


An Efficient, High-order, Adaptive Finite Volume Solver for Modelling Capillary Effects in Advanced Manufacturing Processes

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Abstract—We present a free and open source implementation of the Finite Volume Method that captures thermal and solute-driven free-surface flows, more commonly known as Marangoni flows. This type of physics commonly appears within many manufacturing processes as well as microfluidic devices and is known to be challenging to resolve. The application includes adaptive mesh refinement coupled with high order WENO schemes in order to substantially save computational cost in multiphase flows where adequate tracking of the phase boundaries is crucial. By accurately simulating the temporal evolution of multiphase flow patterns, unique insight into e.g. the physics of melt pool formation in metal additive manufacturing is possible that is otherwise hard to gain experimentally. We demonstrate our work by a simple two-dimensional benchmark case and outline possible applications with a mesoscopic model of Laser Powder Bed Fusion additive manufacturing.

Index Terms—finite volume method, adaptive refinement, marangoni flow, open source software

I. INTRODUCTION

Accurate modelling of complex physics has received considerable attention within the last two decades. This can mostly be attributed to the fact that an increased understanding of the phenomena relevant for manufacturing directly leads to more finely tuned or even new processes that tend to increase output quality.

Two such examples that shall serve as a guideline for this work are microfluidic applications and additive manufacturing of metals. Both groups of processes are characterized by complex flow phenomena that involve several physical phenomena on a microscopic scale [1], [2]. The fact that those effects tend to have fast dynamics further complicates the analysis [3]. This leads to an increased amount of empirical effort necessary to capture and quantify the flow patterns involved.

Hence, simulation has become an important alternative to extensive experimental research. Within computational simulation frameworks, the involved physics can be precisely monitored, even down to a nanoseconds scale, which would oftentimes be prohibitively expensive to do experimentally.

At the same time, a considerable drawback of this approach is the additional amount of work that has to be put into the mathematical modelling of the involved physics. Especially within fluid dynamics, gaining a stable as well as accurate solution can oftentimes be challenging, as the resulting set of partial differential equations can have a small stability region in the temporal domain [4].

Therefore in this work, we propose and demonstrate a performant, free and open source software framework based on the Finite Volume method that is able to capture complex, non-isothermal surface flows. We discuss the necessary mathematical modelling in order to extend the existing solver ecosystem towards incorporating non-isothermal surface tension driven flows. We showcase the capabilities of the developed application using two example cases with different geometric and physical complexity.

II. RELATED WORKS

Modelling field-dependent capillary convection, also known as Marangoni convection, in itself is not a new field of research. Various works exist that address the need to develop robust numerical methods in order to resolve the complex flow patterns that arise. Those are mostly driven by application, i.e. they are tailored towards simulating a particular set of physics.

Early works include the investigation of marangoni effects in microgravity [5], [6], buoyancy-driven flows [7] and welding [8], [9].

After that, Marangoni convection in the microfluidic regime has received increasing attention within the last ten years [10]. Numerous scenarios have been investigated since including engineering of traps and pumps, stability in microchannels and particle accumulation in microfluidic flows [11]–[13].

Another field where the effects of capillary convection have become a dominant part of the involved physics is within metal additive manufacturing. Here, the focus lays on resolving temperature-dependent surface tension flows, as melt pools exhibit large spatial and temporal temperature gradients

[2]. Various methods have been employed to resolve the thermo-fluid dynamics on the powder scale including Arbitrary Lagrangian Eulerian [14], [15], Finite Difference [16], Finite Element [17], [18], Lattice Boltzmann [19], Smoothed Particle Hydrodynamics [20] and the Finite Volume Method which shall be used for this work as well [21]–[23].

However, the mentioned works exhibit some shortcomings when compared to the general current state of numerical modelling. Especially in applications where highly dynamic physics are strongly localized in the computational domain, adaptively refining the domain is a very desirable property that is not widely employed yet. Furthermore, the use of high order spatial approximations remains largely restricted to few numerical schemes of Galerkin type. However, achieving high order accuracy can help alleviate the need for overly fine computational meshes and hence is also desirable. Lastly, few of the presented solution along with their source codes are freely available. To the authors' best knowledge, none of the current works on the presented topic incorporate all three properties. Thus, we present a framework that allows for grid adaptivity, high order schemes and is openly available (see section VII for references).

