣ ph.x      phonons & electron-phonon coupling
- ▣ dos.x     density of states
- ▣ bands.x   plot band structure
- ▣ projwfc.x projection onto atomic orbitals
- ▣ pwcond.x transmittance
- ▣ cp.x      molecular dynamics
  
- ▣ more tools available:  
    *x-ray spectra, optics ,EELS, superconductivity,*  
    *magnetic resonance (NMR) etc..*

# Self-consistent iterative solution of Kohn-Sham equations



How to solve the Kohn-Sham eqns. for a set of fixed nuclear (ionic) positions.

# Input file structure

**&control**

```
calculation = 'scf'  
prefix = 'silicon',  
pseudo_dir = './'  
outdir = './',
```

/

**&system**

```
ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,
```

/

**&electrons**

/

**ATOMIC\_SPECIES**

```
Si 28.086 Si.vbc.UPF
```

**ATOMIC\_POSITIONS {crystal}**

```
Si 0.00 0.00 0.00
```

```
Si 0.25 0.25 0.25
```

**K\_POINTS automatic**

```
2 2 2 0 0 0
```

**NAMELIST**

Variables in any order

If not given, default values are used

**NAMELIST**

The 3 NAMELISTS and INPUT\_CARDS shown here are **always** present

**NAMELIST**

**INPUT\_CARD**

Fixed format inside each INPUT\_CARD

**INPUT\_CARD**

Order of INPUT\_CARDS is not crucial

**INPUT\_CARD**

# 1. CONTROL namelist

**&control**

```
calculation = 'scf'  
prefix = 'silicon',  
pseudo_dir = './'  
outdir = './',
```

/

**&system**

```
ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,
```

/

**&electrons**

/

**ATOMIC\_SPECIES**

Si 28.086 Si.vbc.UPF

**ATOMIC\_POSITIONS {crystal}**

Si 0.00 0.00 0.00

Si 0.25 0.25 0.25

**K\_POINTS automatic**

2 2 2 0 0 0

- Used to specify what kind of calculation to be run:  
calculation = scf/relax/bands/nscf
- Location of pseudopotential, output, temporary directories
- Convergence parameters on forces and energies
- Controls writing of data to disk or memory

## 2. SYSTEM namelist

**&control**

calculation = 'scf'  
prefix = 'silicon',  
pseudo\_dir = './'  
outdir = './',

/

**&system**

ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,

/

**&electrons**

/

**ATOMIC\_SPECIES**

Si 28.086 Si.vbc.UPF

**ATOMIC\_POSITIONS {crystal}**

Si 0.00 0.00 0.00

Si 0.25 0.25 0.25

**K\_POINTS automatic**

2 2 2 0 0 0

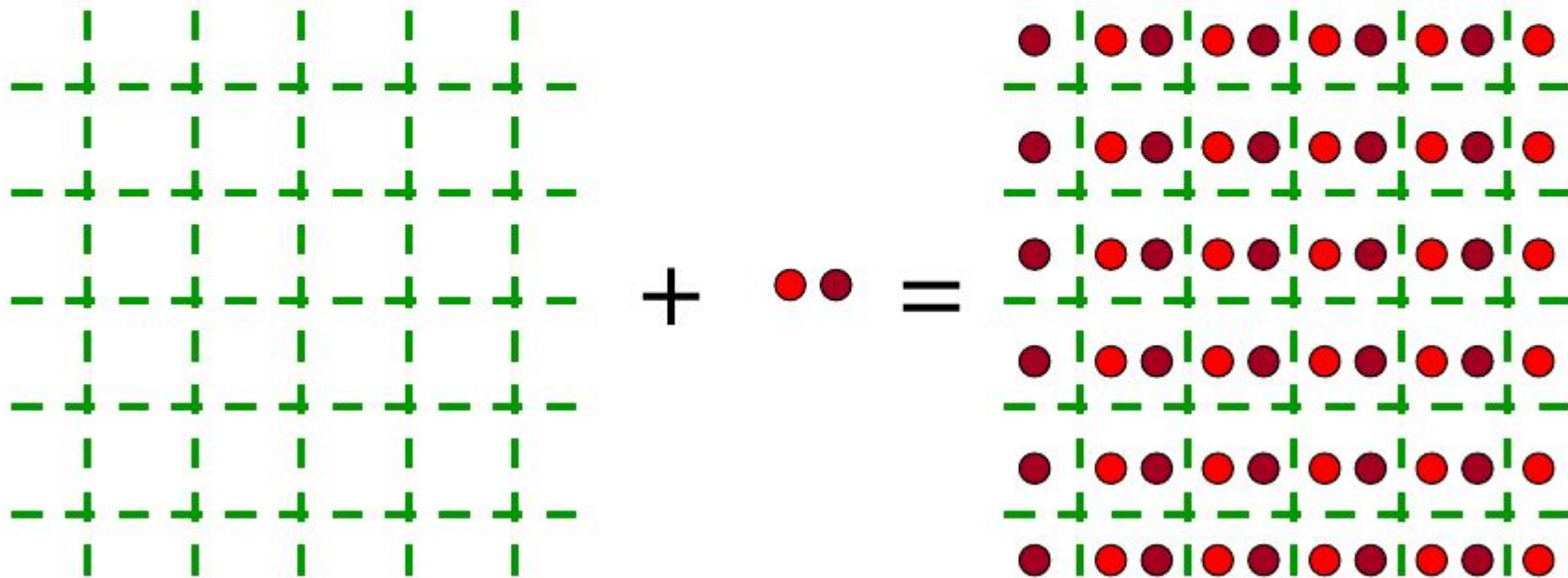
Define the periodic system:

- Bravais lattice
- Lattice parameters
- Number of atoms
- Number of elements
- Expansion over planewaves

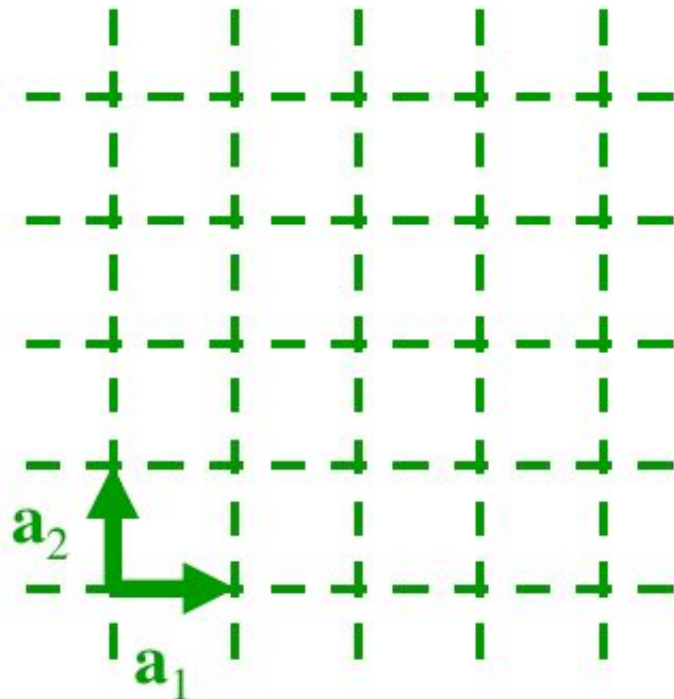


# How to Specify the System

- All periodic systems can be specified by a **Bravais Lattice** and an **atomic basis**.



# How to Specify the Bravais Lattice / Unit Cell



## Input parameter **ibrav**

- Gives the type of **Bravais lattice** (SC, BCC, Hex, etc.)

## Input parameters { **celldm(i)** }

- Give the lengths [& directions, if necessary] of the lattice vectors  $a_1$ ,  $a_2$ ,  $a_3$

- Note that one can choose a non-primitive unit cell (e.g., 4 atom SC cell for FCC structure).



# Bravais Lattices: **ibrav** flag

[https://en.wikipedia.org/wiki/Bravais\\_lattice](https://en.wikipedia.org/wiki/Bravais_lattice)

**ibrav** = 1, 3, 2

**ibrav** = 6, 7

**ibrav** = 8, 9, 10, 11

**ibrav** = 4, 5, -5

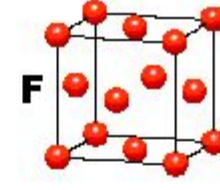
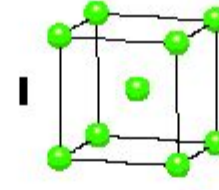
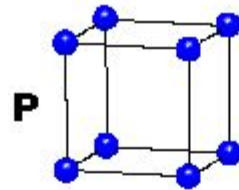
**ibrav** = 12, -12, 13