III. THEORY

To describe the necessary linear algebra, we will use the index notation of tensors in a cartesian coordinate system alongside with the Einstein summation convention where the sum over matching indices is implied.

From here on, we assume a system of partial differential equations that involves the transient evolution of multiple fluid phases in a joint domain. This means we are tasked to solve at least the Navier-Stokes equations in order to capture velocity, pressure and temperature fields.

In addition, we need to properly discretize the different phases in the model. One way to achieve this in a Finite Volume framework is through the Volume-of-Fluid method [24]:

$$\frac{\partial \alpha_j}{\partial t} + \frac{\partial u_i \alpha_j}{\partial x_i} = 0 \quad (1)$$

Where α_j represents the j -th phase involved. From this, we obtain a set of scalar, hyperbolic conservation laws that track the cell-wise volumetric fraction of each phase over the domain.

After that, we can consider the effects happening at the interface of such phases, i.e. within this work we investigate surface tension. With no other physics involved, this leads to a contractional movement normal to the interface. If however varying fields, such as temperature are present that influence the surface tension coefficient, additional forces interact with the interface. This is shown in Fig. 1.

The theory of the proposed implementation of those surface tension effects relies on the Continuum Surface Stress Method initially presented by Lafaurie et al. [25], extending the works on modelling surface tension with the Volume of Fluid Method

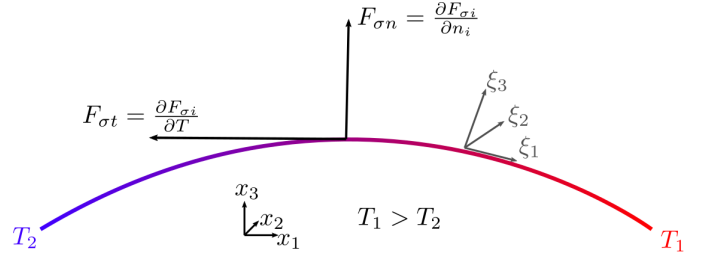


Fig. 1. Forces acting on an interface between two immiscible phases subject to a temperature gradient $\Delta T = T_2 - T_1$. The overall surface tension Force F_σ can be divided into a normal component $F_{\sigma n}$ and a tangential component $F_{\sigma t}$ in the coordinate system ξ_i proprietary to the interface. The tangential component is also called Marangoni force and depends on the gradient of external fields.

as described by Brackbill et al. [26]. First, the capillary stress tensor is introduced:

$$T_{ij} = -\sigma \delta_s (\delta_{ij} - n_i n_j) \quad (2)$$

Where σ is the generally non constant coefficient of surface tension. δ_s is the interface delta function that serves as an indicator function of where the phase boundaries are located. δ_{ij} is the Kronecker Delta function and n_i is the interface unit normal vector. The local coordinate system of the interface needs to be an orthogonal system in order to separate the purely geometry based capillary force and the tangential Marangoni forces (c.f. Fig. 1). The resulting force accounting for all surface tension effects can now be expressed as the divergence of the capillary stress tensor, yielding:

$$\frac{\partial T_{ij}}{\partial x_j} = \frac{\partial \sigma}{\partial x_j} [\delta_s (\delta_{ij} - n_i n_j)] + \frac{\partial \delta_s}{\partial x_j} [\sigma (\delta_{ij} - n_i n_j)] + \sigma \delta_s \frac{\partial}{\partial x_j} (\delta_{ij} - n_i n_j) \quad (3)$$

One can show that Eq. 3 can be re-arranged using some tensor algebra and geometric identities into a much shorter and more useful form. The derivation is given in more detail by Lafaurie et al. [25]. By performing those rearrangements, we can recover a form that separates the normal from the tangential components of the divergence vector:

$$\frac{\partial T_{ij}}{\partial x_j} = \frac{\partial \sigma}{\partial x_i} [\delta_s (\delta_{ij} - n_i n_j)] - \sigma \kappa n_i \delta_s \quad (4)$$

Here, the interface curvature κ is introduced. The second term of the right hand side corresponds to the normal capillary force (Fig. 1, $F_{\sigma n}$) directed in the normal direction of the interface, effecting a contractional movement of the interface. The first term resembles the marangoni-type forces (Fig. 1, $F_{\sigma t}$) present. By taking the derivative of the surface tension coefficient, this term does not vanish if and only if spatial gradients exist at the interface. This is normally the case when there are multiple species involved or temperature gradients present [27]. Note that we then simply evaluate the ordinary form of surface tension without any additional physics present.