**ibrav** = 14

## CUBIC

$$a = b = c$$

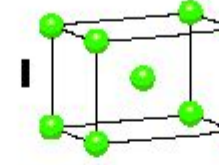
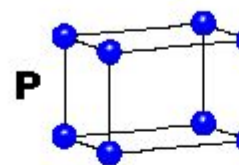
$$\alpha = \beta = \gamma = 90^\circ$$



## TETRAGONAL

$$a = b \neq c$$

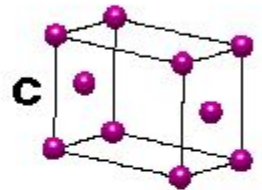
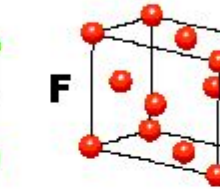
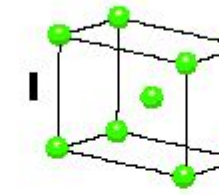
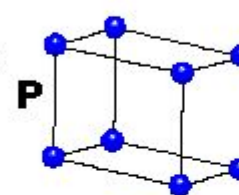
$$\alpha = \beta = \gamma = 90^\circ$$



## ORTHORHOMBIC

$$a \neq b \neq c$$

$$\alpha = \beta = \gamma = 90^\circ$$

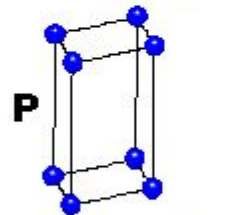


## HEXAGONAL

$$a = b \neq c$$

$$\alpha = \beta = 90^\circ$$

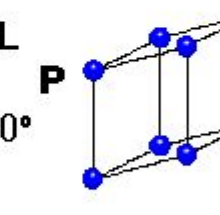
$$\gamma = 120^\circ$$



## TRIGONAL

$$a = b = c$$

$$\alpha = \beta = \gamma \neq 90^\circ$$

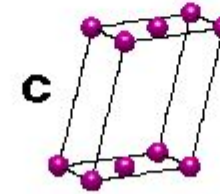
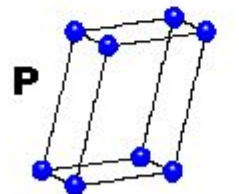


## MONOCLINIC

$$a \neq b \neq c$$

$$\alpha = \gamma = 90^\circ$$

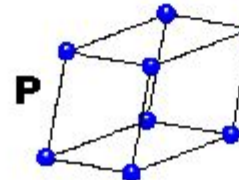
$$\beta \neq 120^\circ$$



## TRICLINIC

$$a \neq b \neq c$$

$$\alpha \neq \beta \neq \gamma \neq 90^\circ$$



### 4 Types of Unit Cell

**P** = Primitive

**I** = Body-Centred

**F** = Face-Centred

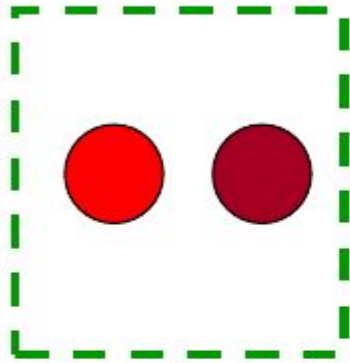
**C** = Side-Centred

+

7 Crystal Classes

→ 14 Bravais Lattices

# Atoms Within Unit Cell – How many, where?



Input parameter **nat**

- Number of atoms in the unit cell

Input parameter **ntyp**

- Number of types of atoms

## FIELD **ATOMIC\_POSITIONS**

- Initial positions of atoms (may vary when “**relax**” done).
- Can choose to give in units of lattice vectors (“**crystal**”) or in Cartesian units (“**alat**” or “**bohr**” or “**angstrom**”)