However, as the surface tension coefficient is otherwise not a direct local variable, one must further differentiate the term:

$$\frac{\partial \sigma(c, T)}{\partial x_j} = \frac{\partial \sigma(c, T)}{\partial T} \frac{\partial T}{\partial x_j} + \frac{\partial \sigma(c, T)}{\partial c} \frac{\partial c}{\partial x_j} \quad (5)$$

Additionally, the delta function δ_s needs to be discretized in a suitable manner in order to capture the physics within the Finite Volume framework. This can be done using the Volume of Fluid Method by taking the gradient of the phase volume fraction α [24], [28]:

$$\delta_s = \left| \frac{\partial \alpha}{\partial x_i} \right| \quad (6)$$

The interface unit normal vector n_i can be computed in a similar manner by using the previously computed interface function:

$$n_i = \frac{1}{\delta_s} \frac{\partial \alpha}{\partial x_i} \quad (7)$$

IV. IMPLEMENTATION

The presented theoretical foundation is modified and implemented based on the open source library *OpenFOAM*, which is a popular software library that implements the Finite Volume Method [29]. The modifications rely on the works of Gueyffier et al. in order to account for the additional Marangoni stresses [28].

The proposed code computes the divergence of the capillary stress tensor for finite volume cell centers. However, to handle source terms within the PIMPLE loop, it is necessary to also supply the surface tension force as a scalar interpolated to the cell face. Therefore, an additional method is implemented that computes the L_2 norm of the surface tension force vector and interpolates the scalar value via finite differencing.

A very useful property of this approach is that not only thermally driven surface tension effects can be incorporated, but also gradients arising from any kind of inhomogeneity. This means that, among others, solute-driven effects can also be modelled. This scenario is common in manufacturing processes involving mixing of multiple solvents. Within the context of additive manufacturing, in-situ alloying is a scenario where such effects are not negligible [30].

To account for thermal and solute-driven effects simultaneously, we need to incorporate the temperature dependence of the surface tension as well as interfacial tension between the involved phases. This allows us to capture not only flow patterns on liquid-gas interfaces, but also between liquid interfaces of dissimilar species. We may find accurate values for temperature dependent surface tension in the literature. However, determining interfacial tension poses a non-trivial problem that has been discussed in the literature [31].

In a pursuit to keep model evaluation simple and not rely on Molecular Dynamics calculations, we use the following, relatively simple approximation for interfacial tension given by Girifalco and Good [32]:

TABLE I

MATERIAL PROPERTIES OF THE OSCILLATING DROPLET CASE [35]. THE INDICES INDICATE EITHER A PROPERTY OF THE SURROUNDING MEDIUM (1) OR OF THE DROPLET (2).

Quantity	Value [Unit]
Density ρ_1	250 $\frac{\text{kg}}{\text{m}^3}$
Density ρ_2	$2\rho_1$
Heat capacity $c_{p,1}$	$5 \cdot 10^{-5} \frac{\text{J}}{\text{kg K}}$
Heat capacity $c_{p,2}$	$2c_{p,1}$
Viscosity μ_1	0,012 $\frac{\text{kg}}{\text{m s}}$
Viscosity μ_2	$2\mu_1$
Thermal conductivity λ_1	$1,2 \cdot 10^{-6} \frac{\text{W}}{\text{m K}}$
Thermal conductivity λ_2	$2\lambda_1$
Surface tension coefficient σ	0,1 $\frac{\text{N}}{\text{m}}$
Marangoni coefficient σ_T	0,02 $\frac{\text{N}}{\text{m K}}$

$$\sigma_{1,2} = \sigma_1 + \sigma_2 + 2C(\sigma_1\sigma_2)^{1/2} \quad (8)$$

Here, $\sigma_{1,2}$ denotes the interfacial tension, σ_1 and σ_2 are the surface tensions of the involved species and C is a constant which we assume to be one as a reasonable approximation [31].

In order to approximate the differential terms in the governing equations efficiently, we use high order weighted essentially non-oscillatory (WENO) schemes that have been developed separately for OpenFOAM by Gärtner et al. [33], [34]. In this implementation, the numeric stencil coefficients to specified order are pre-computed and locally stored to alleviate the harsh overhead in memory requirements. The theory behind the derivation of high order WENO stencils is given in Gärtner et al. [33].

V. NUMERICAL EXPERIMENTS

First, we replicate and examine the numerical setup proposed by Ma et al. that models the flow around a stationary bubble subject to a temperature gradient [35]. We aim to accurately predict the velocity field that arises due to marangoni convection at the interface.

We then proceed to investigate the mesoscopic nanosecond dynamics during selective laser melting of steel, a common problem within additive manufacturing. Here, we show that our implementation is able to handle geometrically and physically complex simulations in 3D.

A. Marangoni Flow around a droplet

We now investigate a simple, two-dimensional benchmark case that illustrates the ability to capture Marangoni effects for the present work.

An overview of the material parameters is given in table I.

We initialize the droplet in the center of a rectangular domain of length $4a \times 4a$ where $a = 1,44 \cdot 10^{-3} \text{m}$ with a radius of a . The droplet is subject to a temperature gradient of $200 \frac{\text{K}}{\text{m}}$ where the bottom temperature is fixed to 290 K.

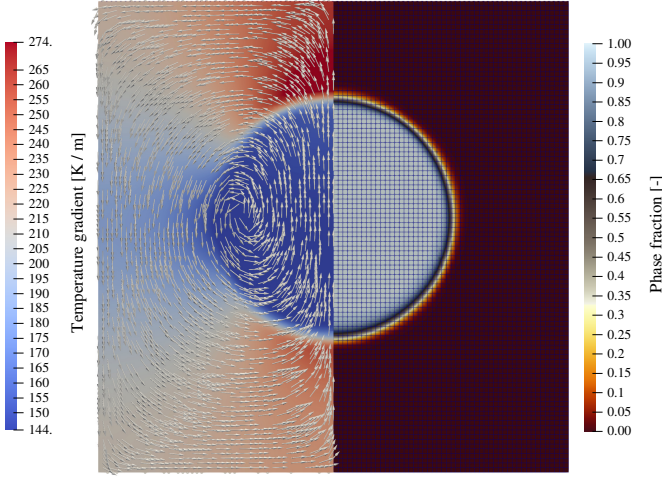


Fig. 2. Flow field around the droplet at final time $t = 0.12$ s. The resulting velocity vector field (arrows) and temperature gradient field are shown on the left slice of the domain. The right side shows the phase fraction of the droplet as well as the spatial discretisation.

We discretize the domain using a coarse grid of 100×100 quadrilaterals. The spatial discretization of the divergence is done using a mixture of first order bounded Gaussian schemes and fourth order WENO schemes. We impose no-slip boundary conditions at the walls as well as Neumann type boundaries for the temperature at the left and right walls. The temperature gradient is enforced during the entire simulation using Dirichlet boundaries at the bottom and top. We simulate the evolution of the involved fields up to $t = 0.12$ s.

The remainder of the involved numerics and parameters, as well as all boundary conditions can be accessed via the online repository given in section VII.

The resulting fields at the final time step $t = 0.12$ s is shown in Fig. 2. We can clearly observe the field of vortices that form around the bubble, initiated from varying pressures along the interface. This is due to the Marangoni effect, resulting in an uneven pressure distribution in the domain. Following equation 4, the first term involving the gradient of the surface tension coefficient - i.e. the marangoni coefficient times the gradient of the temperature field - does not vanish here and hence induces a non homogeneous pressure contribution. The resulting two large vortices drive the droplet to an upwards motion, in direction of increasing temperature gradient.

Despite the comparatively low mesh resolution, the phase boundary is conserved reasonably well after solving for 6000 time steps. A comparison of the sharp interface between the two phases at the beginning and end of the simulation is given in Fig. 3. During the solution process, only minimal numerical diffusion of the interface occurs, indicated by the slightly tapered off edges of the rectangular profile at end time. In addition, we don't observe any oscillatory behaviour in the fields which is essential for obtaining a stable solution.