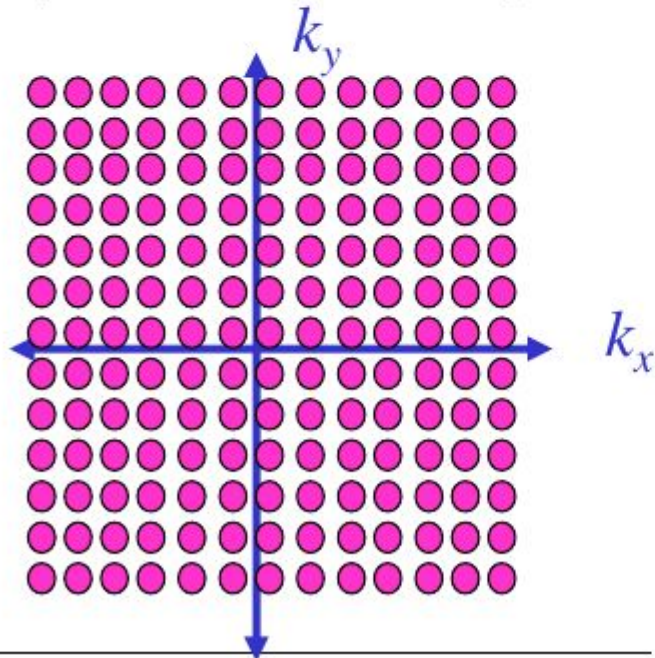
# Plane Waves & Periodic Systems

- For a periodic system:

$$\psi_{\mathbf{k}}(\mathbf{r}) = \frac{1}{\Omega} \sum_{\mathbf{G}} c_{\mathbf{k},\mathbf{G}} e^{i(\mathbf{k}+\mathbf{G})\cdot\mathbf{r}}$$

where  $\mathbf{G}$  = reciprocal lattice vector

- The **plane waves** that appear in this expansion can be represented as a grid in k-space:



- Only true for periodic systems that grid is discrete.
- In principle, still need **infinite number of plane waves**.

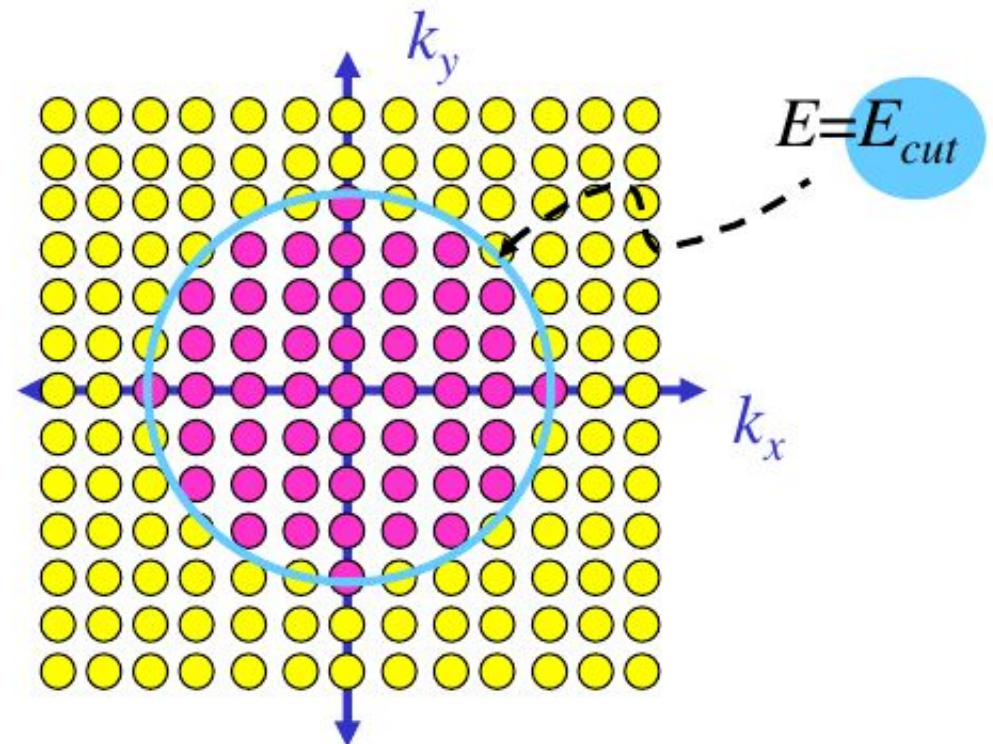


# Truncating the Plane Wave Expansion

- In practice, the contribution from higher Fourier components (large  $|\mathbf{k}+\mathbf{G}|$ ) is small.
- So truncate the expansion at some value of  $|\mathbf{k}+\mathbf{G}|$ .
- Traditional to express this cut-off in energy units:

$$\frac{\hbar^2 |\mathbf{k} + \mathbf{G}|^2}{2m} \leq E_{cut}$$

Input parameter **ecutwfc**



### 3. ELECTRONS namelist

**&control**

calculation = 'scf'  
prefix = 'silicon',  
pseudo\_dir = './'  
outdir = './',

/

**&system**

ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,

/

**&electrons**

/

ATOMIC\_SPECIES

Si 28.086 Si.vbc.UPF

ATOMIC\_POSITIONS {crystal}

Si 0.00 0.00 0.00

Si 0.25 0.25 0.25

K\_POINTS automatic

2 2 2 0 0 0

Parameters that define the self  
consistent iterative solution

For typical use, the default  
parameters are sufficient



**&control**

calculation = 'scf'  
prefix = 'silicon',  
pseudo\_dir = './'  
outdir = './',

/

**&system**

ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,

/

**&electrons**

/

**ATOMIC\_SPECIES**

Si 28.086 Si.vbc.UPF

**ATOMIC\_POSITIONS {crystal}**

Si 0.00 0.00 0.00

Si 0.25 0.25 0.25

**K\_POINTS automatic**

2 2 2 0 0 0

## 4. ATOMIC\_SPECIES input card

- Element symbols
- Atomic weights (only used in molecular dynamics)
- Pseudopotentials...

## PSEUDOPOTENTIALS

More about pseudopotentials

SSSP on Materials Cloud

Pseudo DoJo

ONCV Potentials

SCAN pseudopotentials

PSlibrary table

Original QE PP table

Hartwigesen-Goedecker-Hutter PP table

Old FHI PP table

## PSLIBRARY

Ready-to-use pseudopotentials from **PSlibrary** (*recommended*). For other ready-to-use tables, follow the links of the menu at the left. For more info, see [here](#).

*Please cite the pseudopotentials used and give proper credit to their authors (see [this page](#) for a rather complete list of acknowledgments).*

1 H																	2 He		
3 Li	4 Be													5 B	6 C	7 N	8 O	9 F	10 Ne
11 Na	12 Mg													13 Al	14 Si	15 P	16 S	17 Cl	18 Ar
19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr		
37 Rb	38 Sr		39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe	
55 Cs	56 Ba	57-70 *	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn	
87 Fr	88 Ra	89-102 **	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt										

*	57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb
**	89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No

# Pseudopotentials for Quantum-ESPRESSO

Name: **Oxygen**

Symbol: **O**

Atomic number: **8**

Atomic configuration: **[He] 2s2 2p4**

Atomic mass: **15.9994 (3)**

Available pseudopotentials:

**O.blyp-mt.UPF** (details)

Becke-Lee-Yang-Parr (BLYP) exch-corr  
Martins-Troullier

**O.pbe-rrkjus.UPF** (details)

Perdew-Burke-Ernzerhof (PBE) exch-corr  
Rabe Rappe Kaxiras Joannopoulos (ultrasoft)

**O.pbe-van\_bm.UPF** (details)

Perdew-Burke-Ernzerhof (PBE) exch-corr  
Vanderbilt ultrasoft  
author: bm

**O.pz-mt.UPF** (details)

Perdew-Zunger (LDA) exch-corr  
Martins-Troullier

**O.pz-rrkjus.UPF** (details)

Perdew-Zunger (LDA) exch-corr  
Rabe Rappe Kaxiras Joannopoulos (ultrasoft)

**O.blyp-van\_ak.UPF** (details)

Becke-Lee-Yang-Parr (BLYP) exch-corr  
Vanderbilt ultrasoft  
author: ak

Pseudopotential's name  
gives information about :

- type of exchange-correlation functional
- type of pseudopotential
- e.g.:

**O.pbe-rrkjus.UPF** (details)

Perdew-Burke-Ernzerhof (PBE) exch-corr  
Rabe Rappe Kaxiras Joannopoulos (ultrasoft)

- We will use only norm-conserving pseudo types: mt, vbc, hgh, dojo
- NB: Use the same XC type for all elements!