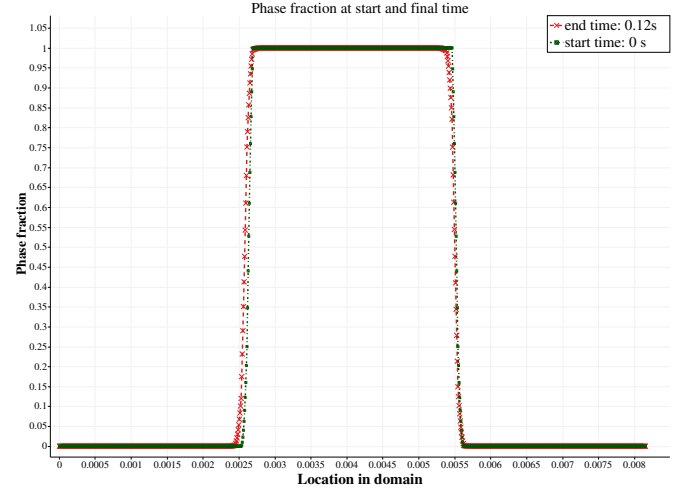


Fig. 3. Phase fraction profiles of the droplet at beginning and end of the investigated temporal domain. This cross-section represents a diagonal slice from the left top to the right bottom of the grid shown in Fig. 2.

B. Powderbed-scale physics during selective laser melting

We now turn to a more complex and relevant simulation setup that appears regularly in additive manufacturing. We aim to investigate the melting, solidification and evaporation behavior of molten metal during the Laser Powder Bed Fusion Process (PBF-LB/M).

This case shall serve as a showcase of a physically complex problem that involves phase changes, simultaneous handling of solid, liquid and gaseous phases (and hence strong pressure gradients) as well as spatially and temporally localized heat transfer.

The computational domain with the initial conditions for the phase fractions is shown in Fig. 4. We create the packing of the metallic powder particles using a rain drop model implemented in the open source Discrete Element Method software Yade [36]. Overall, the domain spans a cuboid geometry of size $0.5 \text{ mm} \times 0.3 \text{ mm} \times 0.7 \text{ mm}$ and contains roughly 800.000 hexahedral cells arranged in a cartesian grid. The inhomogeneous temperature field is generated by a laser heat source of diameter $50 \mu\text{m}$ that inputs an amount of heat equivalent to $67 \frac{\text{J}}{\text{mm}^3}$.

The physical parameters of the PBF-LB/M model are given in table II. The numerical parameters are given in the case files referenced in section VII.

In Fig. 5, we can observe the complex flow field that evolves within the molten material. The melt region largely penetrates the already solidified material below the powder bed, which agrees well with experimental observations done on laser melting of 316L stainless steel [39]. Examining the cross-section further, we observe high flow velocities at the surface of the molten material, indicated by arrows in Fig. 5. The large spatial temperature gradients produce variations in surface tension and hence lead to Marangoni convection at the surface. This current of liquid metal helps to distribute the heat that is input by the laser over the entire melt pool.

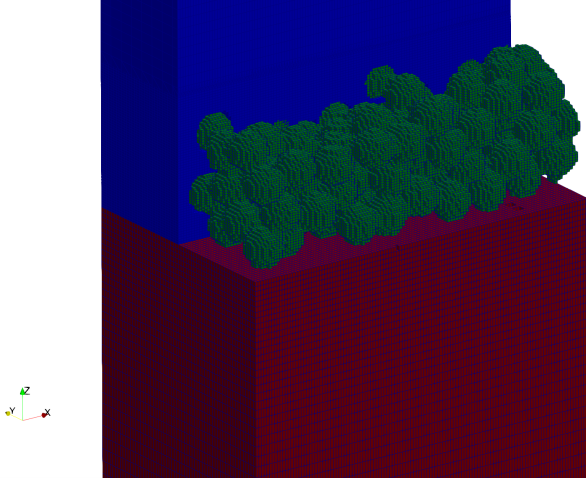


Fig. 4. Composition of the computational domain at the initial time step. Solidified build platform (red), discretised powder bed in solid state (green) and gaseous argon atmosphere (blue). As reference, the individual cell boundaries are marked with blue lines. The initial grid is finest where the temperature gradients are expected to be highest.

TABLE II

SELECTION OF RELEVANT SIMULATION PARAMETERS OF THE LASER POWDER BED FUSION TEST CASE. THE SUBSCRIPTS L, S AND G SIGNIFY THE LIQUID, SOLID AND GASEOUS PHASES OF 316L STAINLESS STEEL INVOLVED IN THE MODEL. [37], [38]

Symbol	Value [Unit]
Laser Power	300 W
Laser scan speed	$0,9 \frac{\text{m}}{\text{s}}$
Density ρ_s	$7950 \frac{\text{kg}}{\text{m}^3}$
Heat capacity $c_{p,s}$	$412 \frac{\text{J}}{\text{kg K}}$
Thermal conductivity λ_s	$1,2 \cdot 10^{-6} \frac{\text{W}}{\text{m K}}$
Enthalpy of fusion H_f	$2,6 \cdot 10^5 \frac{\text{J}}{\text{kg}}$
Density ρ_l	$6881 \frac{\text{kg}}{\text{m}^3}$
Heat capacity $c_{p,l}$	$790 \frac{\text{J}}{\text{kg K}}$
Viscosity μ_l	$5,85 \cdot 10^{-3} \frac{\text{kg}}{\text{m s}^2}$
Thermal conductivity λ_l	$6,6 \frac{\text{W}}{\text{m K}}$
Heat capacity $c_{p,g}$	$1900 \frac{\text{J}}{\text{kg K}}$
Viscosity μ_g	$1,8 \cdot 10^{-5} \frac{\text{kg}}{\text{m s}^2}$
Thermal conductivity λ_g	$6,6 \frac{\text{W}}{\text{m K}}$
Surface tension coefficient σ	$1,908 \frac{\text{N}}{\text{m}}$
Marangoni coefficient σ_T	$-1,622 \cdot 10^{-5} \frac{\text{N}}{\text{m K}}$

Furthermore, we can examine a small eddy forming within the melt pool, indicated by the circularly arranged arrows in Fig. 5. This is also to be expected, albeit not due to Marangoni convection, and assists the homogenization of the temperature field within the melt pool.

VI. SUMMARY AND FUTURE WORK

In this work, we proposed a numerical solver based on the Finite Volume method that is able to incorporate various physics that are relevant for modelling non-isothermal surface flows of arbitrary many phases. We demonstrated the capabilities of the framework using one simple, two-dimensional case

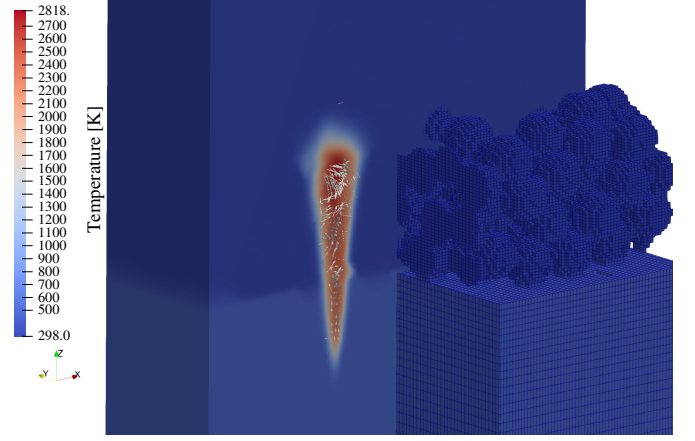


Fig. 5. Temperature field of the investigated Laser Powder Bed Fusion process. The velocity field of the melt pool is indicated by arrows. Additionally, the discretised powder bed is shown along with the computational grid as a reference.

showing the effects of marangoni convection and subsequently with a considerably more complex and application-driven process modelling a scenario common in additive manufacturing.

We expect that the results of this work will help to improve understanding of the complex physics of non-isothermal surface tension driven flows substantially. Furthermore, the high fidelity data that can be generated using the proposed solver can be used in order to train physics informed machine learning models [40]. It has previously been shown that such models can speed up simulations by order of magnitudes by execution of a simple forward pass. However, accurate and rich training data are needed in order to train such models [41], [42]. We anticipate that this work can help supply this data and further enhance the field of modelling for complex manufacturing processes, as experimental data for the applications and regimes discussed are not easy to generate.

Further means of extending the proposed framework include the incorporation of ray tracing algorithms in order to accurately predict heat source dynamics, further tweaking of the underlying volume of fluid formulation for robustness and accuracy and improving parallel solving capabilities on specialized hardware, such as General Purpose GPU architectures (GPGPU).

VII. DATA AVAILABILITY

The presented solver is available via Github at [pzimbrod/thermocapillaryInterFoam](https://github.com/pzimbrod/thermocapillaryInterFoam). All data used in this article along with the case files discussed in sections V-A and V-B can be accessed at [pzimbrod/ieee-icccme-2022](https://github.com/pzimbrod/ieee-icccme-2022).

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