# 5. ATOMIC\_POSITIONS input card

**&control**

```
calculation = 'scf'  
prefix = 'silicon',  
pseudo_dir = './'  
outdir = './',
```

/

**&system**

```
ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,
```

/

**&electrons**

/

**ATOMIC\_SPECIES**

```
Si 28.086 Si.vbc.UPF
```

**ATOMIC\_POSITIONS {crystal}**

```
Si 0.00 0.00 0.00
```

```
Si 0.25 0.25 0.25
```

**K\_POINTS automatic**

```
2 2 2 0 0 0
```

Card: **ATOMIC\_POSITIONS** { **alat** | **bohr** | **angstrom** | **crystal** | **crystal\_sg** }

IF *calculation == 'bands' OR calculation == 'nscf' :*

Specified atomic positions will be IGNORED and those from the previous scf calculation will be used instead !!!

ELSE

Syntax:

**ATOMIC\_POSITIONS** { **alat** | **bohr** | **angstrom** | **crystal** | **crystal\_sg** }

X(1)   x(1)   y(1)   z(1)   { if\_pos(1)(1)   if\_pos(2)(1)   if\_pos(3)(1)   }

X(2)   x(2)   y(2)   z(2)   { if\_pos(1)(2)   if\_pos(2)(2)   if\_pos(3)(2)   }

...

X(nat)   x(nat)   y(nat)   z(nat)   { if\_pos(1)(nat)   if\_pos(2)(nat)   if\_pos(3)(nat)   }

List of **nat** atoms in terms of some choice of units (see INPUT\_PW.html)

The best choice depends on the system being studied, e.g. Angstrom for molecules, alat for slabs, crystal for bulk



## 6. K\_POINTS input card

**&control**

calculation = 'scf'  
prefix = 'silicon',  
pseudo\_dir = './'  
outdir = './',

/

**&system**

ibrav= 2,  
celldm(1)=10.2,  
nat= 2,  
ntyp= 1,  
ecutwfc = 12.0,

/

**&electrons**

/

**ATOMIC\_SPECIES**

Si 28.086 Si.vbc.UPF

**ATOMIC\_POSITIONS {crystal}**

Si 0.00 0.00 0.00

Si 0.25 0.25 0.25

**K\_POINTS automatic**

2 2 2 0 0 0

Possibilities are:

- Ask the code to generate a regularly spaced grid
- Specify a list of k-point coordinates by hand
- Use the gamma point only



# Types of k-point meshes

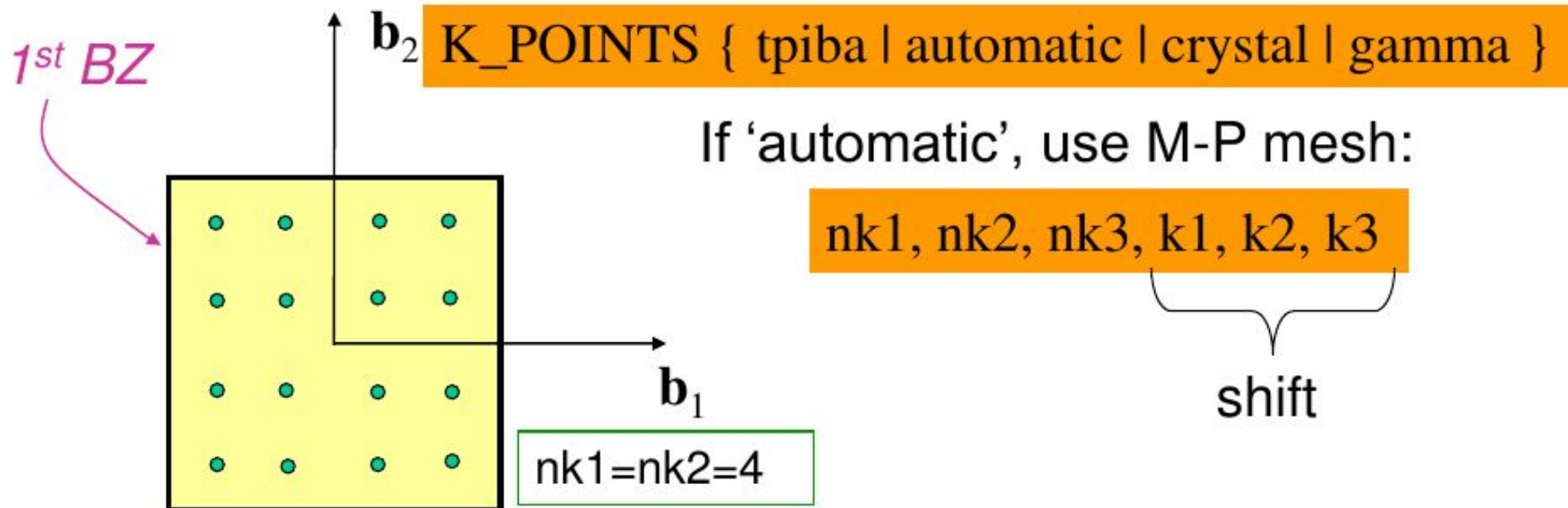
- **Special Points:** [Chadi & Cohen]

Points designed to give quick convergence for particular crystal structures.

- **Monkhorst-Pack:**

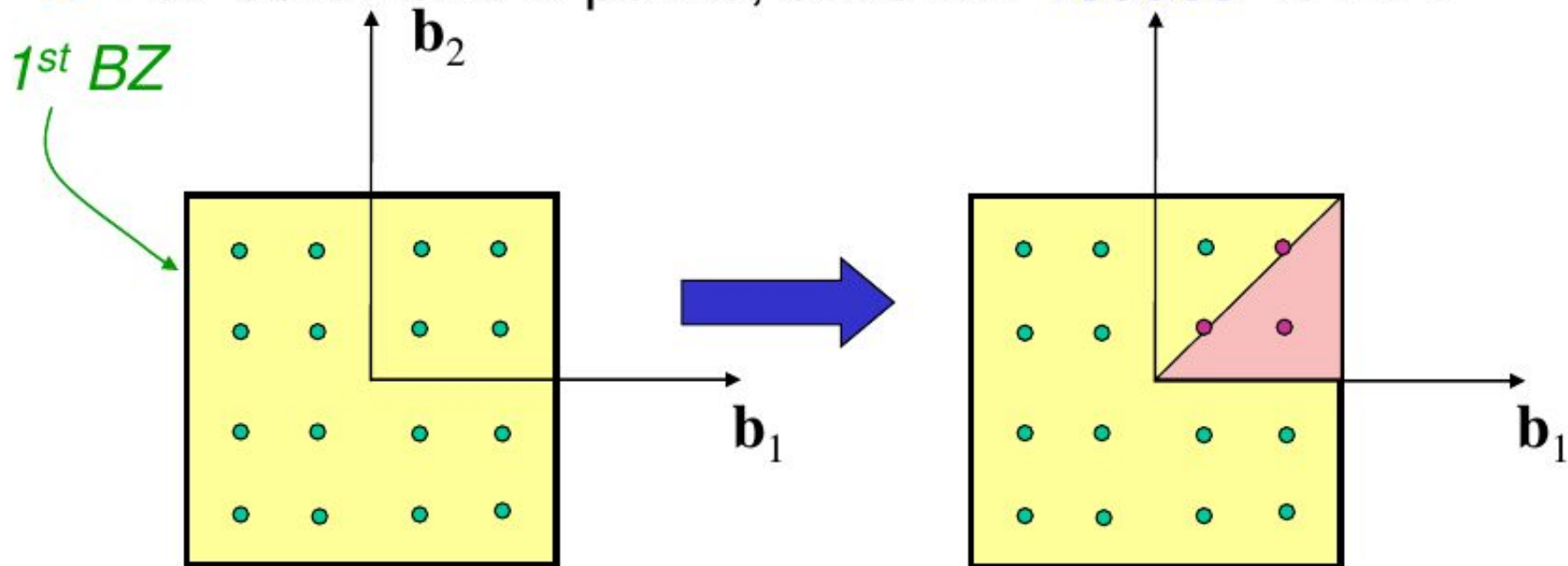
Equally spaced mesh in reciprocal space.

May be centred on origin ['non-shifted'] or not ['shifted']



# Irreducible Brillouin Zone

- IBZ depends on symmetries of system.
- Can save computational time by using appropriately weighted **k-points from IBZ** alone.
- For 'automatic' k-points, code will 'reduce' to IBZ.



- May not want to maintain symmetries in relaxation/MD.

Input parameter `nosym`